FWP 66059

Solid

Phase

PROCESSING



Solid State Joining of Creep Strength Enhanced Ferritic Steels

Glenn J. Grant, Dalong Zhang, Jens Darsell, Chris Smith

2021 DOE FE NETL High Performance Materials Program Virtual Meeting 6/10/2021



PNNL is operated by Battelle for the U.S. Department of Energy



Motivation and Project Objective



- Creep Strength Enhanced Ferritic Steels are workhorse alloys at lower end of elevated temperature applications in Thermal Power plants
 - Low-cost, low thermal expansion, high conductivity alloys for applications 570 C to 620 C (piping, waterwall / membrane wall, superheaters (SC), reheaters (SC))
- Problem
 - Base materials have good creep performance, but they are compromised by welding
 - Microstructure instability over time leads to earlier than expected creep failure in the HAZ of weldments





- WSRF can be as low as 0.50 at long creep times.
- This leads to greater allowances in pipe and tube wall thicknesses (higher material cost and heat transfer inefficiency)
- Forces reductions in operating temperature and/or pressure, which leads to a reduction in plant efficiency.



Parker J, International Journal of Pressure Vessels and Piping (2012), http://dx.doi.org/10.1016/j.ijpvp.2012.11.004

 Creep "softness" occurs in the fine grained heat affected zone adjacent to the weld nugget

> Can a new welding process create a microstructure that will show reduced long term microstructure degradation compared to conventional fusion weldments?

What is Friction Stir Welding ?



Solid-phase joining processes (no melting)

- Spinning, non-consumable tool is plunged into the surface of a material.
- Friction and plastic work energy heats the material sufficiently to lower the flow stress.
- When material softens, the tool is then translated along the joint line causing material in front of the pin to be deformed around to the back, and forged into the gap behind the traveling pin
- The resulting joint is characterized by:

Pacific Northwest

- Fine-grained "nugget" composed of recrystallized grains (d)
- Surrounded by a mechanically deformed region (c) and a heat affected zone (b)







Tools for Steels

Steel Friction Stir Welding – State of the Art



Can it weld the alloys? Yes

Pacific Northwest

> Almost all commercially relevant ferrous and nickel based alloys have had some degree of FSW welding development (High and low carbon steels, 9Cr, 12Cr, 20Cr steels, HSLA, super martensitic, DP, austenitic/ferritic stainless, Ni alloys 600, 718, 282, C22



Can it weld the thicknesses? Some Limits, but tools are improving

- We can now reach 0.375" (9.5mm) to 0.5" (13mm) with commercial tooling in most steels and Nickel alloys
- Welds up to 0.67" single pass in HSLA steels have been demonstrated with PCBN/W-Re tooling
- Welds up to 1.1" single pass in API 5L X70 steels have been demonstrated with W-Re tooling



Typical macrostructure of a fully consolidated, indication-free FSW weld in Gr 91

Potential Benefits of Friction Stir Welding

- Friction Stir Welding can introduce a significantly lower energy input to the weld than fusion welding
 - Welds can be made at peak temperatures just above Ac3, between Ac1 and Ac3, or in some cases below Ac1 depending on process parameters
 - Lower energy results in lower peak temperature, small PAG, and smaller martensite lath width
- FSW allows for much more control over the weld thermal cycle.
 - Could this allow for a "tunable" carbide precipitation sequence?
- FSW produces a unique weld microstructure where the nugget and the TMAZ has been strained above the austenite temperature.
 - Can straining, a dislocation-rich substructure, or dynamic recrystallization promote fine MX, prevent carbide coarsening and the development of new phases over time?

Can the overall WSRF can be improved by using Friction Stir Welding?









Friction Stir Welding of Creep Strength Enhanced Ferritic Alloys

- Gr91, P92, CPJ-7 have been shown to be friction stir weldable
- Defect free welds, 6mm to 10mm penetration can be made at a wide range of process parameters.
- Most welds achieved using 6-7mm Megastir Q70 or Q60 tools
- Processing Parameters
 - Welds made under temperature feedback controlled conditions at tool temperatures from 715-1000C
- Welds made in each of the alloys, and at 3 to 5 weld temperatures, depending on the alloy, were subjected to creep rupture testing







Gr91 Friction Stir Welds vs Fusion Welds in cross weld tensional creep at 625C

Pacific Northwest

First observation: Welds fail in the heat affected zone in a similar way to fusion welds with subtle differences

Creep Rupture Testing of Friction Stir Welds



WSRF improved from 0.77 for fusion welds to 0.80 for FSW at high load, low time

At longer times (4000 hrs) FSW performance is also better at 0.68 for fusion welds and 0.82 for FSW

FSW weld in Gr91 shows ~2X improvement in creep life over fusion welds





Grade 92 625°C Transverse Creep





Pacific Northwest

Same is true for P92

High load creep performance is very good

WSRF improved from 0.75 for fusion welds to 0.97 for FSW at high load, low time

(Base load plant might have to load cycle due to renewables!)

At longer times (10,000 hrs) FSW performance is closer to fusion welds

<u>CPJ-7</u> 625°C Transverse Creep





Pacific Northwest

- FSWs in CPJ-7 behave similar to P-92
 - High load performance is very good compared to base metal
 - No data yet on fusion welded CPJ-7

Creep cavitation begins in the FGHAZ very close to Solid the ICHAZ, very close to the base metal HAZ interface Phase Northwest PROCESSING



Macrograph comparison of welded specimen and after creep fracture

Pacific

• The creep failure appears to occur away from the nugget boundary, right on the edge of the HAZ, near the FGHAZ/ICHAZ boundary







The TMAZ is unique to FSW





What drives differences in creep in these alloys ?



Fine carbides and carbonitirdes (M₂₃C₆ and MX) precipitates play a critical role in creep strength.

Pacific Northwest

- Where these precipitates are located and their size are key parameters.
- Is there a difference in the distribution of these important carbides between FSW and melt based welding processes?



Hawk,JA, Jablonski,PD,2018, CREEP RESISTANT 9%CR STEEL FOR BOILER AND TURBINE APPLICATIONS Proceedings of the 2018 ASME on Elevated Temperature Applications of Materials for Fossil, Nuclear and Petrochemical Industries ETAM2018 April 3-5, 2018, Seattle, WA, USA



FSW weld have a different carbide precipitation sequence and different amounts of carbides, esp MX



HAZ of Fusion Welded P91



Before Creep testing

Fusion weld HAZ before creep shows very low MX

After creep testing

Fusion weld HAZ after creep shows low MX and Fast coarsening of M23C6 and Laves phase

TMAZ of Friction Stir Weld in P91





FSW TMAZ before creep: More MX, even than base metal Less Cr₂₃C₆ than base metal High density of dislocations



T. Watanabe et al. / International Journal of Pressure Vessels and Piping 83 (2006) 63–71

The higher density of MX in FSW TMAZ prior to and after creep test could be a "good player" in making the FSW TMAZ less susceptible to Type-IV failure than that in a fusion weld



Why is there a high density of MX in the TMAZ of a

FSW weld? Is this why FSW better?



- Strain induced dislocations from FSW in the austenite phase field may be nucleation sites for MX and may help to promote a dispersed MX distribution prior to Martensite start.
- This may be the difference between a FSW and a fusion weld. There is coarsening of Cr23C6 and nucleation/coarsening of Laves phase in the HAZ, just like in a fusion weld,
 - But, the higher density of MX in FSW HAZ prior to and after creep test could be a "good player" in making the FSW HAZ less susceptible to Type-IV failure than that in a fusion weld.
- It is also possible that the highly dislocated microstructure even allows for MX nucleation in the solutionized nugget during service life (explanation of the excellent weld nugget-only tests during creep).



Time (hrs)

Next Steps- Putting FSW to the test

- Last task of the project is to show commercial potential by using FSW to fabricate a prototypic part – a P-92 membrane wall
- Possible Advantages

Pacific Northwest

- Better creep performance of weldments
- Lower distortion of panel wall from lowered residual stresses during welding
- Better properties in fatigue
- Less weld penetration into tube wall and so less property depression in wall section













Membrane Wall Welding



Approach

- Weld four lengths of ASTM A213-T92 tube (Vallourec, ~62" long x 1.75" OD x 0.3" wall thickness) from GE/Alstom Power Inc. to Grade 91 webbing strip
- Weld development for the Gr91 web to P-92 tube (tool design, weld process development, microstructure, hardness)
- Two FSW tool types will be utilized:
 - W-Re (W-4HfC-25 Re)
 - PCBN (70PCBN 30 W-Re

 Measure distortion during fabrication









Membrane Wall Welding



 Fixturing has recently arrived and Friction Stir welding will begin to fabricate "short section" multi-tube membrane wall prototypes





Conclusions



- CSEF steels are Friction Stir weldable
- Creep performance is very good, both of the weld metal and in cross-weld tension
 - FSW in P91 can reach a WSRF at 625C of 0.73 to 0.82 over 4000 hrs, significantly higher than fusion welds (0.65)
 - FSW in P92 and CPJ-7 show excellent WSRF at high load, but the performance trends toward fusion weld performance towards lower loads and long times 10,000 hrs
- It is possible that WSRF can be raised by more than 10% from fusion welded equivalents, esp for high load creep
- FSW allows for enough knobs to be turned in the process to customized heat input





It may be possible with FSW to follow a path through thermo-mechanical space that will leave the weld region, and especially the strained part of the HAZ, with a customizable carbide distribution appropriate for better creep resistance, and much closer to the parent microstructure than if it is fusion welded.



Acknowledgment



Acknowledgment: This material is based upon work supported by the Department of Energy Award Number: FWP 66059 Office of Fossil Energy National Energy Technology Laboratory Vito Cedro - Technical Manager

Disclaimer: This report was prepared as an account of work sponsored by an agency of the United States Government. Neither the United States Government nor any agency thereof, nor any of their employees, makes any warranty, express or implied, or assumes any legal liability or responsibility for the accuracy, completeness, or usefulness of any information, apparatus, product, or process disclosed, or represents that its use would not infringe privately owned rights. Reference herein to any specific commercial product, process, or service by trade name, trademark, manufacturer, or otherwise does not necessarily constitute or imply its endorsement, recommendation, or favoring by the United States Government or any agency thereof. The views and opinions of authors expressed herein do not necessarily state or reflect those of the United States Government or any agency thereof.





end

