Computational Screening and Design of Sensing Materials for Harsh Environment Applications

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Introduction

- For advanced real-time monitoring and control of gas species in combustion environments, development of efficient sensing platforms and new sensor materials able to work under **harsh environments** are required.
- Semiconducting optical-based sensor platforms show promise. MO_x metaloxides and ABO₃ perovskite-oxides can be attractive for high-temperature applications due to their high decomposition temperatures and structural stability; ABO₃ allows tunable electronic and optical properties owing to flexible choices of A, B dopants and forming oxygen non-stoichiometric point-defects



• Combining first-principles modeling with machine learning is a powerful tool to screen and design functional materials for sensing application under harsh environments. It can be also used to determine the sensing mechanisms and the performance of the high-temperature sensors.

Methods

Density functional theory (DFT): PAW-PBE(+U) XC in GGA Energies of formation of point defects (La and Mg dopants, O vacancies)

$$\Delta H(SrTiO_{3,def}) = E(SrTiO_{3,def}) - E(SrTiO_{3}) - \sum n_i \mu_i$$

> Optical properties calculated from frequency-dependent dielectric function

$$\varepsilon(\omega) = \varepsilon_1(\omega) + i\varepsilon_2(\omega) = (n + ik)^2 = \frac{4\pi i}{\omega}\sigma(\omega)$$

• Allen-Heine-Cardona (AHC) theory

- Lattice thermal expansion and, most dominantly, electron-phonon coupling O induces *n*-type conductivity, evident in DOS and Drude peak in imaginary contribute to renormalization of the band gaps component of dielectric matrix; causes peak shift in optical conductivity to Prediction of electron-phonon coupling effect on eigenstate energy level shifts lower wavelength (higher photon energy)
- $\Delta \epsilon_{nk}(T)$, in harmonic phonon approximation

$$\Delta \epsilon_{nk}(T) = \frac{1}{N_q} \sum_{\boldsymbol{q},\nu} \frac{a_{\boldsymbol{q}\nu;\boldsymbol{q}\nu}^{(2)}}{\omega_{\boldsymbol{q}\nu}} \Big[\frac{1}{2} + n_B \big(\omega_{\boldsymbol{q}\nu}, T \big) \Big]$$

• Fit to empirical O'Donnell model

$$E_g = E_O - S < \hbar\omega > \left[\coth\left(\frac{<\hbar\omega>}{2k_BT}\right) + \right]$$

 P. B. Allen, V. Heine, J. Phys. C 9(1976)2305-12 • P. B. Allen, M. Cardona, Phys. Rev. B 23(1981)1495-1505

Machine Learning

Principal component analysis applied to combine 37 features into one reduced feature for each parameter in the O'Donnell model

E_0	S	$<\hbar\omega>$
Formation energy , molar density, melting point	Atomic mass, Electron-Phonon Debye temperature , thermal conductivity, heat of vaporization, speed of sound, boiling point, melting point	Entropy of formation heat capacity, volume density

 \blacktriangleright Bolded features were found to have the highest absolute value of the eigenvector (high relative importance in their respective principal component) \blacktriangleright Gaussian process (GP) regression models were trained separately for each of the three parameters



Doped Perovskite Sensing Layers on Optical Fiber

- K.P. O'Donnell, X. Chen, Appl. Phys. Lett. 58(1991)2924-26.





Interstitial Hydrogen and Oxygen Impurities

- Interstitial H atom preferably binds to O atom in STO leading to breaking of \checkmark octahedral symmetry
- Interstitial O atom bonds to apical O causing distortion of Ti octahedral but does not break symmetry of the crystal



- Incorporation of H, O interstitials alters SrTiO₃ electronic, optical properties as both can act as electron donors to system
- H introduces defect states at VBM above Fermi level without significant changes to bandgap, optical absorption, or the dielectric matrix



Publications

- J. Park *et al.*, Phys. Chem. Chem. Phys. 22(2020) 27163-72; ACS Appl. Mater. Interfaces 13(2021) 17717-25; J. Phys. Chem. C 125(2021) 22231-38; 126(2022)8832-38; Chem. Mater. 34(2022)6108-15
- Y.-N. Wu et al., J. Phys. Chem. C 122(2018) 22642-49; J. Phys. Chem. Lett. 11(2020) 2518-23; J. Phys. Condens. Matter 32(2020) 405705.
- T. Jia et al., RSC Adv. 7(2017) 38798-804; Phys. Chem. Chem. Phys. 22(2020) 16721-26; Applied Energy 281 (2021)116040; J. Phys. Chem. C 125(2021) 12374-81; 126(2022)11421-25
- Y. Duan *et al.*, J. Solid State Chem. 256(2017) 239-251.
- S. Nations, *et al.*, **RSC Adv. 11**(2021) 22264-72; **Mater. Adv. 3**(2022)3897-3905; **Nanomaterials 13**(2023)276 • T. Nandi, L. Chong, J. Park, W. A. Saidi, B. Chorpening, S. Bayham, Y. Duan, AIP Advances 14(3)(2024)035231.

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Applicability of AHC Theory via Machine Learning

- Assess the consensus between the AHC theory and the measurements on temperature dependence of the band gaps in MO_x and ABO_3 - In conjunction with O'Donnell model to quantify the temperature dependence
- of band gaps using well-defined parameters



• Machine learning (ML) for predicting O'Donnell model parameters in MO_x - Ultimately enables ML prediction of the temperature dependence of band gaps

800 C
750 C
700 C

- $Sr_8Ti_8O_{24}$ ----H + $Sr_8Ti_8O_{24}$ ---O + $Sr_8Ti_8O_{24}$ ---- $Sr_8Ti_8O_{24}$ ----H + $Sr_8Ti_8O_{24}$ ----O + $Sr_8Ti_8O_{24}$ ---- $Sr_8Ti_8O_{24}$ ----H + $Sr_8Ti_8O_{24}$ ----O + $Sr_8Ti_8O_{24}$ ---- $Sr_8Ti_8O_{24} \longrightarrow$ H + $Sr_8Ti_8O_{24} \longrightarrow$
- O + Sr₈Ti₈O₂₄ -

Machine Learning

- O'Donnell parameters have been calculated by DFT for 54 metal oxides and are accepted as ground truth for the development of the ML models
- Most of the training data as well as the test predictions fall within the predicted mean $\mu \pm 2\sigma$ region
- Region of uncertainty is observed to be larger where the training points are populated sparsely and significantly for increases regions outside of the training dataset

	RMSE	MAE
E _o (eV)	0.8863	0.6729
S	1.142	0.8881
ħω (meV)	6.943	5.414



SnO₂ used as an example test data point for the combined prediction of parameters to calculate band gap

Predicted	Observ
2.8101 ± 0.8618	1.99
3.8541 ± 2.1652	2.3
30.2017 ± 6.3966	40.7
2.2555	1.68
	2.8101 ± 0.8618 3.8541 ± 2.1652 30.2017 ± 6.3966

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