High Throughput Computational Framework of Materials Properties for Extreme Environments

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Acknowledgment and Disclaimer

- This material is based upon work supported by the Department of Energy Award Number DE-FE0031553.
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Outline

- Overview of proposed goal and technical approaches
- Results and discussion
 - A comprehensive DFT study of key fundamental properties: Ideal shear strength of pure elements, binary Ni-X (Ni₁₁X), and ternary Ni-X-Y (Ni₃₄XY) systems
 - A combined DFT/FEM approach to study constitutive behavior of single crystals and polycrystals
 - Machine learning models of materials properties: Illustrated with stacking fault energy
 - Development of hardness model
- Summary of achievements

Goal ⇒ To establish an open-source infrastructure for predictive modeling of materials properties at extreme environments, including:

- High throughput DFT-based first-principles calculations for properties of interest, including phonon, thermodynamic, elastic, ideal shear strength, and stable/unstable stacking fault properties related to Ni-based superalloys (e.g., Inconel 740);
- Exploration of machine learning models to predict the above properties;
- High throughput CALPHAD modeling of properties using our unique capability;
- New capabilities to predict the tensile stress-strain behavior of single crystals and polycrystals using finite element method (FEM) analysis; and
- Modeling of hardness with attributes from plastic deformations.

Overview of high throughput computational framework for materials properties under extreme environments



- ✓ Predict proto data using DFT Tool Kit (<u>www.DFTTK.org</u>) and machine learning models (<u>https://phaseslab.com/sipfenn/</u>)
- ✓ Apply FEM (ABAQUS) to predict tensile strain-stress curve
- ✓ Use CALPHAD approach (PyCalphad.org and ESPEI.org) to model processed data
- ✓ Validate results and improve models

DFT calculations of generalized stacking fault energy



 γ_{SF} : Stacking fault energy γ_{US} : Unstable stacking fault energy τ_{IS} : Ideal shear strength



Shang et al. J Phys CM 24 (2012) 155402



Ideal shear strength of pure elements (GPa)

Li 3	Be 4	Green (low value) – Yellow (middle) – Red (high)										C 6 (79.9)	
0.20	14.00							•	-	•	•		
0.15	5.22	Elems 60 (Diamond: C, Si, Ge)											
0.40	15.09			bcc	18								
Na 11	Mg 12			fcc	45							Al 13	Si 14 (7.2)
0.44				hcp	50								
0.09	0.61											2.66	
0.19	1.50											8.07	
К 19	Ca 20	Sc 21	Ti 22	V 23	Cr 24	Mn 25	Fe 26	Co 27	Ni 28	Cu 29	Zn 30		Ge 32 (5.0)
0.11				5.16	19.01		7.56						
0.05	0.22	1.26	1.78			5.88	10.12	4.95	5.39	2.33			
0.09	0.42	2.81	4.39			6.05	11.34	6.55	3.56	2.42	2.18		4.01
Rb 37	Sr 38	Y 39	Zr 40	Nb 41	Mo 42	Tc 43	Ru 44	Rh 45	Pd 46	Ag 47	Cd 48		
0.07	0.10			5.86	15.44								
0.01	0.11	1.49	1.37			7.70	14.26	10.47	1.84	1.27			
	0.29	2.18	2.38			7.17	15.88	5.57	0.88	1.58	0.50		
Cs 55	Ba 56	La 57	Hf 72	Ta 73	W 74	Re 75	Os 76	lr 77	Pt 78	Au 79	Hg 80	TI 81	Pb 82
0.03	0.35			5.30	16.97						0.21	0.84	
0.01	0.01	1.33	2.41			11.11	19.97	15.47	1.50	0.72			1.02
0.02	0.27	0.63	5.08			10.65	21.77	6.65		0.95	0.22	0.79	0.23

Ce 58	Pr 59	Nd 60	Pm 61	Sm 62	Eu 63	Gd 64	Tb 65	Dy 66	Ho 67	Er 68	Tm 69	Yb 70	Lu 71
					1.01								
1.27	1.35	1.48	1.60	1.68	1.73	1.78	1.80	1.82	1.82	1.85		0.26	
0.51	0.54	0.55	0.55	0.62	0.83	1.09	1.48	1.85	2.16	2.47	2.70	0.43	2.85

Ideal shear strength: Binary Ni-X (Ni₁₁X)

- Alias shear deformation to Ni-X (Ni₁₁X) solutio
- 26 alloying elements X
- Use alloying element descriptors to examine the variations of ideal shear strength



(GPa)

5

4.5

Ni

Ideal shear strength: Ternary Ni-Z-X (Ni₃₄ZX)

- Supercell: 36-atom with 3 {111} layers
- Total Ni₃₄FeX configurations \rightarrow 39
- 0.05 (Si) ~ 0.25 (Mn) GPa spread due to positions of Z and X in the supercell





Multiscale predictive simulation: Single crystal



arXiv:2002.08552

Key contributions: Flow resistance

- Predicted by elastic properties and ideal shear strength based on the Peierls-Nabarro model
- At low strains: Edge dislocation

$$\tau_c^{\alpha} = \tau_{\rm P}^{edge}$$

• At large strains: Junctions with screw components

$$\tau_{c}^{\alpha} = \left(1 - w\gamma^{\beta}\right)\tau_{\mathrm{P}}^{edge} + w\gamma^{\beta}\tau_{\mathrm{P}}^{screw}$$



Stress-strain behavior of Ni: Single crystal

- $h_{\alpha\beta} = q_{\alpha\beta} \left[h_0 \operatorname{sech}^2 \left| \frac{h_0 \gamma}{\tau_s \tau_0} \right| \right]$
 - $\tau_0 \& h_0$ from DFT predictions
 - w and τ_s calibrated to Yao data
- Predict large strain response of various orientations (Haasen data)







Haasen, *Philos. Mag. 1958*, 328, 384–418. Yao et al., *J. Nucl. Mater. 2003*, 323:2–3, 388– 393. ¹²

Ni-X stress-strain behavior

Crystal plasticity finite element method results for single crystals

Engineering Strain [m/m]



First-principles results considering pre-stra Resolved shear stress calculated using: $\tau_c^{\alpha} = (1 - w\gamma^{\beta})\tau_P^{edge} + w\gamma^{\beta}\tau_P^{screw}$

Polycrystal hardening moduli

- Pierce-Asaro-Needleman
 - Function of $\gamma = \Sigma \gamma^{\beta}$
 - Three parameters: τ_0 , h_0 , τ_s
 - Limited hardening capacity
- Bassani-Wu
 - Function of γ^{β}
 - Two additional parameters
 - Higher hardening capacity
- Modified PAN
 - Function of γ^{β}
 - Three parameters
 - Higher hardening capacity



Polycrystal models

- Simplified polycrystal models
 - Accurate to larger microstructural models
 - Much more efficient
- Automated parameter search
 - Existing open-source optimization library, scikit-optimize (scikit-optimize.github.io)
 - No gradient information necessary (Bayesian optimization via Gaussian processes)
 - Independent of hardening model details
- Hardening model conclusions
 - PAN increased hardening capacity when using γ^{eta}
 - Grain size dependence from literature on Ni, Cu



Correlation analysis of materials properties by machine learning: Stacking fault energy in dilute fcc-based alloys



J Phys: Cond. Matter, 2021, DOI: 10.1088/1361-648X/ac0195

Temperature-dependent hardness model $H_{v} = 0.1615 G \left(\left(1 + 5200 \frac{b}{(1-v)^{2j} w(T)} \left(\frac{\tau_{PN}}{G}(T) \right)^{2} \right)^{\frac{1}{1.4}} \right)^{2} e^{-2.2k} \left(\frac{b}{(1-v)^{j} w(T)} \right)^{4} \tan^{3/2} \alpha \cos \alpha$

- Thermally activated dislocation motion: Correlated to diffusion
- For most FCC metals
 - At high-T, dislocation climb
 - At low-T, dislocation glide

Expt. Au

Dislocation climb

0.8

bv 2V

0.6

нv (GPa)

0.01

0.0

Dislocation glide

0.4

T/T_M

0.2

Disl. modeled

2V modeled

1V modeled

(GPa)

₹

1.0

0.1-

0.0



Hardness model for twin boundary

• For FCC metals:

$$b\left(\frac{1}{2}[110]\right) \rightarrow b_1\left(\frac{1}{2}[110]_T\right) + b_2\left(\frac{1}{6}[11\overline{2}]\right)$$

Dislocation through twin boundary → Shockley partial dislocations (¹/₆[112]) activated near twin boundaries → screw characteristics





Summary

Multiscale approach from electron, atoms, to phase, and applications

High-throughput calculations and modeling for efficient data generation

Molecular Dynamics

- DFT-based first-principles calculations of thermodynamic/mechanical properties
- Machine learning models to predict and analyze properties of interest
- CALPHAD modeling to develop the databases

Density Functional Theory

- Python-based open-source codes: SIPFENN, DFTTK, ESPEI, and PyCalphad
- A combined DFT/FEM approach to study tensile stress-stain behaviors for both single crystals and polycrystals
 - Crystal plasticity finite element method (CPFEM) for single crystals
 - CPFEM for polycrystals (Modified PAN model)
 - Key fundamental → Ideal shear strength of pure elements, Ni₁₁X, Ni₃₄ZX by DFT calculations and machine learning understanding
 - Hardness model
 Considering both elastic and plastic deformations temperature, and cross-slip mechanism of dislocation

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CALPHAD