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Experimental and Computational Investigation of High-Entropy Alloys (HEAs) for Elevated-Temperature Applications

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Acknowledgements

We are very grateful to:

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Outline of Presentation

- (1) Potential Significance
- (2) Background and Unique Behavior of HEAs
- (3) Project Objectives
- (4) Proposed Work
- (5) Preliminary Results
- (6) Conclusions
- (7) Future Work
- (8) Published Papers and Presentations

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Potential Significance

- Develop a suitable HEA for steam and gas turbine components operating at temperatures higher than 760° C.
- Increase the thermal efficiency of steam turbines and reduce the costs of fuel and emissions.
- Apply a computer-aided approach for designing new types of alloys applicable for the design and development of other high-temperature materials.



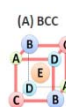
<http://mediaset.gibbs.com/en/using-en/for-steam-turbines-to-help-lead-utility-boost-power-output-and-efficiency-2/>

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Background and Unique Behavior

- Most alloy systems are based on a single principal element to form the matrix.
- Multiple principal elements were expected to yield many intermetallic compounds creating a complex microstructure.
- However, simple face-centered cubic (FCC) and body-centered cubic (BCC) structures are possible and thermodynamically favorable.

(A) BCC



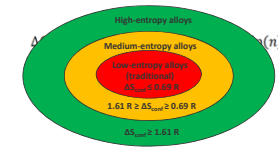
(B) FCC



[1] Yeh JW, Chen SK, Lin SJ, Gan YJ, Chin TS, Shun TY, Trau CH, Chang SY. *Advanced Engineering Materials*, 2004; 6: 299-303.
[2] Cantor B, Chang ITH, Knight P, Vincent AJB. *Mater Sci Eng A Struct Mater*, 2004; 375: 213-218.

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Background and Unique Behavior (Cont'd)



High entropy of mixing and sluggish diffusion yield simple face-centered cubic (FCC) and body-centered cubic (BCC) solid solutions with dispersed nano-phases.

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Background and Unique Behavior (Cont'd)

- HEAs are composed of five or more elements in equimolar ratios.
- A large number of HEAs are potentially possible for various applications. For example, 13 elements could generate 7099 HEAs.

$$C_5^{13} + C_6^{13} + C_7^{13} + C_8^{13} + C_9^{13} + C_{10}^{13} + C_{11}^{13} + C_{12}^{13} + C_{13}^{13} = 7099$$

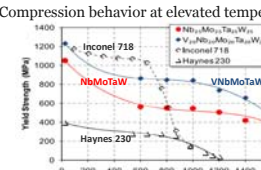
where C_n^m is the number of combinations of m elements taken n times

- Show great high-temperature strengths, corrosion, wear, and fatigue resistance.

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Background and Unique Behavior (Cont'd)

Compression behavior at elevated temperatures



The yield stress of both alloys dropped by 10 - 40% between room temperature and 600 °C, but was relatively insensitive to temperature above 600 °C, comparing favorably with conventional superalloys.

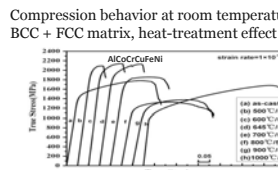
Sadok, O.N., G.B. Wilks, J.R. Scott, and D.R. Miracle. *Intermetallics*, 2011, 19(2): pp. 180-186.
Sadok, O.N., G.B. Wilks, D.R. Miracle, C.F. Chang, P.K. Liaw. *Intermetallics*, 2010, 18: pp. 1758-1760.

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Background and Unique Behavior (Cont'd)

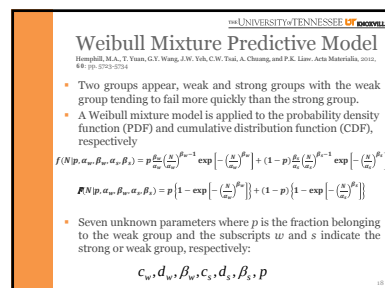
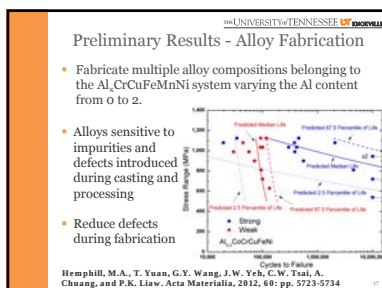
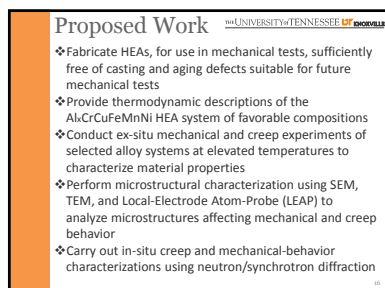
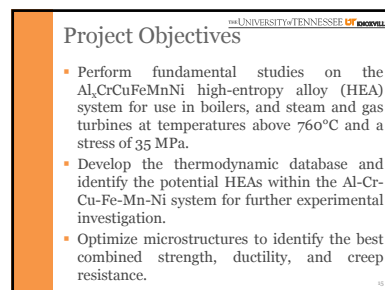
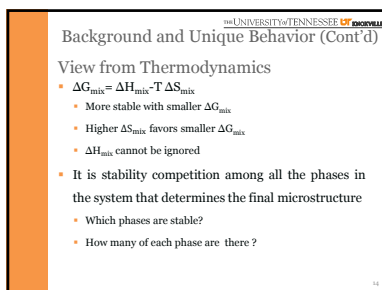
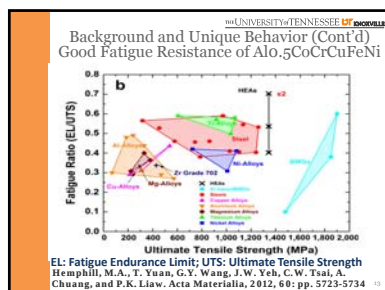
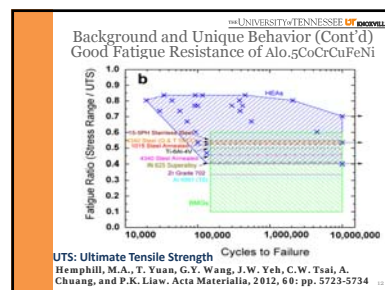
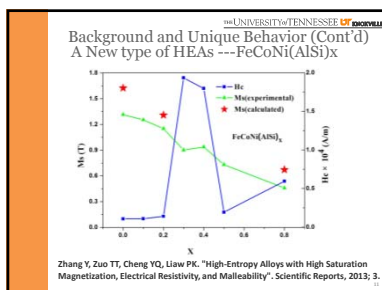
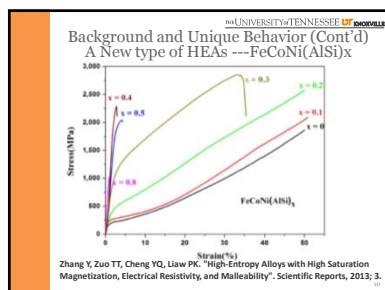
Compression behavior at room temperature

BCC + FCC matrix, heat-treatment effect



Compressive true stress-strain curves of AlCoCrCuFeNi after heat treatment for 5 h at different temperatures. The mechanical properties of this alloy have a strong correlation with the aging temperature.

Wen, L.H., H.C. Kou, J.S. Li, H. Chang, X.X. Xue, and L. Zhou. *Intermetallics*, 2009, 17(4): pp. 266-269.



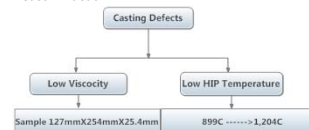
Preliminary Results - Alloy Fabrication (Cont'd)

- ❖ Al_{0.8}CrCuFeMnNi
- ❖ Φ 25.4 mm (1 inch) $\times$ 762 mm (30 inch)
- ❖ Vacuum Cast $\rightarrow$ Hot Isostatic Pressing
899 °C, 103MPa, 2h.
- ❖ Differential Thermal Analysis (DTA)
100 °C---1,600 °C
 - No thermal transitions: 100 to around 980 °C.
 - Three transitions occurred at 980 °C, 1,173 °C, and 1,358 °C.
 - Discernible liquidus is not observed up to 1,600 °C.

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Preliminary Results - Alloy Fabrication (Cont'd)

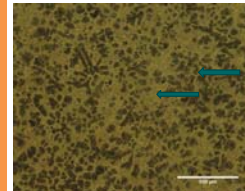
- ❖ Cast in vacuum



- ❖ Arc-Melting (Φ 10 mm $\times$ 50 mm)
- ❖ Vacuum Induction Melting (VIM) + Hot Working/Cold Work and Full Annealing Process

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Preliminary Results - Microstructures (Cont'd)

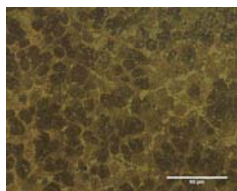


Dendrite
Interdendrite

OM images of Al_{0.8}CrCuFeMnNi

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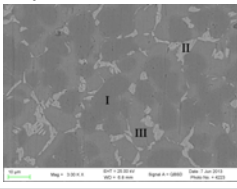
Preliminary Results - Microstructures (Cont'd)

OM images of Al_{0.8}CrCuFeMnNi

white-phase layers appear along the grain boundary

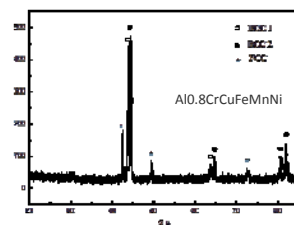
22

Preliminary Results - Microstructures - SEM



Atomic %	Al	Cr	Mn	Fe	Ni	Cu
I	8.12	31.88	15.47	29.04	9.05	6.44
II	6.97	1.65	19.12	3.11	9.9	59.25
III	20.97	4.86	19.31	6.59	28.13	20.14

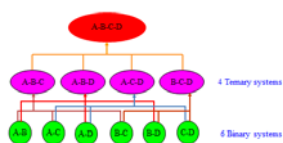
Preliminary Results - XRD Patterns



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Preliminary Results - Thermodynamic Modeling

Construct A Database



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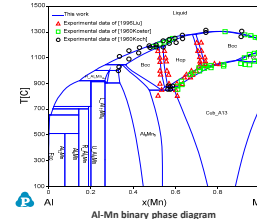
Preliminary Results - Thermodynamic Modeling Constituent binary and ternary systems of the Al-Cr-Cu-Fe-Mn-Ni system

Binary Systems					
Al-Cr	Al-Cu	Al-Fe	Al-Mn	Al-Ni	
Cr-Cu	Cr-Fe	Cr-Mn	Cr-Ni	Cu-Fe	
Cu-Mn	Cu-Ni	Fe-Mn	Fe-Ni	Mn-Ni	
Ternary Systems					
Al-Cr-Cu	Al-Cr-Fe	Al-Cr-Mn	Al-Cr-Ni	Al-Cu-Fe	
Al-Cu-Mn	Al-Cu-Ni	Al-Fe-Mn	Al-Fe-Ni	Al-Mn-Ni	
Cr-Cu-Fe	Cr-Cu-Mn	Cr-Cu-Ni	Cr-Fe-Mn	Cr-Fe-Ni	
Cr-Mn-Ni	Cu-Fe-Mn	Cu-Fe-Ni	Cu-Mn-Ni	Fe-Mn-Ni	

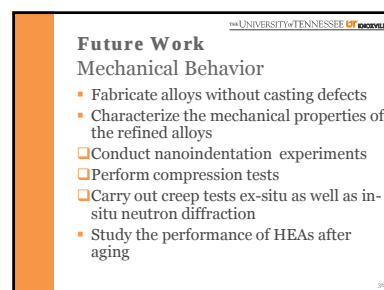
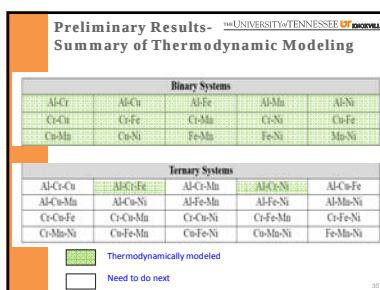
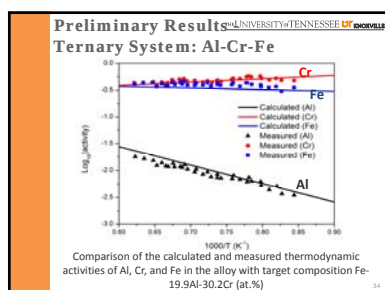
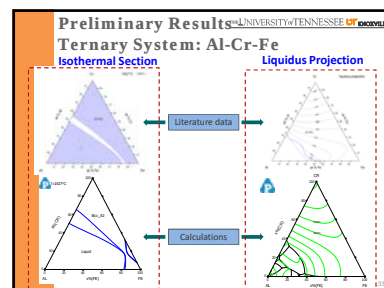
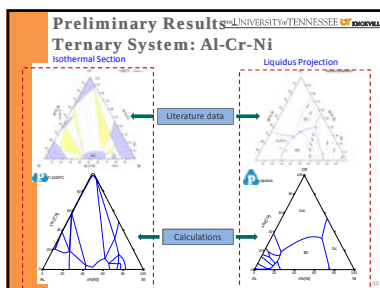
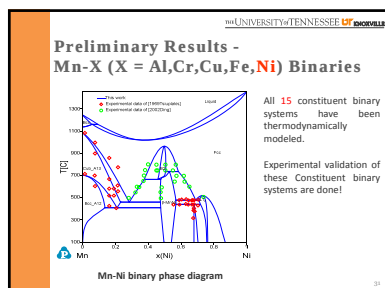
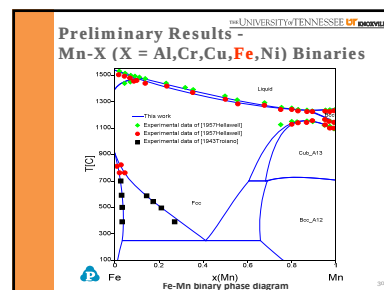
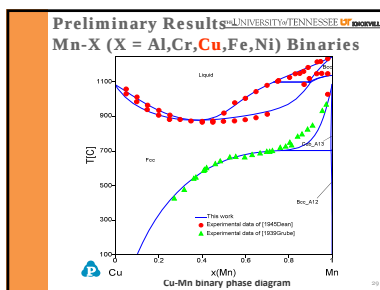
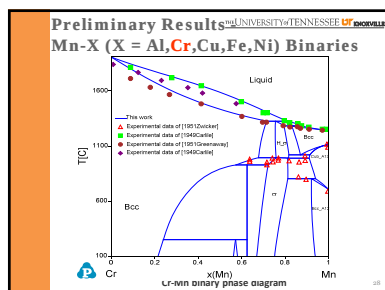
- There are **15** constituent binary systems.
- There are **20** constituent ternary systems.

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Preliminary Results - Mn-X (X = Al, Cr, Cu, Fe, Ni) Binaries

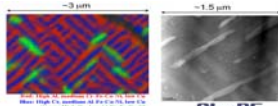


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Future Work (Cont'd)

- Conventional and high-resolution transmission electron microscope (TEM) to determine the microstructure features.



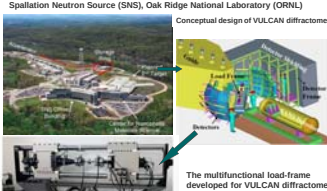
A TEM image reveals copper-rich nano-particles in $Al_{0.8}CoCrCuFeNi$.

- Shape and average size of the main phases and deformation mechanisms, such as the interaction between dislocations and phase boundaries during mechanical-property tests

Future Work

- In-situ neutron-diffraction experiments

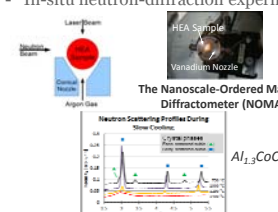
Spallation Neutron Source (SNS), Oak Ridge National Laboratory (ORNL)



The multifunctional load-frame developed for VULCAN diffractometer

Future Work

- In-situ neutron-diffraction experiments



The Nanoscale-Ordered Materials Diffractometer (NOMAD)

$Al_{1.3}CoCrCuFeNi$

Future Work

Thermodynamic Modeling

- Continue to work on the thermodynamic modeling of the other constituent ternary systems.
- Provide experimental validation of the developed thermodynamic database based on the experimental data from both literature and our team members.
- Provide thermodynamic predictions to identify potential HEAs within the Al-Cr-Cu-Fe-Mn-Ni system.

Published Papers and Presentations

Papers:

- [1] Zhang Y, Zuo TT, Cheng YQ, Liaw PK. "High-Entropy Alloys with High Saturation Magnetization, Electrical Resistivity, and Malleability". *Scientific Reports*, 2013; 3.
- [2] Guo W, Dmowski W, Noh JY, Rack P, Liaw PK, Egami T. "Local Atomic Structure of a High-Entropy Alloy: An X-Ray and Neutron Scattering Study". *Metallurgical and Materials Transactions a-Physical Metallurgy and Materials Science*, 2013; 44A: 1994.
- [3] Hemphill MA, Yuan T, Wang G, Yeh J, Tsai C, Chuang A, Liaw PK. "Fatigue Behavior of $Al_{0.5}CoCrCuFeNi$ High-Entropy Alloys". *Acta Materialia*, 2012.

Published papers and presentations

- [4] Yang X, Zhang Y, Liaw PK. "Microstructure and Compressive Properties of NbTiV-TaAlx High Entropy Alloys". *Procedia Engineering*, 2012; 36: 292.
- [5] Zhang Y, Yang X, Liaw PK. "Alloy Design and Properties Optimization of High-Entropy Alloys". *JOM*, 2012; 64: 830.
- [6] Zhang Y, Zuo T, Liao W, Liaw PK. "Processing and Properties of High-Entropy Alloys and Micro-and Nano-Wires". *ECS Transactions*, 2012; 41: 49.
- [7] Zhang C, Zhang F, Chen SL, Cao WS. "Computational Thermodynamics Aided High-Entropy Alloy Design". *JOM*, 2012; 64(7): 839-845.

Published Papers and Presentations

Presentation:

- 2013 TMS Meeting, San Antonio, TX, USA, March 3-9, 2013
- Automatic Fabrication of High-Entropy Alloys and Their Properties. Y. Yokoyama, X. Xie, J. Antonaglia, M. Hemphill, T. Zhi, T. Yuan, G. Wang, C. Tsai, J. Yeh, A. Chuang, K. Dahmen, P. K. Liaw (invited)
- Non-Equilibrium and Equilibrium Phases in $AlCoCrFeNi$ High-Entropy Alloys. Z. Tang, O. Senkov, C. Parish, L. Santodonato, D. Miracle, G. Wang, C. Zhang, F. Zhang, P. K. Liaw
- Ordering Behavior in the $Al(x)CoCrCuFeNi$ High-Entropy Alloys. L. Santodonato, Y. Zhang, M. Gao, C. Parish, M. Feygenson, Z. Tang, J. Neugebauer, R. Weber, P. K. Liaw
- Computational Modeling of High-Entropy Alloys. M. Gao, D. Tafen, J. Hawk, Y. Wang, M. Widom, L. Santodonato, P. K. Liaw (invited)
- Statistical Fatigue - Life Modeling for High-Entropy Alloys. T. Yuan, M. Hemphill, Z. Tang, G. Wang, A. Chuang, C. Tsai, J. Yeh, P. K. Liaw (invited).

Published Papers and Presentations

- A Combinatorial Approach to the Investigation of Metal Systems that Form Both High Entropy Alloys and Bulk Metallic Glasses. B. Welk, P. K. Liaw, M. Gibson, H. Fraser
- Minor Phase and Defect Effects on Fatigue Behavior of Wrought $Al_{0.5}CoCrCuFeNi$ High-Entropy Alloys. Z. Tang, M. Hemphill, T. Yuan, G. Wang, J. Yeh, C. Tsai, P. K. Liaw
- Phase Separation and Intermetallic Formation in "High-Entropy" Alloys. C. Parish, M. Miller, L. Santodonato, Z. Tang, P. K. Liaw
- Computational Thermodynamics Aided High-Entropy Alloy Design. C. Zhang, F. Zhang, S. Chen, W. Cao, J. Zhu, Z. Tang, P. K. Liaw
- The 9th International Conference on Bulk Metallic Glass (BMG-IX) 2012, Xiamen, China
 - Zhang C, Zhang F, Chen SL, Cao WS, Tang Z and Liaw PK. "Computational Thermodynamics Aided High-Entropy Alloy Design".

Conclusions

- We have fabricated bulk $Al_{0.8}CrCuFeMnNi$ samples, even though there are some cast defects existing.
- The $Al_{0.8}CrCuFeMnNi$ alloy has typical dendritic grains. White-phase layers appear along the grain boundary.
- The $Al_{0.8}CrCuFeMnNi$ alloy has three phases, disordered BCC1, BCC2, and FCC phases.
- All the constituent binary systems have been thermodynamically modeled.
- Al-Cr-Fe and Al-Cr-Ni ternary phase diagrams are in good agreement with the literature data.
- Future work has been identified.

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BIG ORANGE
UT BIG IDEAS

Thank You!

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Background and Unique Behavior

Selection Criteria in Literature

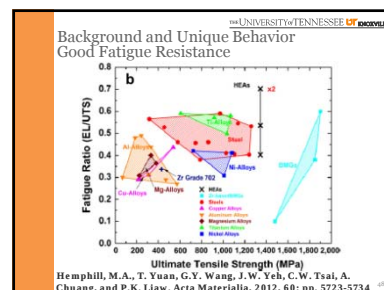
- Higher entropy of mixing^[1] ($\Delta S_{mix} > 1.61R$)
- Smaller enthalpy of mixing^[2] ($-15 < H_{mix} < 5 \text{ kJ/mol}$)
- Less atomic size difference ($1 \leq \delta \leq 6$)^[1]

$$\delta = \sqrt{\sum_{i=1}^n c_i (1 - \frac{r_i}{r})^2}, \bar{r} = \sum_{i=1}^n c_i r_i$$

- Valence electron concentration^[1] ($\frac{VEC}{\text{atom}} < 8$ (fcc))

**Semi-empirical, system dependent, limited predictability*

[1] Y. Zhang and Y.J. Zhou, Materials Science forum 2007, 561-565: 1337
[2] S. Guo, C. Ng, J. Lu and C.T. Liu, JAP 2011, 109: 103505



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Background and Unique Behavior

Thermodynamic description

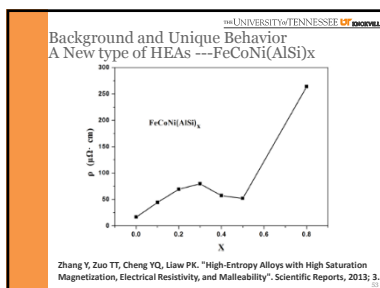
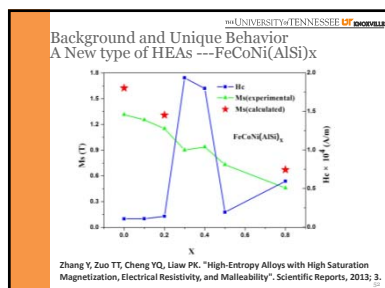
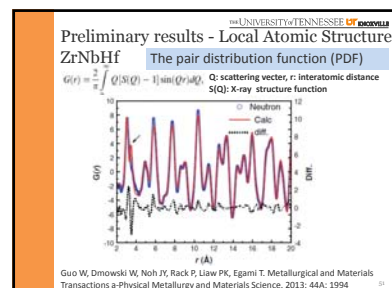
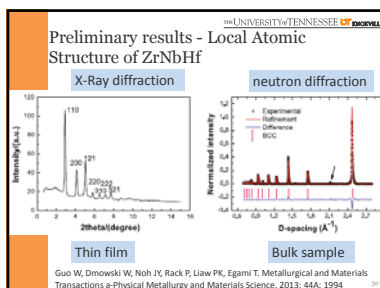
- When we mix n elements together to form a disordered phase, the entropy of mixing is:

$$\Delta S_{mix} = -R \sum_{i=1}^n x_i \ln x_i$$

where x is the mole fraction of element i.

- When equimolar of every element is used, the entropy of mixing: ΔS_{mix} reaches a maximum

Design of HEA: Mix equimolar multi-components to form disordered phase



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Approach

Alloy System

System

Phase I Phase II Phase III Phase ..

Component: A, B, C, D, ...

$$G^E = \sum x_i \cdot G_i^E + \sum w_{i,j}^E \cdot G_{i,j}^E + \sum w_{i,j,k}^E \cdot G_{i,j,k}^E$$

