

# GPU-Enhanced DFTB Approaches for Probing Multi-component Alloys

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# Project Description and Objectives

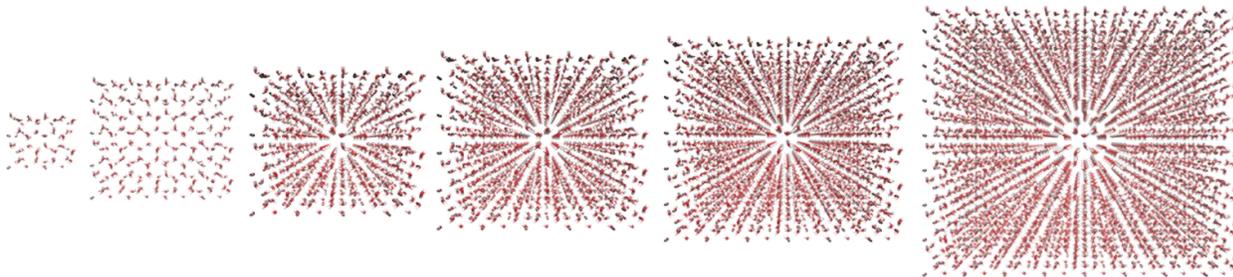
- **Purpose:** Incorporate GPU into DFTB to *accelerate calculations of multi-component alloys*
- Alignment to Fossil Energy objectives:  
GPU-enhanced DFTB enables *fast/accurate predictions of complex, structural materials used in fossil energy power plants*
- **Current Methods in the Field**
  - **Classical force-field based theory:** can handle large systems but poorly describe the electronic level of details in materials/alloys
  - **Density functional theory (DFT):** efficiently captures the quantum-mechanical nature of alloys but cannot handle large sizes relevant to alloys
  - **Density functional tight binding (DFTB):** can probe large systems at quantum level, significantly faster than DFT



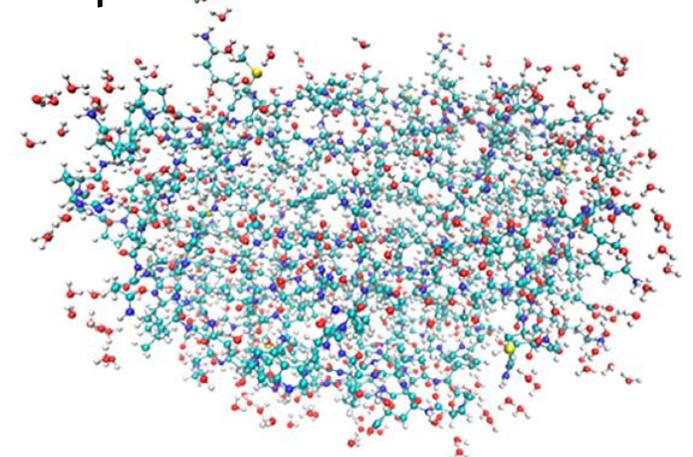
# Project Description and Objectives

- Current Status of project

- Successfully incorporated GPUs into DFTB for extremely fast calculations of large systems
- Successfully parameterized DFTB and used in crystal structure prediction



up to 15,000 atoms!



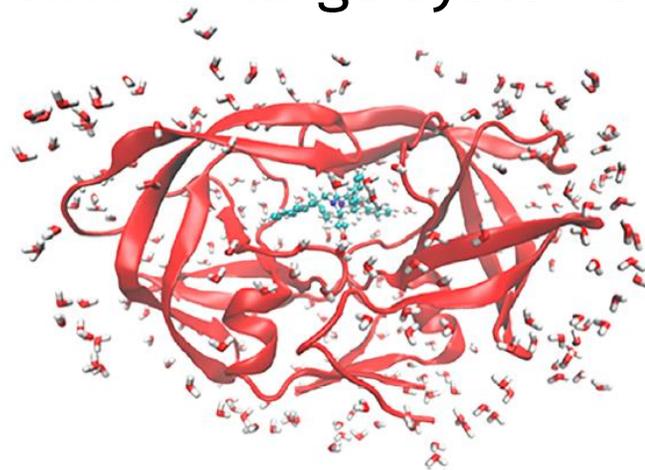
around 5,000 atoms

- Project on-track to meet deliverables: goals/objectives have not changed

**Phase 1:** GPU parallelization done; **Phase 2:** DFTB parameterization done

# Why Use DFTB for Alloys?

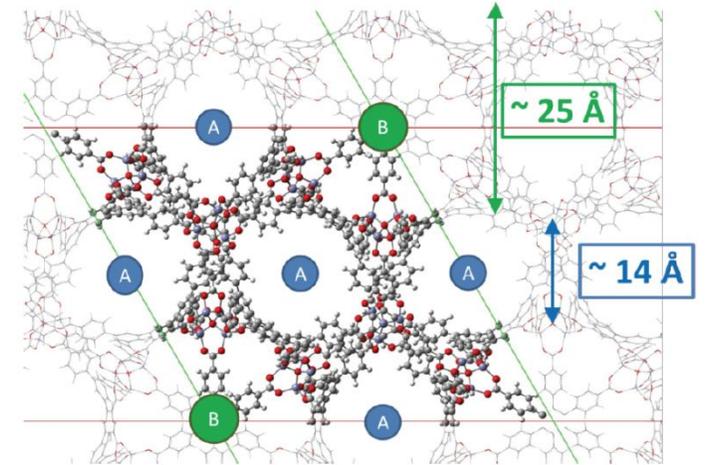
- *Density Functional Tight Binding (DFTB)*: parameterized DFT with atomic-centered basis functions
- DFTB *extremely fast* for large systems



S. Allec, Y. Sun, J. Sun, C.A. Change, B.M. Wong  
*Journal of Chemical Theory and Computation*, 15, 2807-2815 (2019)

~ 5,000 atoms in unit cell

- Implemented **GPU-enhanced DFTB** in this project to accelerate dynamics calculations of alloys



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*J. Mater. Chem. A* 2, 3389 (2014)

> 1,000 atoms in unit cell  
(geometry optimizations ~ minutes)

# GPU Implementation

$$\hat{H}\Psi = E\Psi$$

Solves the eigenvalue problem by diagonalizing the Hamiltonian

- 3 eigensolvers in DFTB+:
  - QR
  - DivideAndConquer
  - RelativelyRobust

## MAGMA

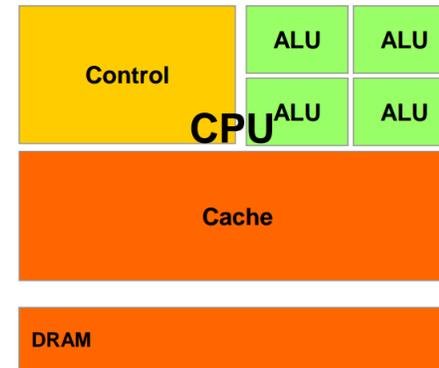
- Linear algebra library similar to LAPACK
- Used for hybrid “Multi-core +GPU” architectures.



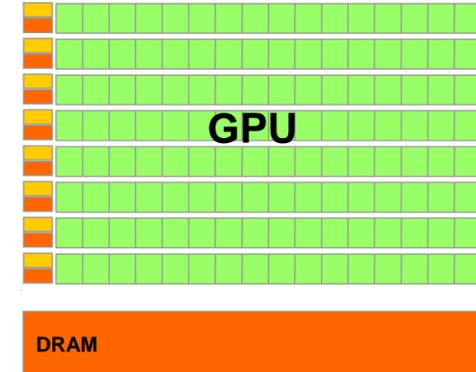
P's columns are the eigenvectors of A

$$P^{-1}AP = \begin{pmatrix} \lambda_1 & & & \\ & \lambda_2 & & \\ & & \ddots & \\ & & & \lambda_n \end{pmatrix}$$

Matrix with eigenvalues on diagonal



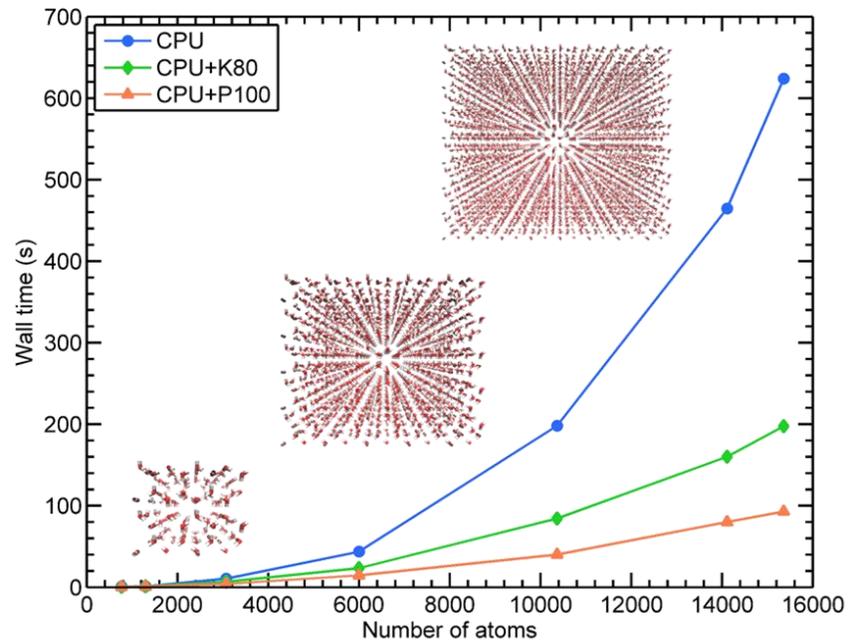
- Several Cores
- Low Latency
- Good for serial processing
- Can do handful of operations at once



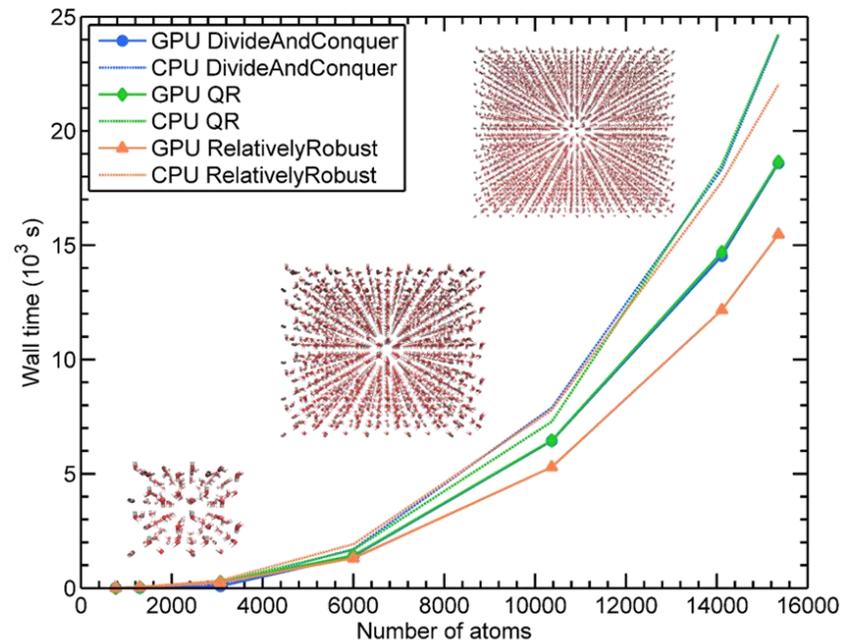
- Many Cores
- High Throughput
- Good for parallel processing
- Can do thousands of operations at once

# Results of Phase 1: GPU Acceleration

- Performed CPU/GPU benchmarks on large systems (~15,000 atoms!)
- Different algorithms scale with varying performance
- Relatively Robust has the best performance on GPU
- Large complex systems (i.e. alloys) can be modelled



CPUs vs GPUs

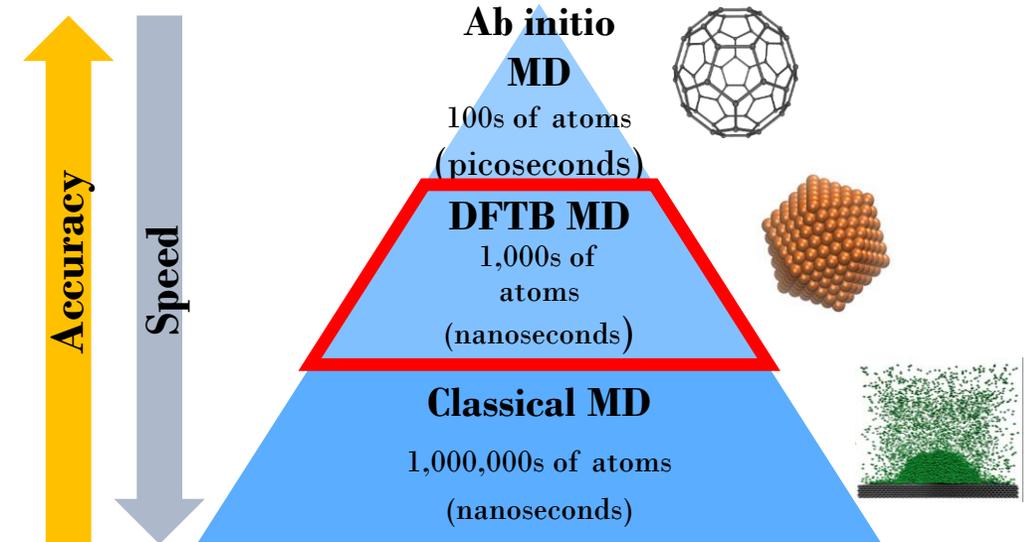


algorithm comparison

# GPU-enabled DFTB

## • Why use DFTB ?

- DFT good for small systems
- Classical methods do not consider quantum nature of chemical/material systems.
- DFTB merges reliability of DFT with computational efficiency of tight binding
- Slater-Koster files used instead of DFT functionals



## Challenges

- DFTB limited by set of parameters for elements in periodic table (Slater – Koster files)

## Goal

- Create Slater – Koster files for missing element pairs
- Use DFTB to calculate phases/properties of multi-component alloys

# GPU-enabled DFTB

## DFTB THEORY

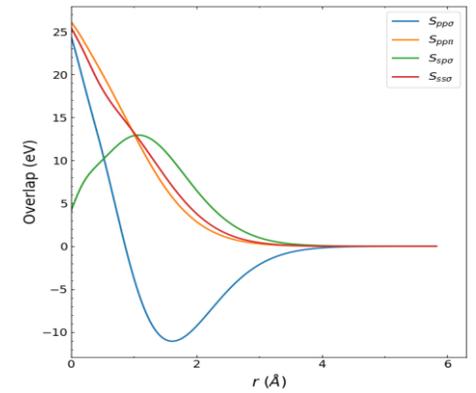
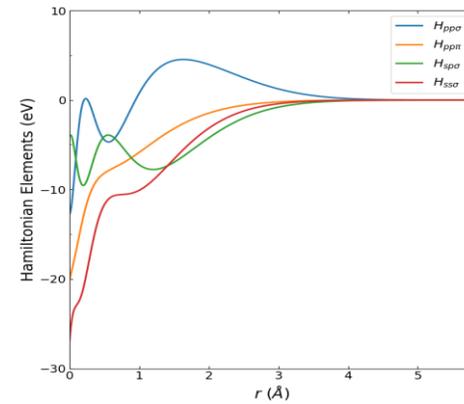
$$E_{\text{DFTB}} = \sum_i^{\text{occ}} \langle \phi_i | H_0 | \phi_i \rangle + \frac{1}{2} \sum_{\alpha, \beta}^N \gamma_{\alpha\beta} \Delta q_{\alpha} \Delta q_{\beta} + E_{\text{rep}}$$

non-SCC  $H$   
(parametrized)

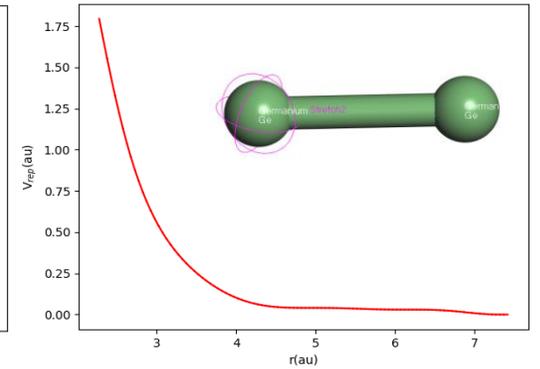
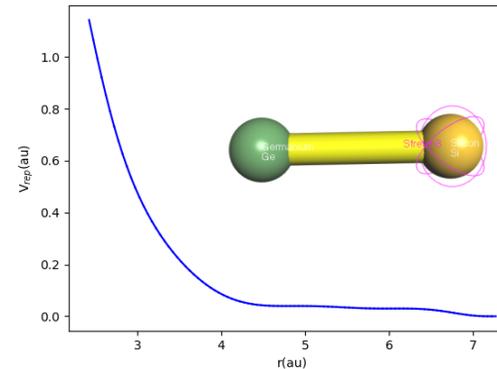
Long-range  
electrostatic  
interactions

Short-range  
repulsion

- $E_{\text{rep}}$  lumps together many-body effects (e.g., exchange-correlation)
- $H_0$  and overlap *matrix elements parametrized beforehand* from DFT calculations



Hamiltonian (H) and Overlap (S) from Ge-Ge Slater-Koster files.



Repulsive Potentials for Ge-Si and Ge-Ge

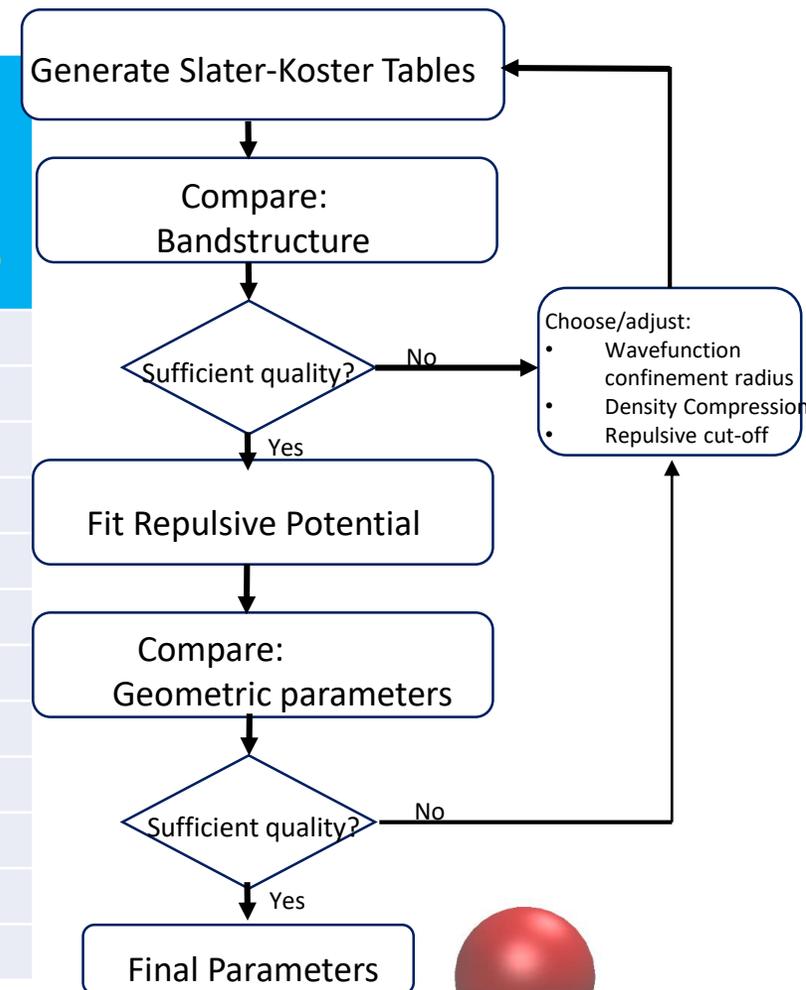
$$H_{\mu\nu}(\mathbf{r}) = \langle \phi_{\mu}(\mathbf{r}) | H_0 | \phi_{\nu}(\mathbf{r} - \mathbf{r}_o) \rangle, \quad S_{\mu\nu}(\mathbf{r}) = \langle \phi_{\mu}(\mathbf{r}) | \phi_{\nu}(\mathbf{r} - \mathbf{r}_o) \rangle$$

$$H_0 = -\frac{1}{2} \nabla^2 + v_{\text{eff}}[\rho^a(\mathbf{r})] + v_{\text{eff}}[\rho^b(\mathbf{r} - \mathbf{r}_o)]$$

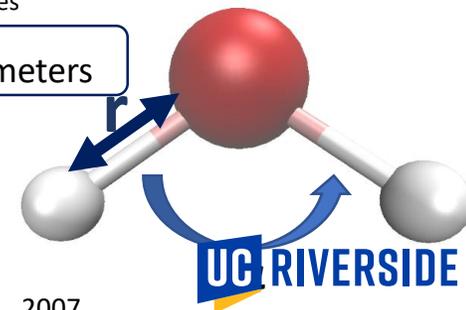
$$E_{\text{rep}} = E_{\text{DFT}}^{\text{tot}} - E_{\text{DFTB, bare}}^{\text{tot}} = \sum_{i < j}^{\text{pairs}} U_{\text{type}(ij)}(r_{ij})$$

# Results of Phase 2 : DFTB Parameterization

Compound	Multiplicity	Parameter*	DFT	DFTB	$\Delta(\text{DFTB-DFT})$	$\Delta(\text{DFTB-DFT})\%$
FeH <sub>2</sub>	3	r	1.54	1.503	-0.037	-2.40
		$\alpha$	102.365	110.34	7.975	7.79
	5	r	1.66	1.58	-0.08	-4.82
		$\alpha$	177.3	179.9	2.6	1.47
FeO	1	r	1.54	1.6	0.06	3.90
	3	r	1.58	1.68	0.1	6.33
	5	r	1.61	1.75	0.14	8.70
Fe(CH <sub>3</sub> ) <sub>2</sub>	1	r	1.92	1.89	-0.03	-1.56
		$\alpha$	117.7	125.71	8.01	6.81
	3	r	1.94	2.13	0.19	9.79
	5	r	2.05	1.96	-0.09	-4.39



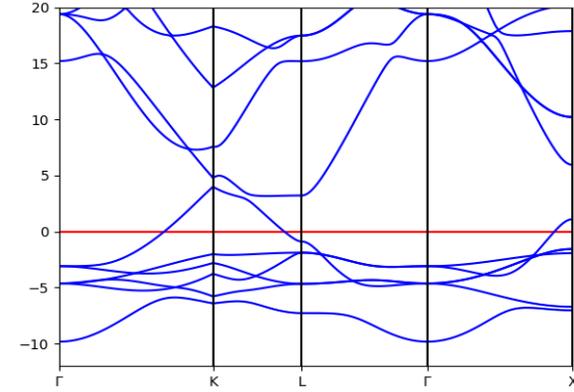
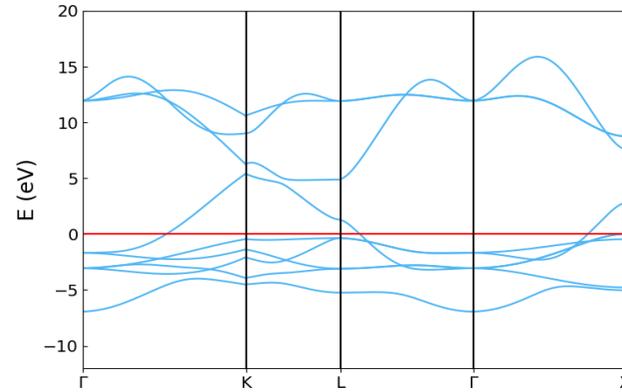
\*r in Angstrom,  $\alpha$  in degrees



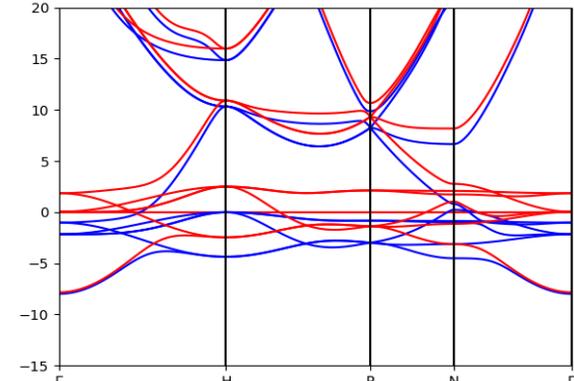
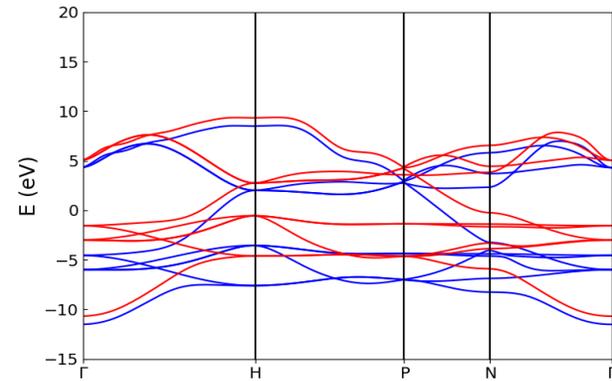
# Results of Phase 2 : DFTB Parameterization

## Bandstructures

Au (FCC)



Fe (BCC)



Using DFTB

Using DFT(pbe)

- Valence and conduction bands calculated via DFTB and DFT, show almost the same dispersion/delocalization

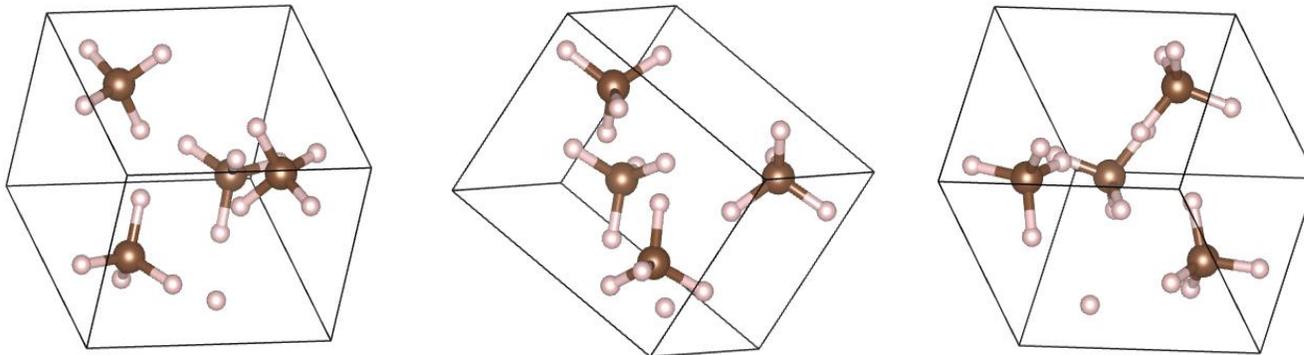
# Results of Phase 3 : Crystal Structure Prediction

## Constrained Evolutionary Scheme for Structure Prediction of Molecular Crystal (CH<sub>4</sub>)

- Variable-cell structure prediction of CH<sub>4</sub> with cell size containing 4 formula units at 20 GPa (4 molecules/unit cell).
- Use USPEX's evolutionary structure prediction algorithm to investigate possible structures of Methane (at T=0, P=20 GPa).

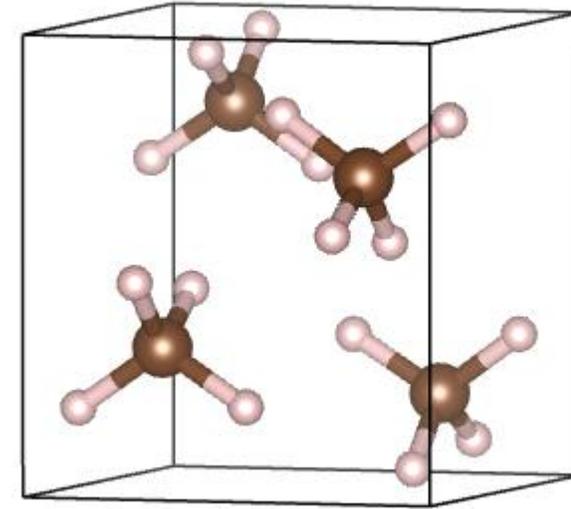
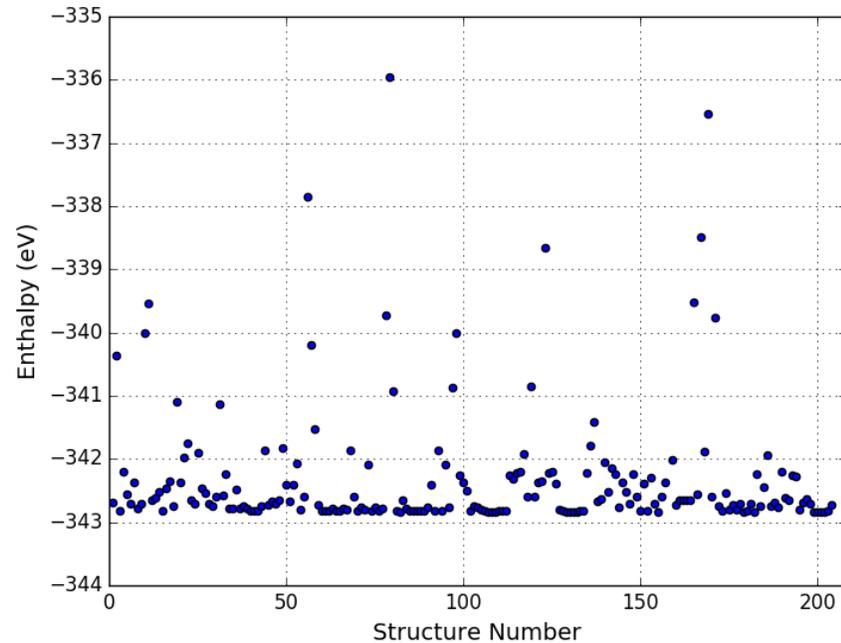
### METHOD:

- Initial population of 20 structures and part of new generations are produced using space-group symmetry combined with random cell parameters and, random positions and orientation of molecular units.
- Structure relaxation via GPU enabled DFTB3 (3ob-3-1 parameters) with Lennard-Jones potential.
- Every subsequent generation is produced from the best 60% of the previous generation.
- New population/structures created via variation operators (heredity, lattice mutation, soft mutation, permutation)
- Ran till 9 generations, 204 total structures were ranked based on enthalpy.



A few crystal structures of CH<sub>4</sub> produced using space group symmetry during the 1<sup>st</sup> generation in USPEX

# Results of Phase 3 : Crystal Structure Prediction

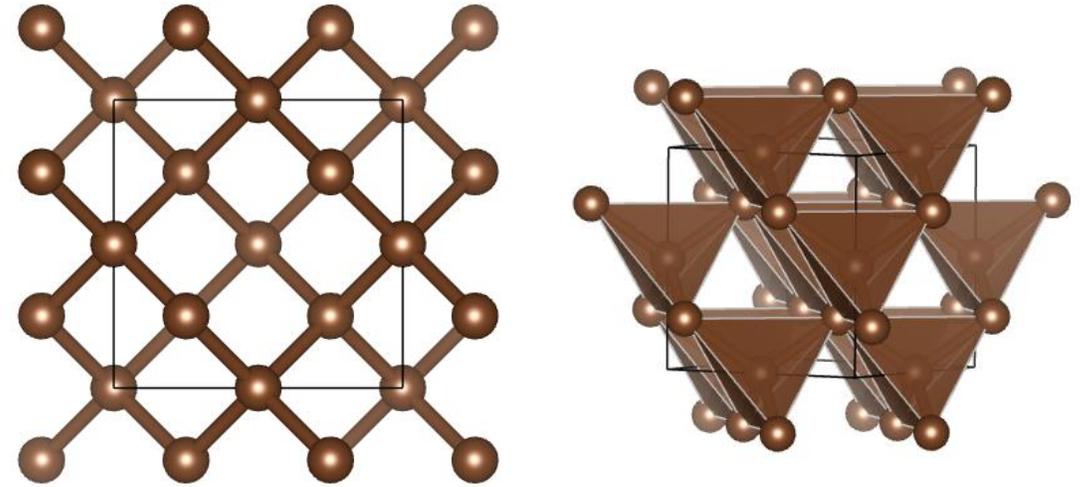
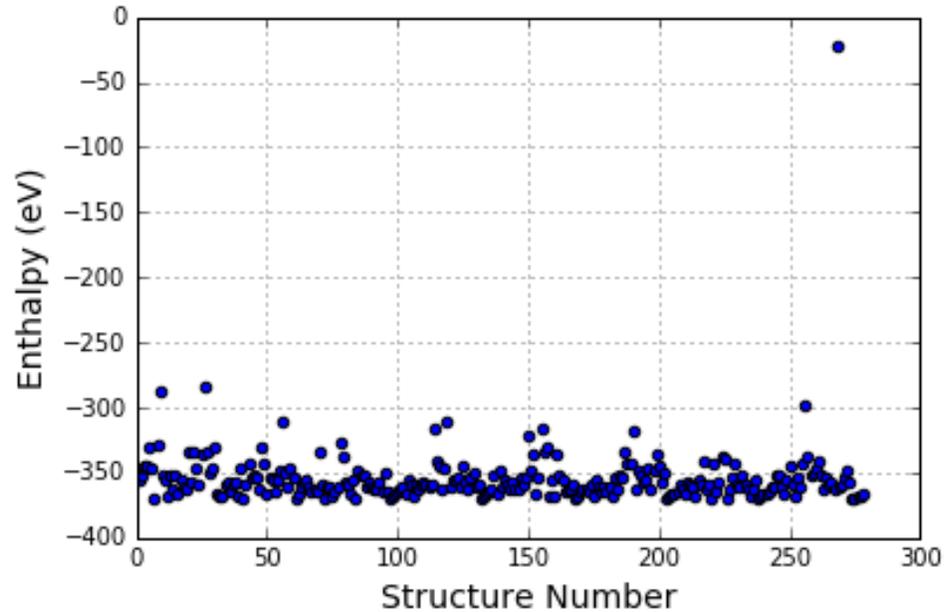


Molecular crystal structure of CH<sub>4</sub> as predicted by GPU-DFTB at 20 GPa

Enthalpy for distinct structures of CH<sub>4</sub> during various stages (generation) in the evolutionary algorithm

- Calculated enthalpy on distinct configurations of CH<sub>4</sub>.
- In each generations distinct structures of CH<sub>4</sub> are generated via variation operators.
- At pressure 20 GPa, the low enthalpy structures are all made of well-separated CH<sub>4</sub> molecules.
- Each CH<sub>4</sub> molecule forms a slightly distorted tetrahedron with C-H distance about 1.06-1.08 Å and H-C-H angles in the range 108.70° – 110.4°.
- At low pressure methane keeps its molecular state.

# Results of Phase 3 : Crystal Structure Prediction



Diamond crystal structure of C as predicted by GPU-DFTB

Enthalpy for distinct structures of C during various stages(generation) in the evolutionary algorithm

- Enthalpy on distinct configurations of C .
- In each generations distinct structures of C are generated via variation operators.
- Diamond crystal structure is predicted for C, and the energy difference between graphite and diamond seems to be overestimated by DFTB.
- Future goal include exploring crystal structures of Fe, Ni, Al alloys at different temperature and pressure.

# Preparing Project for Next Steps

- **Market Benefits/Assessment**

- **Current market gap:** existing simulation tools (i.e. MD/DFT) not capable of predicting dynamics of large alloy systems
- **Benefits:** project goals directly translate to understanding (1) structural deformation in complex alloys & (2) reactive processes in these complex systems

- **Technology-to-Market Path**

- **Technology transfer is high:** many technologies depend on structural materials, including furnaces and structural composites in buildings
- **New research:** first demonstration of accelerating DFTB with GPUs for large systems

# Concluding Remarks

- Applicability to Fossil Energy and alignment to strategic goals
  - GPU-enhanced DFTB enables fast predictions of complex, structural materials used in fossil energy power plants
  - **Better than Classical methods & orders of magnitude faster than DFT**
- **Next step:** incorporate GPU-DFTB for obtaining phase diagrams of alloy systems
- **First demonstration** of accelerating DFTB-based dynamics with GPUs for large systems
  - Published in *Journal of Chemical Theory & Computation* (IF: 5.4)
  - **Featured as cover for the May 14, 2019 issue**



# Acknowledgement and Disclaimer

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