

National Energy Technology Laboratory (NETL)
Office of Fossil Energy and Carbonization Management (FECM)

Ceramic Matrix Composites for H₂ Combustion (FE003228)

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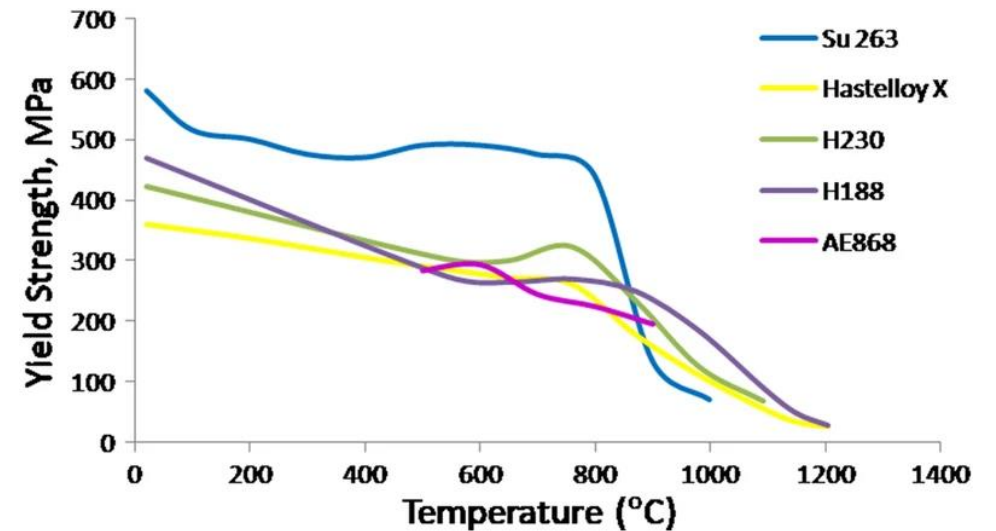
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University of Miami, Miami, FL

University Turbine Systems Research (UTSR) & Advanced Turbines Project Review Meeting
University of Alabama at Tuscaloosa, September 24-25, 2024

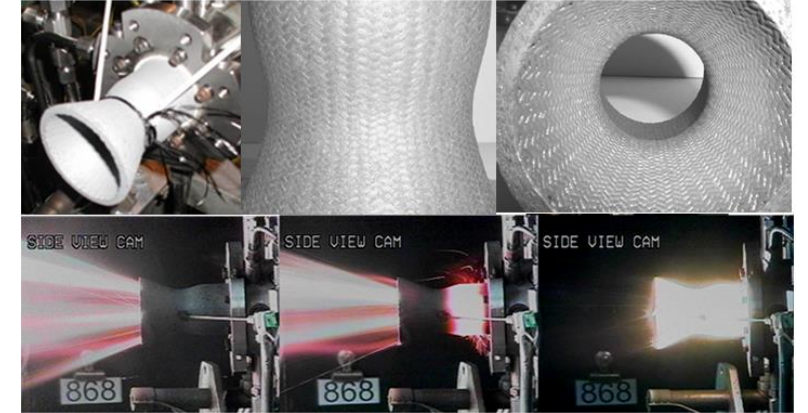


- Gas turbine engines for power generation are under transition to hydrogen-based combustion systems to achieve net-zero or net-negative carbon emissions.
- A transition to hydrogen-based fuel combustion systems heavily relies on advancements in materials technology.
 - Hydrogen burns $\sim 250^{\circ}\text{C}$ hotter than natural gas.
 - Current metal-based components are often operated very close to their melting points (within 100°C)
 - Large amounts of water vapor production oxidizes current metal materials.
 - Small molecular size of hydrogen interacts metals to hydrogen embrittlement and can cause dangerous fuel leaks.



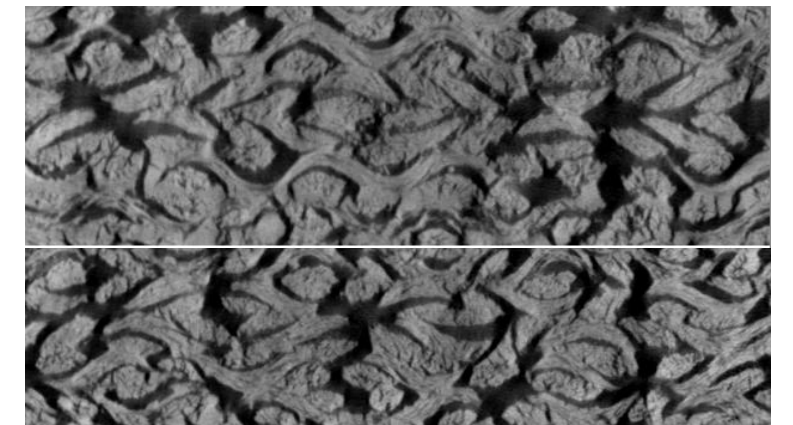
Strength of various superalloys in combustor and afterburner application.
(<https://doi.org/10.1007/s41745-022-00295-z>)

- CMCs are investigated as a possible alternative to metal alloy components in H_2 gas turbine engines for their thermal and chemical resistivity as well as high customizability.
 - Modular nature of composites allow for selection of fiber, matrix, and additives that tailor the material to the intended application.
 - Have seen success in use as refractory materials in gas turbine engines using traditional fuels – must be adapted to address unique challenges posed by H_2 combustion.
- CMC production does not come without its own set of challenges:
 - High processing temperatures during pyrolysis result in shrinking and thermal warping that impacts final part geometry.
 - Outgassing of volatiles during pyrolysis result in highly porous matrix (densification via multiple re-infiltrations is necessary).
 - Brittleness of CMC materials make them difficult to machine after manufacturing.

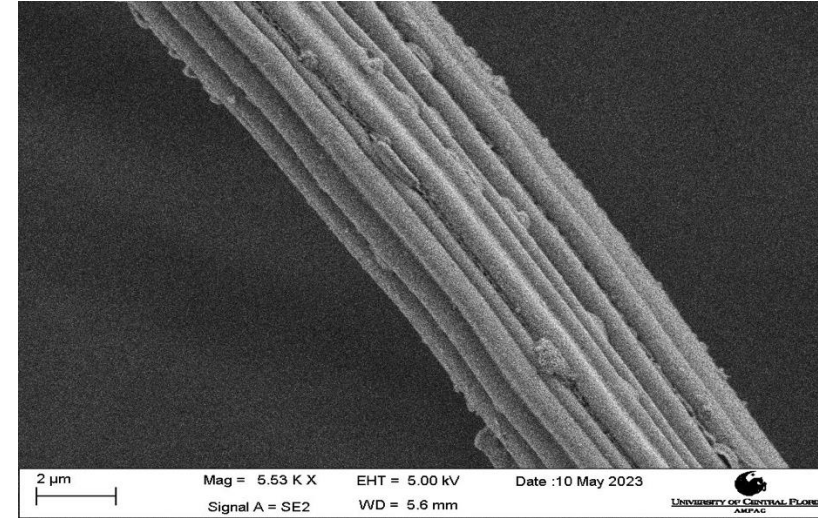
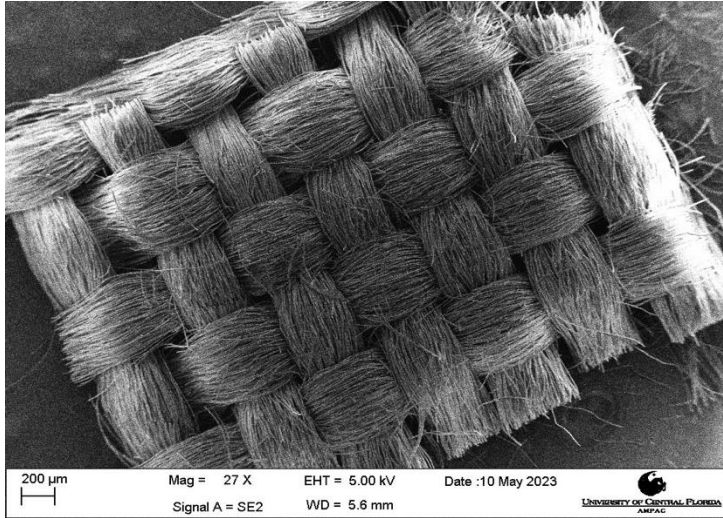


Cf/ZrC CMC combustion chamber undergoing oxygen-hydrogen hot fire testing.

(<https://ultramet.com/ceramic-matrix-composites/ceramic-matrix-composites-performance/>)

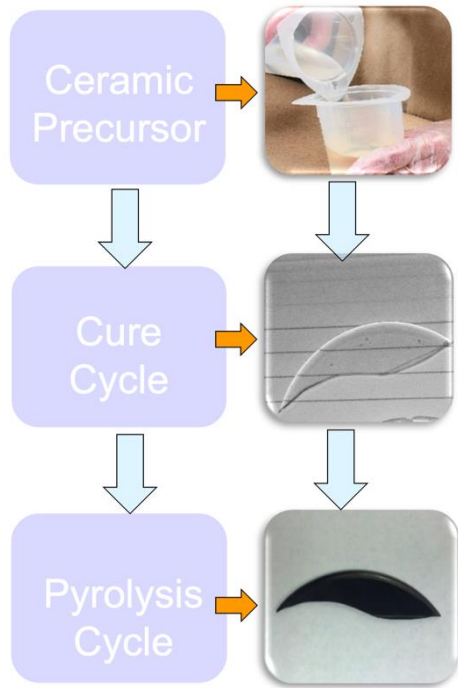


Micro-CT image of porous CMC cross section before densification.



- **Yttria-Stabilized Zirconia (YSZ) is the ceramic fiber used in the CMCs**
 - Currently used in thermal barrier coatings.
 - Melting point of 2,590°C with continuous use limit of 2,200°C.
 - Excellent performance in corrosive & oxidizing environments.
 - High porosity of woven YSZ results in effective wetting and solution retention.
 - Phase-stabilized with Yttria eliminates disruptive phase transitions.
- **Zirconium oxide rigidizer contains sub-micron particles of YSZ in a zirconium acetate aqueous solution**
 - Used in fabrication to provide dimensional stability and mechanical strength to laminates while increasing YSZ content.

- When selecting ceramic precursor, factors such as ceramic yield, processing ability, and thermal performance should be considered.

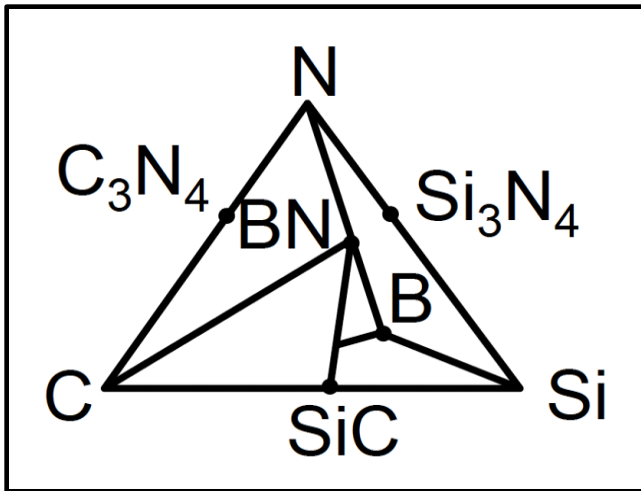


Polymer to Ceramics:

- Low cost
- Near net shape manufacturing
- Outstanding thermo-chemical stability

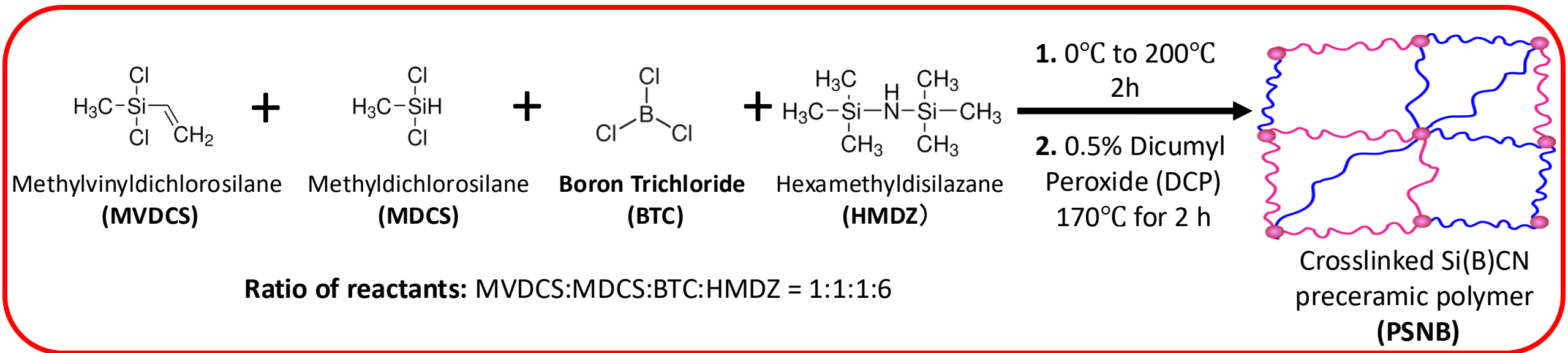
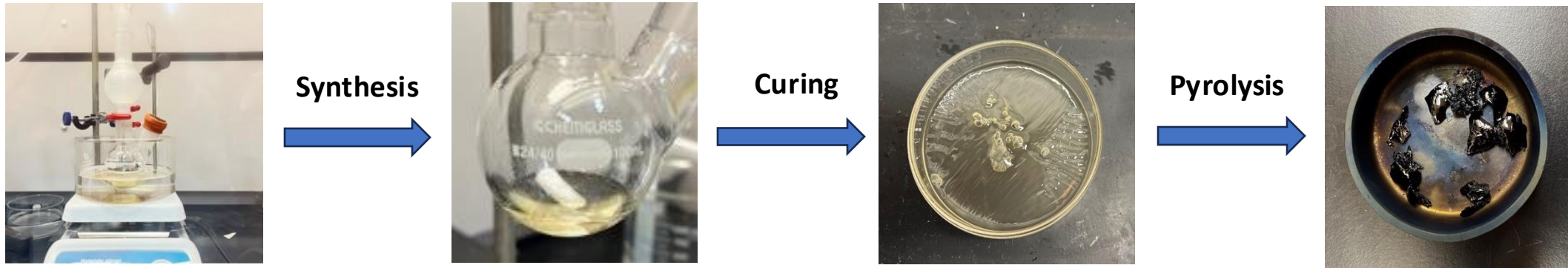
Ceramic Precursor	Resulting Ceramic	Ceramic Yield (Literature)	Ceramic Yield of Precursor (Experimental)
SPR-688	SiOC	65-85%	79.08%
SMP-10	SiC	72-78%	73.02%
Durazane 1800	SiCN	80-90%	82.45%

Commercially available preceramic polymers



**Amorphous Ceramic
Si(B)CN**

- **Boron** significantly enhances the **high-temperature stability**.
- The presence of **boron delays** the onset of **crystallization**, enabling the material to maintain its amorphous structure at higher temperatures and maintaining **structural integrity** of the CMC.
- By forming protective borosilicate glass layer, enhancing **resistance to oxidation**.
- The incorporation of boron leads to the formation of stronger bonds, providing Si(B)CN ceramics with **high flexural strength**, even at elevated temperatures.

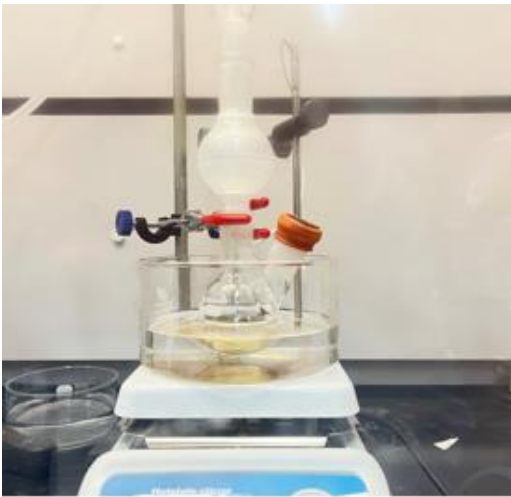


- Synthesis procedures of the Si(B)CN preceramic polymer precursor in the
- Curing was conducted in presence of 0.5% Dicumyl Peroxide (DCP) at 170°C for 2 hours.
- Pyrolysis was conducted at 850°C for 2 hours.

PSNB : A liquid polyborosilazane precursor for Si-B-C-N ceramic by co-condensation reaction

❑ Synthesis (Nitrogen)

- Boron Trichloride (BTC)
- Methylvinylchlorosilane (MVDCS)
- Methylchlorosilane (MDCS)
- Hexamethyldisilazane (HMDZ)



❑ Drying (Nitrogen)

- At 200 °C for 1 h in vacuum
- PSNB was obtained as a light yellow viscous liquid

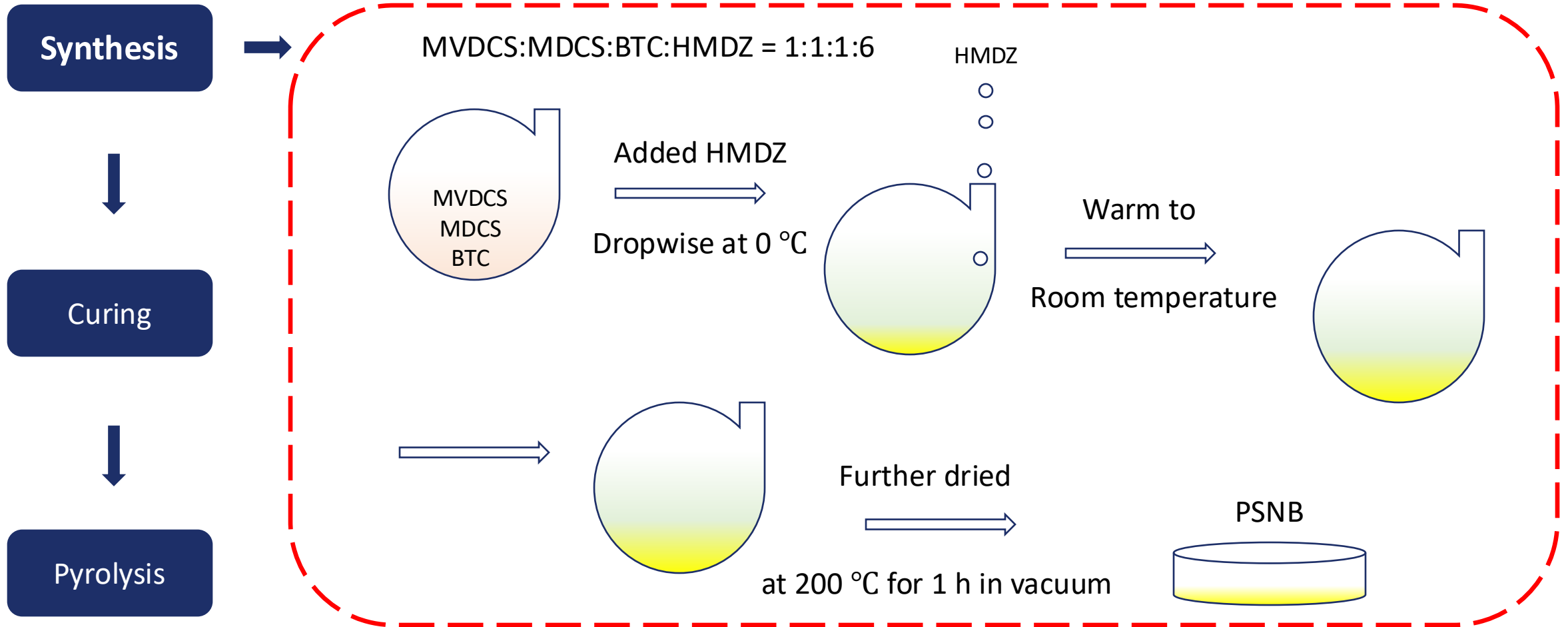
❑ Curing (Nitrogen)

- Dicumyl Peroxide (DCP) of 0.5 wt% was added
- Obtained a hard bulk cured PSNB with light yellow color



❑ Pyrolysis (Argon)

- At a heating rate of 5 °C/min
- 850 °C for 2 h
- Got black and glassy sample



Synthesis of Si(B)CN preceramic polymer using the Schlenk Line technique

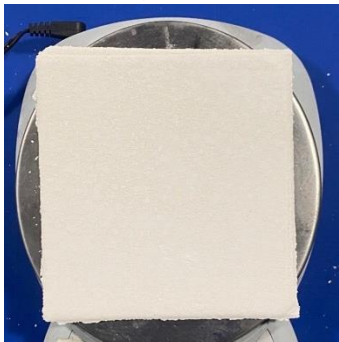
- Polymer Infiltration and Pyrolysis (PIP) is the manufacturing method used in this study
 - Relative ease and low cost of manufacturing make PIP an attractive option compared to other CMC manufacturing methods
 - Volatilization of organic compounds in ceramic precursors result in very high initial porosity: multiple re-infiltrations required for dense samples
 - Initial samples underwent multiple cycles of PIP



Reinfiltrated samples are pyrolyzed again under same conditions

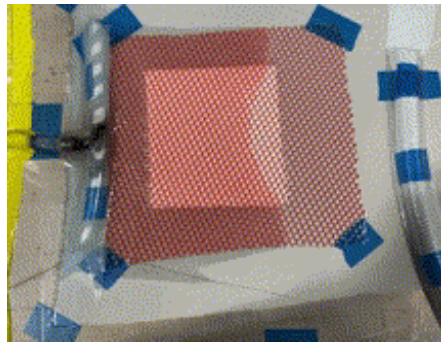


Hand layup of YSZ 'preform' consisting of 8 layers of YSZ fiber saturated in YSZ rigidizer



The 'preform' is dried in autoclave for 2 hours at 180°C

The preform is then saturated with pre-ceramic polymer via vacuum infusion

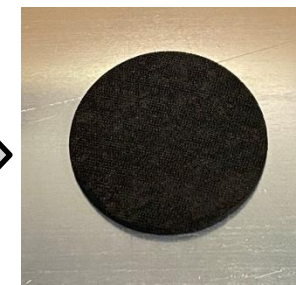


The polymer-infused laminate is cured in autoclave for 1 hour at 180°C then 2 hours at 200°C

The 'green body' material is waterjet cut into desired geometry before undergoing pyrolysis



Samples undergo pyrolysis at 950°C for 2 hours in N₂ atmosphere

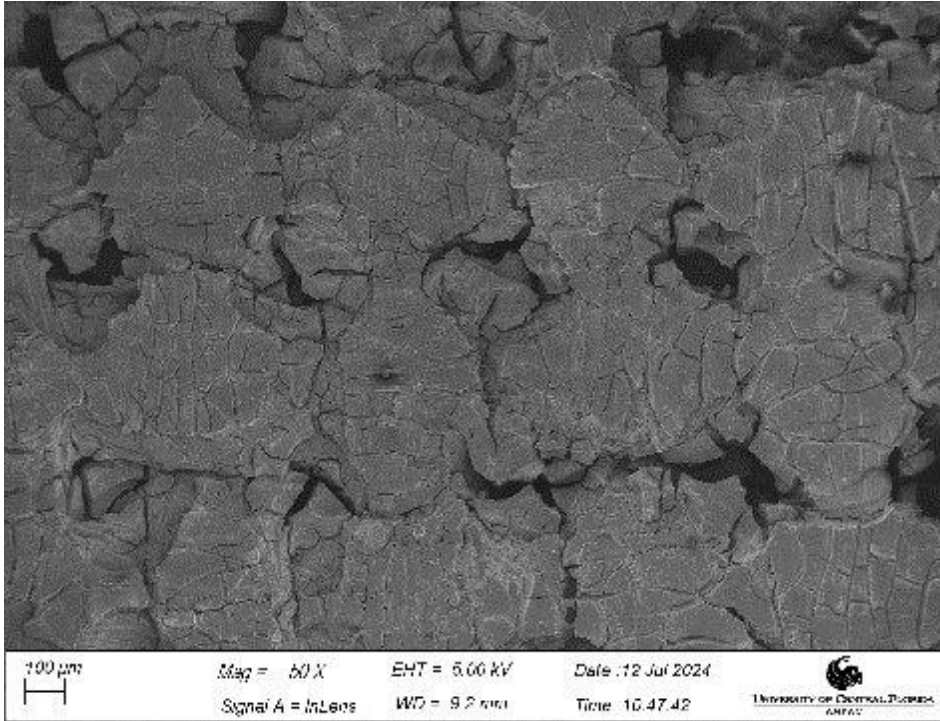


Resulting matrix phase is amorphous

CMCs are re-infiltrated with more pre-ceramic polymer and cured in autoclave again



This 'densifies' the composite and reduces porosity

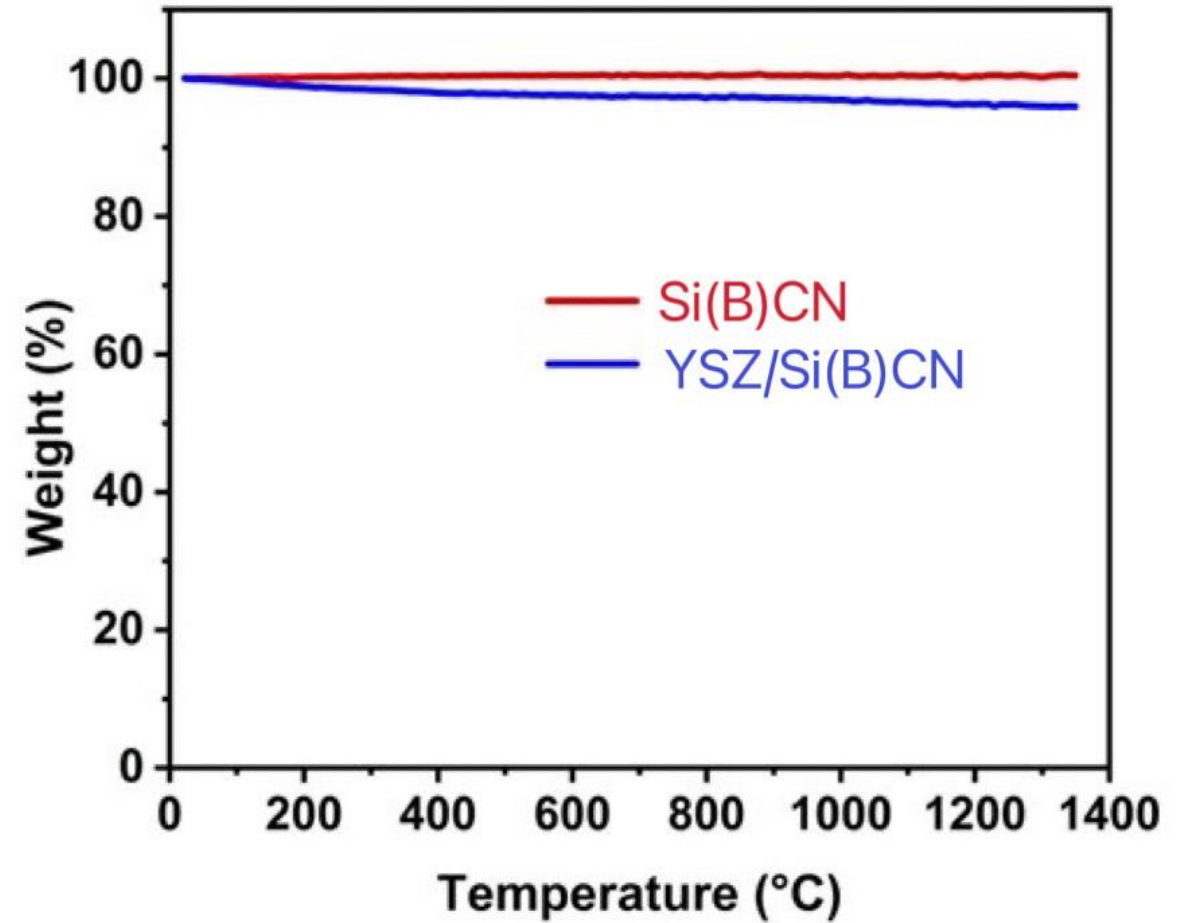


Element Line	Weight %	Weight % Error	Atom %	Atom % Error
<i>B K</i>	2.1	± 0.1	5.4	± 0.4
<i>C K</i>	9.3	± 0.2	21.4	± 0.4
<i>N K</i>	2.9	± 0.3	5.8	± 0.6
<i>O K</i>	22.2	± 0.2	38.3	± 0.4
<i>Si K</i>	14.6	± 0.2	14.4	± 0.2
<i>Zr L</i>	48.8	± 0.9	14.7	± 0.3
<i>Zr M</i>	---	---	---	---
Total	100.0		100.0	

- The SEM and EDS analysis has confirmed the presence of elemental Zr, **B**, Si, C, and N.
- The fabrication of YSZ/Si(B)CN ceramic matrix composites was successful.

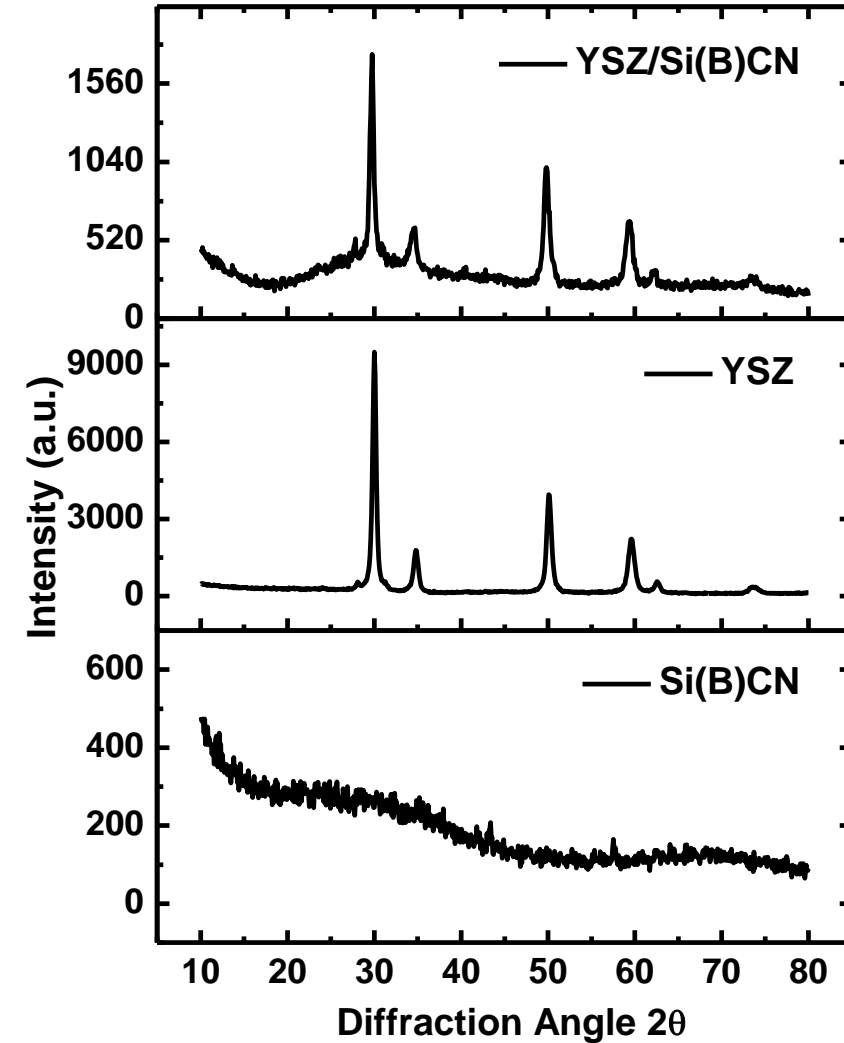
Thermal Stability of Ceramic Matrix and CMC

- Si(B)CN: almost no weight loss was observed at 1,350°C.
- YSZ/Si(B)CN: ~4% weight loss was measured at 1,350°C.

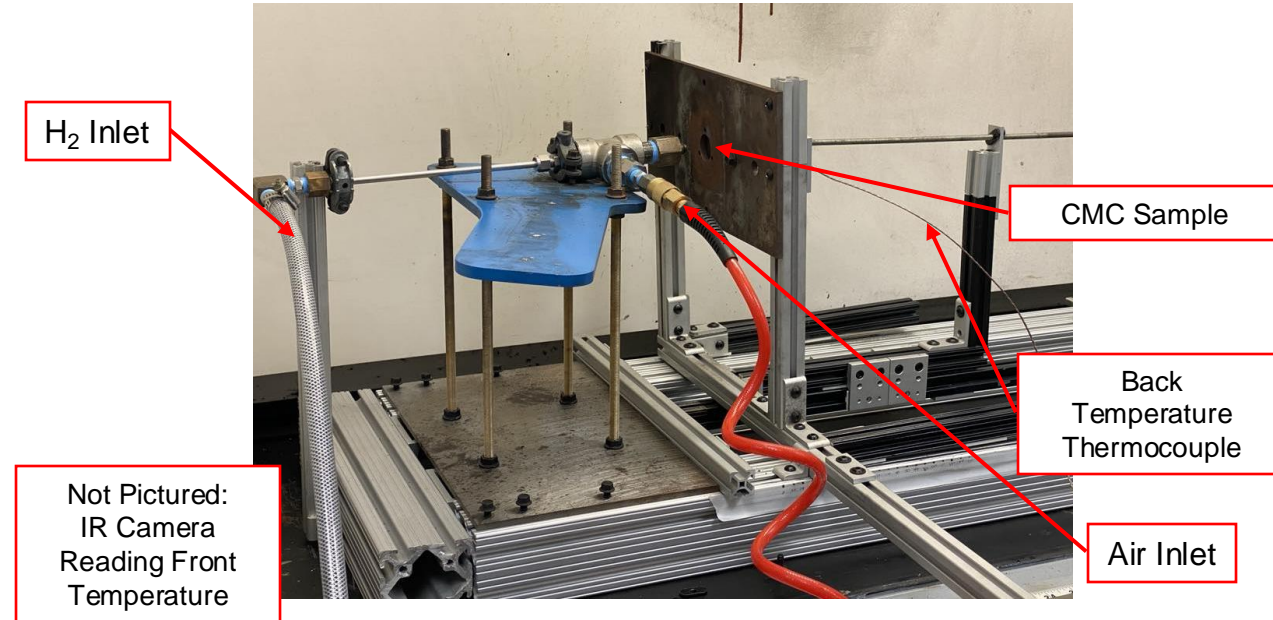
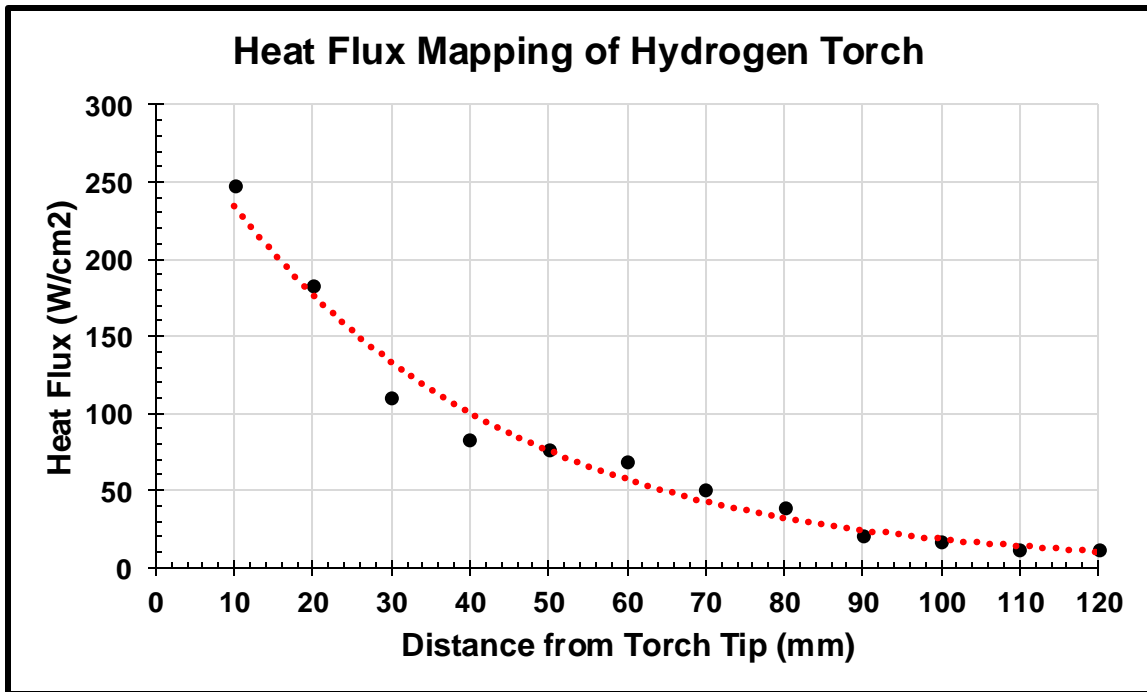


Phase Analysis of Ceramic Matrix and CMC

- ❑ The XRD analysis suggested the Si(B)CN matrix after the pyrolysis remained in an amorphous state
- ❑ The crystallinity in the YSZ/Si(B)CN composite arose from the presence of the YSZ fiber.

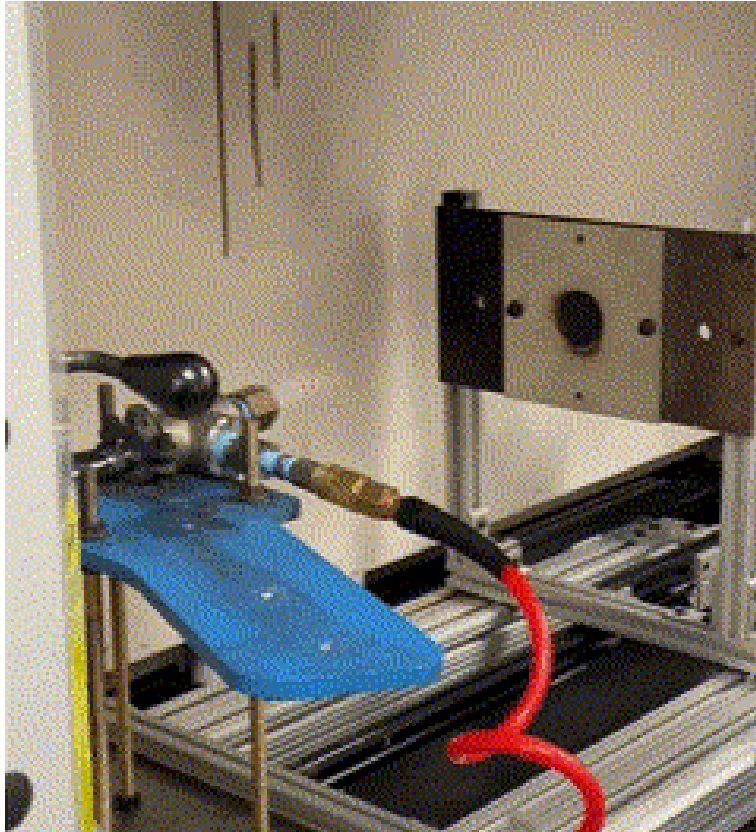


- Air and fuel flow rates measured with control orifices and upstream pressure regulators
- Heat flux is mapped at various distances from the torch tip
- Hydrogen torch gives us insight on how material behaves in hydrogen- and water vapor-rich erosion environment

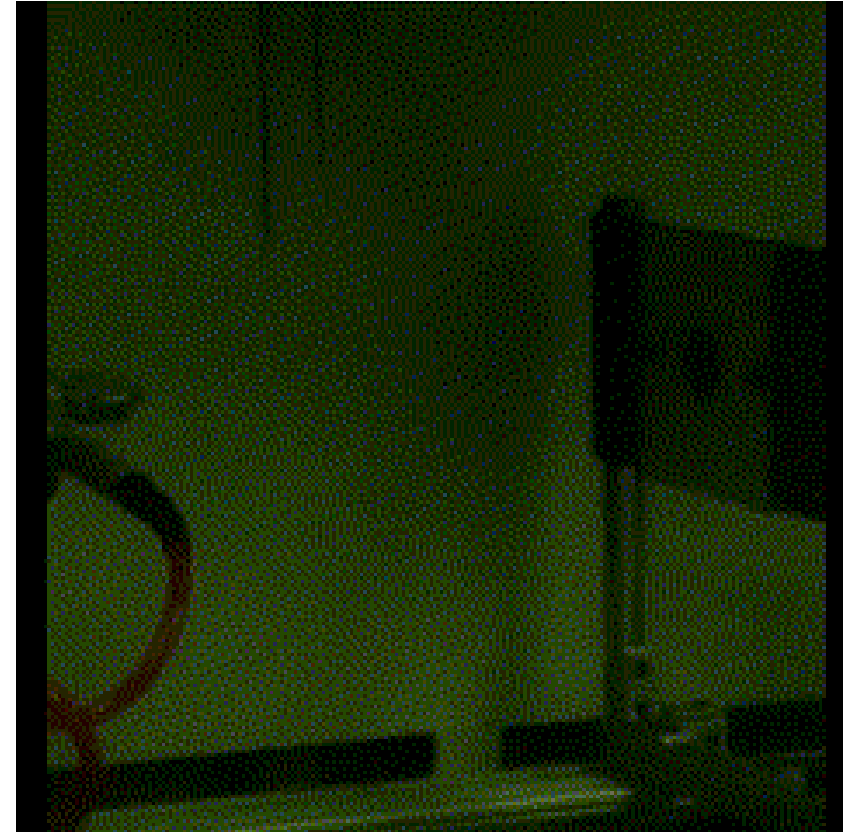


Test Conditions	
Heat Flux (W/cm ²)	183.3
Flame Temperature (°C)	2,000
Exit Velocity (m/s)	30
Equivalence Ratio	>1
Exposure Duration (s)	600

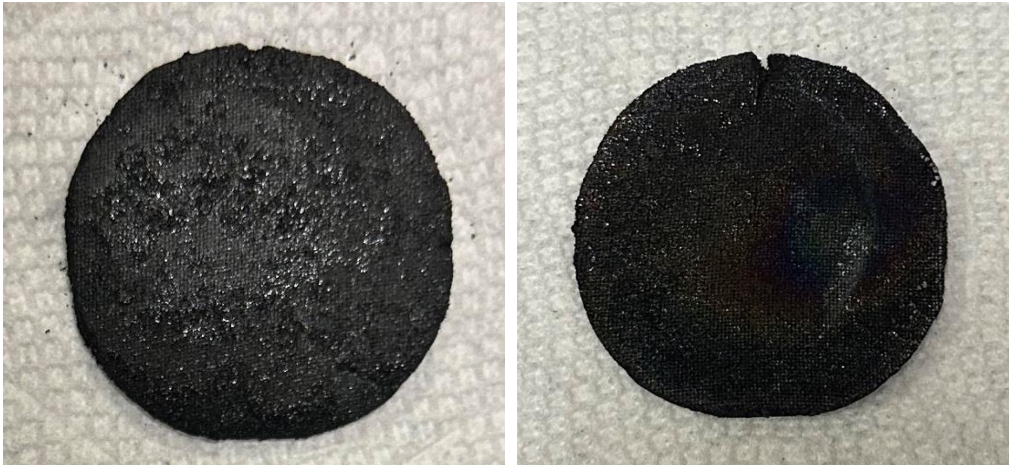
Hydrogen/Air Torch Test



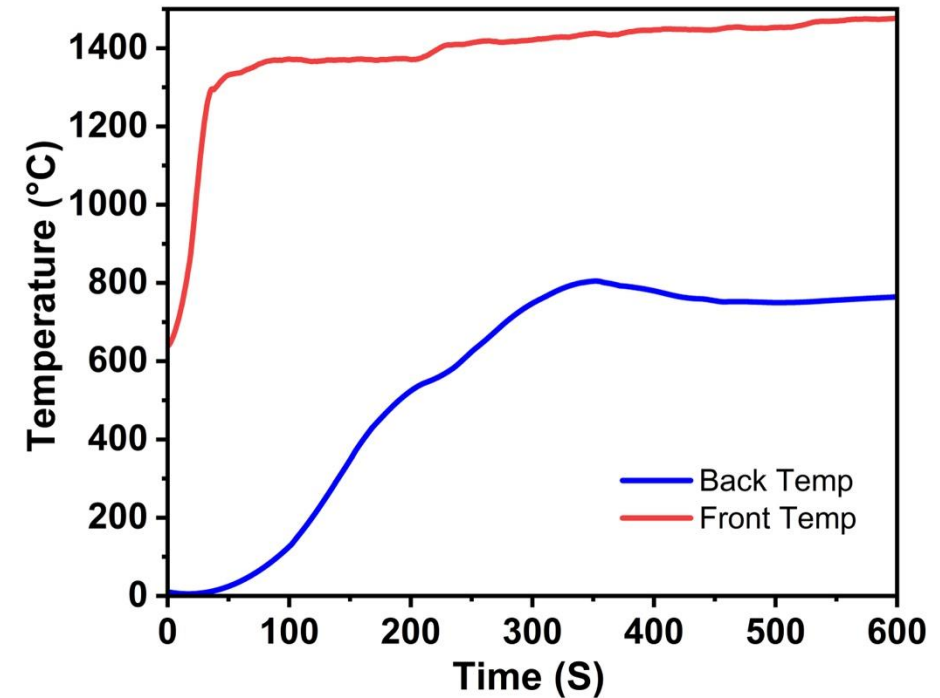
Oxyacetylene Torch Test



H₂/Air Torch Test



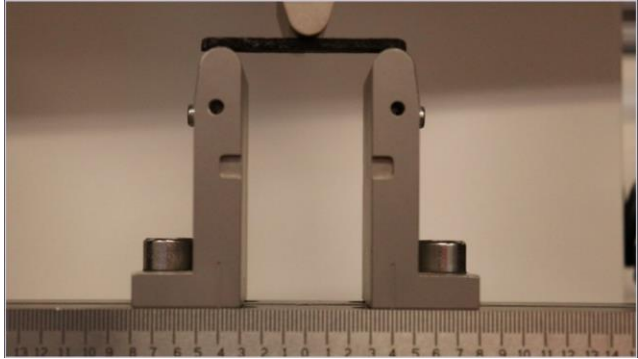
The CMC sample before and after the 10-minute H₂/Air torch test.



The front and back temperature plots of the CMC during 10-minute H₂/Air torch test.

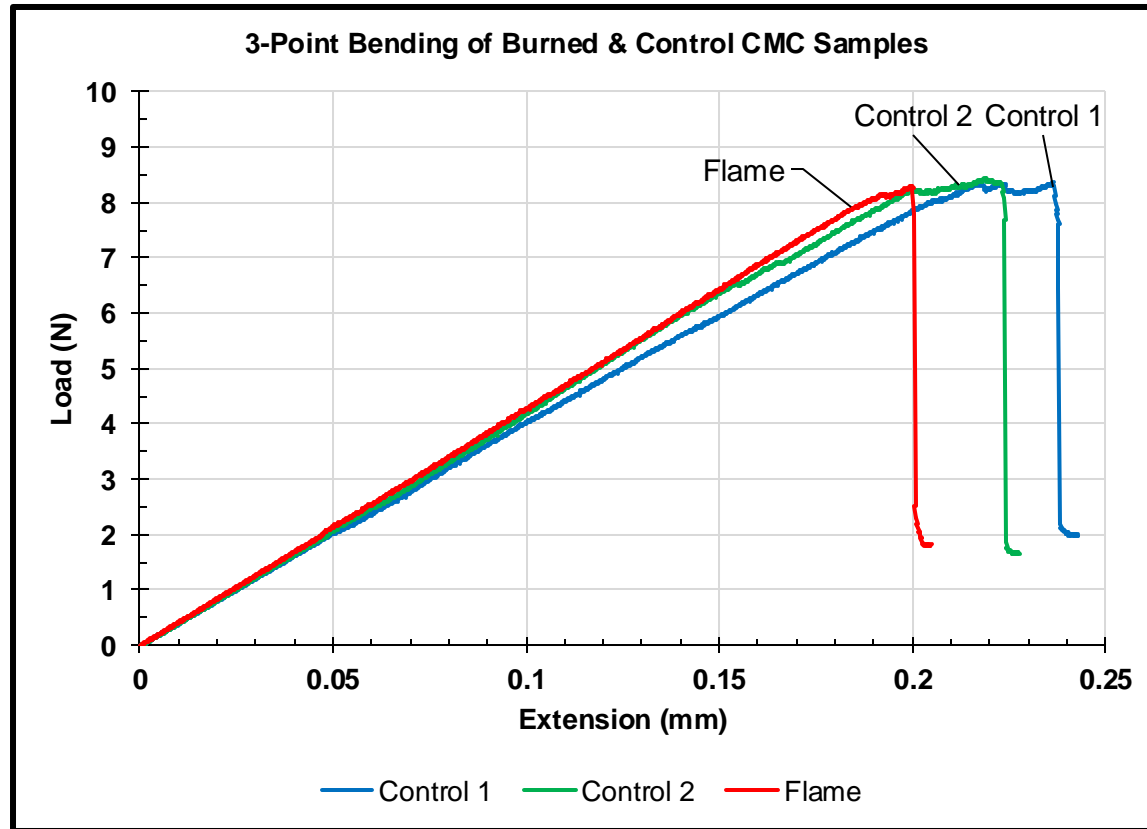
- The YSZ/Si(B)CN ceramic matrix composites withstood 10 minutes of continuous hydrogen combustion for multiple trials without visible damage.

Post-Torch Test: Mechanical Property Characterization

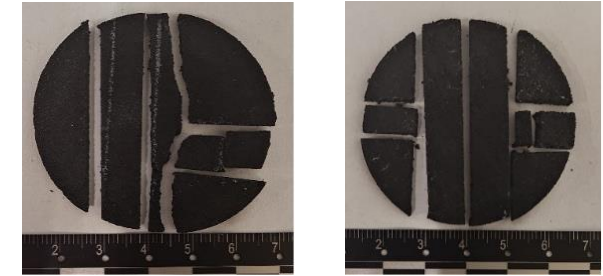


$$E_f = \frac{L^3}{4bh^3} \left(\frac{dP}{dw} \right) \quad \sigma_f = \frac{3PL}{2bh^2}$$

σ_f	Stress required to fracture the sample
E_f	Flexural modulus of elasticity
L	Support span
b	Width of specimen
h	Thickness of specimen
P	Force
w	Cross head displacement
$\Delta P/\Delta w$	Initial stiffness
$\Delta w/\Delta t$	Deflection rate
$\Delta P/\Delta t$	Initial loading rate
P_{max}	Max load
$w(P_{max})$	Deflection at max load

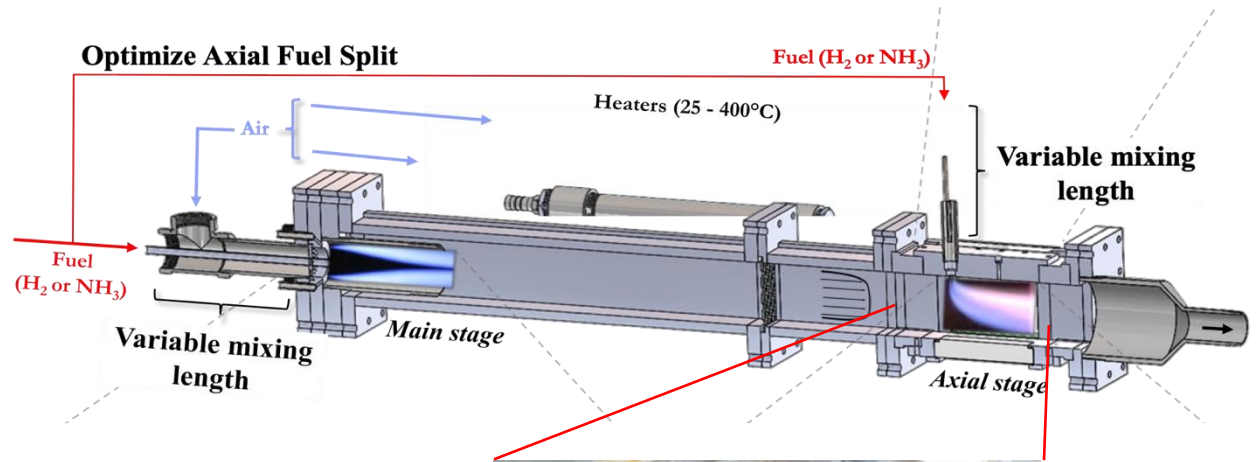


Sample	$\Delta P/\Delta w$ (N/mm)	P_{max} (N)	$w(P_{max})$ (mm)	σ_f (MPa)	E_f (GPa)
Control 1	40.819	8.363	0.168	7.73	3.78
Control 2	43.909	8.433	0.170	8.01	4.19
Flame	43.419	8.288	0.153	7.88	4.17

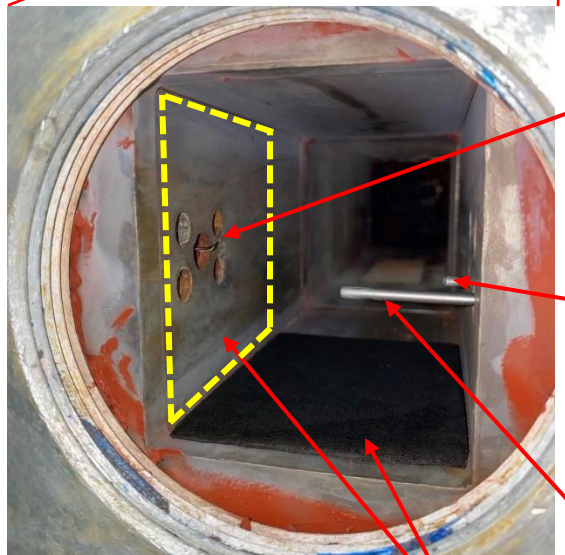


- Hydrogen flame damage had minimal effect on mechanical strength
- Matrix enhanced fracture strength (YSZ Tow $\sigma_f = 1.5 \text{ MPa}$)
- Average Values
 - $\sigma_f = 7.87 \pm 0.16 \text{ MPa}$
 - $E_f = 4.05 \pm 0.26 \text{ GPa}$
 - $P_{max} = 8.36 \pm 0.08 \text{ N}$

3-point bending test of flame-damaged and control samples were compared to understand effect of H₂ flame exposure on mechanical strength.



Not Pictured:
Thermocouples
Measuring
Outside Wall
Temperatures



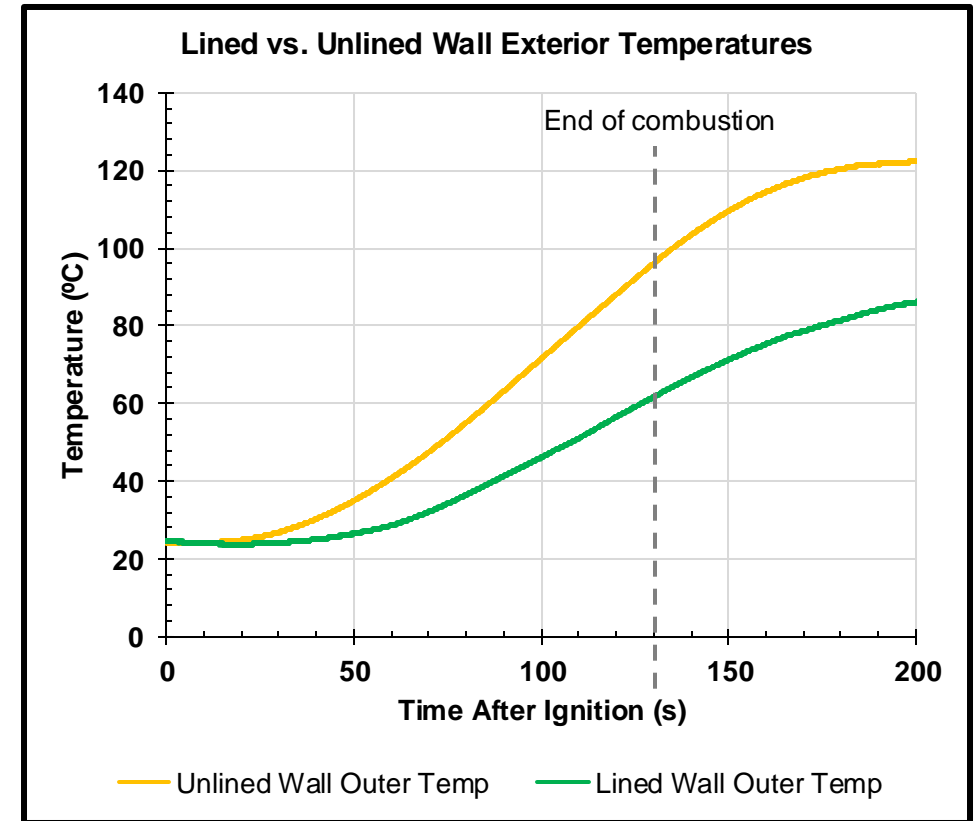
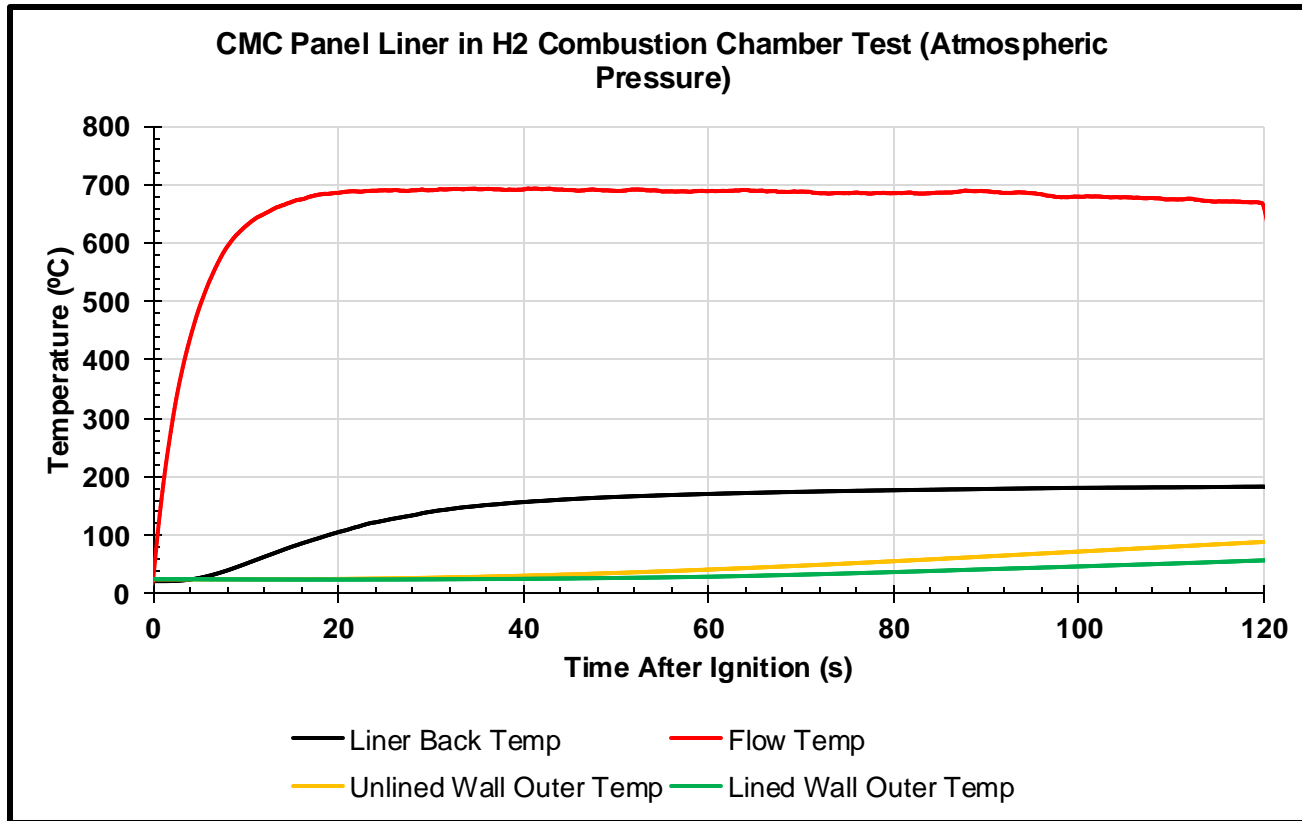
CMC Back
Temperature
Thermocouple

Inside Wall
Temperature
Thermocouple

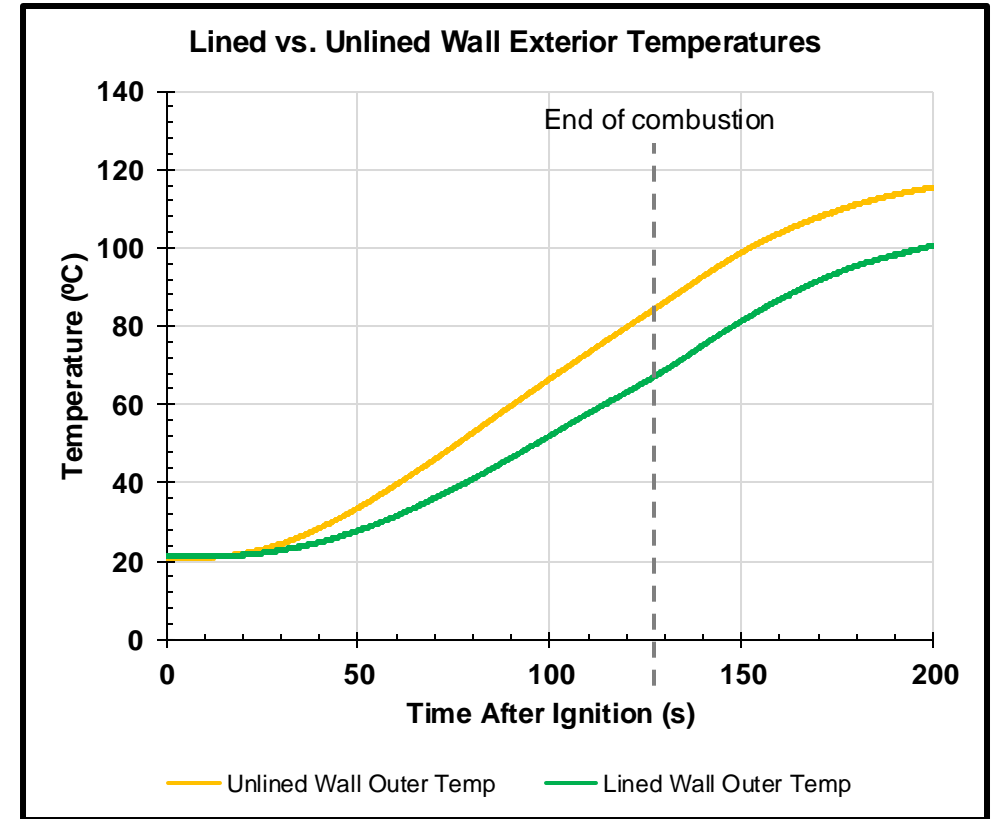
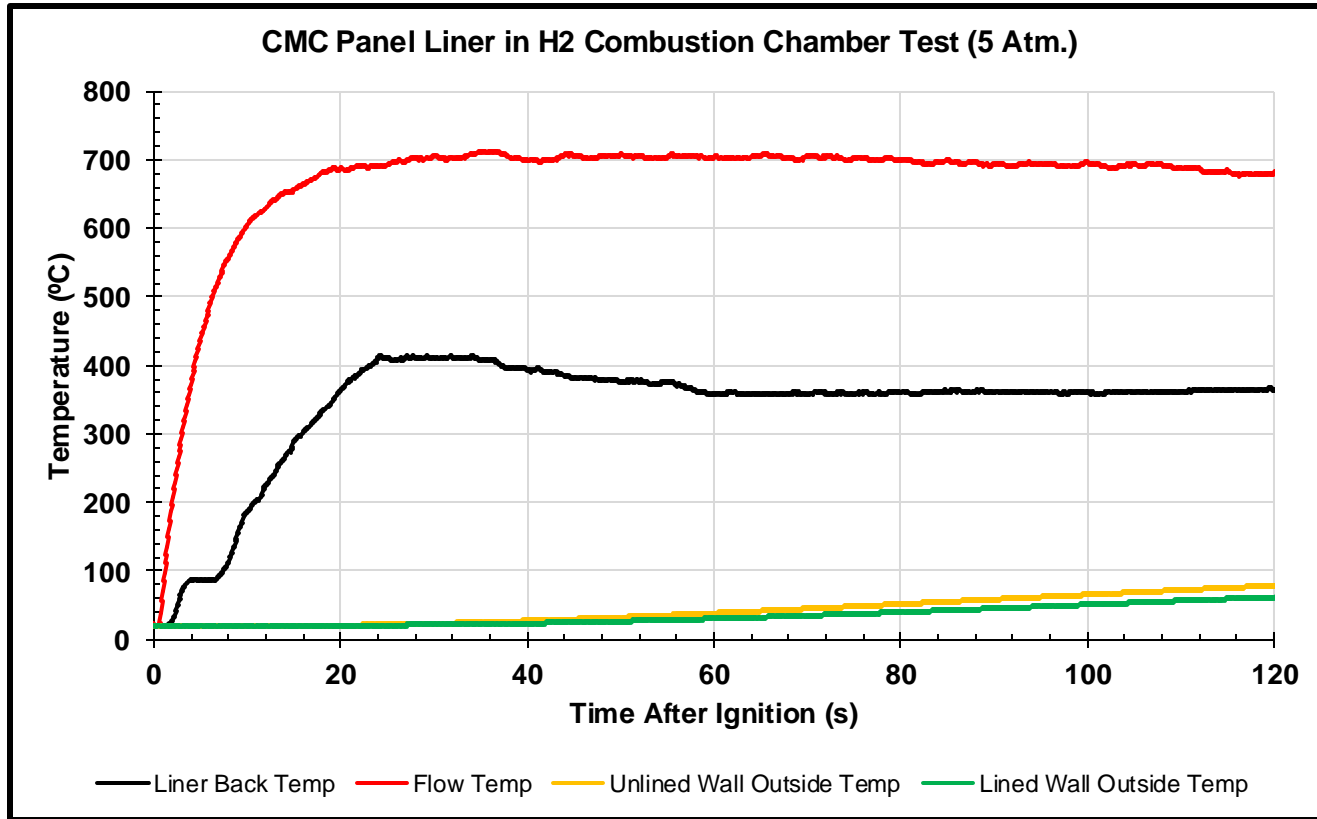
Flow
Temperature
Thermocouple

CMC Panel
Liners

- Preliminary testing, targeting low temperatures (~700°C) at atmospheric and pressurized conditions.
- Investigating effects of hydrogen flame traveling parallel to CMC (as opposed to through-thickness as in the torch test).
- Combustion testing duration of 2 minutes.



- The CMC liner withstood 2 minutes of continuous hydrogen combustion for multiple trials with no visible damage
- Flow temperature stabilized at ~680°C while liner back face temperature did not increase past 185°C
- Successfully protected the stainless-steel walls of the combustion chamber facility
 - Lined wall was ~32°C cooler than unlined wall by end of combustion



- While CMC liner still did not experience acute or visible damage, temperature profile was very different from atmospheric test
- Flow temperature stabilized at ~700°C and liner back face temperature rose beyond 400°C before lowering and stabilizing ~360°C
- CMC still protected stainless-steel walls of the combustion chamber, however to a much lesser degree
 - Lined wall was only ~17°C cooler than unlined wall by end of combustion
 - Could be explained by porosity in composite allowing pressurized hot gas to travel through voids and heat up backside of CMC

Ongoing Work: Fabrication of the CMC Liner for Long Duration Testing

**Aluminum mandrel /
YSZ Preform**



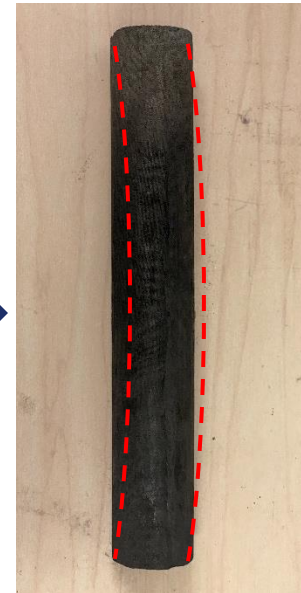
2 sheets of YSZ fiber
(18" x 44" total) saturated with
rigidizer and wrapped around
mandrel to create cylindrical
porous preform

**Resin infiltration and curing
of the green body**

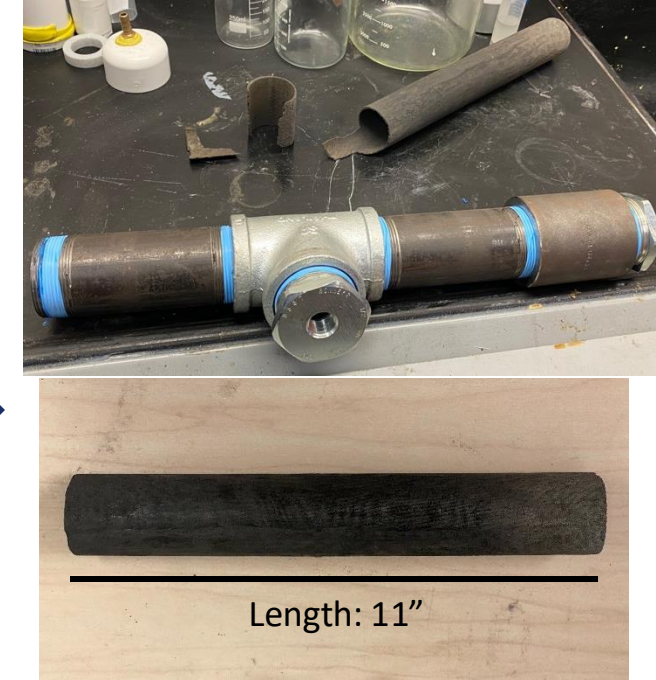


Air-tight PVC infiltration chamber
was designed to accommodate large
part geometry

Pyrolysis



OD: 48.51 mm
The CMC line had a slight warping
after pyrolysis



Curvature of liner caused stress
fracture when fitting inside
chamber

Liner length reduced to 11", combustion
chamber length can be adjusted

❑ Combustor firing - Pressure validation

- Ran H₂-air mixture at a set equivalence ratio of 0.34 resulting in a calculated crossflow temperature of 1000°C
 - The current condition at 5 atm pressure would have a flow velocity of about 24 m/s

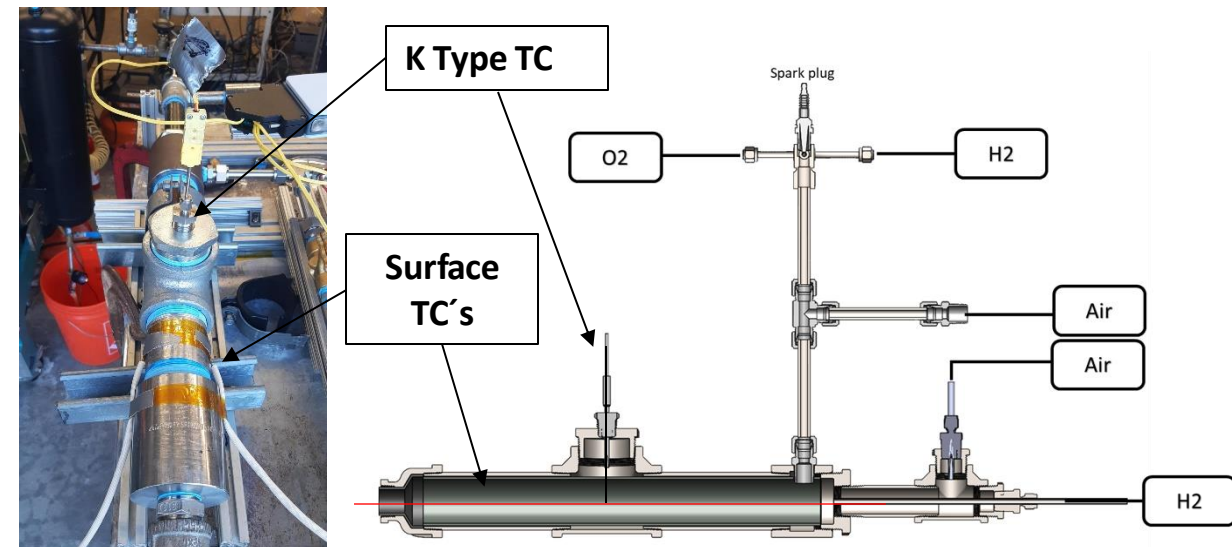
Results

- The facility was pressurized up to 70 psi (4.7 atm)



❑ Combustor firing - Temperature validation

- Set 2 surface thermocouples outside the combustor
- Set 1 K type thermocouple in the crossflow.
- The K Type thermocouple was located at the centerline of the combustor





Conclusions & Future Work



- The Si(B)CN preceramic polymer formulation and its CMC processing technique show a promise for H₂ combustion environments.
- Direct H₂ flame exposure at high heat flux resulted in minimal damage to the CMC.
- The reduced insulation effectiveness of the CMCs at higher pressures suggest the need to further densify the material through more PIP cycles, reducing porosity and increasing thermal performance.
- A densification study will be carried out to identify optimal number of PIP cycles by measuring mass gained per subsequent cycle and using Micro-CT to assess porosity at each step.
- A full-sized CMC combustion liner will be manufactured using the material formulation presented, and long-duration testing will be conducted to investigate long-term H₂ combustion effects and survivability.



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Christopher Varela, MS Student/UCF
Michael Tonarely, PhD Student/UCF
Yiting Wang, PhD Student/UCF
David Barnhard, PhD Student/UM

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