Mechanisms Governing Mixed Conducting SOFC Cathodes

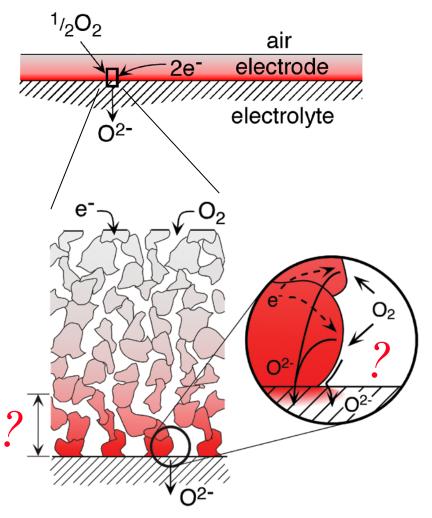
Stuart B. Adler, Dinesh Baskar, Xiyong Chen, Yunxiang Lu, Jamie R. Wilson

University of Washington

8th Annual SECA Workshop August 9, 2007

Mixed-conducting SOFC Cathodes

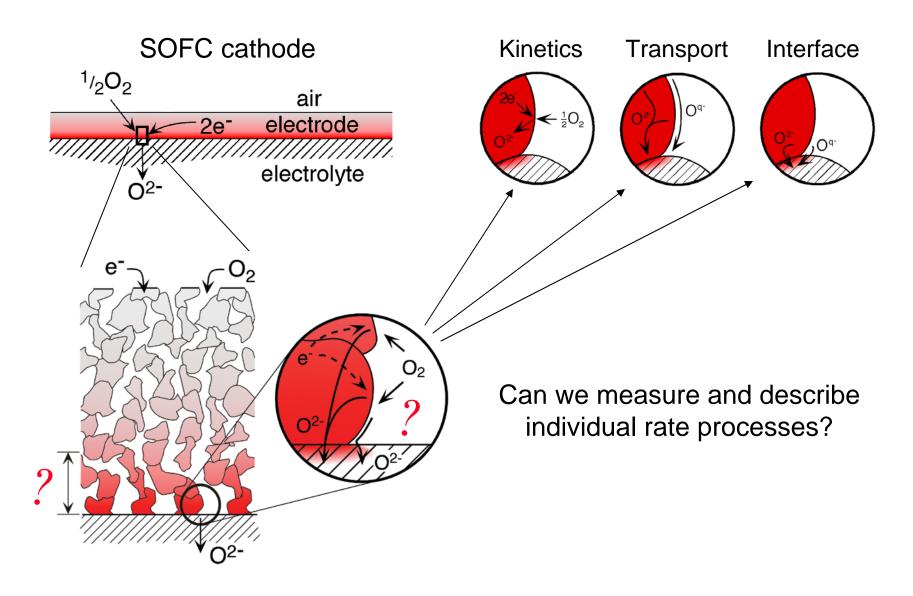
SOFC cathode



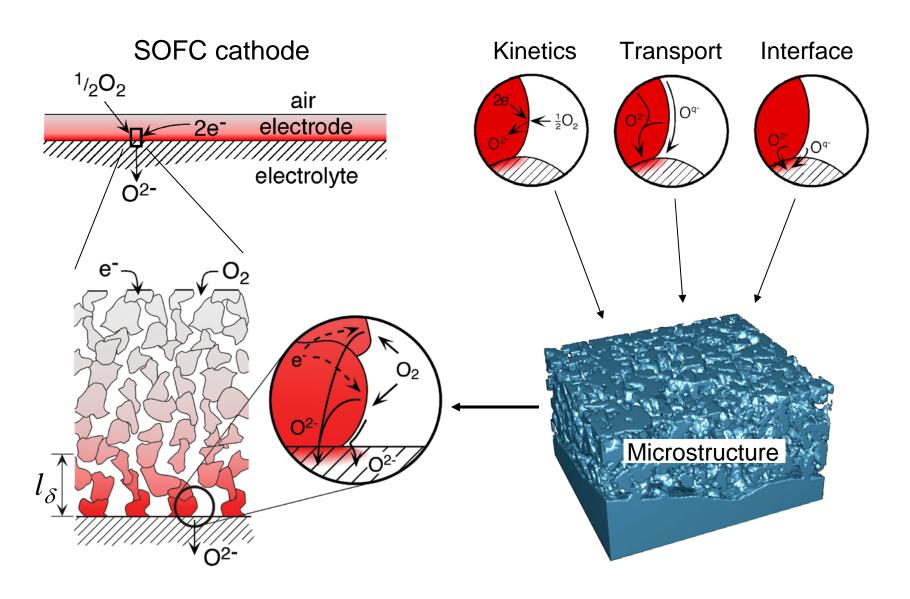
Why our understanding remains largely empirical:

- Many models fit i-V and impedance data equally well.
- Poor *ex-situ* understanding of individual rate processes.
- Inherent convolution of rates.
- Lack of quantitative information about the microstructure.

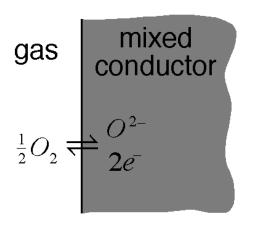
Mixed-conducting SOFC Cathodes



Mixed-conducting SOFC Cathodes

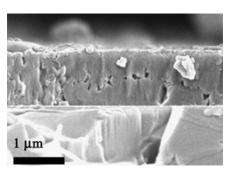


Isolating the Kinetics



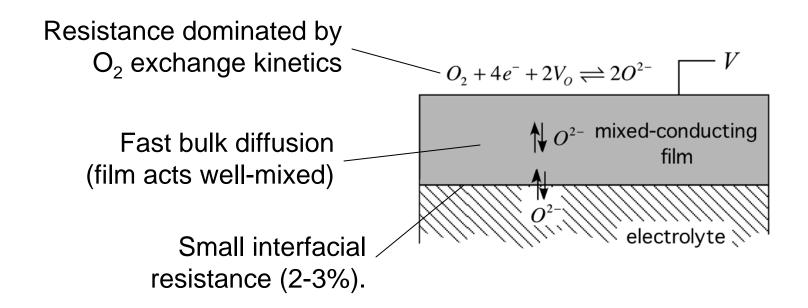
This is a difficult reaction to measure and describe.

Isolating the Kinetics

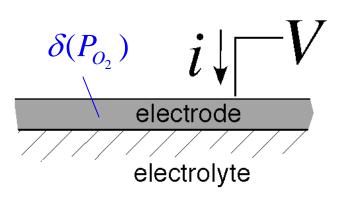


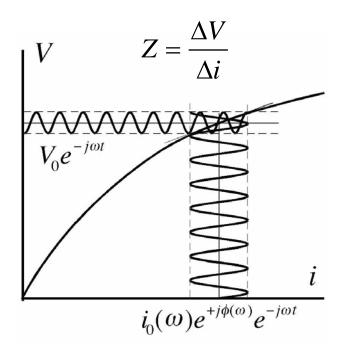
Tohoku University (Maya Sase, Tatsuya Kawada)

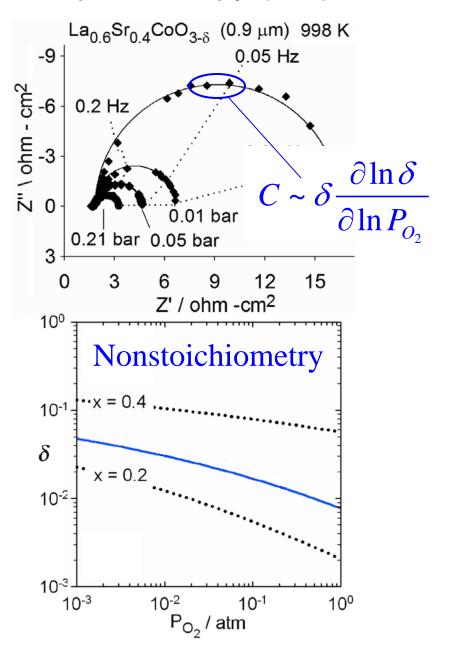
- Pulsed laser-deposited films of La_{0.6}Sr_{0.4}CoO_{3-δ} (LSC-64) on Gd-doped ceria (GDC).
- Uniform thickness (500~1000 nm) and composition.
- Laterally polycrystalline, but 100 oriented (pseudocubic).



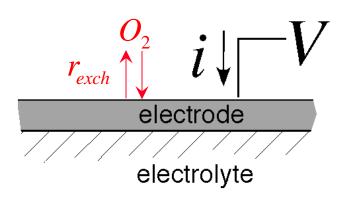
Electrochemical Impedance Spectroscopy (EIS)

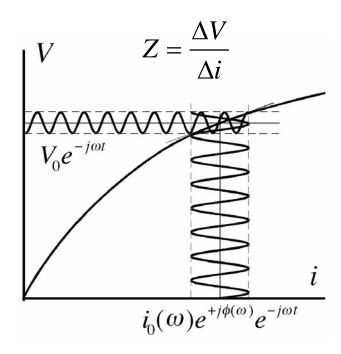


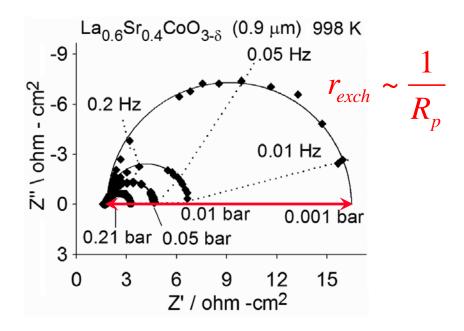


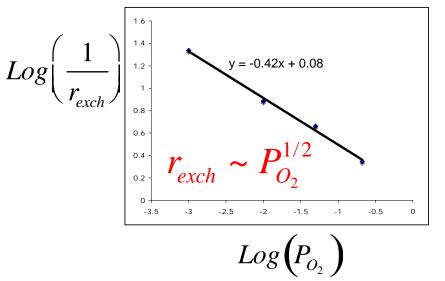


Electrochemical Impedance Spectroscopy (EIS)



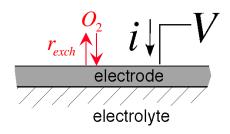


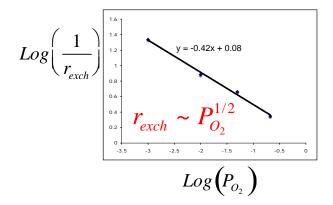




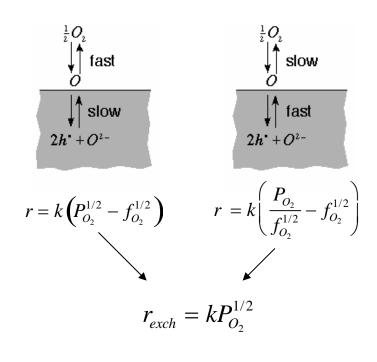
Limitations of linearized rate measurements

What mechanism explains a power-law P_{O2} dependence?



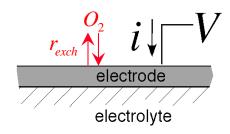


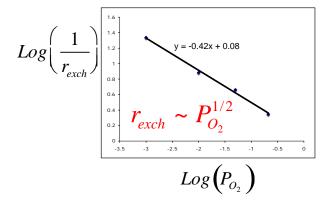
Multiple models predict similar T and $P_{\rm O2}$ dependence of $r_{\rm exch}$.



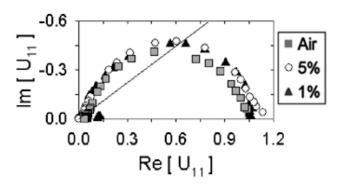
Limitations of linearized rate measurements

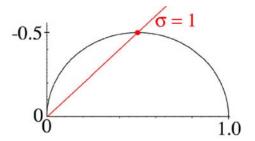
What mechanism explains a power-law $P_{\rm O2}$ dependence?



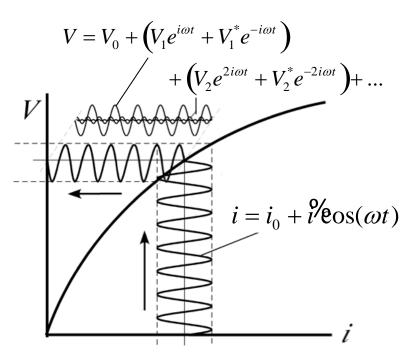


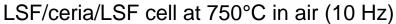
EIS data lack mechanismspecific characteristics

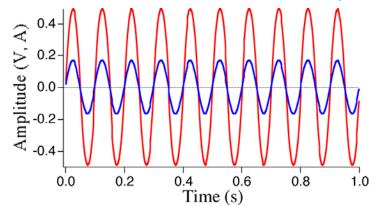


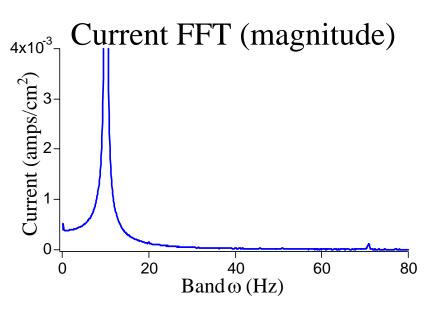


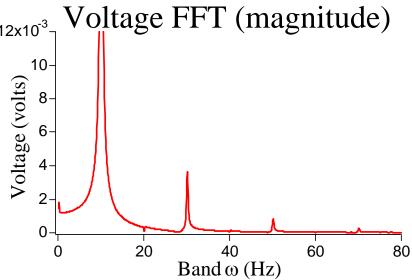
Nonlinear Electrochemical Impedance Spectroscopy (NLEIS)







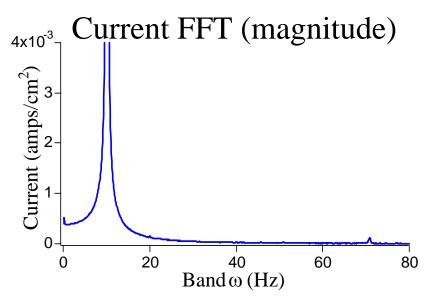


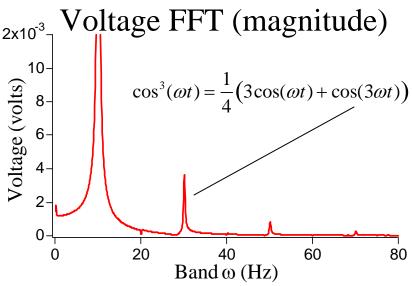


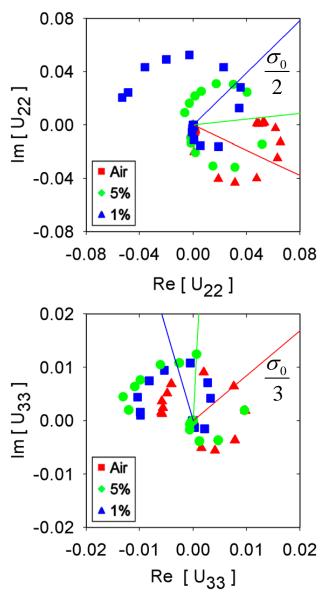
Nonlinear Electrochemical Impedance Spectroscopy (NLEIS)

 Magnitude, sign, and phase of the higher harmonics are tied to nonlinearities of the underlying physics. (analogy: music).

- Small nonlinearities appear as separate peaks, allowing precise quantification.
- Noise and systematic errors uncorrelated to the input perturbation are automatically filtered.







Observations:

- The harmonic signals are small difficult to isolate by other means.
- Unlike EIS, higher harmonics are sensitive to changes in P_{O2} .

How do we interpret the harmonic response in terms of mechanism?

Sources of Nonlinearity in a Mixed Conducting Oxide Film

$$r = \Re_0 \left(P_{O_2}^{gas}, f_{O_2}^{solid} \right) \left\{ 1 - e^{\frac{-\Lambda}{RT}} \right\}$$

$$O_2 + 4e^- + 2V_o \rightleftharpoons 2O^{2-} \qquad V$$

$$\delta(f_{O_2}^{solid}) \text{ obeys rigid band model}$$

$$-\frac{1}{2} \frac{\partial \ln f_{O_2}^{solid}}{\partial \ln \delta} = 1 + \frac{4\delta}{g_0 RT}$$

$$O_2 + 4e^- + 2V_o \rightleftharpoons 2O^{2-} \qquad V$$

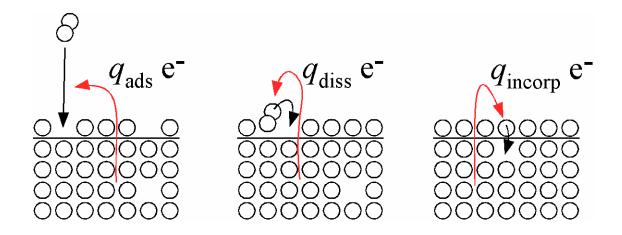
$$film$$

$$film$$

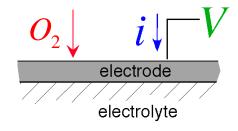
$$O_2^{2-} \text{ mixed-conducting film}$$

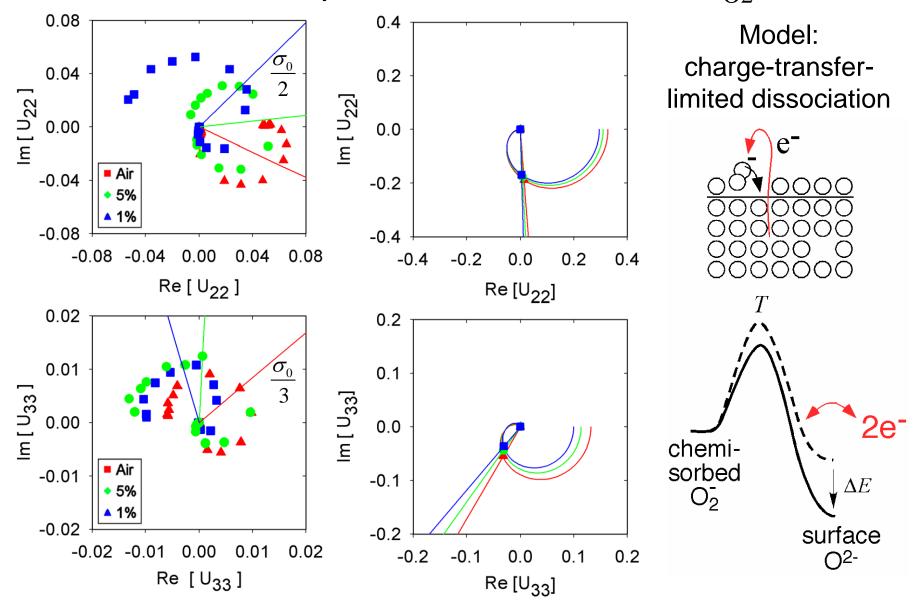
$$O_2^{2-} \text{ electrolyte}$$

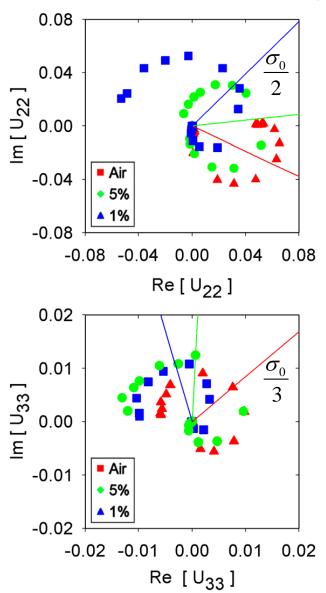
Possible rate-limiting elementary steps



$$\frac{Lc_0}{3} \frac{d\delta}{dt} = -\frac{\frac{\text{%eos}(\text{%b})}{2F}}{2F} - 2\Re_0 \left[1 - e^{\frac{-\Lambda}{\lambda RT}} \right] \qquad O_2 \downarrow \\
V = \frac{RT}{4F} \ln \frac{f_{O_2}(\delta)}{P_{O_2}^{(gas)}} \qquad \text{elec}$$

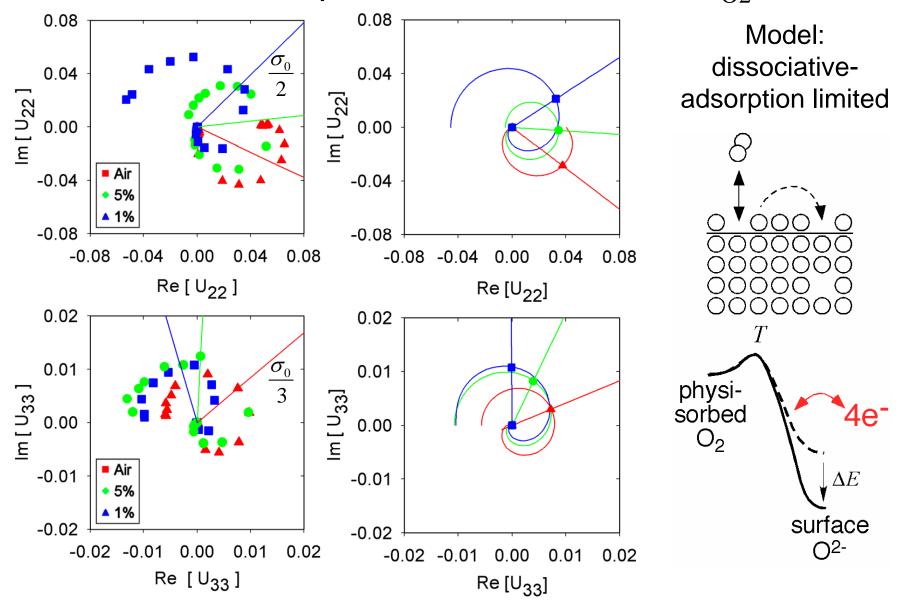


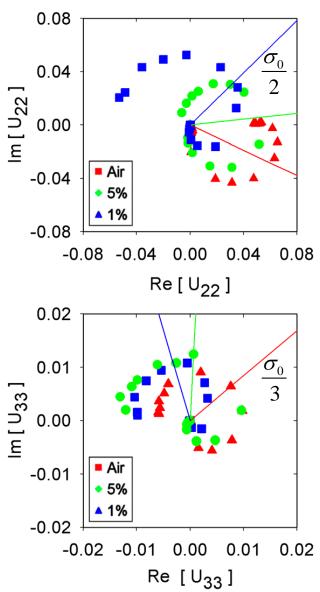




Possible rate limiting phenomena

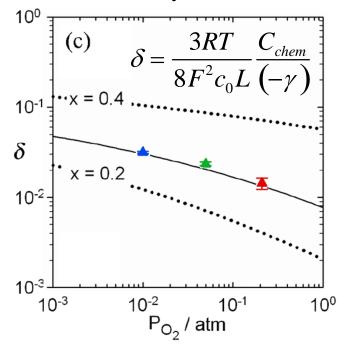
- dissociation of chemisorbed intermediate
- molecular adsorption
- atomic incorporation into solid



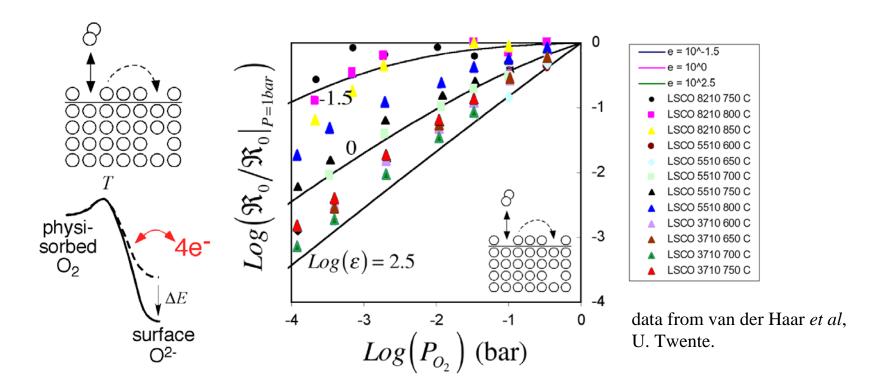


$$\gamma = \frac{\partial \ln \delta}{\partial \ln P_{O_2}} = -0.27 \sim -0.23$$

Nonstoichiometry calculated from the thermodynamic factor

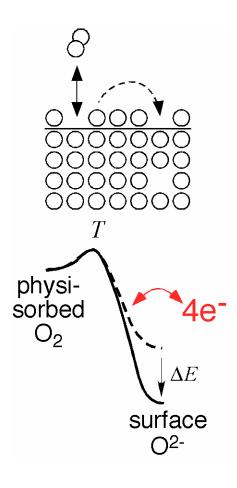


Consistency with equilibrium exchange rates



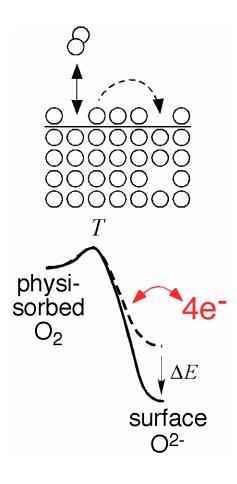
• Dissociative adsorption explains O_2 exchange measurements on bulk $La_{1-x}Sr_xCoO_{3-\delta}$ over a wide range of x, T and P_{O2} .

Physical Interpretation of the Observed Kinetics



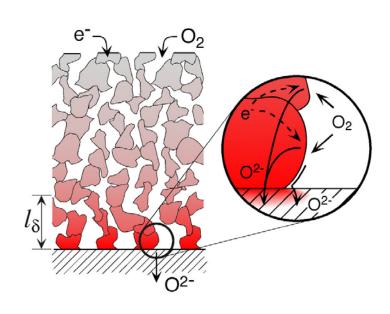
- The bottleneck is not charge-transfer reaction is limited by adsorbate lifetime and configurational probability.
- A strong Arrhenius dependence does not imply an activation barrier! In this case a strong T dependence arises from a large negative enthalpy of adsorption.
- O₂ exchange "power law" varies from zeroth to first order, depending on thermodynamic conditions.

Physical Interpretation of the Observed Kinetics



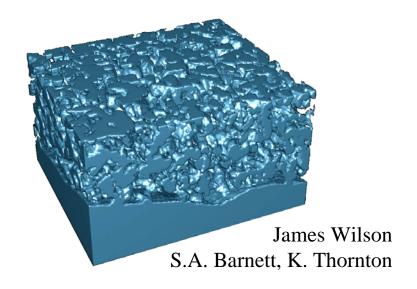
Possible implications for porous electrodes

- Surface and bulk are in equilibrium, even though gas and surface are not.
- Any surface diffusion is downstream of reaction barriers.
- No further reduction occurs at the TPB (only ion transfer).



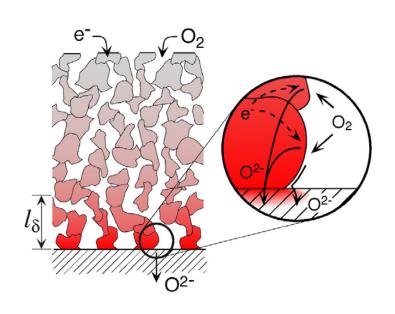
macrohomogeneous model

- microstructure described in terms of average porosity and area (ε, a) .
- requires $l_{\delta} >> d_{p}$.



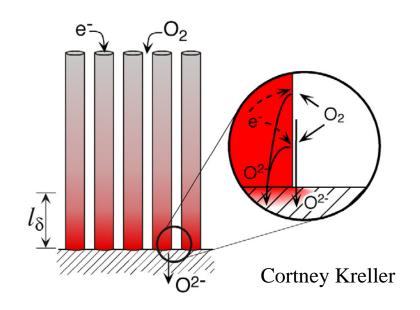
3-dimensional model

- measured microstructure represented by continuous finite element domain.
- can model 3-D gradients.



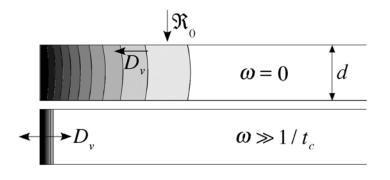


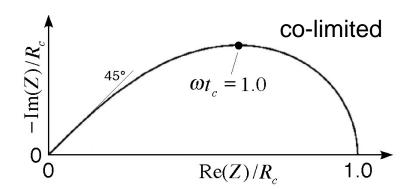
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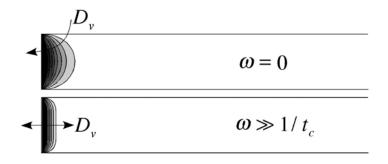


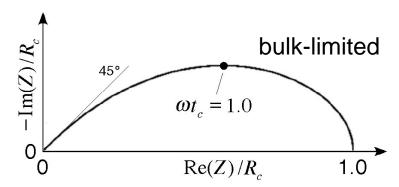
pseudo 3-dimensional model

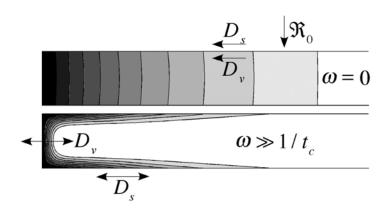
- microstructure replaced by simplified structural elements having same ε , a.
- can model 2-D gradients.

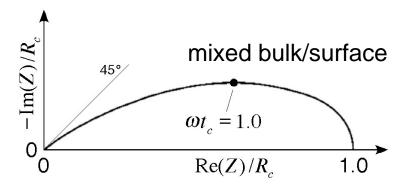






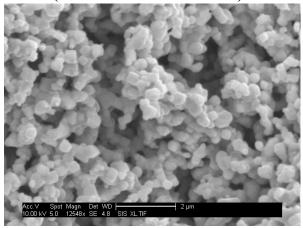




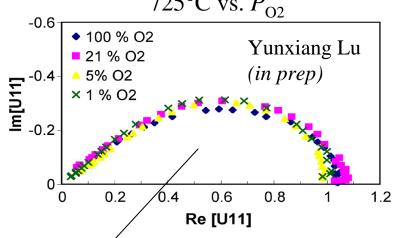


<u>data</u>

La_{0.8}Sr_{0.2}CoO_{3-δ} on SDC (sintered at 1000°C)



1st harmonic (impedance) of porous LSC-82 on SDC at 725° C vs. P_{O2}



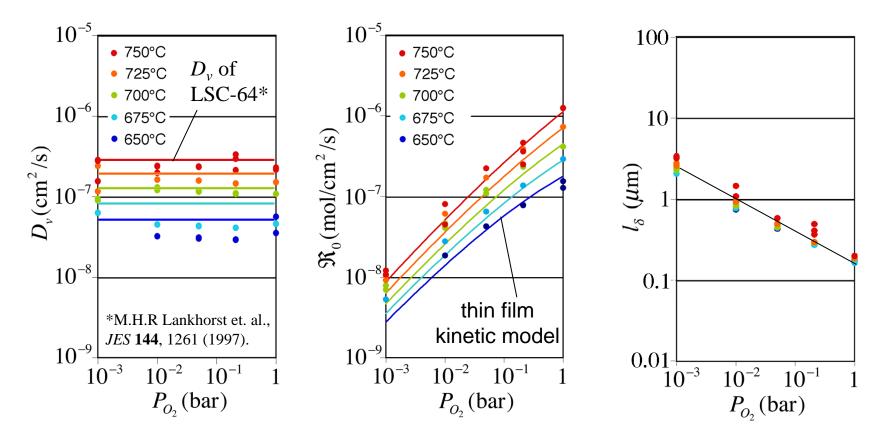
model

$$\begin{bmatrix} R_c \left(\mathfrak{R}_0, D_v, D_s(T), \delta \left(x, T, P_{O_2} \right), \left\{ \varepsilon \right\} \\ t_c \left(\mathfrak{R}_0, D_v, D_s(T), \delta \left(x, T, P_{O_2} \right), \left\{ \varepsilon \right\} \\ a \end{bmatrix} \end{bmatrix} = \begin{bmatrix} R_c \left(x, T, P_{O_2} \right) \\ t_c \left(x, T, P_{O_2} \right) \end{bmatrix}$$

Possible surface diffusion mechanisms:

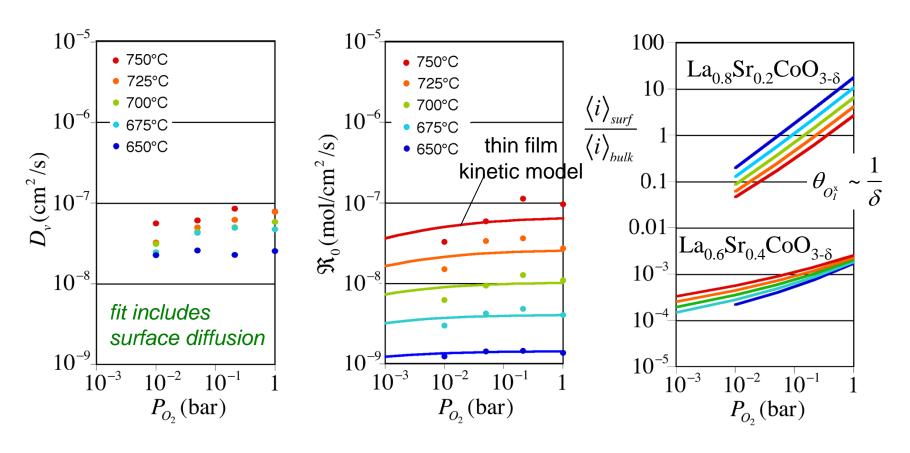
- oxygen vacancies
- oxide ion interstitials

Results: Impedance of Porous La_{0.6}Sr_{0.4}CoO_{3-δ}



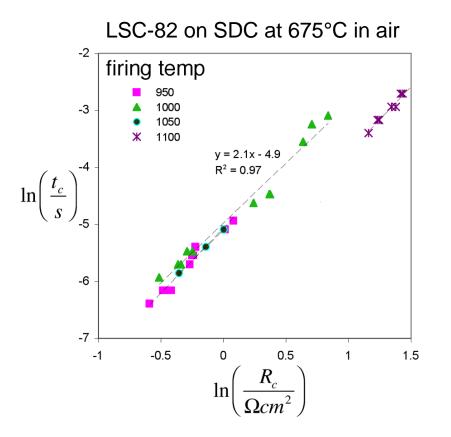
- Bulk transport dominates; D_{v} matches independent measurement.
- Apparent exchange kinetics match thin film within a scaling factor.
- Utilization region is sub-micron under typical IT-SOFC conditions.

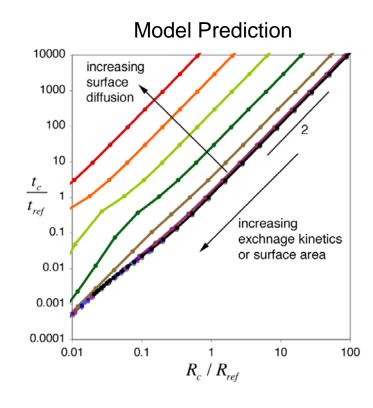
Results: Impedance of Porous La_{0.8}Sr_{0.2}CoO_{3-δ}



- Must include surface diffusion to yield reasonable values of D_{v} .
- Apparent exchange kinetics match thin film within a scaling factor.
- Surface diffusion plays a significant role under IT-SOFC conditions and appears to occur by an "interstitial" mechanism (?!).

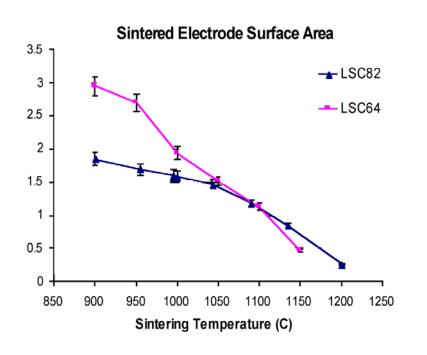
Results: Electrode Firing Temperature



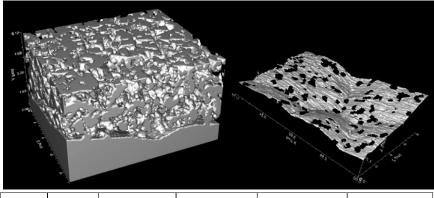


- Above ~950°C, increased firing temperature reduces performance.
- $t_c \sim (R_c)^2$ scaling implies reduced surface area or exchange kinetics.

Results: Electrode Firing Temperature



FIB-SEM reconstruction and analysis

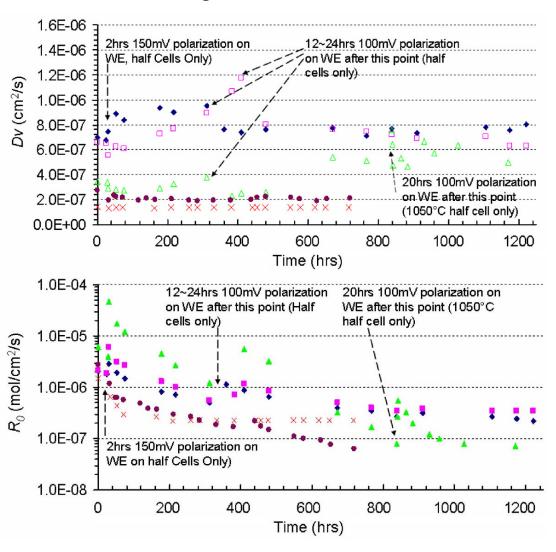


	Firing Temp.	Porosity	Volumetric Surface Area	Fraction of Electrolyte Bonded to LSC	Tortuosity for Bulk Ion Transport in LSC	Tortuosity for Gas-Phase Mass Transfer
	950¼C	42 %	2.7 μm ⁻¹	22 %	1.4	1.6
	1100¼0	46 %	2.8 μm ⁻¹	21 %	1.3	1.7
- 1						

- Changes in surface area cannot explain the 50x variation in rate.
- Sample variability is much greater than effect of firing temperature.
- Firing conditions appear to influence catalyst chemistry much more significantly than morphology.

Results: Electrode Degradation

Degradation of various LSC-82 cells in air at 750°C.



Electrode degradation also appears to be tied to <u>surface kinetics</u>.

Summary

- The rate of O_2 reduction on $La_{1-x}Sr_xCoO_{3-\delta}$ (LSC) has been isolated from other rates using thin film electrodes and NLEIS, and used to rationalize porous electrode performance using a mechanistic model.
- O_2 reduction on LSC appears to be limited by dissociative adsorption. This result explains previous O_2 exchange measurements vs. x, T, P_{O2} , and also explains impedance of porous LSC electrodes over a wide range of conditions.
- Porous LSC electrodes are dominated by a bulk transport path at high vacancy concentration. Lower Sr doping and temperatures lead to significant involvement of surface transport at high $P_{\rm O2}$. Materials like LSM are almost certainly surface dominated.
- Surface catalytic activity is very sensitive to firing conditions and time (degradation). The origin of these effects appears to be mostly surface chemistry, not morphology. Further work that can identify changes in surface structure/composition is needed.

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Tatsuya Kawada (Faculty, Tohoku)

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Hsun Yi (PhD student, Michigan MSE)

Roberto Mendoza, Katsuyo Thornton (Michigan MSE)

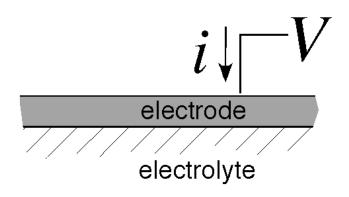
Support

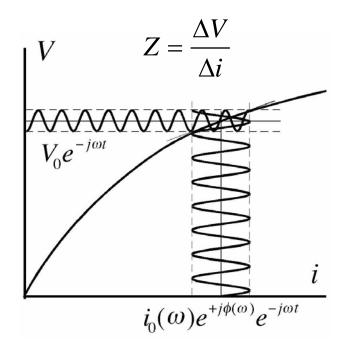
DOE SECA Core Technology Program

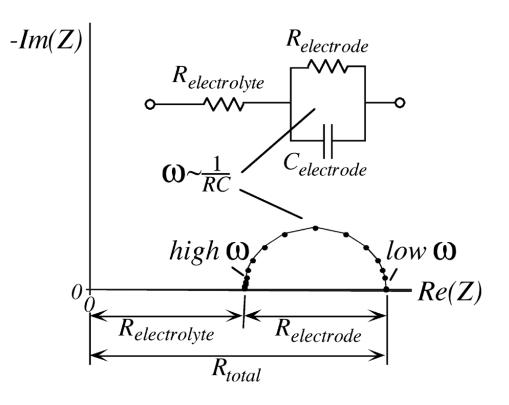
NSF: DMR-Ceramics, CTS



Electrochemical Impedance Spectroscopy (EIS)

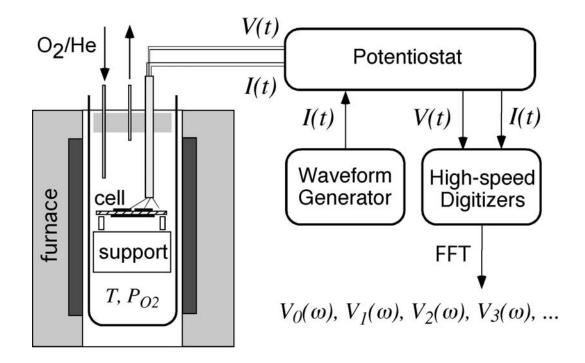




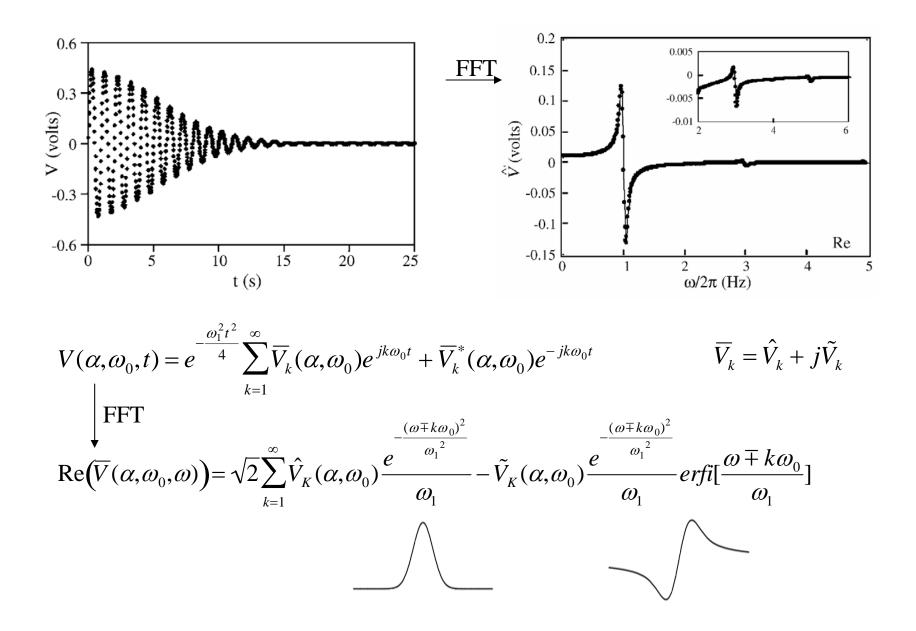


EIS probes linearized rate and faradaic capacitance.

NLEIS measurements



Voltage Signal Involves both Amplitude and Phase Shift



Power Series Expansion of Harmonic Response

$$\overline{V}_{1}(\alpha, \omega_{0}) = \alpha \overline{V}_{1,1}(\omega_{0}) + \alpha^{3} \overline{V}_{1,3}(\omega_{0}) + \alpha^{5} \overline{V}_{1,5}(\omega_{0})
\overline{V}_{3}(\alpha, \omega_{0}) = \alpha^{3} \overline{V}_{3,3}(\omega_{0}) + \alpha^{5} \overline{V}_{3,5}(\omega_{0})$$

Least Squares Fit

$$\overline{V}_{k} = \hat{V}_{k} + jV_{k}^{\prime \circ}$$

 \hat{V}_k = Real Part of the Response

 V_k = Imaginary Part of the Response

