



# Dartmouth



THAYER SCHOOL OF  
ENGINEERING  
AT DARTMOUTH

## Intermetallic Strengthened Alumina-Forming Austenitic Steels for Coal-Fired Power Systems

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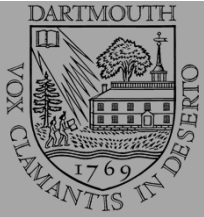
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DOE grant DE-FE0008857



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# Acknowledgement

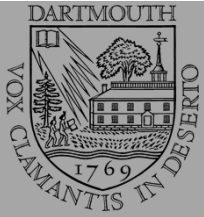
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# Outline

- Introduction
  - Motivation
  - Background
- Results and Discussion
  - Microstructural analysis
  - Thermo-mechanical treatments
  - SEM & TEM characterization
  - XRD analysis
  - Room temperature tensile tests
  - High temperature tensile tests
- Summary

# New Materials for High Temperature Applications

- **Motivation:** Develop materials which can be used at **higher temperature** (>700 °C) and **pressure** (>100 MPa) to enhance efficiency (>50 %) and reduce CO<sub>2</sub> emissions in fossil fired boiler/steam turbine power plants
- **Solutions:**
  - Ni-Base Superalloys: too costly
  - FeCrAl alloys: bcc structure, weak >500 °C
  - Al<sub>2</sub>O<sub>3</sub> coatings or surface treatments
  - Alumina-Forming Austenitic Steels
    - Combination of creep and oxidation resistance
    - Lower cost (Lower nickel content)



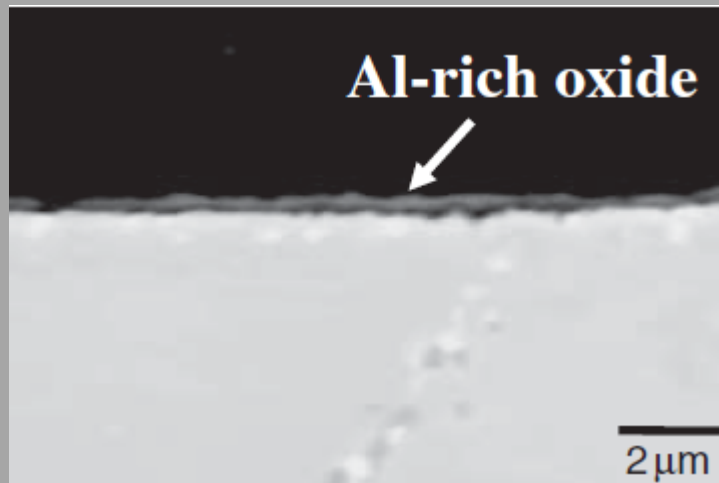
[www.siemens.com](http://www.siemens.com)

Yamamoto, Y., et al.: Science, 2007, vol. 316(5823), pp. 433–36.

Yamamoto, Y., et al., Metallurgical and Materials Transactions A, 2011. 42(4): p. 922-931.

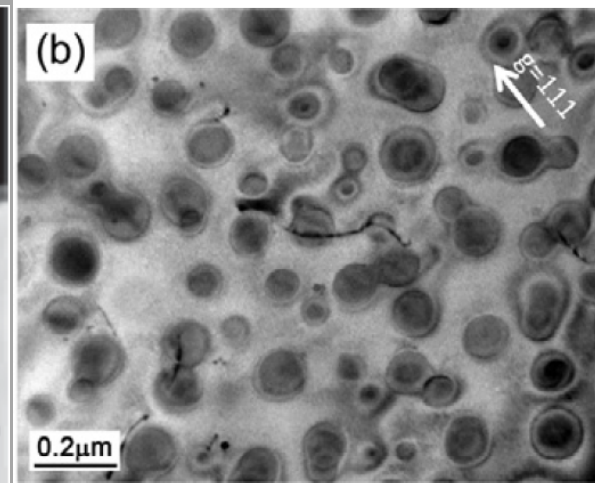
# Alumina-forming Austenitic (AFA) Stainless Steels

- Combination of good **oxidation** & **creep resistance**
  - Oxidation resistance achieved by the formation of protective, external **alumina scale**. (~3 wt.% Al )
  - f.c.c. matrix with **intermetallic** strengthening ( $\text{Ni}_3\text{Al}$  etc.)

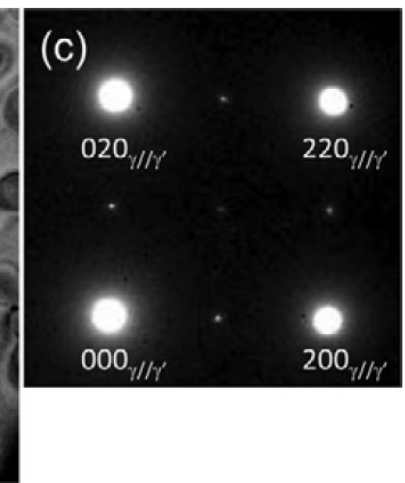


Fe-14Cr-20Ni-0.95Nb-2.5Al-2.5Mo wt. % base alloy (initial developed AFA)

BSE image after 72 hours of oxidation at 800°C in air



Fe-14Cr-32Ni-3Nb-3Al-2Ti wt.% base alloy (recent developed AFA)



TEM BF images of the alloys and SAD pattern

# Oxidation Resistance and Creep Performance of AFA Steels

- Alumina formation in AFA alloys
  - Others: Ti content, C and B addition

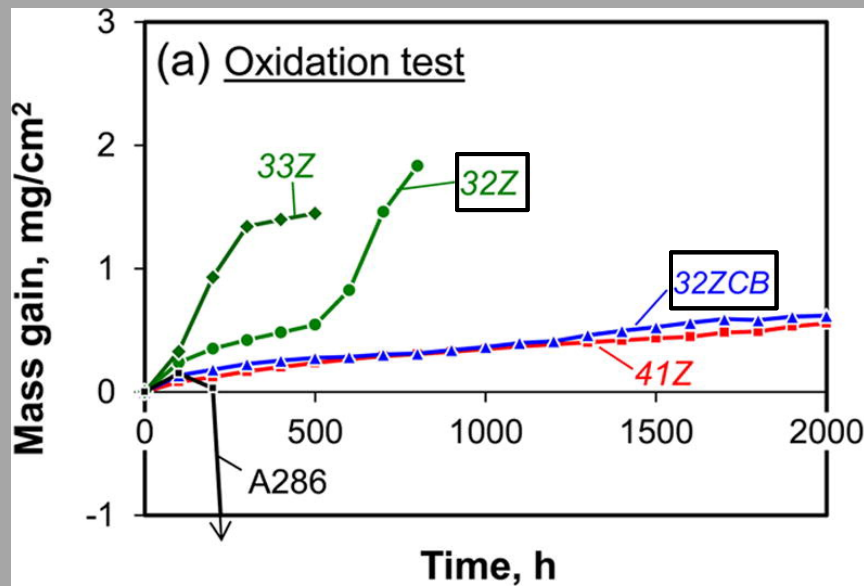
**32ZCB:** Fe-14Cr-32Ni-3Nb-3Al-2Ti-0.27Zr-0.14Si (wt.%)

**41Z:** Fe-14Cr-32Ni-3Nb-4Al-1Ti-0.27Zr-0.12Si (wt.%)

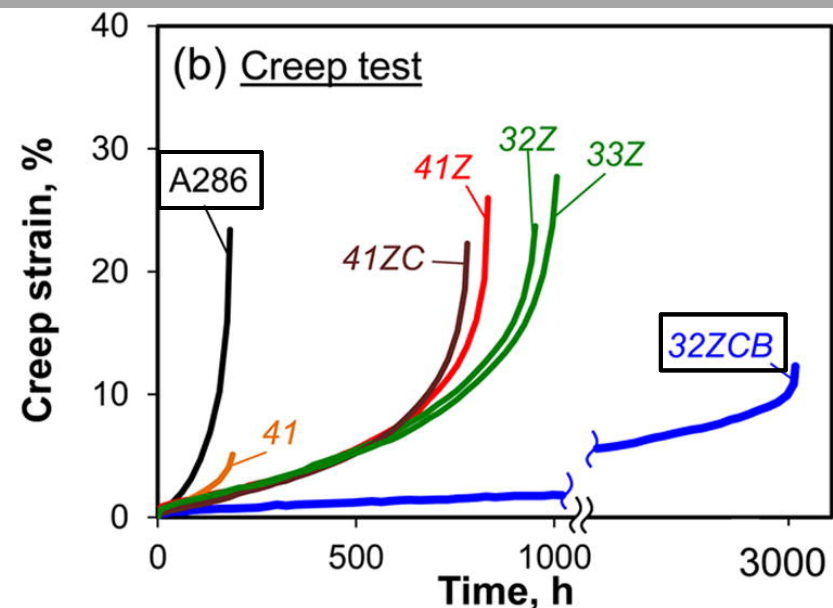
**A286:** Fe-14Cr-25Ni-2Ti-0.15Al (wt.%)

- The best alloy has >7 times longer creep life than A286

(Iron-base superalloy)



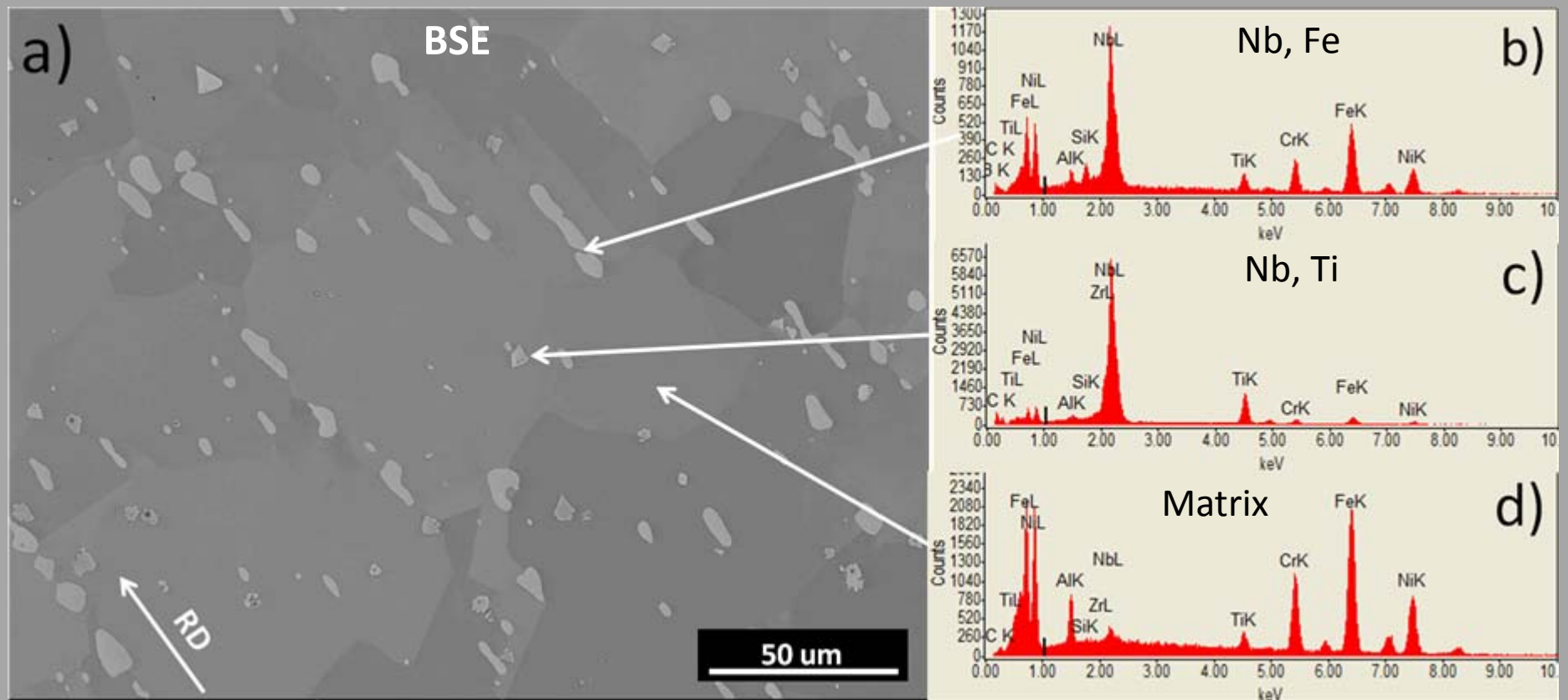
Cyclic oxidation test results at 800 °C in 10% water vapor



creep-rupture curves at 750 °C and 100 MPa.

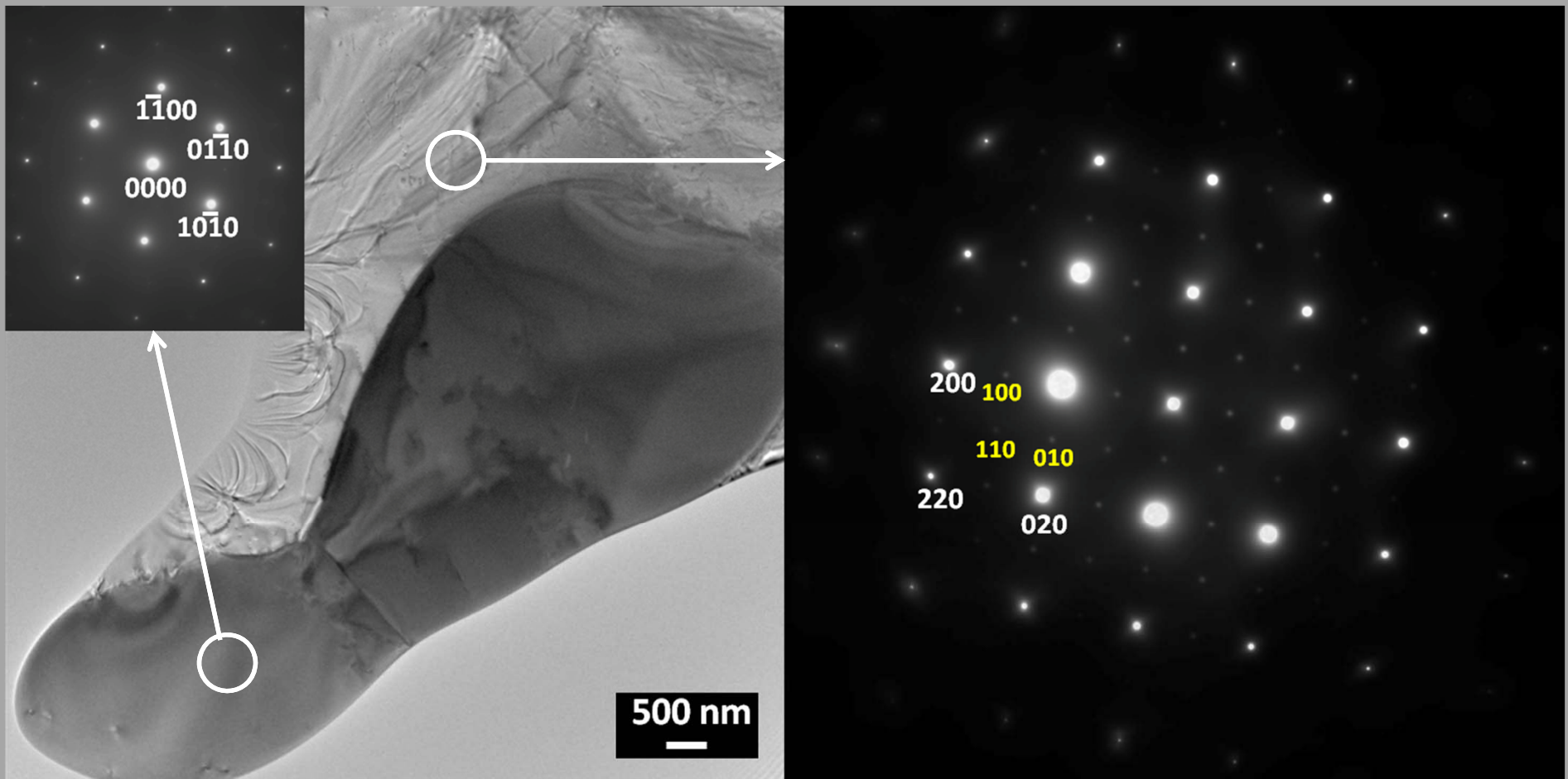
# BSE Image and EDS Results of DAFA29

- DAFA29: Fe-14Cr-32Ni-3Nb-3Al-2Ti-0.3Zr-0.15Si-0.1C-0.01B (wt.%) (as-hot-rolled)
  - Nb enrich precipitates and grain size  $\sim 40 \mu\text{m}$



# BF&SAD of Precipitates in DAFA29

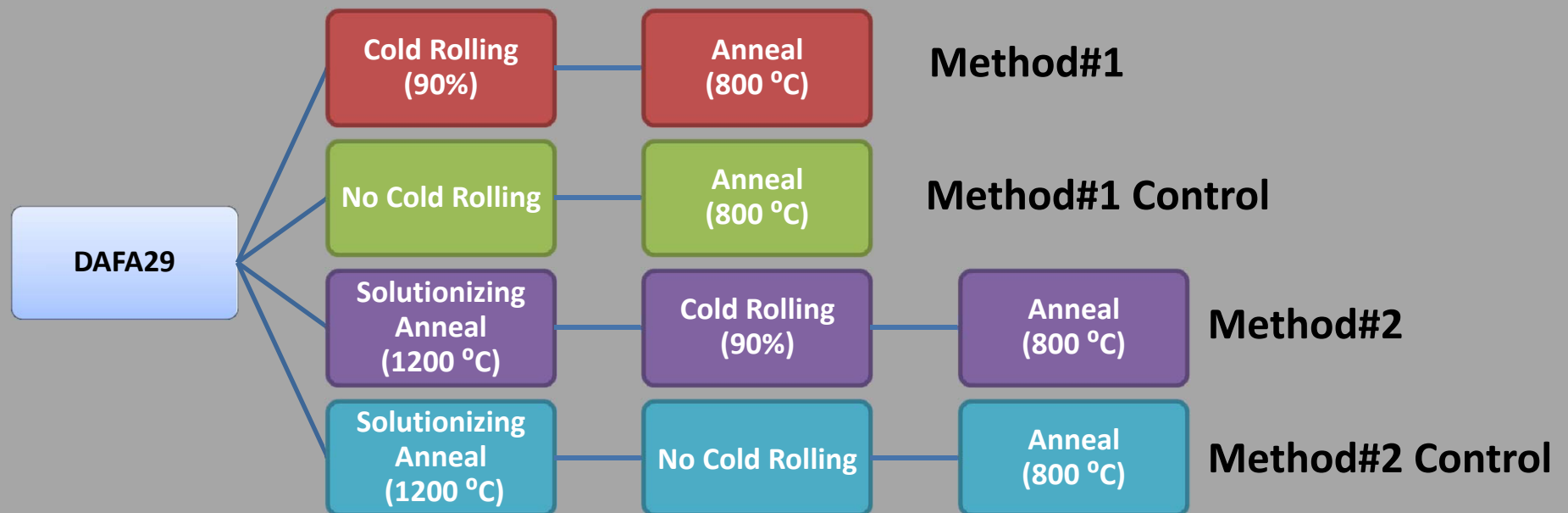
- DAFA29: Fe-14Cr-32Ni-3Nb-3Al-2Ti-0.3Zr-0.15Si-0.1C-0.01B (wt.%) (as-hot-rolled)
  - Fe<sub>2</sub>Nb Laves phase precipitates + L1<sub>2</sub> precipitates in f.c.c. matrix





# Thermo-mechanical Treatments (TMT) Procedures

DAFA29: Fe-14Cr-32Ni-3Nb-3Al-2Ti-0.3Zr-0.15Si-0.1C-0.01B (wt.%) (recent developed)



- Cold rolling 90 % thickness reduction (~4.5 % reduction per pass)
  - ❖ Enhance the creep properties
  - ❖ Introduce dislocations which will act as nucleation sites for fine precipitates

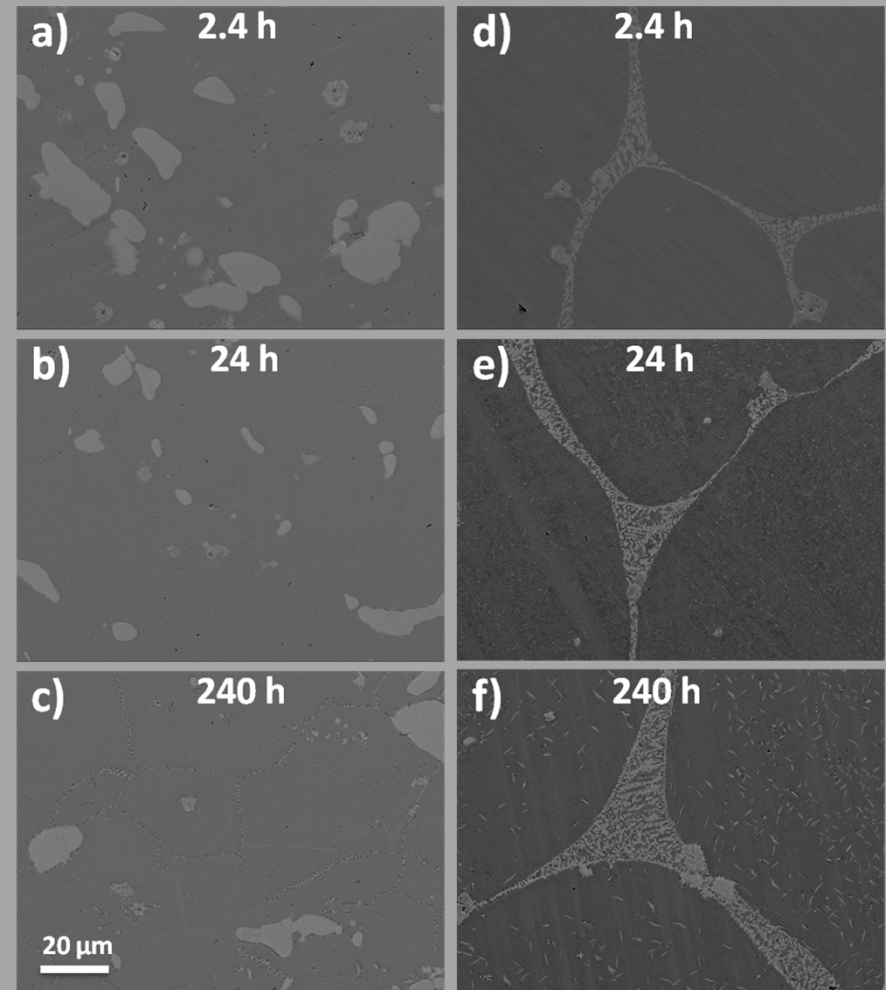
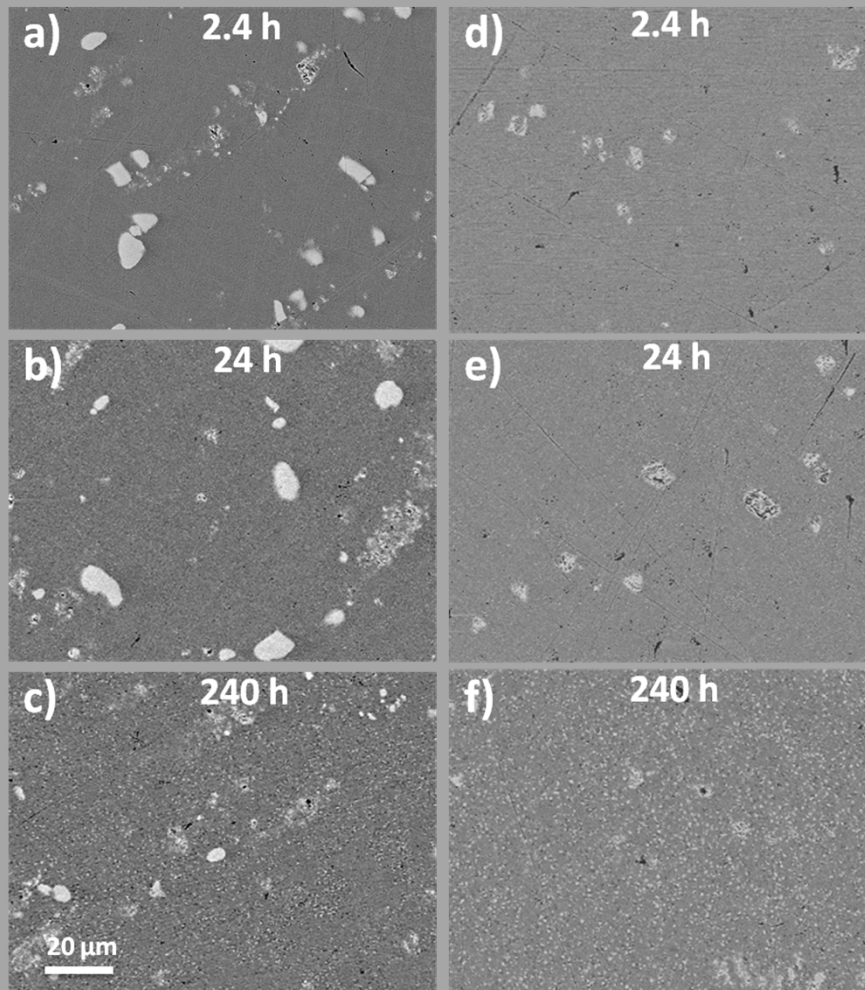
# The Effects of Cold Rolling on The Microstructures of TMT DAFA29

**Method #1**  
Cold rolling + 800 °C

**Method #2**  
1200 °C + Cold rolling + 800 °C

**Method #1 Control**  
800 °C

**Method #2 Control**  
1200 °C + 800 °C



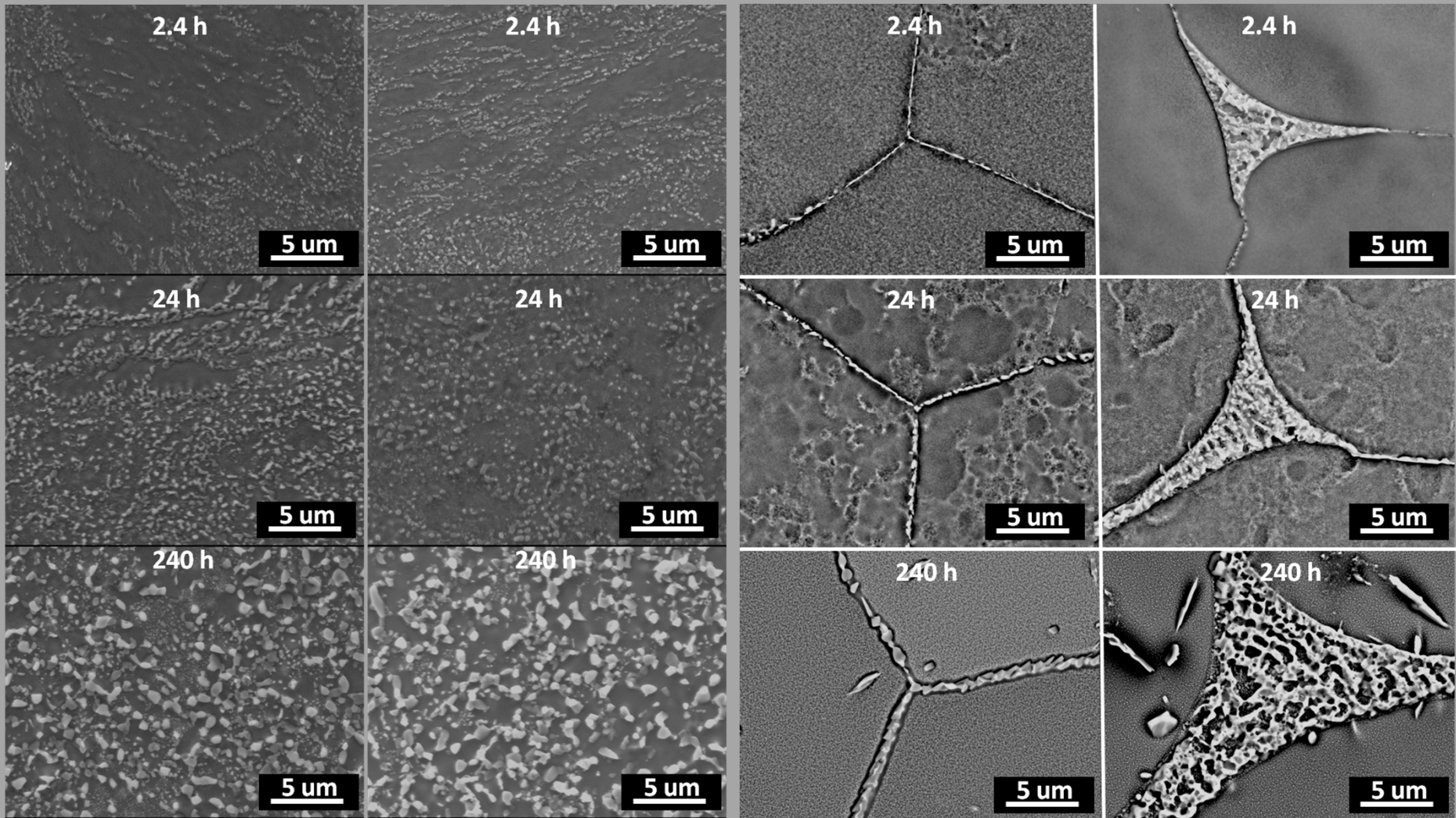
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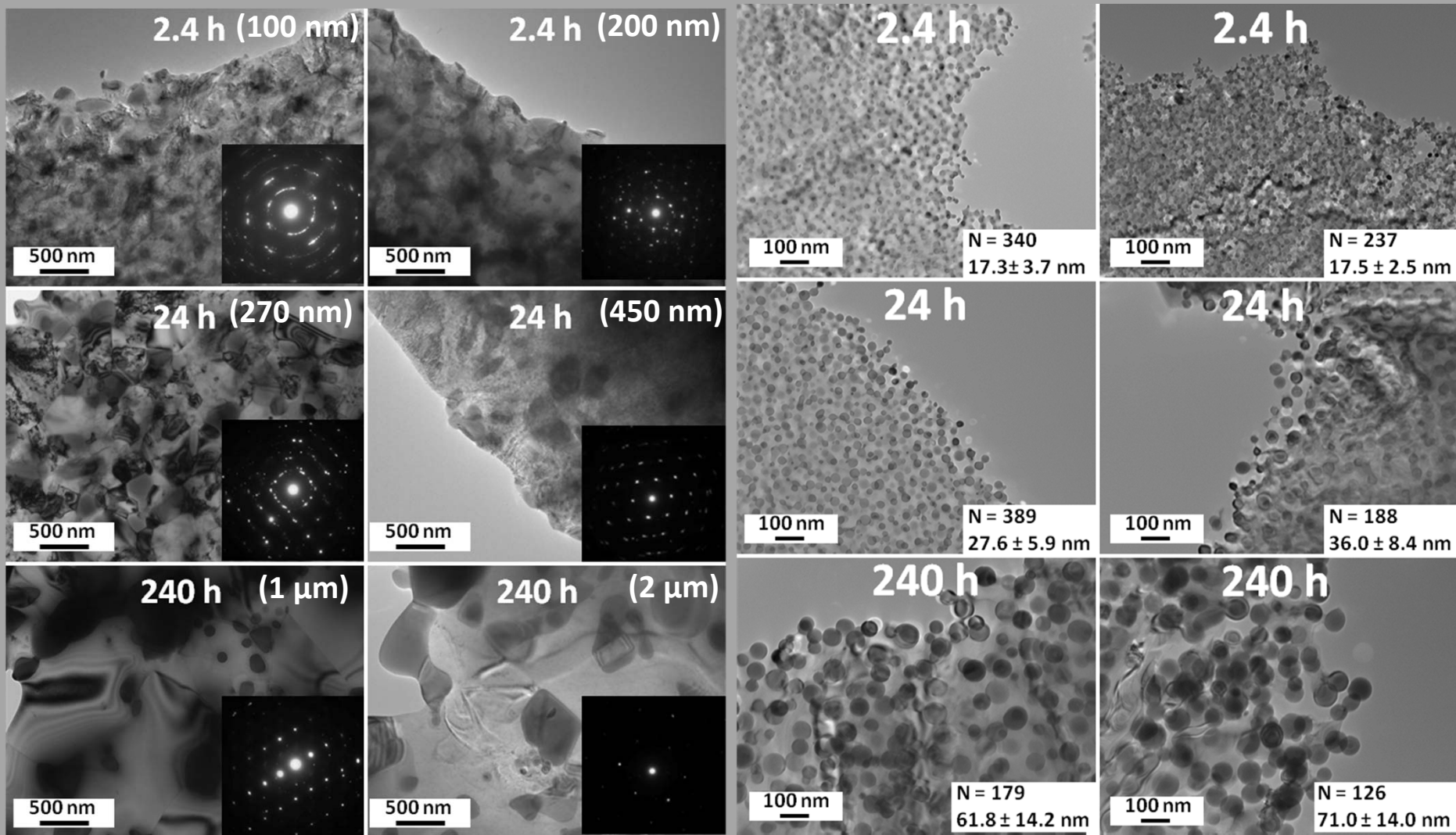
# BF Images and SAD of DAFA29 after Thermo-mechanical Treatments

**Method #1**  
Cold rolling + 800 °C

**Method #2**  
1200 °C + Cold rolling + 800 °C

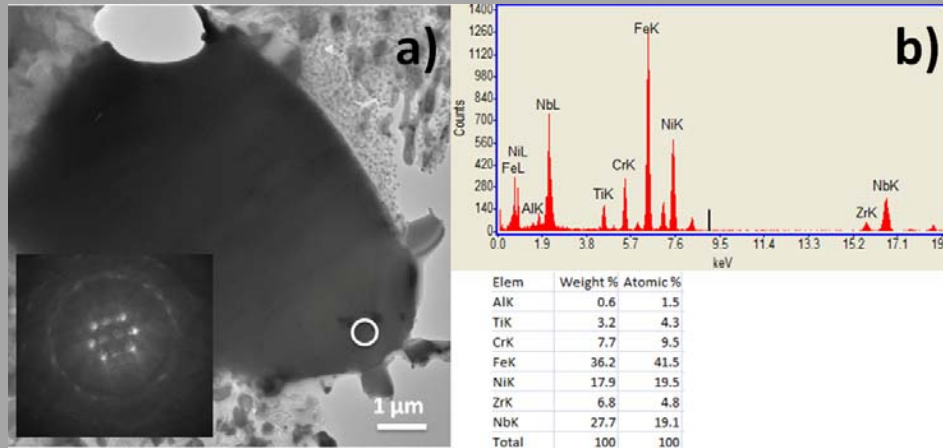
**Method #1 Control**  
800 °C

**Method #2 Control**  
1200 °C + 800 °C

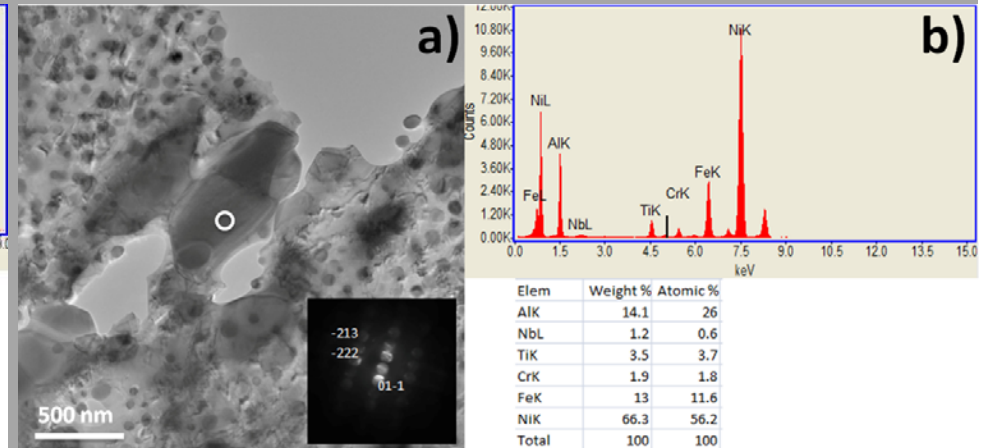


# BF TEM Image, EDS and CBED of Different Precipitates in TMT DAFA29

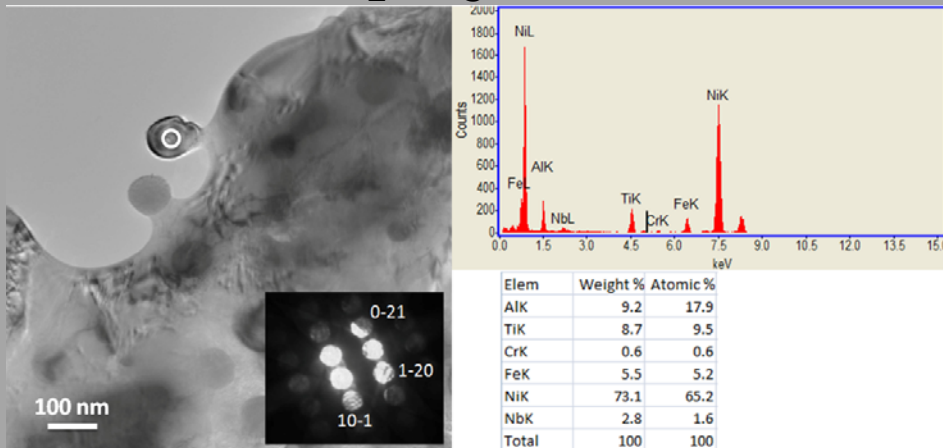
## Laves-Fe<sub>2</sub>Nb (4%)



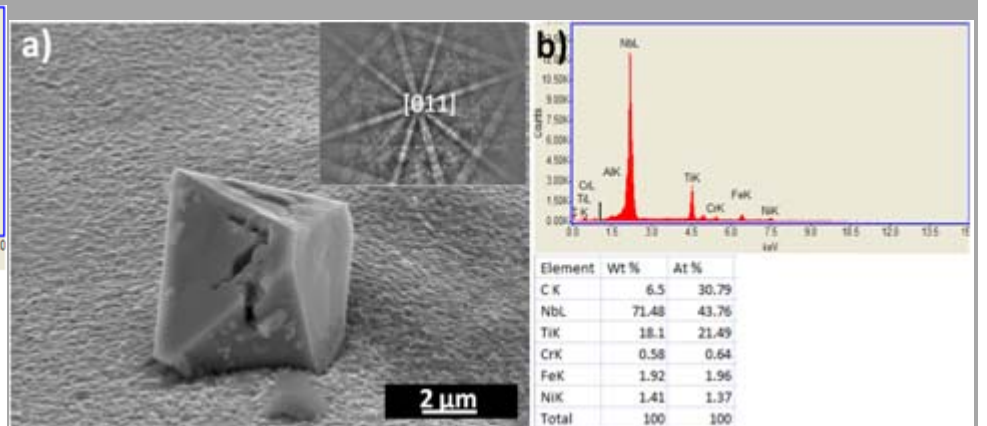
## B2-NiAl (2%)



## L1<sub>2</sub>-Ni<sub>3</sub>Al(Ti) (17%)



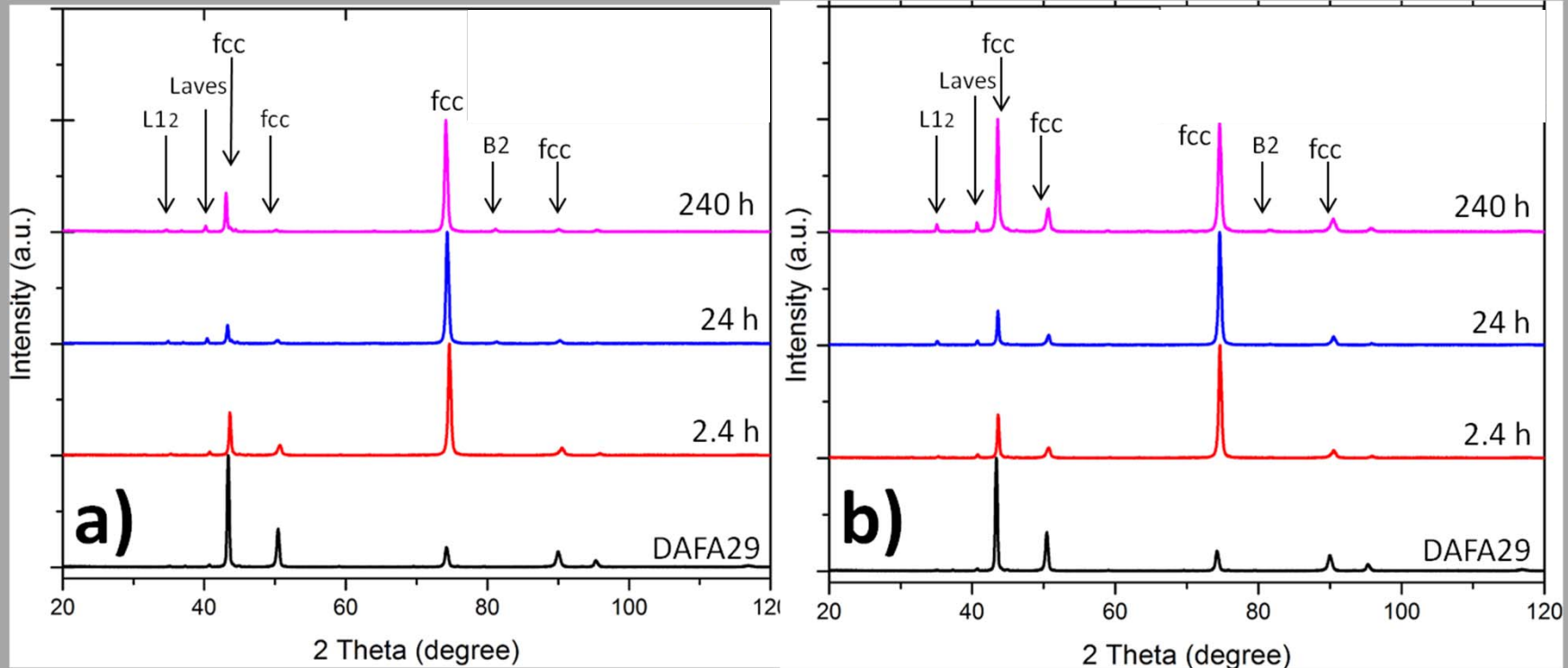
## MC (M:Nb,Ti) (1%)



# Synchrotron XRD Results

**Method#1:**  
Cold rolling + 800 °C

**Method#2:**  
1200 °C + Cold rolling + 800 °C



**Lattice parameters**

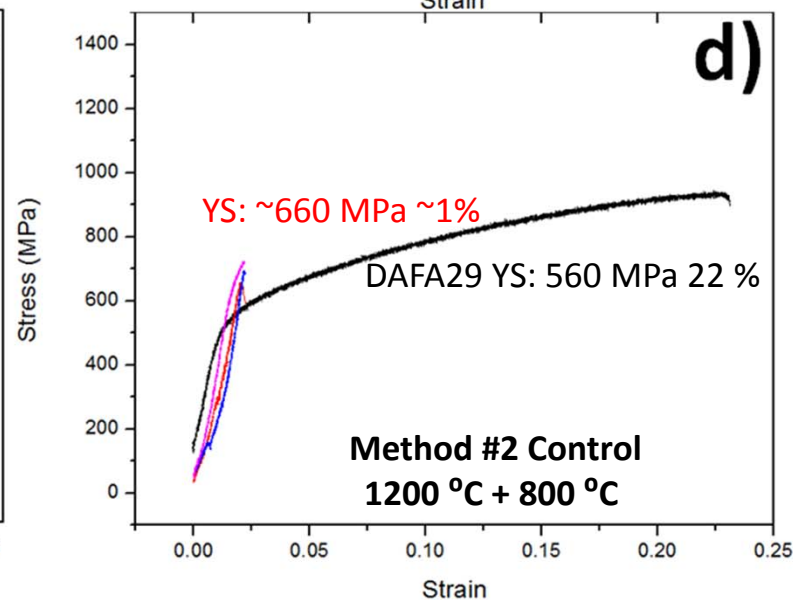
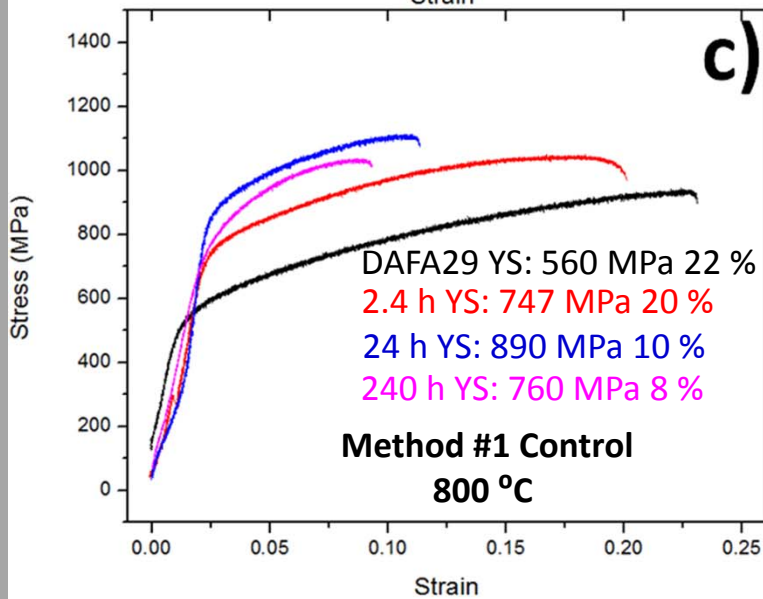
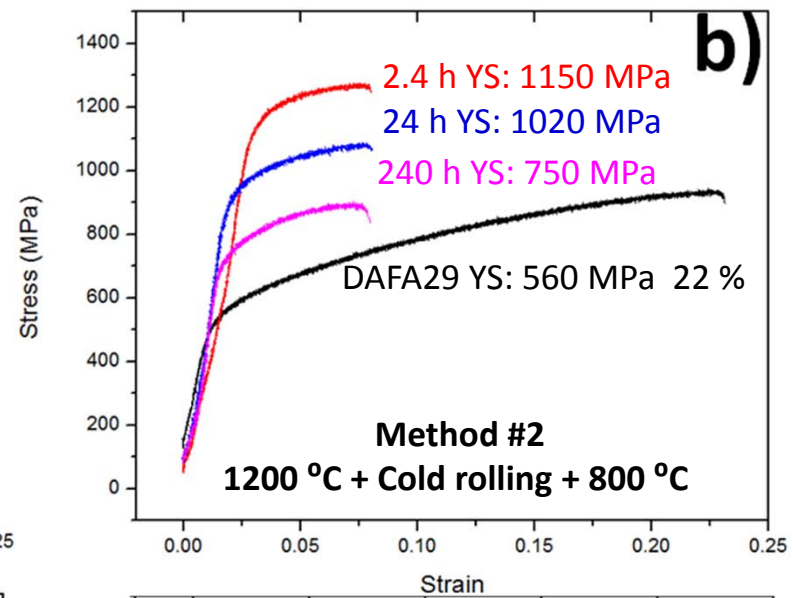
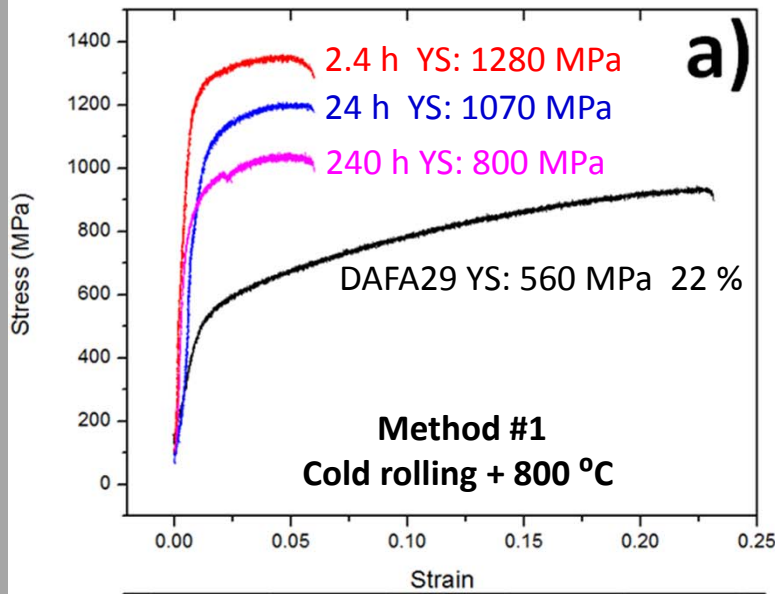
	Fe-f.c.c.	Fe <sub>2</sub> Nb (a)	NiAl	Ni <sub>3</sub> Al
DAFA29	3.611	4.820	2.888	3.604
Method #1 2.4 h	3.599	4.812	2.883	3.590
24 h	3.601	4.853	2.895	3.591
240 h	3.597	4.881	2.900	3.587

**Lattice parameters**

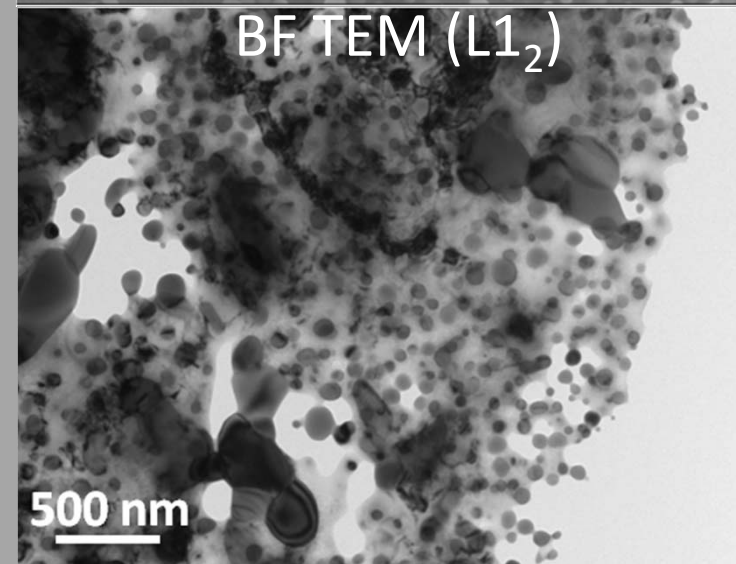
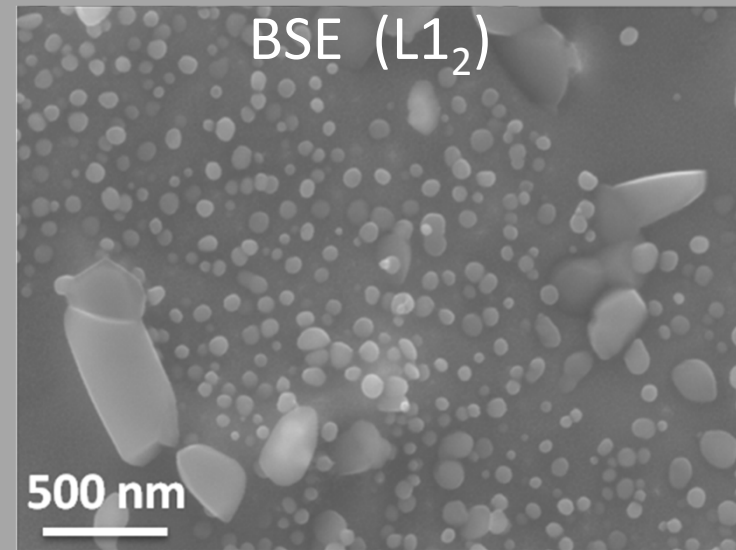
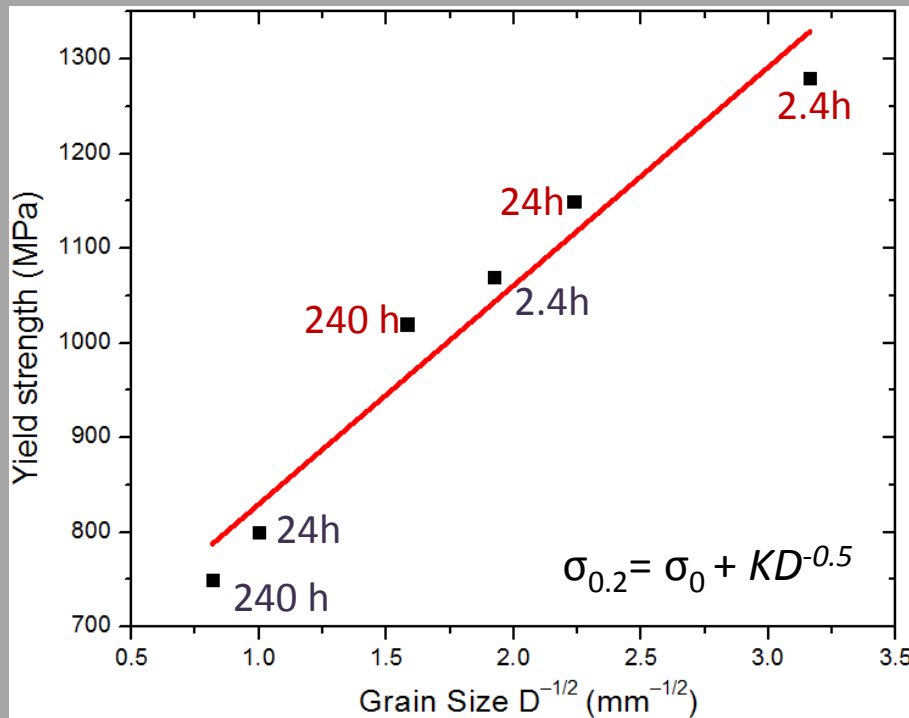
	Fe-f.c.c.	Fe <sub>2</sub> Nb (a)	NiAl	Ni <sub>3</sub> Al
DAFA29	3.611	4.820	2.888	3.604
Method #2 2.4 h	3.601	4.812	2.883	3.591
24 h	3.601	4.817	2.888	3.594
240 h	3.601	4.820	2.890	3.591

Lattice misfit of L1<sub>2</sub> phase with f.c.c. matrix is calculated to be only ~0.28% for both treatments

# Room Temperature Tensile Tests



# Hall-Petch Relationship for TMT DAFA29



Hall-Petch:  $\sigma_{0.2} = \sigma_0 + KD^{-0.5}$ ,

where  $\sigma_0 = 600 \text{ MPa}$ ,  $K = 230 \text{ MPa} \cdot \mu\text{m}^{-0.5}$

$\sigma_0 = \sigma_{\text{ppt}} + \sigma_d + \sigma_{\text{ss}}$



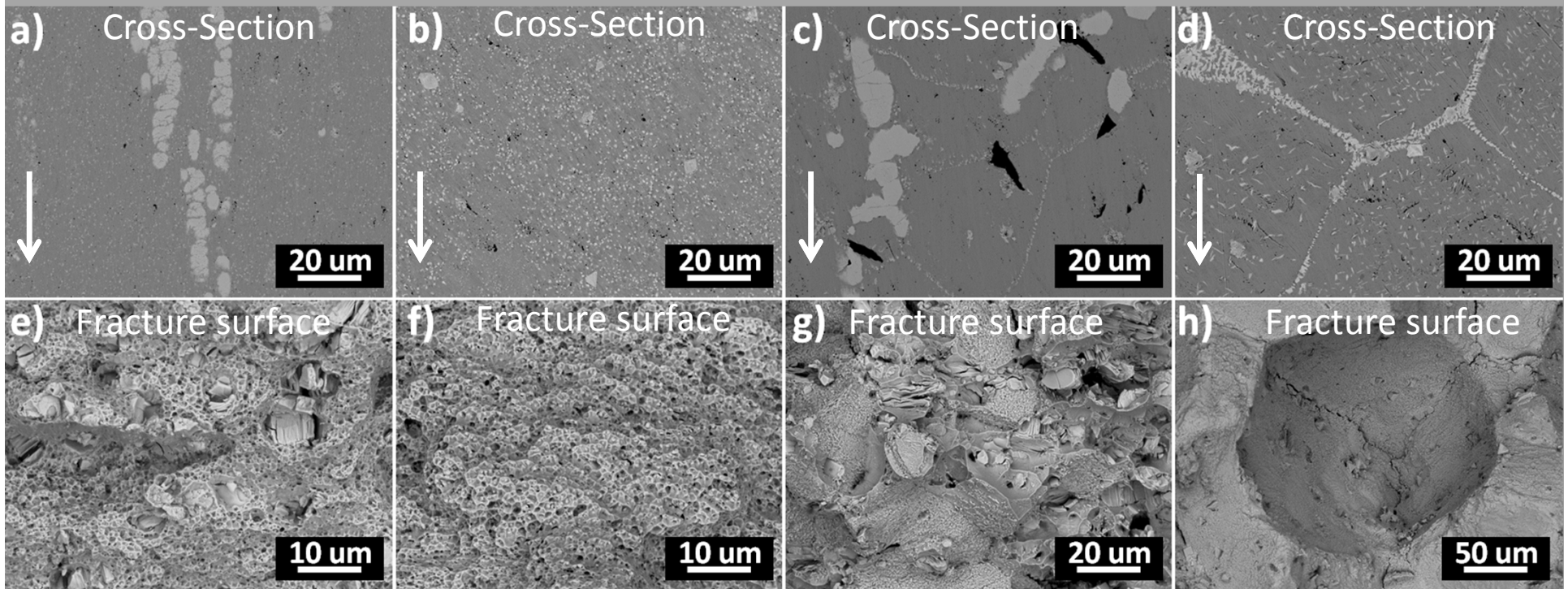
# Cross-Sections and Fracture Surfaces for Samples Treated by TMT Methods

**Method #1**  
Cold rolling + 800 °C

**Method #2**  
1200 °C + Cold rolling + 800 °C

**Method #1 Control**  
800 °C

**Method #2 Control**  
1200 °C + 800 °C



240 h  
YS: 800 MPa  
Elongation: 5.1 %

240 h  
YS: 750 MPa  
Elongation: 6.2 %

240 h  
YS: 760 MPa  
Elongation: 8.0 %

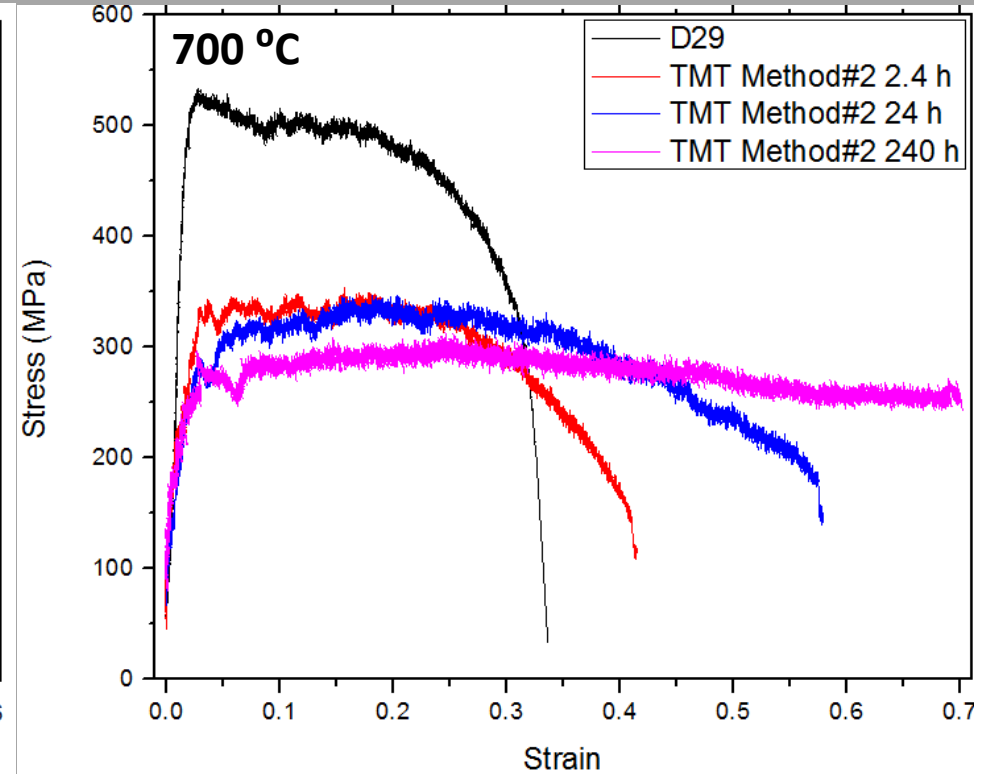
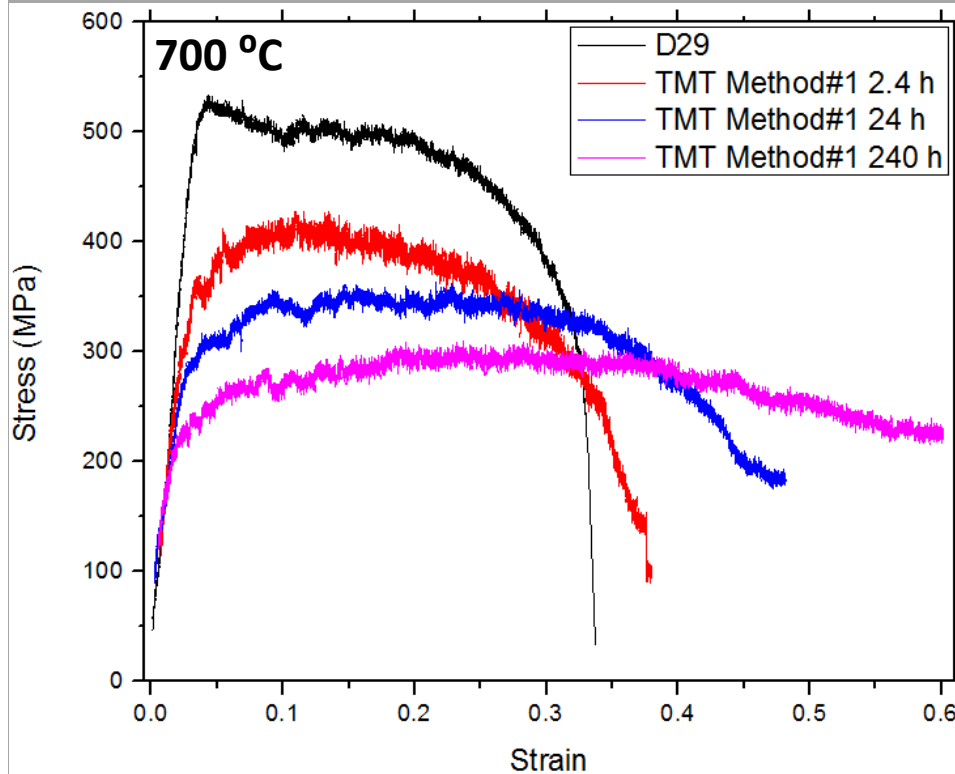
240 h  
YS: 660 MPa  
Elongation: 1.0 %



# Tensile Tests at 700 °C

**Method #1**  
Cold rolling + 800 °C

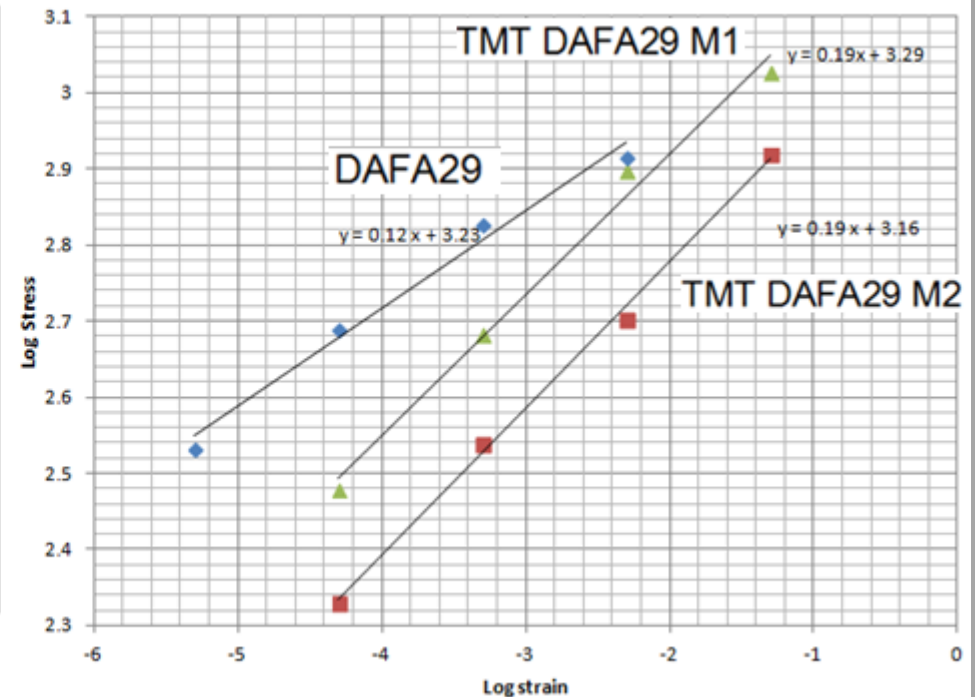
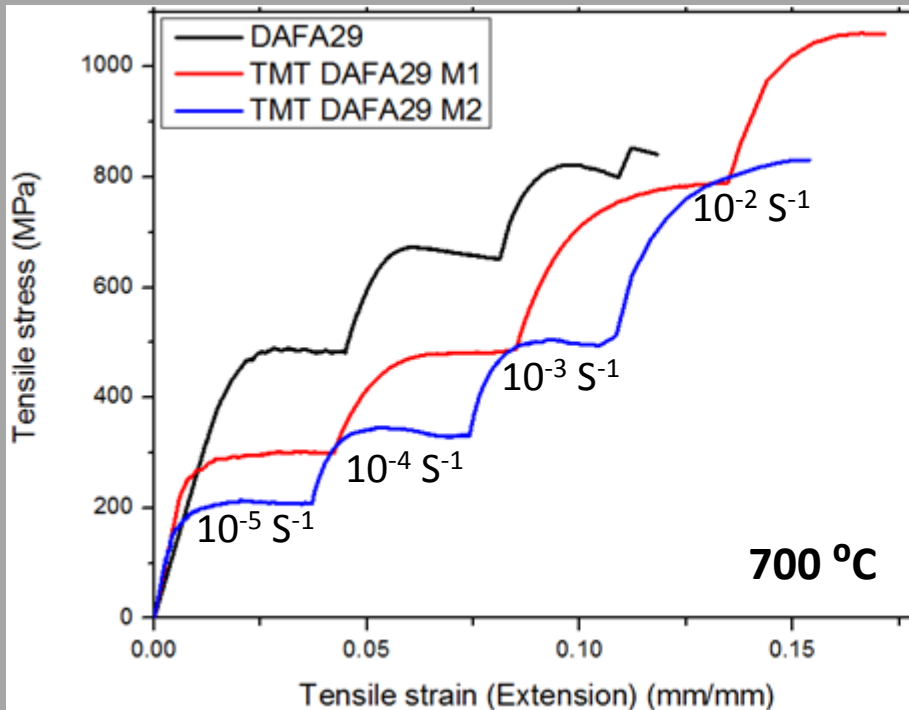
**Method #2**  
1200 °C + Cold rolling + 800 °C



Materials	Yield Strength (MPa)	Elongation (%)
DAFA29	510	31
TMT 2.4h	384	35
TMT 24h	289	46
TMT 240h	218	59

Materials	Yield Strength (MPa)	Elongation (%)
DAFA29	510	31
TMT 2.4h	337	39
TMT 24h	290	55
TMT 240h	262	68

# Strain Rate Jump Tests at 700 °C



$$\sigma_y = C(\dot{\epsilon})^m$$

$$m = \frac{\partial \ln \sigma(\dot{\epsilon})}{\partial \ln(\dot{\epsilon})}$$

As-received DAFA29:  $m = 0.12$

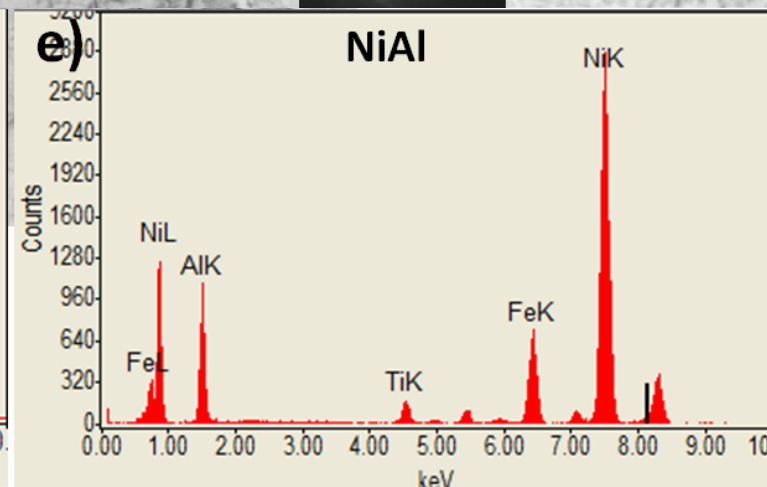
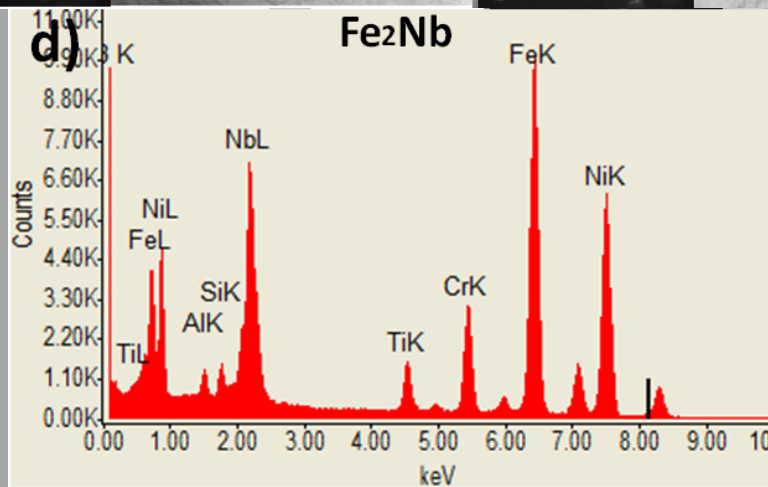
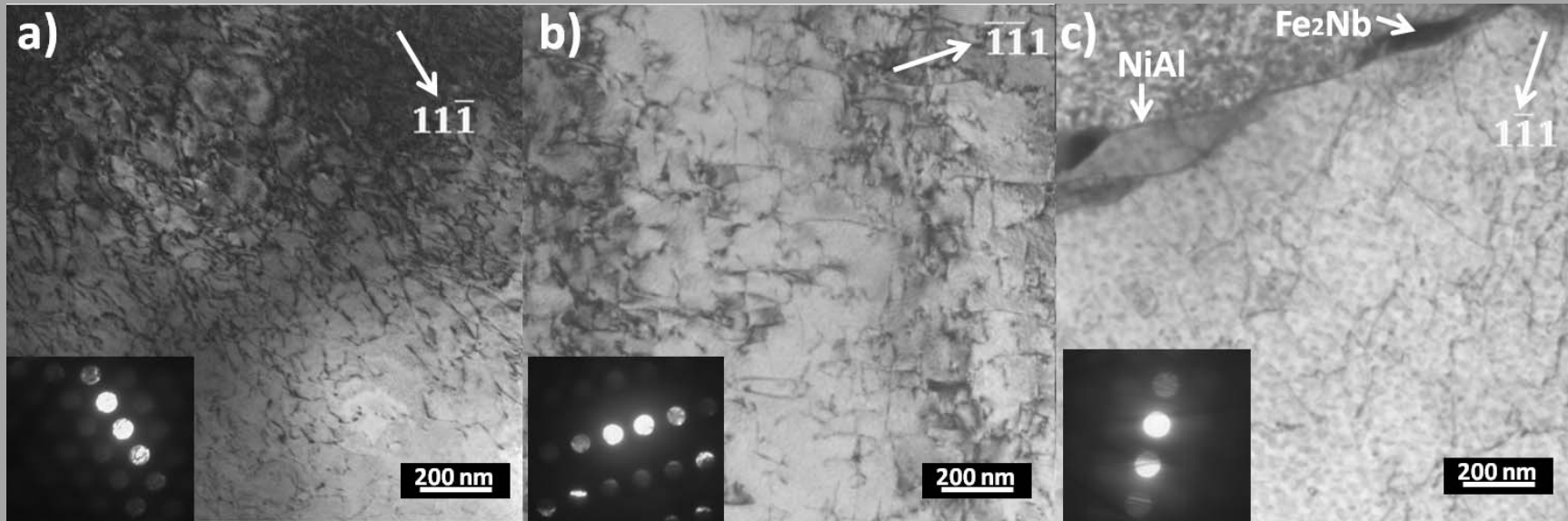
TMT DAFA29:  $m = 0.19$

# Dislocations in DAFA29 Tested in Different Strain Rates at 700 °C

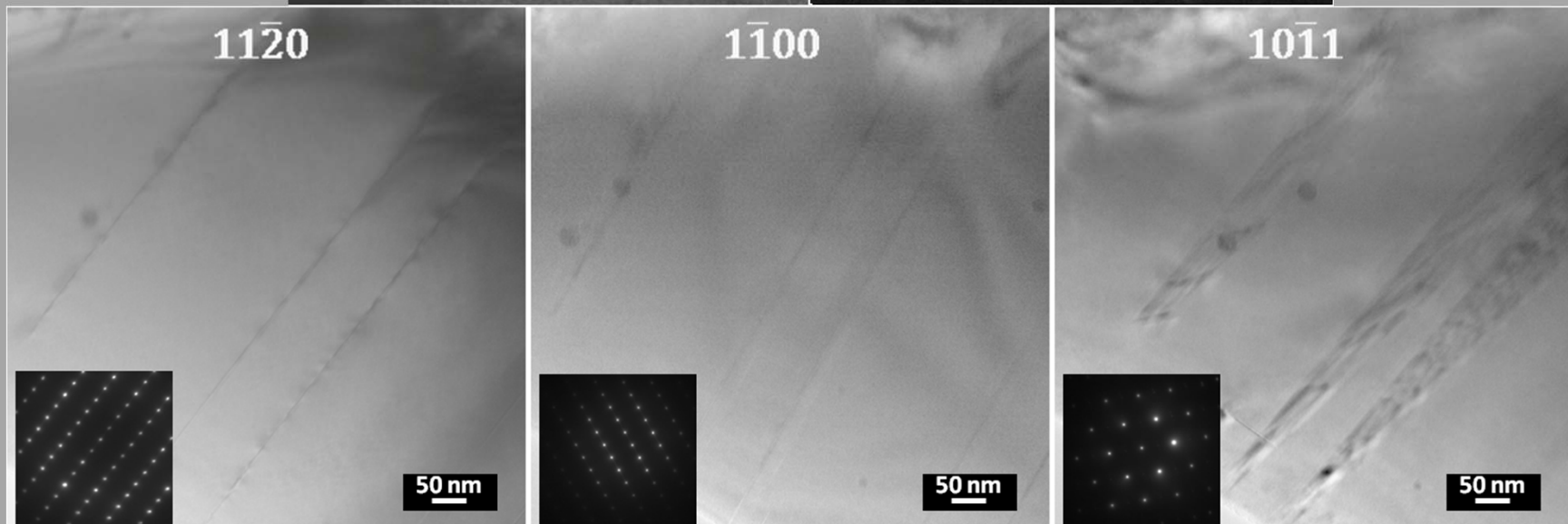
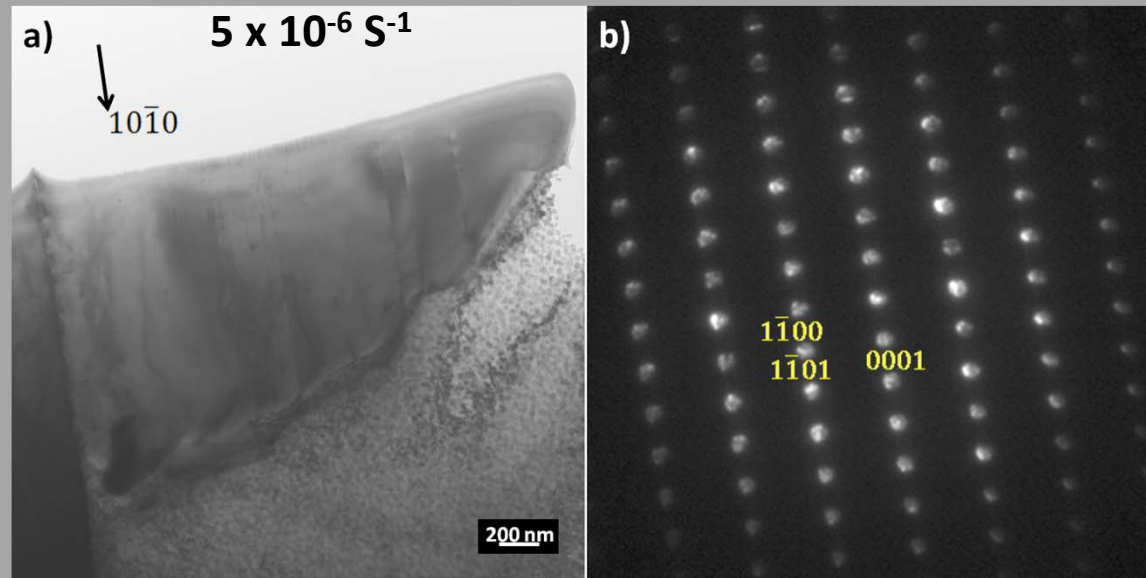
$5 \times 10^{-2} \text{ S}^{-1}$

$5 \times 10^{-4} \text{ S}^{-1}$

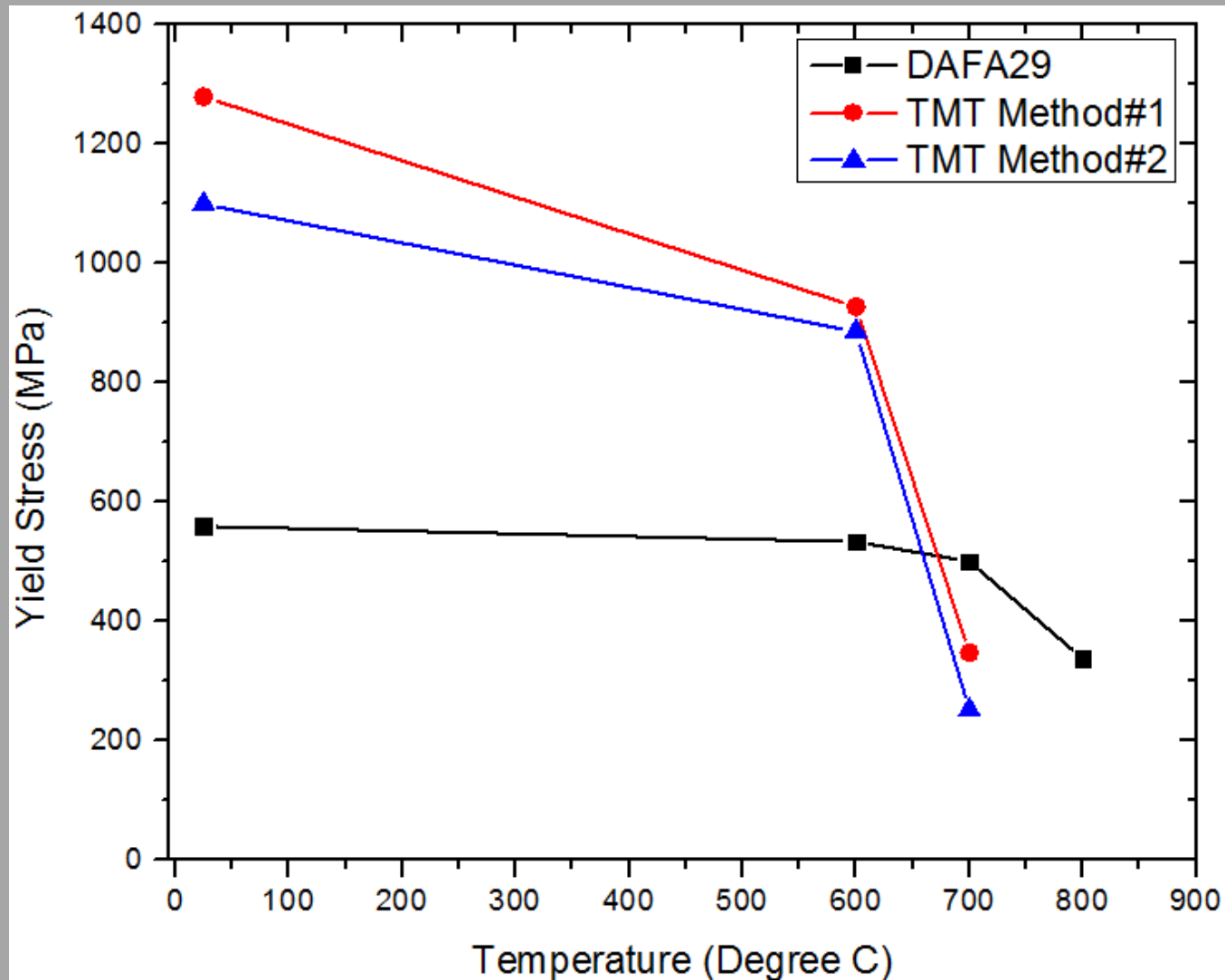
$5 \times 10^{-6} \text{ S}^{-1}$

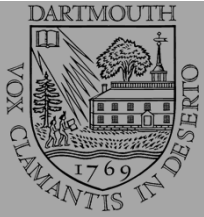


# Stacking Faults on Grain Boundaries Laves Phase Precipitates



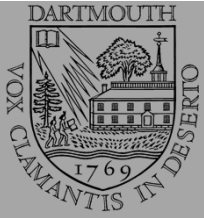
# Stress Versus Temperature for as-received and TMT DAFA29





# Summary

- A solutionizing anneal at 1200°C followed by cold rolling and annealing at 800°C can be used to generate a finer-scale and more uniform distribution of Laves phase precipitates.
- Cold rolling produces a high density of dislocations, which act as nucleation sites for Laves phase, B2, and L1<sub>2</sub> precipitates
- Nanocrystalline steels processed through 90% cold rolling exhibit a dramatic increase in yield strength up to 1280 MPa at RT. The TMT alloys lose stress at 600 -700°C.
- The yield strength of TMT AFA steels exhibits a Hall-Petch relationship with a large value for  $\sigma_0$  that likely arises from precipitate strengthening ( $\sigma_{ppt}$ ).
- The high temperature strength of both as-received and TMT DAFA29 are strain rate dependent at 700 °C

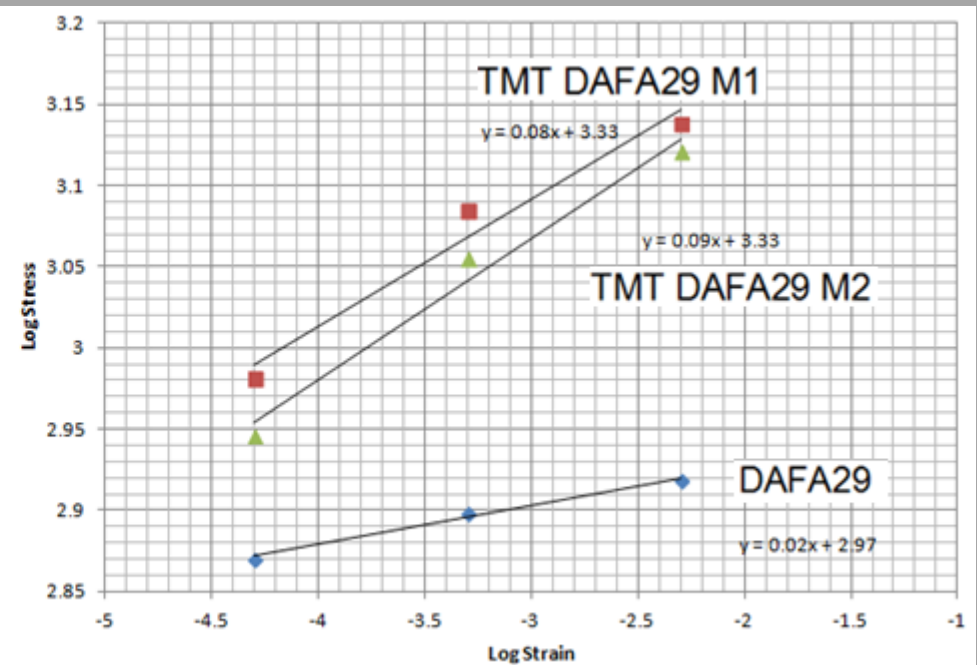
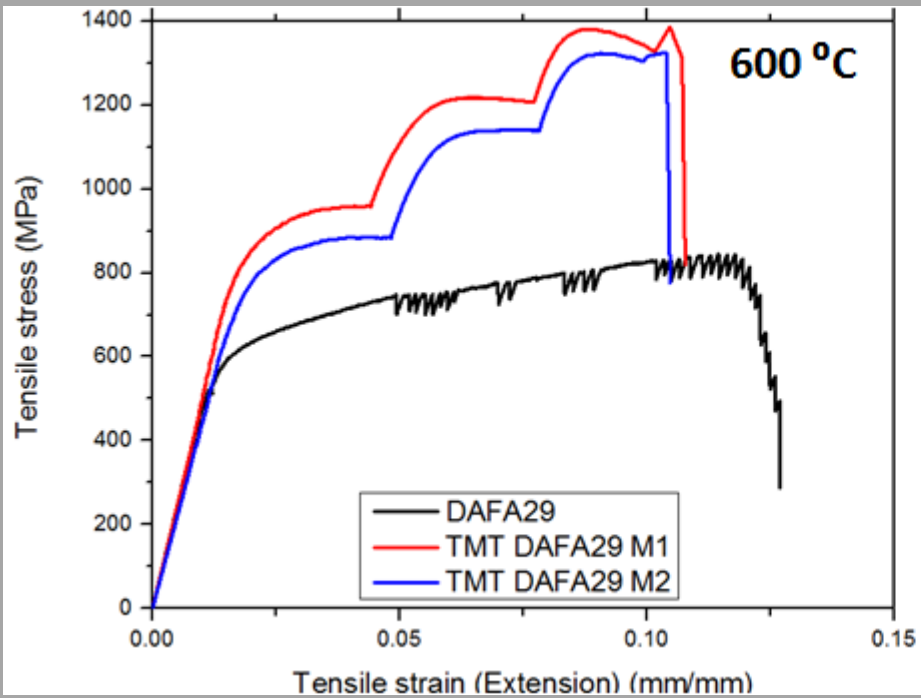


# Future Work

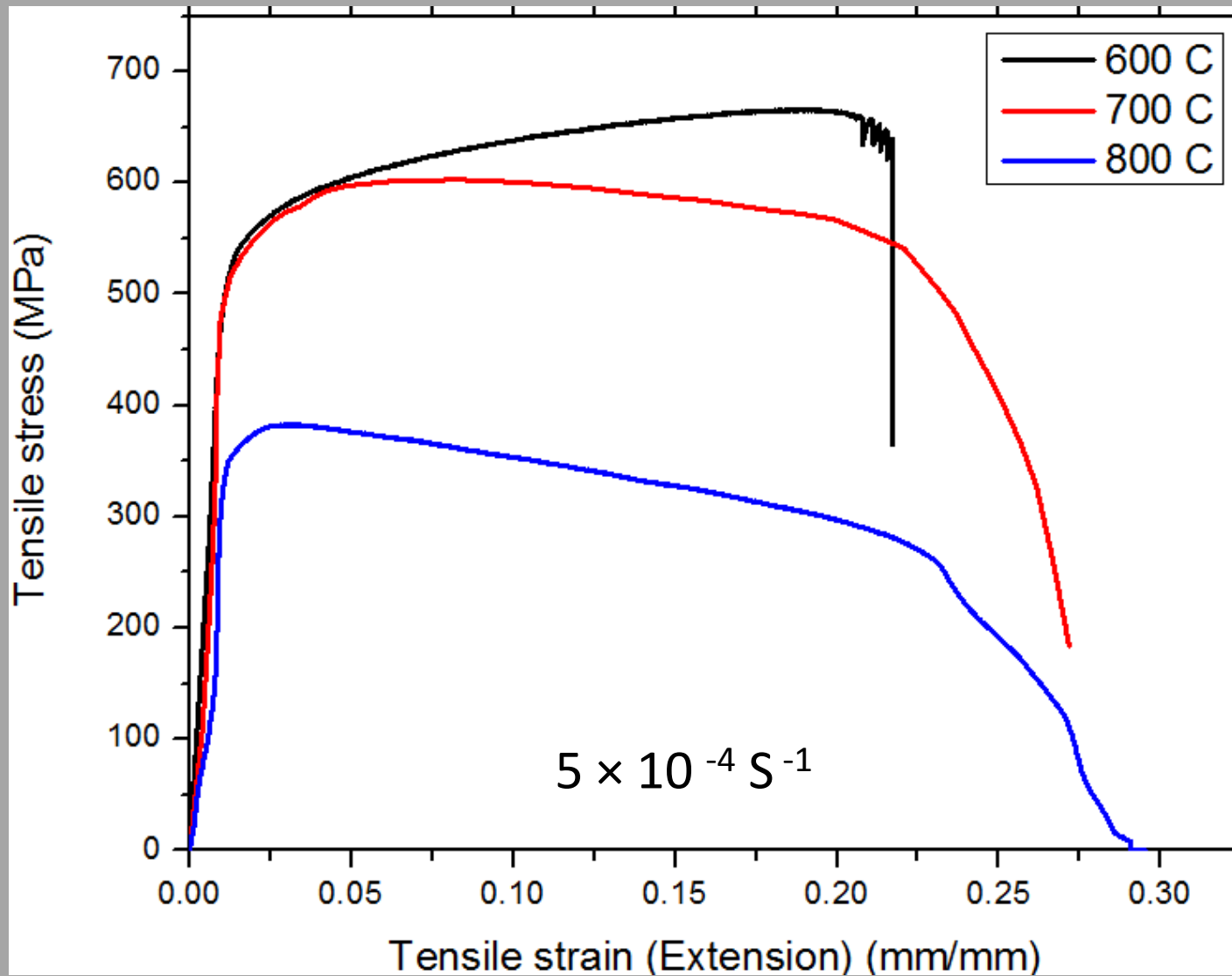
- Continue high temperature tensile tests
  - Tests at different temperatures (600-800 °C)
  - Fracture behavior analysis
- Creep tests of TMT DAFA29
  - Study the creep mechanisms for as-received D29
  - Characterize the deformed creep samples
  - Determine dislocation/precipitate interactions



# Strain Rate Jump Tests at 600 °C



# Tensile Tests of D29 at Different Temperatures



# Stress vs Temperature for DAFA29

