

# Synchrotron X-Ray Studies of SOFC Cathodes

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#### Overview

- Motivation and Background
- Current Results
  - Segregation
  - Lattice parameter vs. electrochemical state
- Summary

# X-Ray Characterization

#### Overview of Synchrotron X-Ray Program

Bulk structure and properties (e.g. thermal expansion)

Literature

Interface structure at operating temperatures in typical atmospheres

**Progress** 

Chemical state of atoms in cathodes under operating conditions

**Progress** 

Dynamic response of cathodes under electrochemical loading

**Latest Results** 

High performance SOFC Cathodes



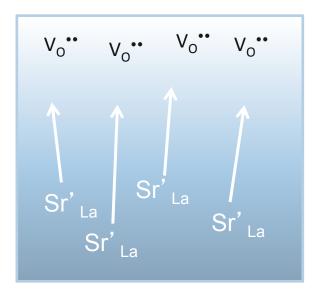
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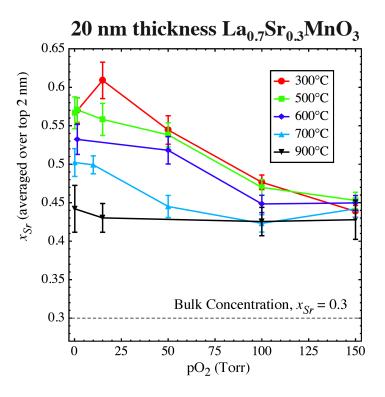


# LSM on DyScO<sub>3</sub>

- Observe that strontium segregation depends on both T and pO<sub>2</sub>
- Charged vacancies are often not considered in surface segregation studies.
  - The concentration of these defects depends strongly on temperature and pO<sub>2</sub>.
- A gradient of V<sub>o</sub>•• near the surface could drive Sr segregation.



Applied Physics Letters 93, 151904 (2008)



#### Change in Sr concentration from bulk

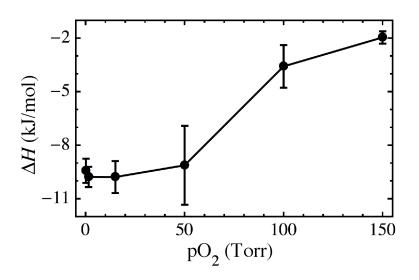
	Operating T (700-1000 C)	Low T (300 C)
Low pO <sub>2</sub> (mTorr)	+35%	+50%
Operating pO <sub>2</sub> (atmospheric)	+21%	+25%

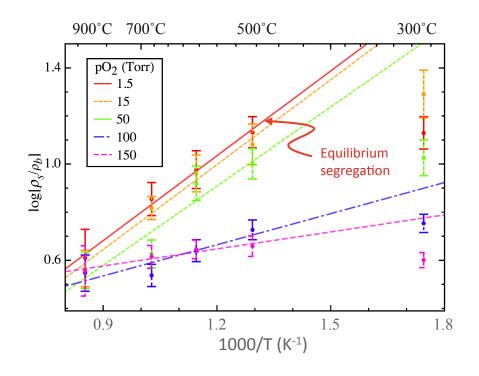
#### **Kinetic limitations**

Equilibrium segregation:

$$\frac{x_{Sr}^s}{x_{La}^s} = \frac{x_{Sr}^b}{x_{La}^b} e^{-\Delta H_{seg}/kT}$$

■ Linearity at high T (above 500°C) indicates equilibrium segregation.



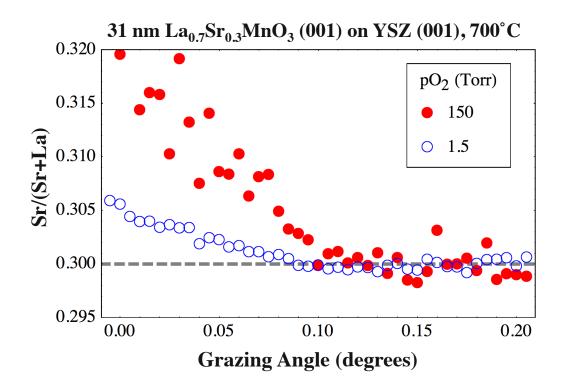


- → Previous room-temperature measurements likely depended on thermal history
- → Further details: T.T. Fister et al. APL, 93, 151904 (2008).



# La<sub>0.7</sub>Sr<sub>0.3</sub>MnO<sub>3</sub> on YSZ

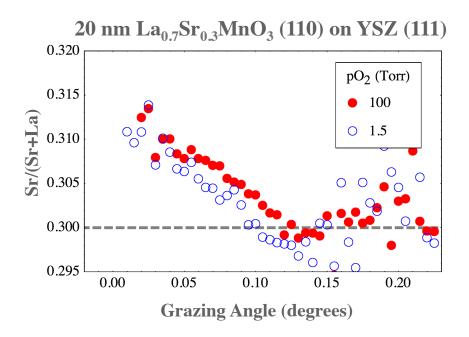
- Reduced segregation compared with LSM/DSO
- Grain boundary segregation may limit surface concentration
- pO<sub>2</sub>-dependence is opposite

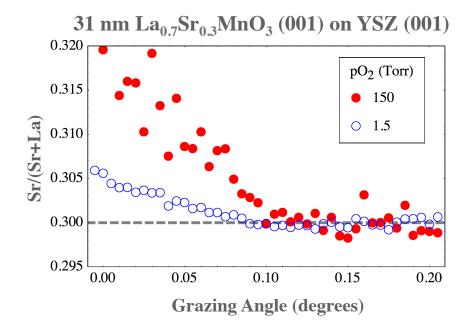


Do more oxygen vacancies in YSZ increase Sr segregation at the YSZ interface?



# La<sub>0.7</sub>Sr<sub>0.3</sub>MnO<sub>3</sub> on YSZ: Orientation Dependence



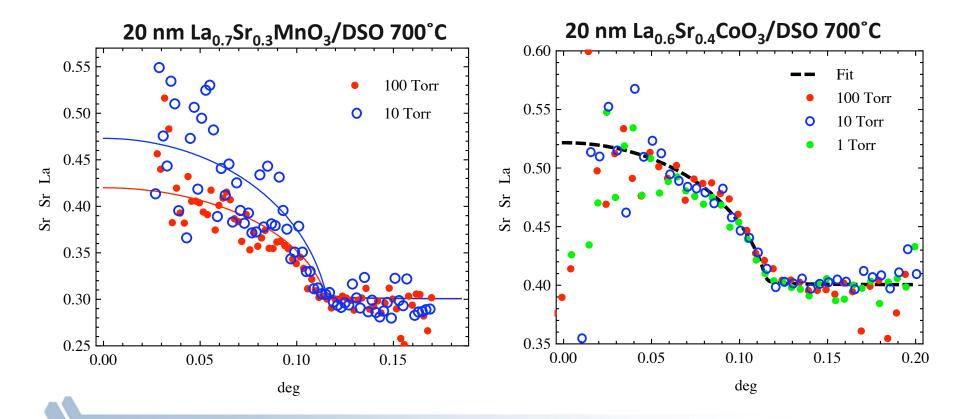


Segregation is observed for both orientations but is not significantly stronger for (110) surface.



### LSC Behaves Differently Than LSM

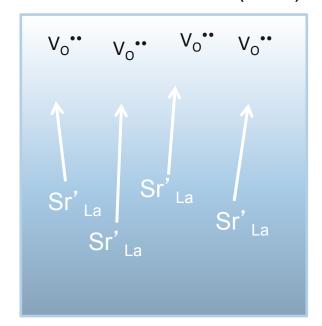
- LSM: surface oxygen vacancies
  - pO<sub>2</sub>-dependent strontium surface segregation
- $La_{0.6}Sr_{0.4}CoO_3$  (LSC) &  $La_{0.6}Sr_{0.4}Co_{0.2}Fe_{0.8}O_3$  (LSCF) : bulk oxygen vacancies
  - Surface strontium enrichment, but no pO<sub>2</sub> dependence



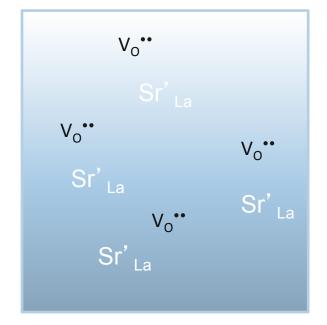
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#### Electronic Conductor (LSM)



#### Mixed Conductor (LSC, LSCF)





#### **Summary of Strontium Segregation**

- For nearly all samples, strontium surface segregation is observed.
- The strontium segregation:
  - → Is approximately independent of strain state (i.e., substrate) and film-thickness
  - → Depends on pO₂ for LSM but not for LSC and LSCF
    - Behavior may dependent on mobility of oxygen vacancies
  - → Depends on temperature and crystal orientation.



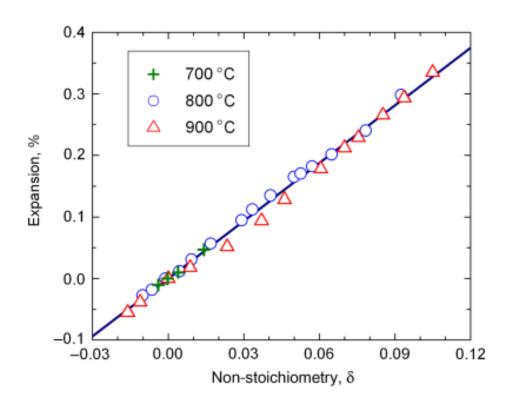
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#### Effect of oxygen stoichiometry on lattice parameter

- Volume expansion of LSCF lattice is linearly proportional to deviation from ideal stoichiometry
- Literature curve is for bulk samples expanding in all three dimensions (have to account for one-dimensional expansion of our constrained films)
- Estimate that 1 V cathodic potential at 600° and  $pO_2 = 150$  Torr gives rise to approximately  $\Delta d = 0.05$  (initial d was not determined)
  - corresponds to ~1 new oxygen vacancy per 20 unit cells

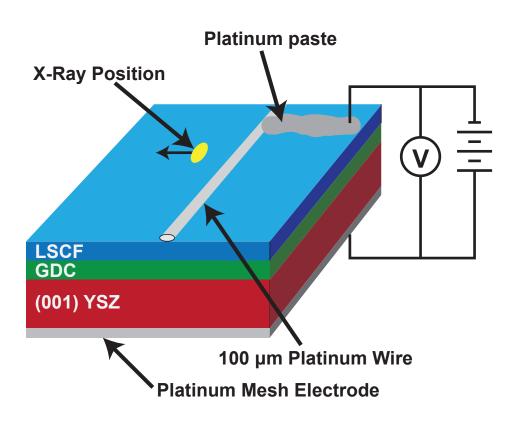


plot from S.R. Bishop, K.L. Duncan, E.D. Wachsman, J. Am. Ceram. Soc. 93 (2010) 4115-4121.



#### **Experiment**

- ~20 nm thickness LSCF and Gd<sub>2</sub>O<sub>3</sub>-doped CeO<sub>2</sub> (GDC) layers by PLD on (001) YSZ; GDC prevents reactions between LSCF and YSZ
- Examined effects of applied DC potential, pO<sub>2</sub>, and T
- Monitored changes in both current (conduction) and outof-plane lattice parameter (of all three materials)
- Investigating ionic component of LSCF conductivity (YSZ blocks electronic component)

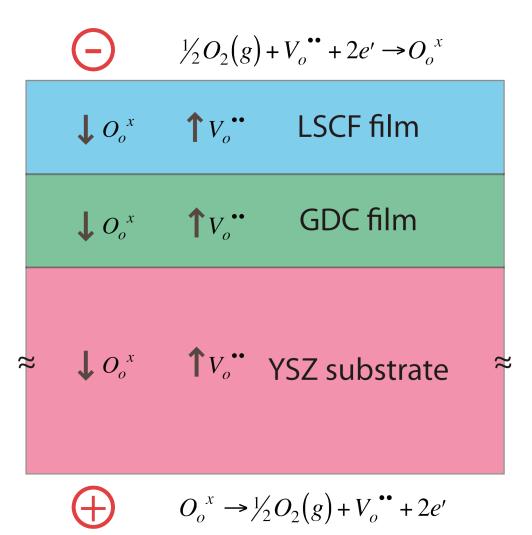


~20 µm wide incident X-ray beam



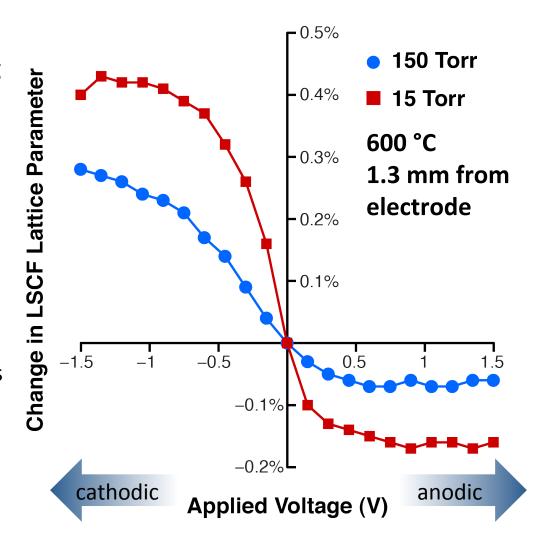
#### **Expected behavior**

- Applying a cathodic (anodic) potential drives oxygen into (out of) the LSCF film at the LSCF/gas interface
- Also drives oxygen into (out of) the GDC film at the LSCF/GDC interface
- If barriers to oxygen vacancy transport across these two interfaces are equal, expect no change in lattice parameter or conduction when field is applied
- If changes in lattice parameter and conduction <u>are</u> observed, we can determine which interface is rate limiting



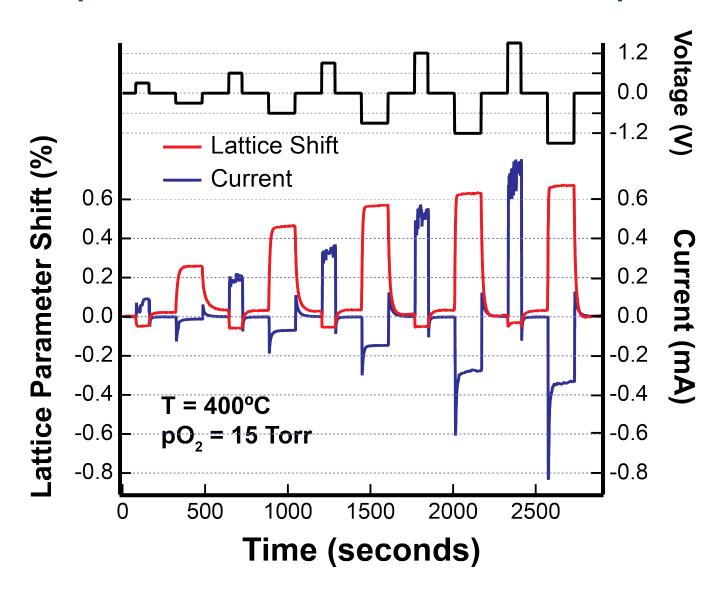
# LSCF lattice parameter shift vs applied potential

- Oxygen transport across the LSCF/gas interface is rate-limiting under both anodic and cathodic conditions
- lacktriangle Cathodic potentials result in larger  $\Delta d$  than anodic potentials
  - larger barrier to oxygen reduction at the LSCF/GDC interface under cathodic conditions than to reverse reaction under anodic conditions
- Stoichiometry changes increase with decreasing pO<sub>2</sub>
  - O<sub>2</sub> reduction barrier increases with decreasing pO<sub>2</sub>



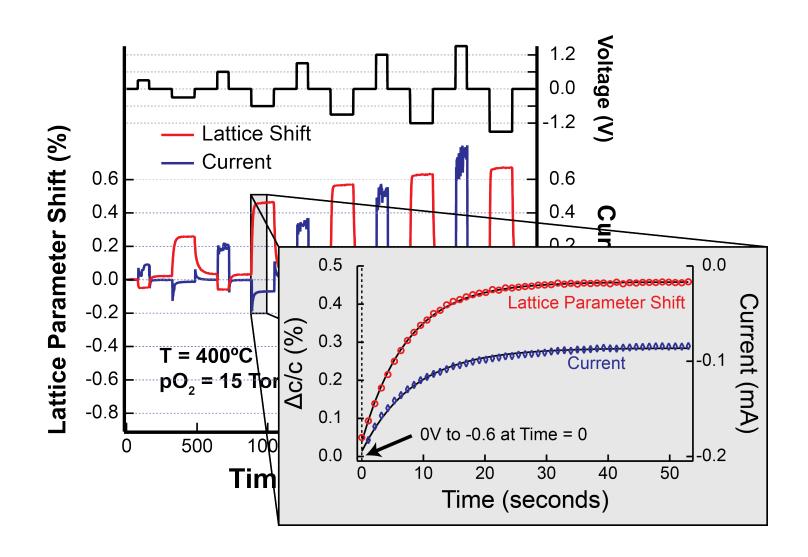


#### Time Dependence of Electrochemical Response



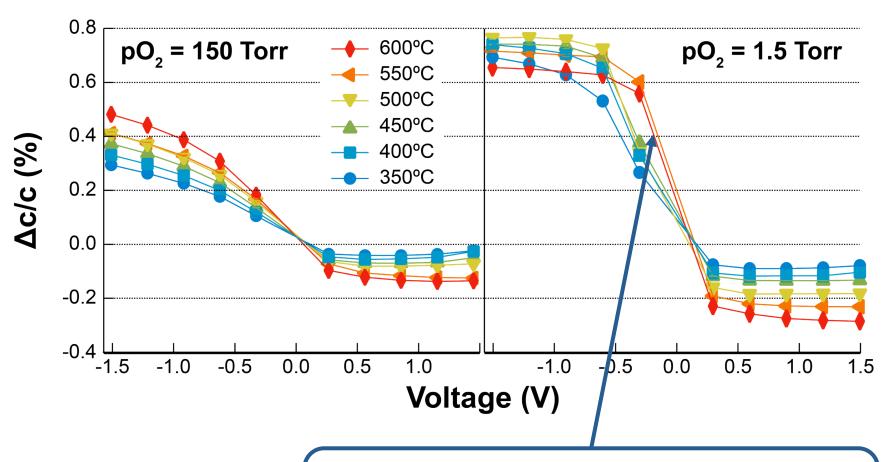


#### Time Dependence of Electrochemical Response





#### C-Lattice Shifts versus Voltage and Temperature



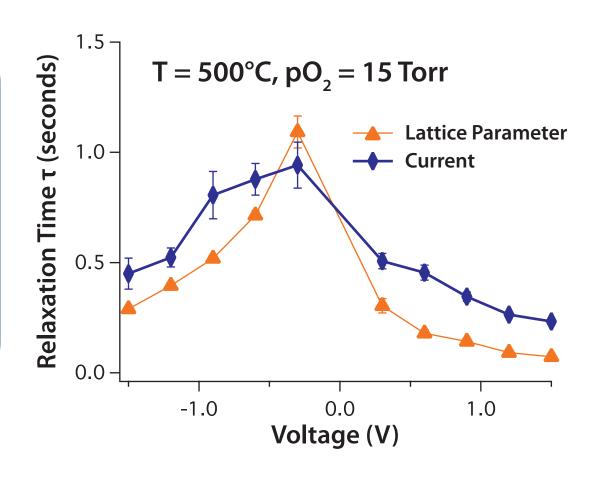
Explanation: LSCF phase change from rhombohedral to cubic between 400-550°C

Solid State Ionics 159 (2003) 71–78 Applied Catalysis B: Environmental 103 (2011) 318–325

#### Comparing Current and Lattice Relaxation

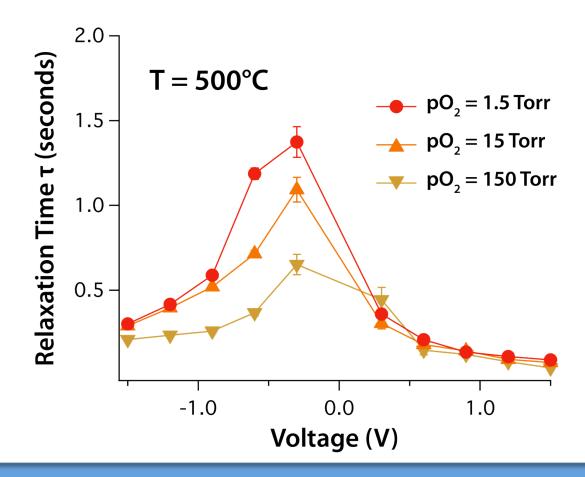
Current averages over the entire sample.

X-rays sample a very small area responds more quickly.



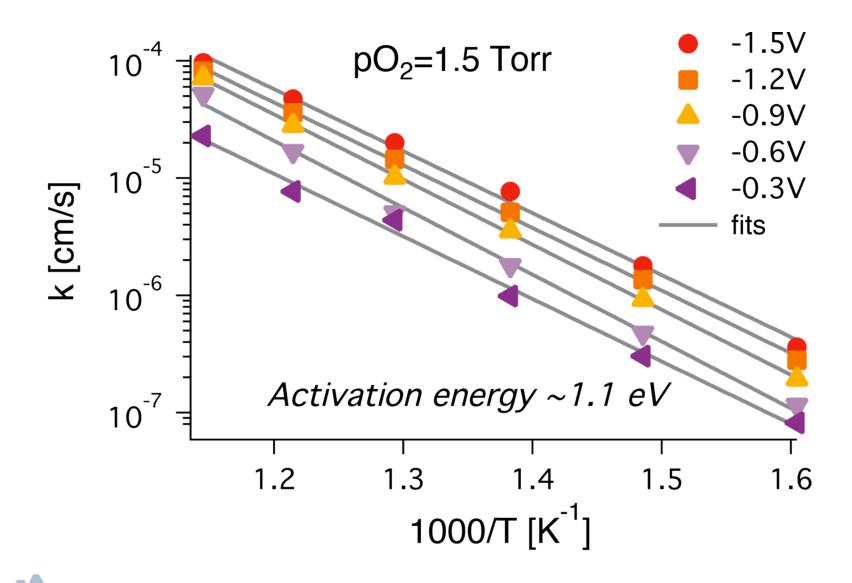


#### Effect of pO<sub>2</sub> on Lattice Relaxation



Not surprisingly, the lattice responds more quickly at higher oxygen partial pressures.

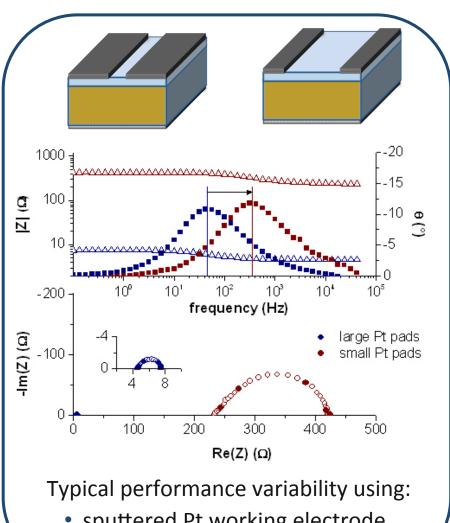
#### **Activation Energy for Oxygen Exchange**





Electrochemical variation due to current collector geometry and morphology

- Current collector geometry is critical
  - Macroscopic area is not linearly related to impedance
- Current collectors affect electrical property measurements (active area)
- Inconsistency in "painting" Pt-paste
  - Area coverage
  - Contact area with film
- Sputtered Pt de-wets film surfaces
  - Contact is lost as Pt loses continuity



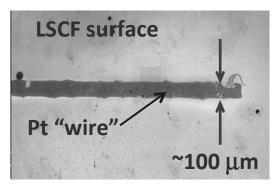
- sputtered Pt working electrode
- Pt paste counter electrode



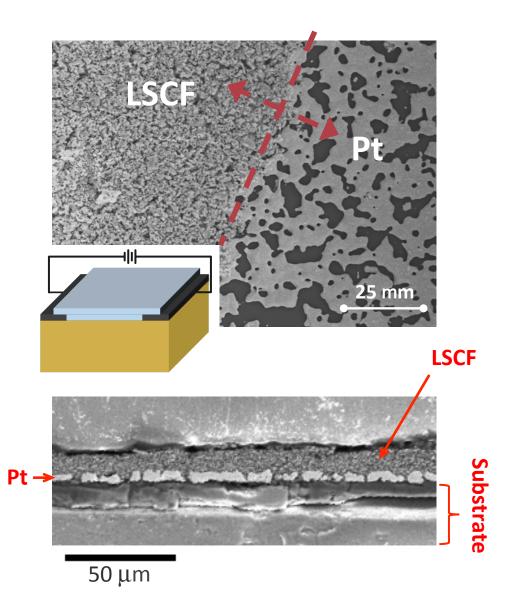
# Screen-Printing of SOFC Materials

# Screen-printed electrodes provide enhanced performance

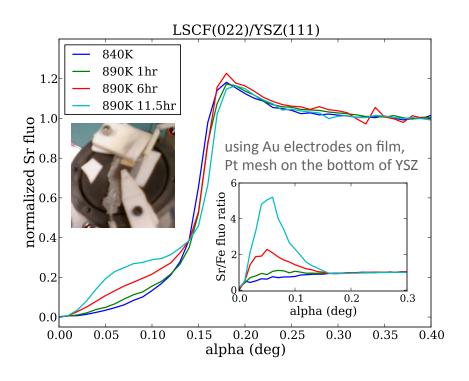
- Total thickness variation < 10%</p>
- Accurate lateral positioning and sizing
- Electrically continuous "wires"
- Complex geometries are possible
- High process control and repeatability



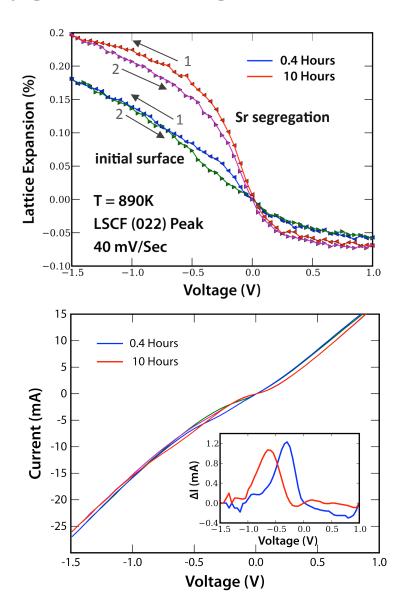
- Buried Pt electrodes / porous LSCF
- Stable microstructure after anneal



# Impact of Sr Segregation on Oxygen Exchange



- Slow increase in Sr segregation during annealing at ~620°C in air
- Lattice expansion is higher and shows more hysteresis on the Sr segregated surface
- IV curve shows only small difference between the initial and Sr segregated surface





#### **Summary**

- In-situ x-ray techniques provide opportunity to understand relationships between film / interface structure and electrical behavior
- Application of DC fields across LSCF/GDC/YSZ heterostructures results in rapid (< 1 sec) changes in LSCF (but not GDC or YSZ) out-of-plane lattice parameters
  - Indicates that oxidation / reduction reactions at the LSCF / gas interface are rate-limiting
- The activation energy is in the order of 1.1eV for the currents and c-lattice parameter shifts.
- The tau values extracted from the current are larger than the ones from the c-shift for high pO<sub>2</sub> and high T.



# The End

