

***Improving the Weldability of Fe-Al-Cr Alloys
Through TiC Additions***

John N. DuPont and Ken D. Adams

***Lehigh University
Department of Materials Science and Engineering***

Fossil Energy Materials Conference

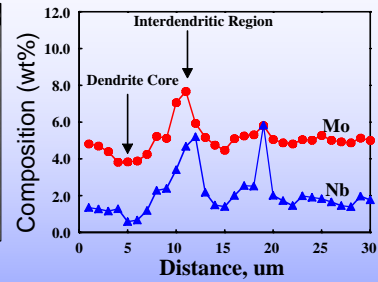
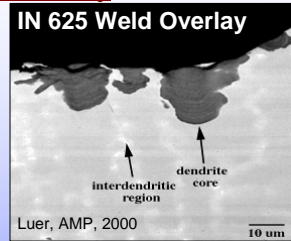
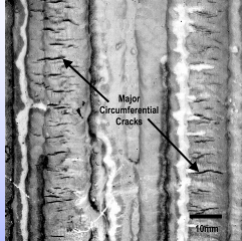
May 12-14, 2009

Financial support provided by the U.S. Dept. of Energy –
Fossil Energy Materials Program



Advantages of FeCrAl Weld Overlays Over Currently Used Alloys in Low NO_x Combustion Conditions

Failed Inconel 625 Weld Overlay



- Cracking in overlay
- Preferential attack due to microsegregation
- Initiation sites for corrosion-fatigue cracking

Drawbacks of Austenitic Overlays:

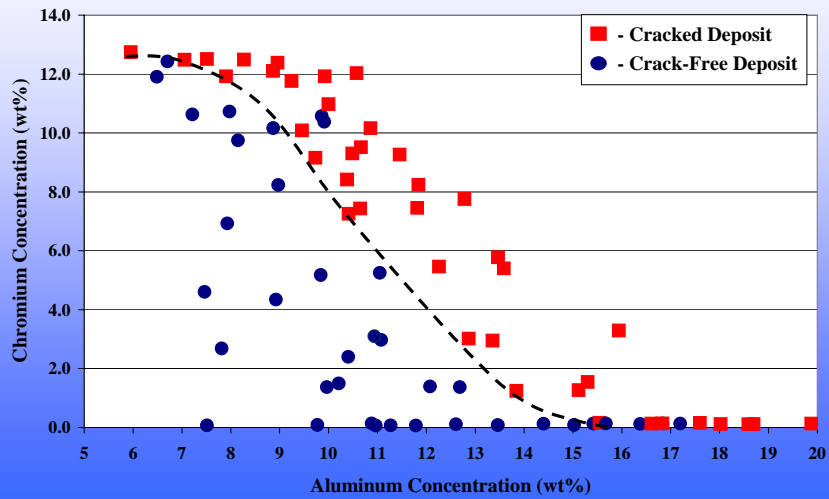
- Expensive: \$15-\$20/lb
- Exhibit microsegregation and preferential corrosion
- Susceptible to corrosion-fatigue cracking

Use of Fe-Al-Cr Weld Overlays:

- Low cost: ~ \$5-10/lb
- No residual microsegregation
- Excellent corrosion resistance in sulfur bearing environments
- Weldability is limited

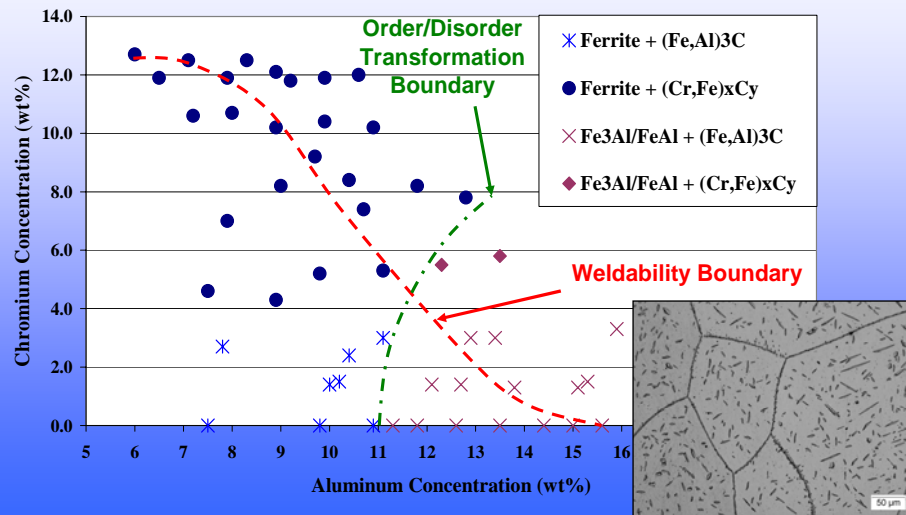
Previous Weldability Results

- Deposition of crack-free welds well beyond 11 wt% Al were possible

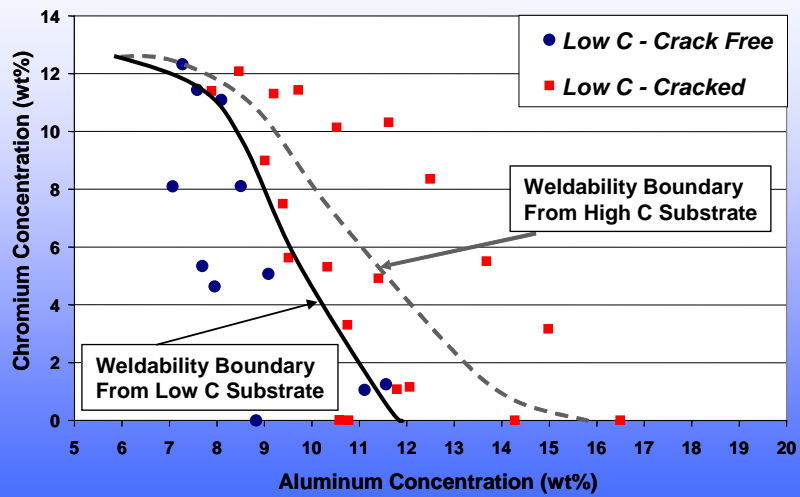


Previous Weldability Results

- Deposition of crack-free welds well beyond 11 wt% Al were possible
- Improved cracking resistance attributed to formation of $(\text{Fe,Al})_3\text{C}$ carbides that act as Hydrogen trap sites



Beneficial effect of carbide additions confirmed indirectly with
welds made on low Carbon Substrates



Technical Approach

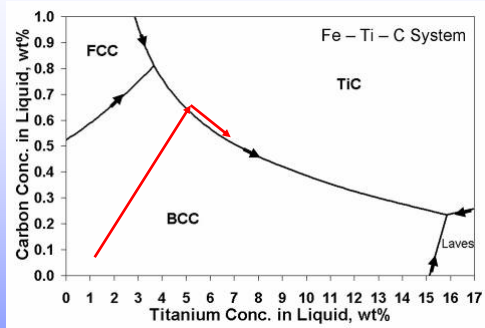
TiC in steels:

TiC has high binding energy with H_2

Trapping Locations	Binding Energy (kJ / mol)	Reference
Dislocation	0 – 59	Choo & Lee 1982, Hirth 1980
Grain boundary	18 – 59	Choo & Lee 1982, Kurnitsk 1980
MnS	72	Lee & Lee 1983
Al ₂ O ₃	79	Lee & Lee 1986
Y ₂ O ₃	80	Maroef & Olson 1999
TiC	87 – 98	Lee & Lee 1984, Pressouyre & Bernstein 1978

TiC in Fe-Ti-C system:

TiC can form in situ during solidification



Technical Approach

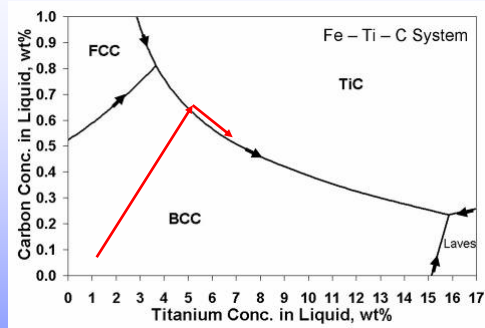
TiC in steels:

TiC has high binding energy with H_2

Trapping Locations	Binding Energy (kJ / mol)	Reference
Dislocation	0 – 59	Choo & Lee 1982, Hirth 1980
Grain boundary	18 – 59	Choo & Lee 1982, Kurnitsk 1980
MnS	72	Lee & Lee 1983
Al ₂ O ₃	79	Lee & Lee 1986
Y ₂ O ₃	80	Maroef & Olson 1999
TiC	87 – 98	Lee & Lee 1984, Pressouyre & Bernstein 1978

TiC in Fe-Ti-C system:

TiC can form in situ during solidification



Overall Objective:

Utilize TiC additions to improve the weldability of FeCrAl weld overlays

Need to:

- Establish method for controlling TiC contents (liquidus projection for FeCrAlTiC system and solidification model for predicting/controlling primary solidification path)
- Verify improvements to weldability
- Ensure corrosion resistance is not adversely affected

Approach

Solidification Modeling

- Calculate liquidus projections using Thermo-Calc
- Develop a solute redistribution model for determining TiC content from knowledge of nominal composition and cooling rate
- Validate with experimental alloys and phase ID

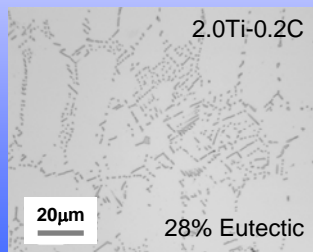
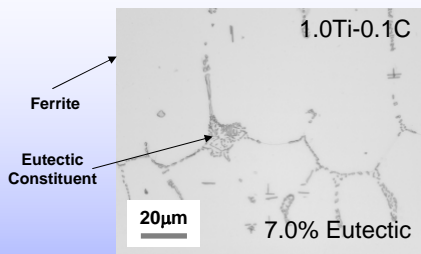
Weldability Testing

- Confirm effectiveness of TiC for mitigating cracking
- Welds prepared with a range of Al, Cr, and TiC contents

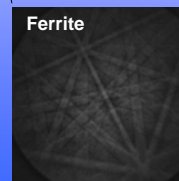
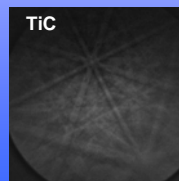
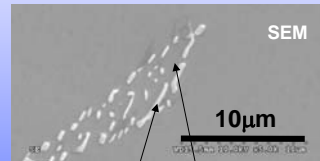
Corrosion Testing

- Thermo gravimetric analysis in simulated Low NOx combustion gas

Experimental Alloy Microstructures / Phase ID



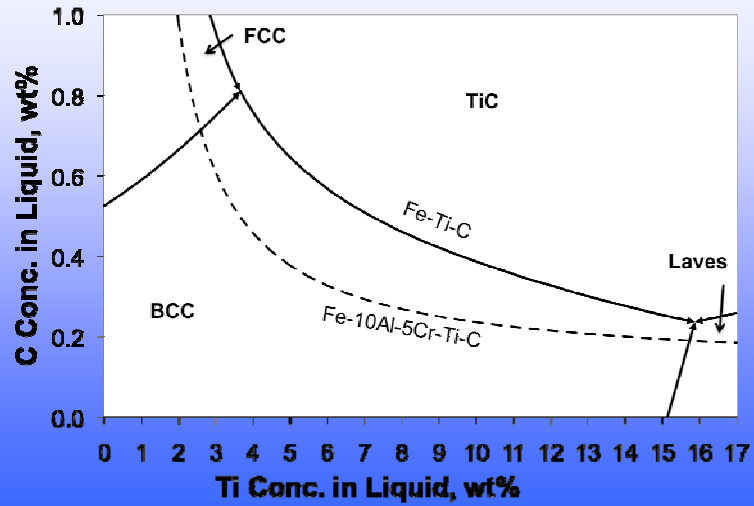
(wt%)	Ti	C
Fe-10Al-5Cr	1.0	0.1
Fe-10Al-5Cr	1.0	0.5
Fe-10Al-5Cr	1.5	0.4
Fe-10Al-5Cr	2.0	0.1
Fe-10Al-5Cr	2.0	0.2



EBSD

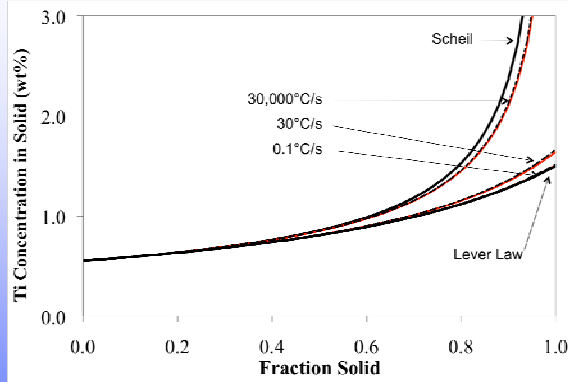
Calculated Liquidus Projection

- Additions of Al and Cr to Fe affect position of the eutectic line
- Additions of Al and Cr lower solubility of Ti and C in Fe, as expected

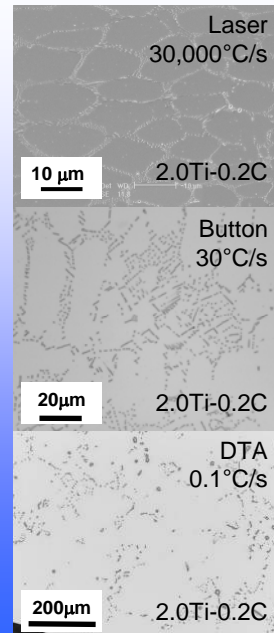


Assessing Solute Redistribution Behavior

Influence of cooling rate on redistribution of Ti

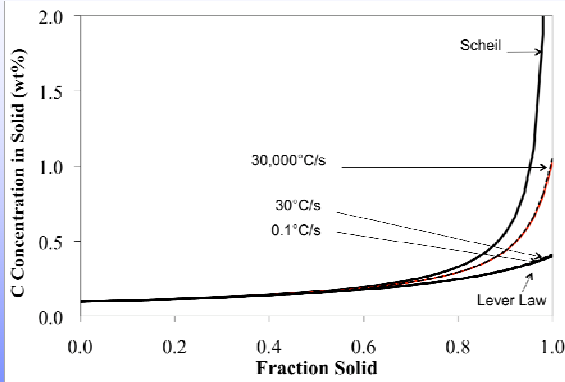


- S. Kobayashi (ISIJ 1988) solute redistribution model
- Range of solute redistribution does not permit use of simplified solidification path calculations (based on the lever law or Scheil approximations)

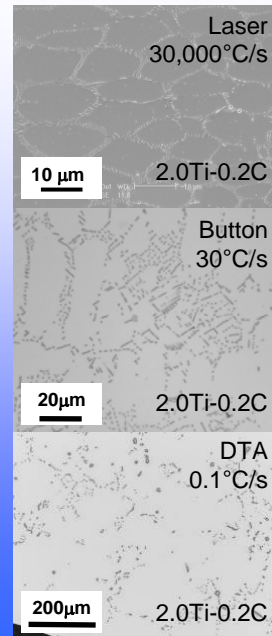


Assessing Solute Redistribution Behavior

Influence of cooling rate on redistribution of C



- S. Kobayashi (ISIJ 1988) solute redistribution model
- Range of solute redistribution does not permit use of simplified solidification path calculations (based on the lever law or Scheil approximations)



Solidification Path Calculations

Need to account for cooling rate and effect on back diffusion in the solid

Adapting Kobayashi's Solute Redistribution Model (ISIJ 1988) for Solidification Path:

$$C_L^i = C_o^i \xi^{\frac{k^i - 1}{1 - \beta k^i}} \left\{ 1 + \Gamma \left[\frac{1}{2} \left(\frac{1}{\xi^2} - 1 \right) - 2 \left(\frac{1}{\xi} - 1 \right) - \ln \xi \right] \right\}$$

$$\xi = 1 - (1 - \beta k^i)(1 - f_L)$$

$$\beta = \frac{4\alpha}{1 + 4\alpha}$$

$$\alpha = \frac{D_{Fe}^i t_f}{L^2} = \frac{D_{Fe}^i \Delta T}{\varepsilon L^2} \quad \leftarrow \text{Solute ability to diffuse}$$

\leftarrow Distance needed to diffuse

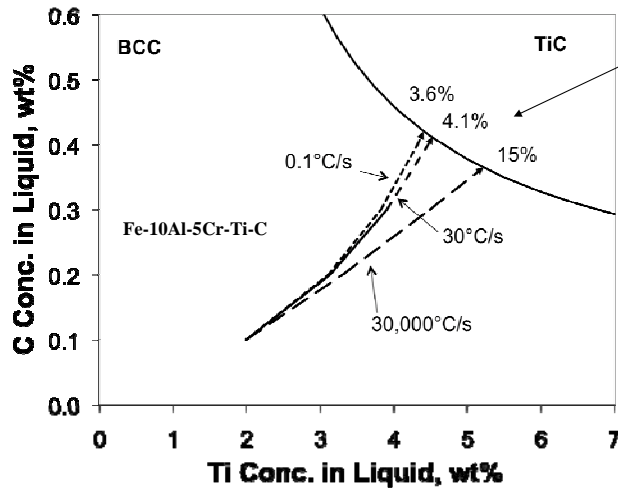
Brody-Flemings (AIME 1966)

Steps in Model

- Starting at $f_L = 1$ at nominal composition
- Solve for f_L for a target solute C_L value
- Insert f_L into C_L for other solute element
- Increase target C_L and repeat to define path

Example Solidification Path Calculations

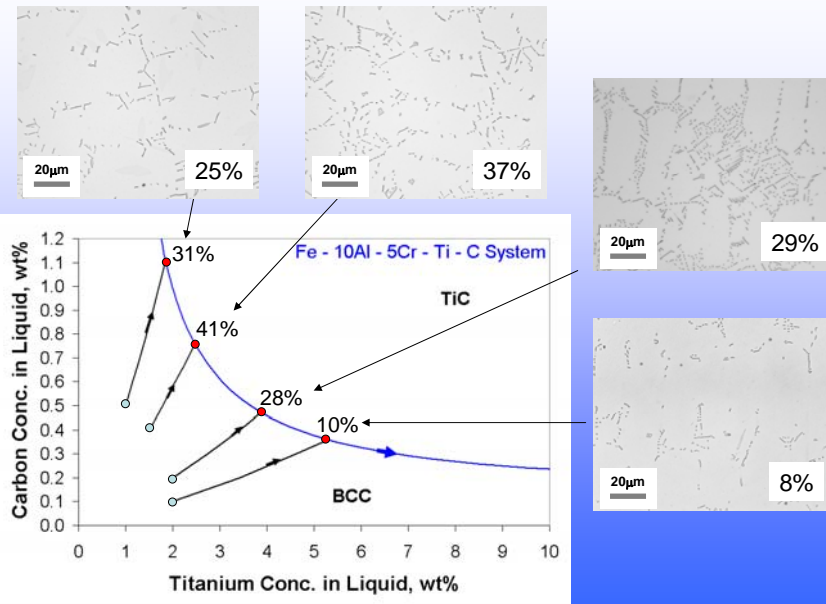
- Solute redistribution model developed to account for back diffusion of solute elements
- Based on Kobayashi model
- First known ternary solidification path model that accounts for back diffusion



Percent liquid remaining when eutectic solidification commences

Higher cooling rates limit back diffusion of Ti, resulting in more Ti enrichment in liquid phase and more eutectic constituent

Comparison of Model and Experimental Results



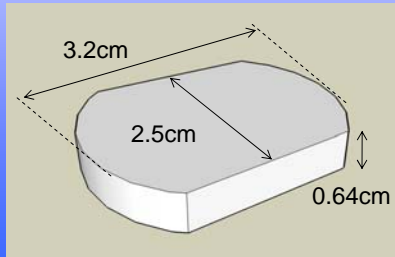
Weldability Testing

Gas Tungsten Arc Welding

- 200 amps
- 12 volts
- 2.8 mm arc gap
- 5 mm/s travel speed
- Cooling rate $\sim 10^2$ °C/s
- Shielding gas:
 - Argon
 - Argon – 5% Hydrogen

Effect of Al and Cr					
(wt%)					(%)
Fe	Al	Cr	Ti	C	% TiC
Bal.	5.0	2.0	2.0	0.3	7.7
Bal.	7.5	3.5	2.0	0.3	6.9
Bal.	10.0	5.0	2.0	0.2	4.7
Bal.	12.5	6.5	2.0	0.2	5.1
Bal.	15.0	8.0	1.5	0.2	3.9
Bal.	17.5	9.5	1.5	0.2	4.5
Bal.	20.0	11.0	1.5	0.2	3.9

* Ti and C chosen for TiC Contents of 4 – 7 %



Fe-15Al-8Cr-1.5Ti-0.2C

Weldability: Effect of TiC

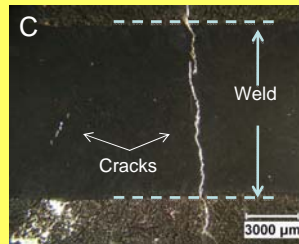
- Ground, polished, etched
- Stereomicroscope Images



Shielding Gas:
Argon

Composition:
Fe-10Al-5Cr-0.05C

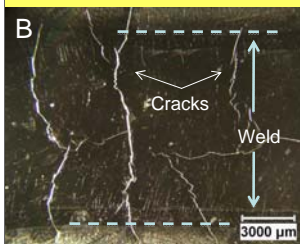
0% TiC



Shielding Gas:
Argon/Hydrogen Mix

Composition:
Fe-10Al-5Cr-
1Ti-0.1C

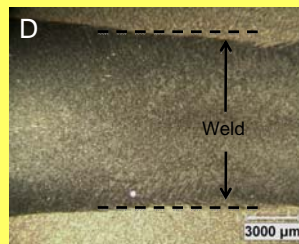
2.2% TiC



Shielding Gas:
Argon/Hydrogen Mix

Composition:
Fe-10Al-5Cr-0.05C

0% TiC

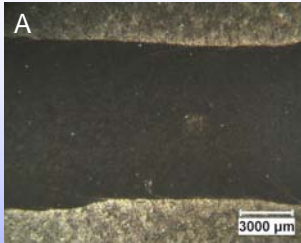


Shielding Gas:
Argon/Hydrogen Mix

Composition:
Fe-10Al-5Cr-
1.5Ti-0.4C

5.7% TiC

Weldability: Effect of Al and Cr

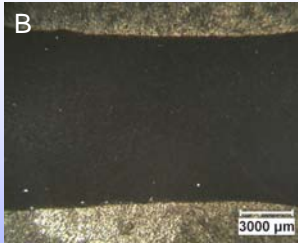


Shielding Gas:
Argon/Hydrogen Mix

Composition:
Fe-5Al-2Cr-2Ti-0.3C

7.7% TiC

No Cracks

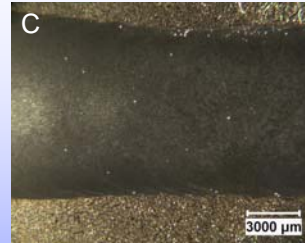


Shielding Gas:
Argon/Hydrogen Mix

Composition:
Fe-10Al-5Cr-2Ti-0.2C

4.7% TiC

No Cracks



Shielding Gas:
Argon/Hydrogen Mix

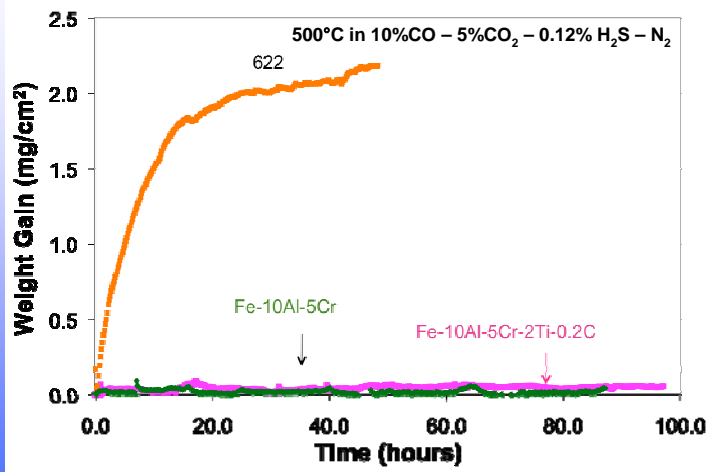
Composition:
Fe-20Al-11Cr-1.5Ti-0.2C

3.9% TiC

No Cracks

Corrosion Resistance

- Thermo gravimetric Analysis:



- Corrosion resistance slightly decreases when TiC present
- Corrosion resistance remains significantly better than Ni-based overlay

Major Project Accomplishments

- A solidification model was developed and validated for controlling the TiC content of Fe-Cr-Al-Ti-C weld overlays
- Resistance to hydrogen cracking in welds was shown to significantly improve with TiC additions
- Corrosion resistance is not significantly affected