

Predictive Design of Novel Ni-based Alloys D.D. Johnson and M.J. Kramer Ames Laboratory, US-DOE, Ames, IA 50011







Project Description and Objectives

Develop new alloys that can perform at elevated temperatures in supercritical steam and CO_2 environment.

Use advanced computational tools, validated by targeted experiments, to increase operating temperature of Haynes-282 by 50°C

Enable AUSC to operate above 760°C and 5000 psi

Provide 'plug-in-play alloy' alloy compatible with current Ni-based alloy production.



Challenge is to develop an efficient, high fidelity multi-element alloy design tool





Current Status of Project

Modeling Approach

- Korringa-Kohn-Rostoker method and coherent potential approximation (KKR-CPA)
 - Highly efficient electronic structure method that allow for complex chemistries using smaller model sizes compared to DFT.
- Mean-field approximation of the T_m
 - Includes short-range ordering and clustering

Accurately models complex chemistries to predict phase stability

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Singh, Prashant, Gupta, Shalabh, Thimmaiah, Srinivasa, Thoeny, Bryce, Ray, Pratik K, Smirnov, Andrei V, Johnson, Duane D & Kramer, Matthew J. Vacancy-mediated complex phase selection in high entropy alloys. *Acta Mater* **194**, 540-546 (2020).



The equation of state E(V) calculation for the fcc, bcc, and hcp phases for Haynes-282:

 $Ni_{0.567}Cr_{0.224}Co_{0.099}Mo_{0.052}Ti_{0.026}AI_{0.032}$



Current Status of Project

Modeling Validation

- Compare predicted values for
 - Phase stability
 - Melting Temperatures (T_m)
 - Elastic Moduli

Alloy Design Criteria

- Identify promising regions of phase space for:
 - $T_m \sim > 50^{\circ}C$ of Haynes 282
 - Elastic Moduli > 10% higher
 - Sufficient Cr, AI , (Si) for oxidation stability
 - Reduce Co (lower cost)

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Current Status of Project

Baseline Characterization of Haynes-282

- Alloy sheet from Haynes (also provided additional data on oxidation and microstructure)
 - Initial oxidation characteristics
 - Phase assemblages and T_m
 - **Elastic Moduli**

Alloy Selection and Testing

- Characterize alloys across prospective phase space
 - DSC, XRD, SEM, Ultrasound
- Further evaluate 'best samples' for

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- Oxidation resistance
- **Mechanical properties**



20



9/6

2/2.3

1.5/1.8

Мо

Ti

AI

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20 (°) XRD of the surface after 100 hrs

oxidation in dry air

700 ° C 100h

As received

70

Model Prediction

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Investigate role of major element substitutions.

 Shaded regions show the extend to solid solution for each element (Ni, Co and Cr) in a fcc matrix compared to bcc and hcp)



Ni > 45 at. %: Co < 40 at. %: Cr < 35 at. %

Energies are shown relative to that of an elemental solid X in Haynes-282



Model Prediction

Investigate role of minor element substitutions.

 Shaded regions show the extend to solid solution for each element (Mo, Ti and Al) in a fcc matrix compared to bcc and hcp)



Mo < 17 at. %: Ti < 11 at. %: Al < 15 at. %

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Role of refractories

Formation Enthalpy



The calculated formation enthalpy (E_{form}) with experimentally-determined melting temperature (T_m) for common Haynes alloys. Similar trends were seen with Mo (less dramatic) and Re (more dramatic).

Role of Fe, Hf, Nb, Si, V incorporated into the 2nd generation



Role of refractories on bulk moduli

Haynes-282 bulk moduli was calculated (●) and compared to experimental data (□).

Calculation overestimated by ~ 8-20 GPa (within 10%)

Model alloy $Ni_{65}Cr_{12}Co_5Al_3Ti_3X_{10} X = Mo, Re, W$

- Understand role of refractory elements
 - Moduli and T_m increased with increasing valance electrons
 - Are there chemical substitutions that can mimic this effect?
 - Necessary to reduce cost and density



Similar trends were seen with Mo (less dramatic).





Role of refractories on phase stability

1st Generation

Experimental (DSC)melting data compared to Haynes-282.

Model alloys selected to validate specific predictions

Ti > 3% and refractories > 5% resulted in bcc phases

Deviation from prediction







Role of refractories on phase stability

Experimental and calculated T_m onsets of 1st generation samples compared to Haynes 282.

Alloys fabricated and characterized to validate model predictions.

- Model didn't correctly predict T_m for phase separate samples
 - Identified limits for the high T solid solution
- Model captured the trends in T_m for Mo, Re and W.







Refined suite of compositions

1st Generation alloy

2nd Generation alloy

2nd generation results

- Target compositions w/ fcc matrix
- Investigate larger range
 of Ni, Co and Cr
- Include B, C, Fe and Si

| | T | 1 | | | | (200) | Ń | I |
|----------|--------------------------------------|-----------|--|----------|------------|-------|---|---|
| | | | | | NISA29A HT | ~ | Ni ₆₇ Cr ₁₇ Co ₄ (AlTiFe | SiC) ₁₂ (Mo _{2.5} W _{2.5}) |
| | 4 | 8 | | | NISA28A HT | | Ni ₆₇ Cr ₁₇ Co ₄ (AlTiM | oFeSiC) ₁₂ |
| NISA12 | A_AC | | Ni ₇₀ Cr ₁₃ Co ₅ Al ₃ Ti ₂ Re ₅ | | NISA27A_HT | | Ni ₆₇ Cr ₁₅ Co ₆ (AlTiM | oFeSiC) ₁₂ |
| NISA11 | A AC | | Ni ₇₀ Cr ₁₃ Co ₅ Al ₃ Ti ₂ W ₅ | | NISA26A_HT | | Ni ₆₅ Cr ₁₅ (CoAlTiMo | FeSiC) ₂₀ |
| NISA10 | NISA10A_AC NISA9A_AC NISA8A_HT | | Ni ₇₀ Cr ₁₃ Co ₅ Al ₃ Ti ₂ Mo ₅ | n (| NISA25A_HT | | Ni ₆₇ Cr ₁₃ (CoAlTiMo | FeSiC) ₂₀ |
| NISA94 | | h | Ni _{75.6} Cr _{11.63} Co _{5.81} Al _{3.5} Ti _{3.5} | a. | NISA24A_HT | | (NiAlTiMoFe) ₇₉ Cr ₁ | ₃Co ₈ |
| | | | Ni ₆₉ Cr ₁₀ Co ₅ Al ₃ W ₅ Re ₅ Ti | ₹ T | NISA23A_HT | | (NiAlTiMoFe) ₇₉ Cr ₁ | ₈ Co ₃ |
| | | | NiCrCo-Al-ReTi | - isc | NISA22A_HT | | (NiAlTiMoFeSi) ₇₉ C | r ₁₃ Co ₈ |
| sit sit | | | | ter – | NISA21A_HT | | (NiAlTiMoFeSi) ₇₉ C | r ₁₆ Co ₅ |
| | HT A | | NI ₆₉ Cr ₁₀ Co ₅ Al ₃ W ₁₀ H ₃ | | NISA20A_HT | A | (NiCrCoAlTi) ₉₃ Mo | Fe ₁ Si _{0.5} C _{0.25} B _{0.25} |
| L NISA5 | | | Ni ₆₅ Cr ₁₂ Co ₅ Al ₃ W ₈ Ti ₇ | | NISA19A_HT | | (NiCrCoAlTi) ₉₃ Mo | Fe ₁ Si _{0.5} C _{0.} |
| NISA4/ | | | Ni ₆₅ Cr ₁₂ Co ₅ Al ₃ Re ₈ Ti ₇ | | NISA18A_HT | | (NiCrCoAlTi) ₉₃ Mo | Fe ₁ Si ₁ |
| NISA34 | _нт | | Ni ₆₅ Cr ₁₂ Co ₅ Al ₃ W ₃ Re ₈ Ti ₄ |] 1 | NISA17A_HT | | (NiCrCoAlTi) ₉₅ Mo | 2.5W _{2.5} |
| NISA24 | НТ | | Ni ₆₅ Cr ₁₂ Co ₅ Al ₃ W ₈ Re ₃ Ti ₄ | | NISA16A_HT | | (NiCrCoAlTi) ₉₅ Mo | |
| NIISA 1/ | | | | - | NISA15A_HT | | (NiCrCoAlTi) ₉₃ Re ₅ I | -e ₂ |
| INISA IA | | | | | NISA14A_HT | | (NiCrCoAlTi) ₉₃ W ₅ F | e ₂ |
| H282 | | / | ~Ni ₅₆ Cr ₂₂ Co ₁₀ Al _{3.3} Mo ₅ Ti _{2.5} | - | NISA13A_HT | | (NiCrCoAlTi) ₉₃ Mo | Fe ₂ |
| 40 | 50 | 60 | 70 | 80 4 | | 50 | 60 | 70 |
| | | 2θ (deg.) | | | | 20 (| dea.) | |

fcc, bcc and $L1_2$ phases present





Nearly single phase fcc, as cast show texturing along [200]

Refined suite of compositions

onset

peak

end

cal

2nd generation results

- Increased T_m
- Narrow the range of melting



| 10 | Compare the second s | | Melting T | | | | | |
|------|---|--------|-----------|--------|-----------------------------------|--|--|--|
| שו | Composition | Onset | Peak | End | | | | |
| H282 | $Ni_{55.3}Cr_{21.9}Co_{9.7}AI_{3.2}Ti_{2.2}Mo_{5.0}Fe_{1.5}Mn_{0.3}Si_{0.3}C_{0.3}$ | 1329.5 | 1362.7 | 1369.9 | | | | |
| | | | | | | | | |
| 13A | (NiCrCoAlTi) ₉₃ Mo ₅ Fe ₂ | 1380.3 | 1404.4 | 1410.6 | | | | |
| 14A | (NiCrCoAlTi) ₉₃ W ₅ Fe ₂ | 1407.0 | 1435.7 | 1441.4 | ⊢ Mo-W-Re | | | |
| 15A | (NiCrCoAlTi) ₉₃ Re ₅ Fe ₂ | 1411.3 | 1441.7 | 1460.5 | Fe, Cr:Co | | | |
| 16A | (NiCrCoAlTi) ₉₅ Mo ₅ | 1362.6 | 1388.8 | 1393.7 | MasiM | | | |
| 17A | (NiCrCoAlTi) ₉₅ Mo _{2.5} W _{2.5} | 1380.4 | 1407.7 | 1413.1 | | | | |
| 18A | (NiCrCoAlTi) ₉₃ Mo ₅ Fe ₁ Si ₁ | 1351.4 | 1386.0 | 1392.9 | | | | |
| 19A | (NiCrCoAlTi) ₉₃ Mo ₅ Fe ₁ Si _{0.5} C _{0.5} | 1356.8 | 1382.5 | 1389.5 | Adding Si-C-B | | | |
| 20A | (NiCrCoAlTi) ₉₃ Mo ₅ Fe ₁ Si _{0.5} C _{0.25} B _{0.25} | 1376.7 | 1389.0 | 1395.0 | | | | |
| 21A | (NiAlTiMoFeSi) ₇₉ Cr ₁₆ Co ₅ | 1352.9 | 1384.3 | 1389.7 | | | | |
| 22A | (NiAlTiMoFeSi) ₇₉ Cr ₁₃ Co ₈ | 1366 | 1383.6 | 1399.6 | Fo Sim/Crico | | | |
| 23A | (NiAlTiMoFe) ₇₉ Cr ₁₈ Co ₃ | 1362.4 | 1388.2 | 1392.9 | Fe, SI W/ Cr:CO | | | |
| 24A | (NiAlTiMoFe) ₇₉ Cr ₁₃ Co ₈ | 1376.8 | 1402.5 | 1408.2 | | | | |
| 25A | Ni ₆₇ Cr ₁₃ (CoAlTiMoFeSiC) ₂₀ | 1357.5 | 1391.5 | 1396.7 | Nii·Cr | | | |
| 26A | Ni ₆₅ Cr ₁₅ (CoAlTiMoFeSiC) ₂₀ | 1350.3 | 1386.1 | 1393.9 | NI.CI | | | |
| 27A | Ni ₆₇ Cr ₁₅ Co ₆ (AlTiMoFeSiC) ₁₂ | 1351.6 | 1387.3 | 1395.5 | Cr:Co Ni:Cr:Co | | | |
| 28A | Ni ₆₇ Cr ₁₇ Co ₄ (AlTiMoFeSiC) ₁₂ | 1356.0 | 1378.8 | 1386.0 | | | | |
| 29A | Ni ₆₇ Cr ₁₇ Co ₄ (AlTiFeSiC) ₁₂ (Mo _{2.5} W _{2.5}) | 1388.7 | 1400.0 | 1403.9 | | | | |

stoichiometries are fixed for elements in the parentheses for each grouping





Refined suite of compositions for down selection process

- Identified a broad range of compositions with T_m > 50°C of Haynes 282
 - RT Moduli is effective criteria for further down selection.
 - Ideal for implementing advanced search algorithm and machine learning for optimization

Samples 20, 24 and 29 were selected for further study

2nd Generation



Highlighted samples have the highest T_m



Down Selection







Baseline Characterization-Haynes 282

Oxidation (TGA)

- Synthetic air, 760, 800, 900 & 1000 °C isothermal holds 20-100 hrs
- Two-step steady state oxidation
 - How does changes in alloy composition alter the transient and steady-state oxidation?

Cross-sectional SEM

- ~10 µm continuous oxide layer
 - Primarily Cr₂O₃, TiO₂ and NiO (XRD)
 - Oxide penetration (~20 μm), mostly Al₂O₃, No MoO₃

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Electron back-scatter image (top) and elemental EDS maps for Haynes-282 after oxidation at 760 °C/100h



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Oxidation: Haynes 282 vs Ames 20, 24 & 29 at 800°C



- Excellent oxidation for Ames #20, 29 samples.
- Even with less Cr, Co, the scale on #20, 29 seems to be more protective at 800°C.
- Ames 24 shows how small changes in Cr, Si can have profound changes in oxidation resistance.





Next Steps

In depth analysis of the down-selected alloys

- Role of Mo:W on:
 - T_m and moduli
 - Relationship between calculated moduli and yield strength
- Role of Ni:Cr:Co
 - Small changes also have large effects on $\rm T_{\rm m}$ and moduli
- Roles of B,C, Si
 - Dramatically effect short term oxidation

High Temperature tensile and creep strength

- Optimize aging protocols
 - Role of t, T and minor alloy content on carbide formation and $\gamma"$

Evaluate oxidation resistance compared to Haynes 282

- Perform 100 hrs oxidation test in dry air at 760, 800 and 900°C
- Extended oxidation test in dry and wet at 800°C





Challenges

Predicting lifetime performance

- KKR-CPA is accurately predicting overall alloy performance for complex chemistries
 - Range of Stability
 - Moduli (as a function of T)
- Can we extend these methods to predict stability at elevated T?
- Need rapid screening methods to predict alloy performance under realistic conditions including:
 - Corrosion/oxidation
 - Creep
 - Aging



Ames has developed a small punch test to rapidly evaluate small sample creep properties up to 1350°C.





Preparing Project for Next Steps

Market Benefits/Assessment

- Increase operating T of Haynes-282 by 50°C
 - Higher operating efficiencies
 - Longer lifetime
- Materials failures represent a significant fraction of power plant operating costs.
- Accurate and efficient modeling can reduce time to market.

Technology-to-Market Path

- Adoption: The optimized alloy's fabrication will fit into existing plants.
- Remaining technology challenges: Life-time assessment.
- New research: Develop methods to predict phase evolution/formations under operating conditions
- Haynes is providing materials and data.



Concluding Remarks

- Computationally efficient multi-elemental approach validated for Ni-based alloys will enable FE to address these challenges:
 - Development of new alloy materials that have the potential to improve the performance and/or reduce the cost of existing fossil fuel technologies.
 - Development of materials for new energy systems and capabilities.
 - Development of refractory alloys to withstand even higher temperatures and aggressive environments.
- Current approach optimizes alloy composition based on phase stability and elastic moduli.
 - Model identified a broad range of promising compositions.
 - Developing suite of characterization tools to rapidly assess promising candidate compositions.





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