

# Defect Thermodynamics and Transport Properties of Proton Conducting Oxide BaZr<sub>1-x</sub>Y<sub>x</sub>O<sub>3-δ</sub> (x≤0.1)

## Evaluated Based on Density Functional Theory Modeling

Research & Innovation Center

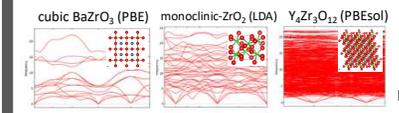


Yueh-Lin Lee<sup>a,b</sup>, Yuhua Duan<sup>a</sup>, Dan Sorescu<sup>a</sup>, Wissam Saidi<sup>a</sup>, Dane Morgan<sup>c</sup>, William K Epting<sup>a</sup>, Thomas Kalapos<sup>a,b</sup>, Gregory A. Hackett<sup>a</sup>, Harry Abernathy<sup>a</sup>  
<sup>a</sup> National Energy Technology Laboratory, Pittsburgh, PA/Morgantown, WV /Albany, OR; <sup>b</sup> NETL Support Contractor, Pittsburgh, PA /Morgantown, WV /Albany, OR; <sup>c</sup> University of Wisconsin-Madison, Madison, WI

### First Principles-Based Modeling of Defect and Transport Properties for Proton Conducting Oxides

A first principles-based model was developed to describe defect chemistry and transport properties of the proton conducting oxide BaZr<sub>1-x</sub>Y<sub>x</sub>O<sub>3-δ</sub> (x≤0.1) for solid oxide cell and other energy applications.[1]

#### Ab Initio Thermodynamics [1-3]



Gas thermodynamics: 
$$\mu^{O_2}(T, p(\text{gas})) = \frac{1}{2} \left[ E_{DFT}(\text{gas}) + (h^{\text{gas}}(T, p(\text{gas})^0) - h^{\text{gas}}(T^0, p(\text{gas})^0) - T \cdot s^{\text{gas}}(T, p(\text{gas})^0)) + k \cdot T \cdot \ln \frac{p(\text{gas})}{p(\text{gas})^0} \right]$$

Metal chemical potentials: 
$$\begin{cases} \mu_{\text{Ba}} = E_{DFT}(\text{BaZrO}_3) + G^{\text{vib}}(\text{BaZrO}_3)(T) - \mu_{\text{Zr}} - 3\mu_{\text{O}}(T, P(\text{gas})) \\ \mu_{\text{Zr}} = E_{DFT}(\text{ZrO}_2) + G^{\text{vib}}(\text{ZrO}_2)(T) - 2\mu_{\text{O}}(T, P(\text{gas})) \\ \mu_{\text{Y}} = \frac{1}{4} [E_{DFT}(\text{Y}_4\text{Zr}_3\text{O}_{12}) + G^{\text{vib}}(\text{Y}_4\text{Zr}_3\text{O}_{12})(T) - 3\mu_{\text{Zr}} - 12\mu_{\text{O}}(T, P(\text{gas}))] \end{cases}$$

#### First principles-based defect model formalism [1,3]

Site balance:  $\sum c(X_A^q) + c(\text{Ba}_A^x) = 1, (A - \text{site})$ ;  $\sum c(Zr_B^q) + c(Zr_B^x) = 1, (B - \text{site})$ ;  $3 \cdot (\sum c(X_O^q) + c(O_O^x)) = 3, (O - \text{site})$

Defect formation:  $\Delta G^{\text{form}}(X^q) = E_{\text{defect}}^{\text{DFT}}(X^q) + G_{\text{bulk}}^{\text{vib}}(X^q) - E_{\text{bulk}}^{\text{DFT}}(X^q) + G_{\text{bulk}}^{\text{vib}} + \sum n_i \mu_i + q\mu_e + E_{\text{corr}}$ ;  $c(X^q) = \frac{\exp(-\frac{\Delta G^{\text{form}}(X^q)}{k_B T})}{\exp(-\frac{\Delta G^{\text{form}}(X^q)}{k_B T}) + 1}$

Free electron concentration:  $c(e^-) = N_c \exp(-\frac{E_{\text{gap}} - E_{\text{Fermi}}}{k_B T})$ ,  $N_c = 2 \left( \frac{2\pi \cdot m_e^* k_B T}{h^2} \right)^{3/2}$

Free hole concentration:  $c(h^+) = N_v \exp(-\frac{E_{\text{Fermi}}}{k_B T})$ ,  $N_v = 2 \left( \frac{2\pi \cdot m_h^* k_B T}{h^2} \right)^{3/2}$

#### Equilibrium constants of key chemical reactions:

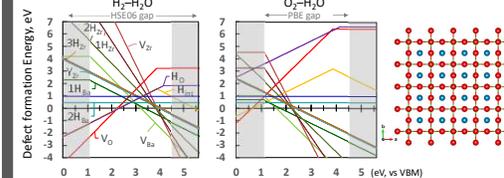
Oxidation Reaction:  $\frac{1}{2} \text{O}_2(\text{g}) + \text{V}_O^{2+} \leftrightarrow 2h_A^+ + \text{O}_O^x$   
 $K_{\text{oxidation}} = \exp\left(\frac{-\Delta E_{\text{oxidation}} + T\Delta S_{\text{oxidation}}}{k_B T}\right) = \frac{O_O^x \cdot h_A^{+2}}{[V_O^{2+}] \cdot P(\text{O}_2)^{0.5}}$

Hydration Reaction:  $\text{H}_2\text{O}(\text{g}) + \text{V}_O^{2+} + \text{O}_O^x \leftrightarrow 2\text{OH}_O^+$   
 $K_{\text{hydration}} = \exp\left(\frac{-\Delta E_{\text{hydration}} + T\Delta S_{\text{hydration}}}{k_B T}\right) = \frac{[\text{OH}_O^+]^2}{P(\text{H}_2\text{O}) \cdot [V_O^{2+}] \cdot [\text{O}_O^x]}$

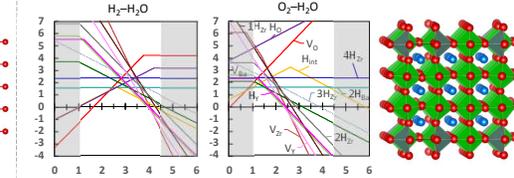
Charge neutrality:  $\sum X_q \cdot n_L \cdot c(X^q) \cdot q + c(h^+) - c(e^-) + c(h_A^+) + c(A') = 0$

Model formalism coded with Octave based on the Density Functional Theory (DFT) results as inputs

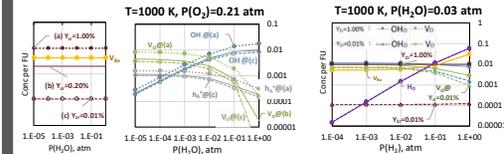
#### BaZrO<sub>3</sub> First Principles Charged Defect Analysis



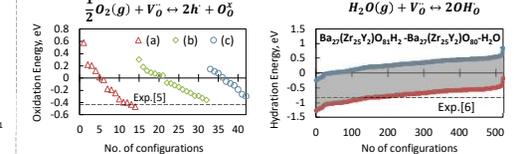
#### BaZr<sub>0.9</sub>Y<sub>0.1</sub>O<sub>3-δ</sub> First Principles Charged Defect Analysis



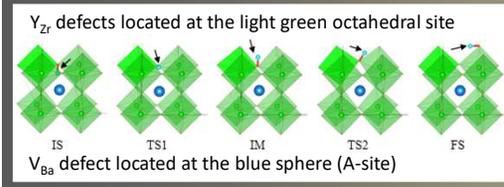
#### BaZrO<sub>3</sub> Brower Diagram with Varying Y<sub>Zr</sub> Conc. (≤1%)



#### Configurational Dependence of Chemical Reaction Energies

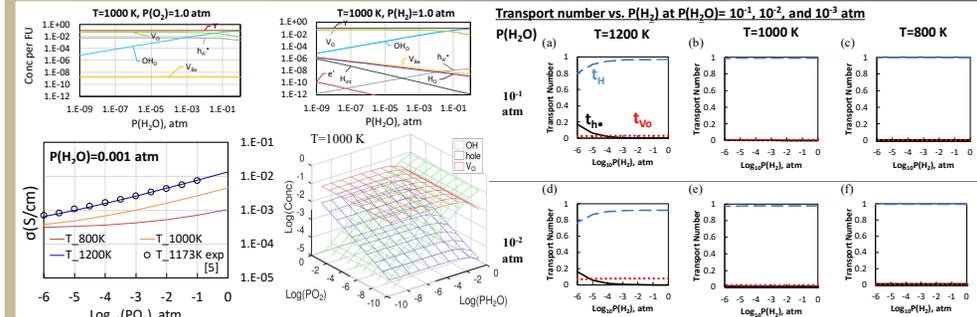


#### H trapping effect of V<sub>Ba</sub> vs. Y<sub>Zr</sub> in BaZrO<sub>3</sub>



The defect equilibria suggests increased OH<sub>O</sub><sup>+</sup> concentration with increased A-site cation deficiency/V<sub>Ba</sub>  
 However, strong H<sup>+</sup> trapping @ V<sub>Ba</sub> may impede H<sup>+</sup> transport

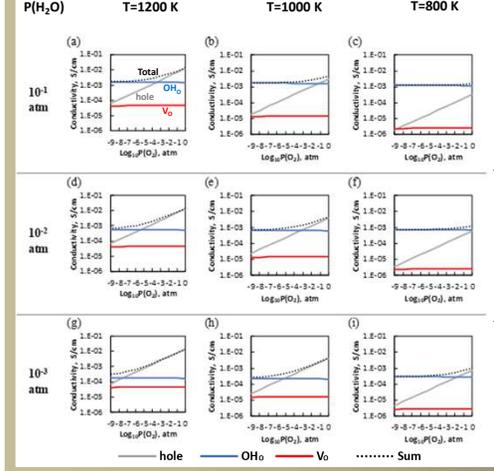
### BaZr<sub>0.9</sub>Y<sub>0.1</sub>O<sub>3-δ</sub> Brower Diagrams, Conductivities, Transport Numbers vs. P(O<sub>2</sub>)/P(H<sub>2</sub>) and P(H<sub>2</sub>O)



#### Activation energies and pre-factors in the transport modeling

	Hole (acceptor-bound hole)	Proton	Oxygen vacancy
Pre-exponential factors, cm/s	3x10 <sup>3</sup>	1.1	2x10 <sup>-2</sup>
Hopping barriers, eV	0.7	0.3	0.6

#### Total conductivity vs. P(O<sub>2</sub>) at P(H<sub>2</sub>O)=10<sup>-1</sup>, 10<sup>-2</sup>, and 10<sup>-3</sup> atm



#### Conclusions

- DFT-based thermodynamic modeling was performed to determine the effect of humidity and H<sub>2</sub>O gas pressure on the defect chemistry and transport properties of proton conducting oxide BaZr<sub>1-x</sub>Y<sub>x</sub>O<sub>3-δ</sub> (x≤0.1) in the temperature range of 800–1200 K for solid oxide cell and other energy applications.
- The first-principles charge defect analysis was carried out to obtain the defect energetic ensembles as a function of Fermi level, as well as the dependence of the hydration and oxidation reaction energies on complex defect-dopant configurations.
- Oxygen vacancies introduced to compensate for the extra Ba vacancies may cause a slight increase in the concentration of hydroxyl and proton species upon hydration of Ba deficient BaZr<sub>1-x</sub>Y<sub>x</sub>O<sub>3-δ</sub>, while the overall proton diffusivity is predicted to decrease with increased proton hopping barriers upon trapping near the Ba vacancies.
- The developed defect model can describe the bulk defect chemistry and transport properties of BaZr<sub>1-x</sub>Y<sub>x</sub>O<sub>3-δ</sub> while providing configurational, energetic, and electronic characteristics as a complementary tool to experimental measurements for testing various mechanistic pathways or to provide essential mechanistic data.

#### References

- V.-L. Lee, Y. Duan, et al. *Defect Thermodynamics and Transport Properties of Proton Conducting Oxide BaZr<sub>1-x</sub>Y<sub>x</sub>O<sub>3-δ</sub>* (2021)
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