



# The Thermodynamic Evaluation and Modeling of Grade 91 Alloy and its Secondary Phases through the CALPHAD Approach

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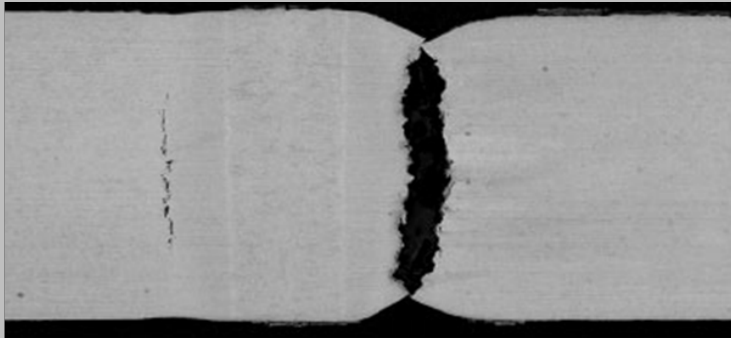
<sup>2</sup>Worcester Polytechnic Institute (WPI)



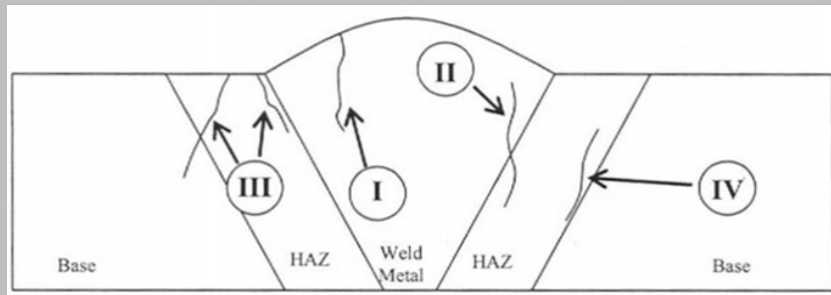
# Outline

- *Creep Resistance--Type IV Cracks*
- *Microstructure*
- *Critical Secondary Phase Formations*
- *Short-Term and Long-Term Creep*
- *Simulation Results*
  - *Baseline*
  - *Modified Composition*
- *Conclusion*

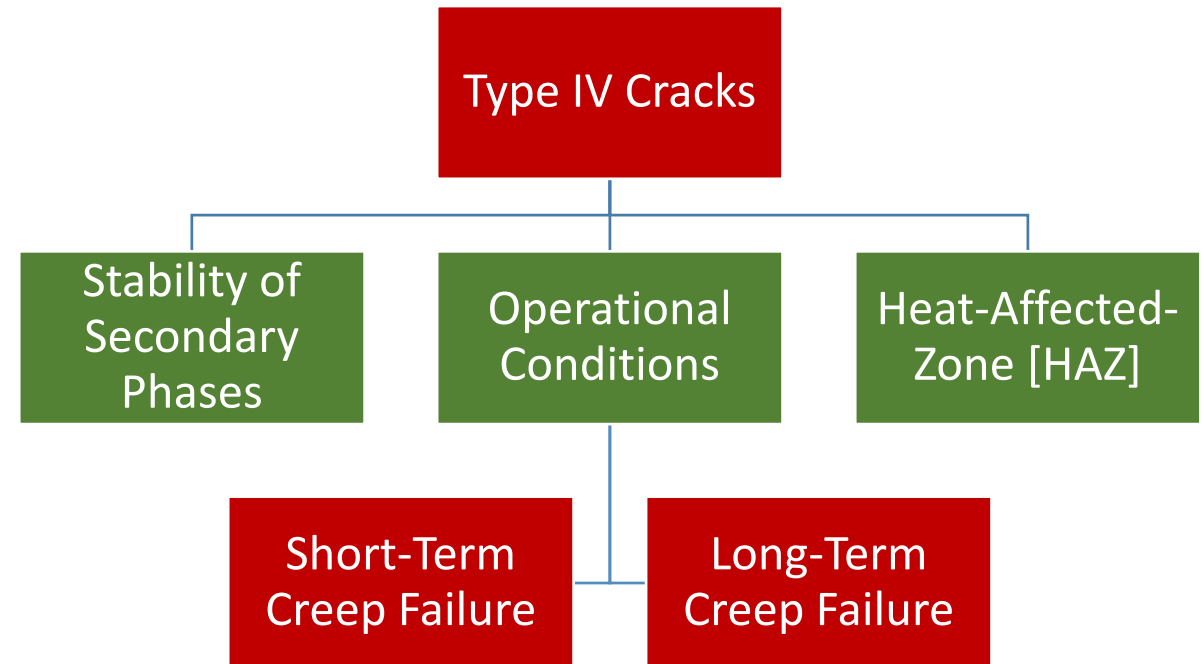
# Creep Failure for Gr.91



*D. J. Abson and J. S. Rothwell 2013*



*J. M. Brear and A. Fleming 2004*



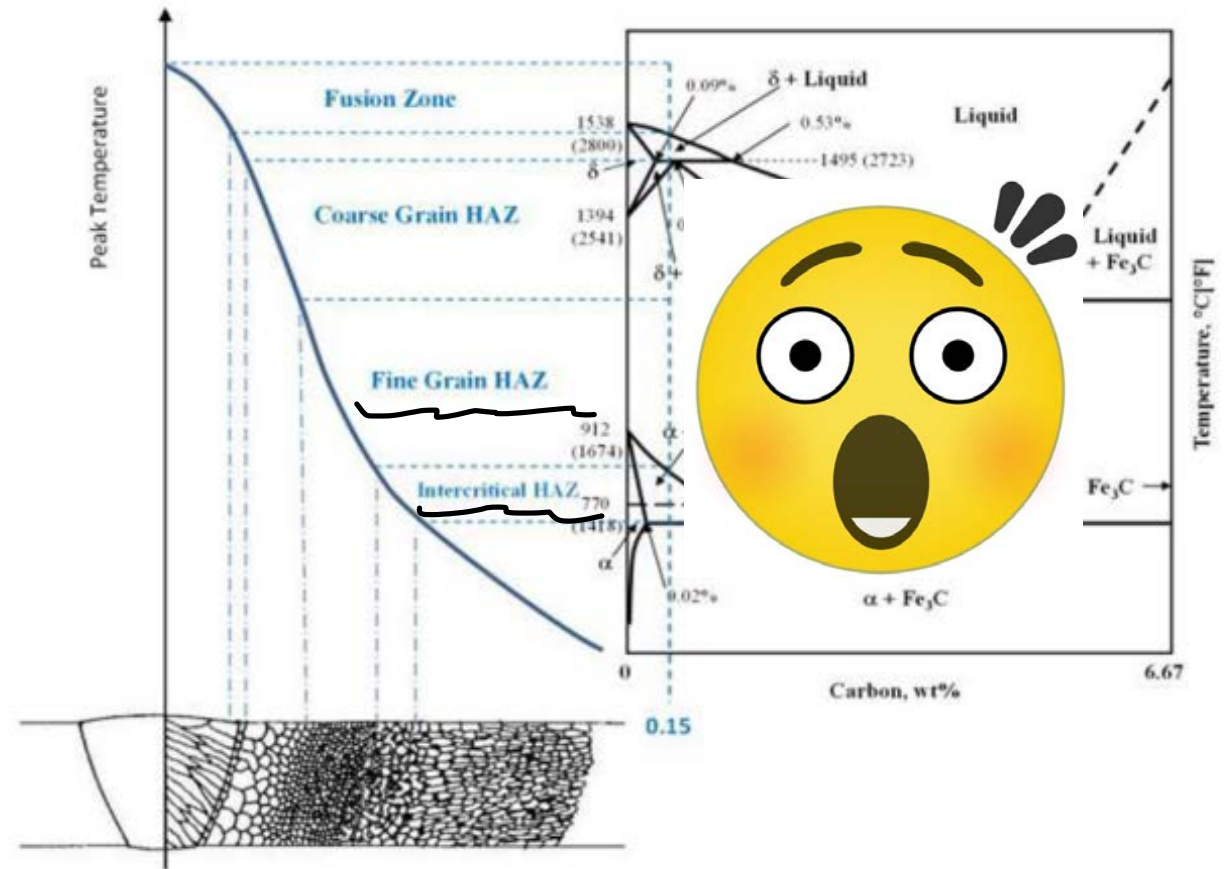
- Extensive Experiments and Modeling effort
- **Anything further help from Thermodynamics?**

# Microstructure

## 3 Subzones:

- Coarse-grain HAZ (CGHAZ)
- Fine-grain HAZ (FGHAZ)
- Intercritical HAZ (ICHAZ)

Most investigation relied on carbon steel phase diagrams?!



*Lippold J.C 2015*

# Critical Secondary Phases

## $M_{23}C_6$ Precipitates

M = Cr, Fe, and Mo, Coarsening during operation

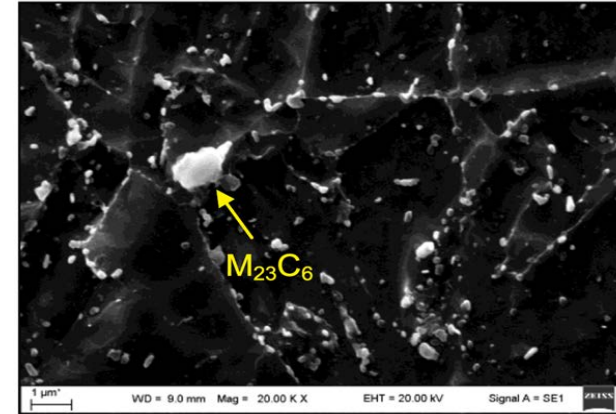
## $MX$ Precipitates

FCC structure, M = V/Nb, X = C/N. Very small and low coarsening rate.

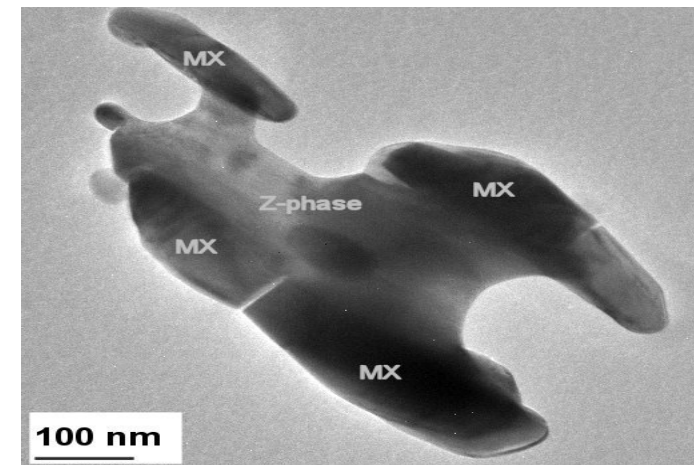
## Z-Phase

Cr(V,Nb)N, detrimental to creep resistance, long term operation.

How to control their stabilities?



A. Grybėnas 2017



H.K. Danielsen 2008



# Questions Related to Thermodynamics

- Creep mechanisms at different conditions?
- Thermodynamics vs. HAZ microstructures?
- The role of elements inside Gr.91?
- Any better creep resistant alloys?
- Choice of processing parameters?

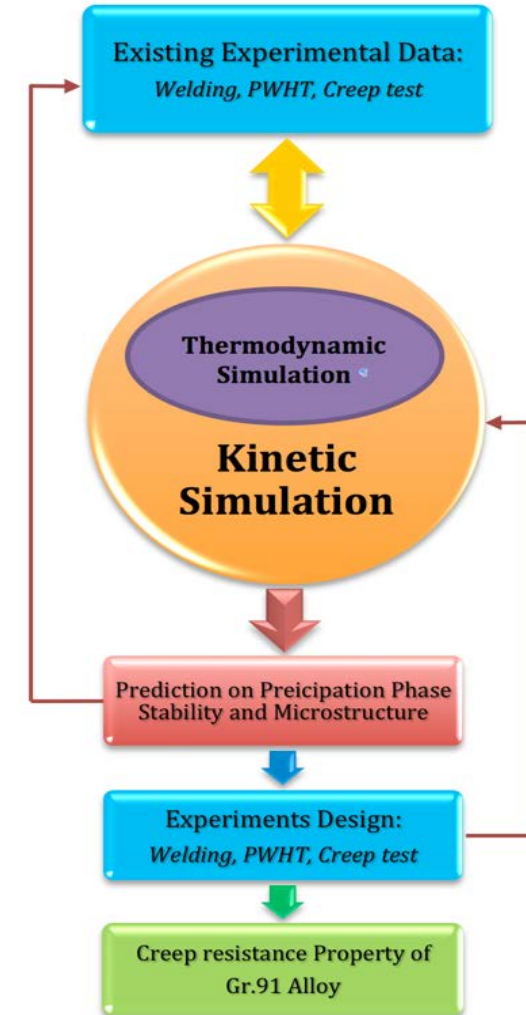
# Approach of Study

## *1<sup>st</sup> Set of Results – Baseline Study (Gr.91)*

- Isopleth Diagrams
- Ac1 and Ac3 Temperatures
- Equilibrium & Scheil Simulations

## *2<sup>nd</sup> Set of Results – Compositional Changes*

- Additional Alloying Element = Mn, Ni, & Ti.
- 3 Different Compositional Changes = V, Nb, & N.





# Part I

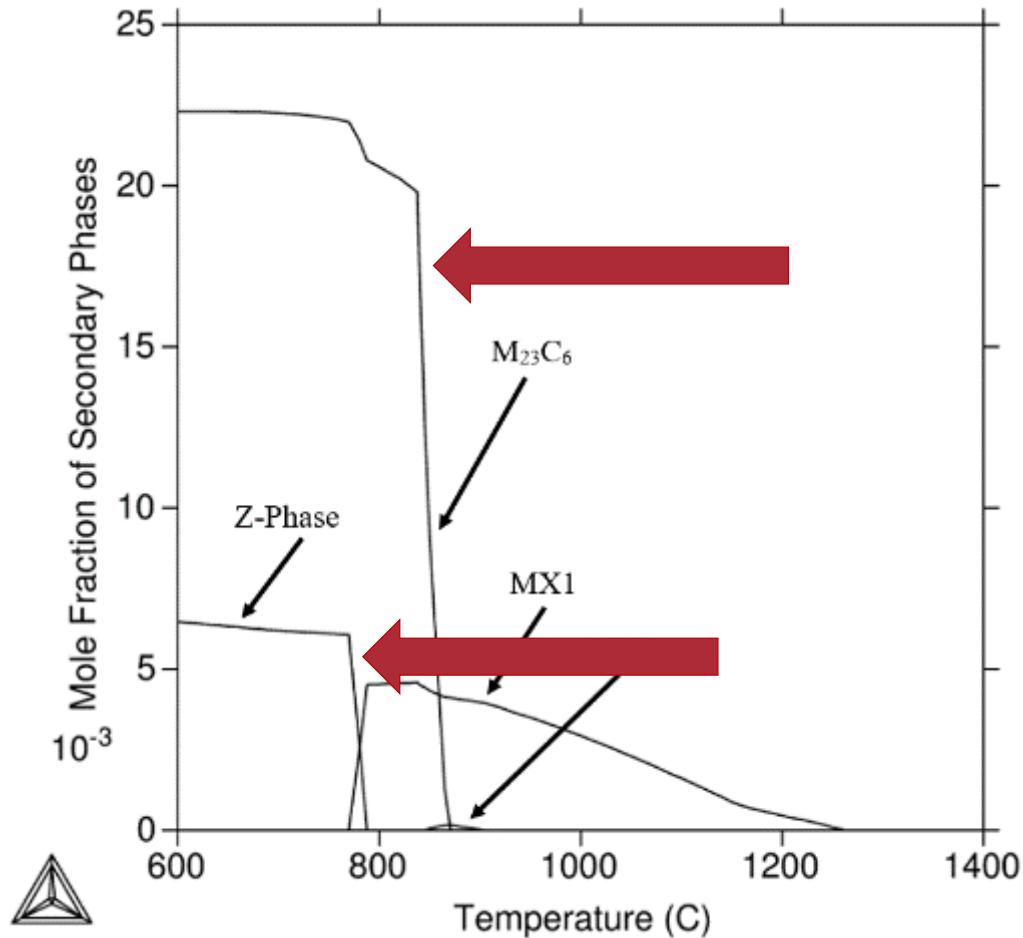
## • **Baseline Study**

- *Factors affecting creep resistance?*
- *Possible ways to improve creep resistance?*
- *Any criteria?*



# Results – Baseline

## Molar Fraction of Secondary Phases



### M<sub>23</sub>C<sub>6</sub>

- Most dominate secondary phase
- <870°C

### Z-Phase

- Stable in lower temperature regions
- <770°C

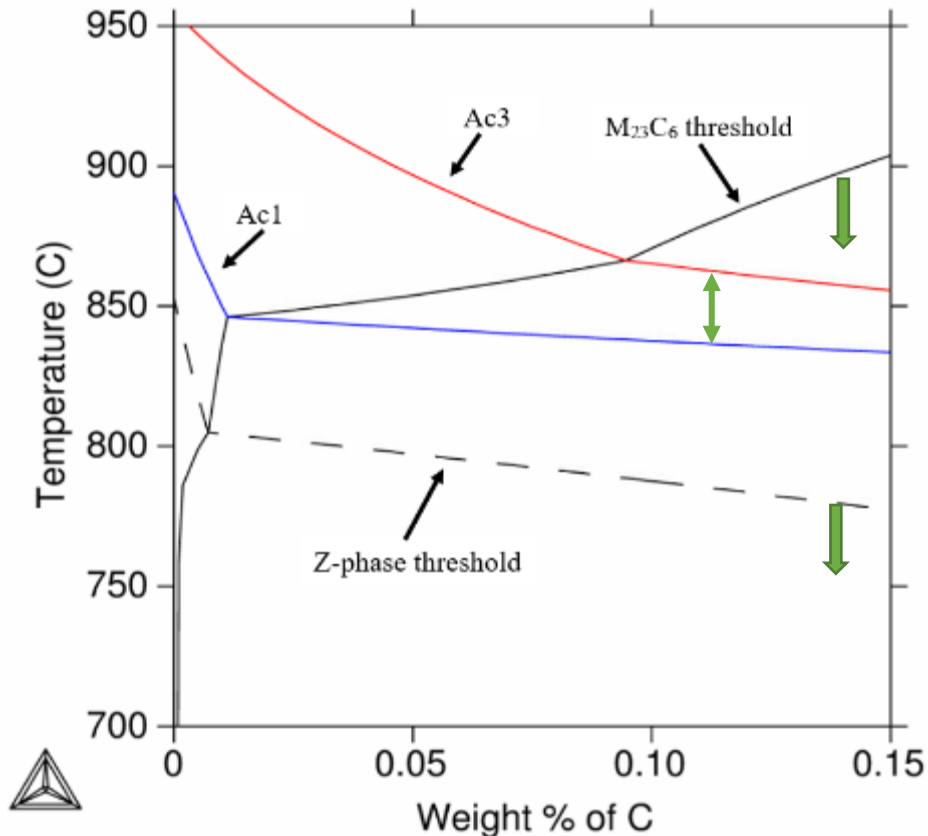
### MX Phases

- MX1 and MX2

### Goal

- **Suppress M<sub>23</sub>C<sub>6</sub> and Z-phases.**
- **Stabilize MX phases.**

## Microstructure



### Threshold Temperatures

- M<sub>23</sub>C<sub>6</sub> (the lower the better)
- Z-Phase (the lower the better)
- Ac1 Temperature
- Ac3 Temperature

### Goal

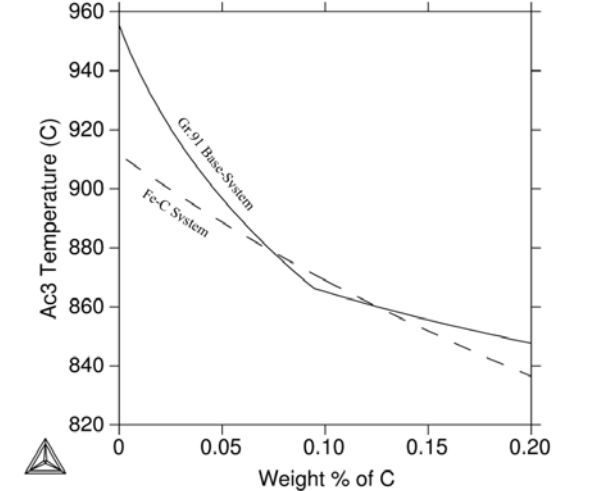
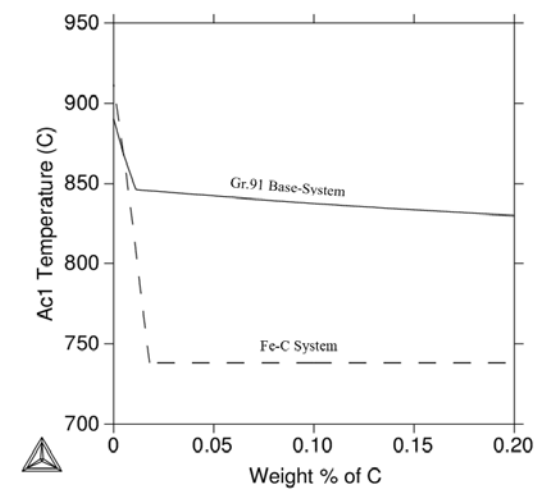
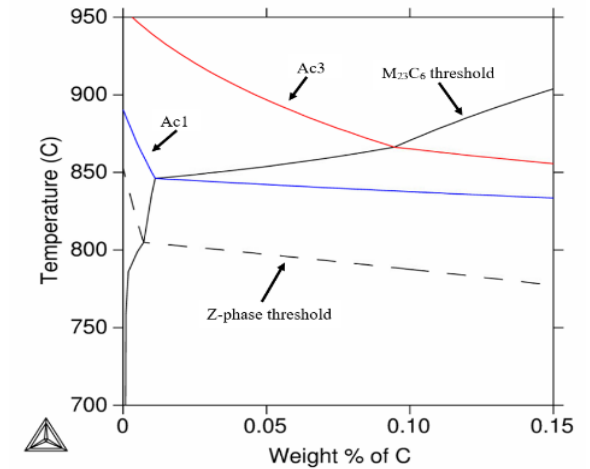
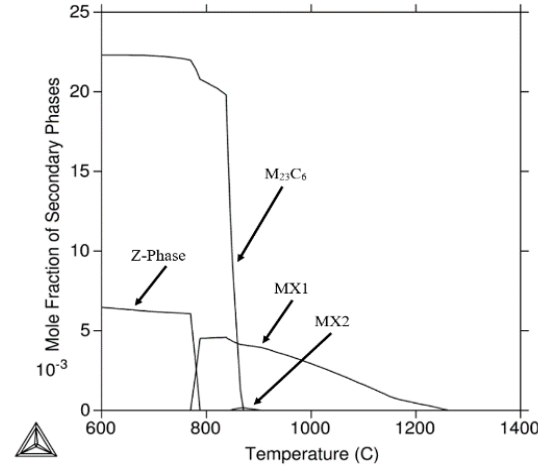
- Change HAZ Microstructure
- Destabilize M<sub>23</sub>C<sub>6</sub>
- Destabilize Z phase

## Baseline Results

- Z phase, M<sub>23</sub>C<sub>6</sub>, MX1 and MX2 are the critical phases
- Ac1, Ac3, M<sub>23</sub>C<sub>6</sub> and Z phase thresholds are critical temperatures and may affect the HAZ microstructure.
- There are both good and bad effects of changing C concentrations.
- Alloying elements greatly changed the Ac1 temperature.

**Baseline Composition**

Elements (wt.%)	Cr	C	V	Nb	Mo	N	Mn	P	S	Si	Al	Ti	Zr
ASME standard	7.90-9.60	0.06-0.15	0.16-0.27	0.05-0.11	0.80-1.10	0.025-0.080	0.25-0.66	0.025 (max)	0.012 (max)	0.18-0.56	0.02 (max)	0.01 (max)	0.01 (max)
Simulation	8.75	.10	.215	.08	.95	.05	-	-	-	-	-	-	-





## Part II

### • **Steel Compositions**

- *Role of each element*
  - *Cr, V, Nb, N, Mn, Ni*
- *Any better compositions?*



# Compositional Changes

## compositional changes:

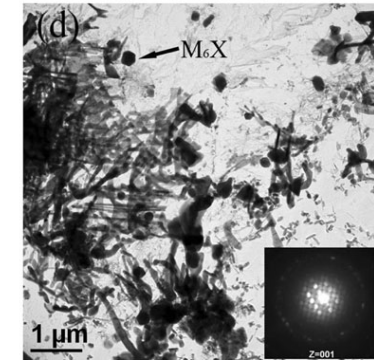
- Cr = 7.0 - 11.0wt.%
- V = .01 - .50wt.%
- Nb = .05 - .25wt.%
- N = .025 - .24wt.%
- Mn = 0.0 - .80wt.%
- Ni = 0.0 - .80wt.%

### Simulation Setup Part 1

Elements (wt.%)	Cr	C	V	Nb	Mo	N	Mn	Ni
ASME standard	7.90-9.60	0.06-0.15	0.16-0.27	0.05-0.11	0.80-1.10	0.025-0.080	0.25-0.66	0.43(max)
Modified Simulation 1	7.0-11.0	.10	.215	.08	.95	.05	-	-
Modified Simulation 2	8.75	.10	.01-.50	.08	.95	.05	-	-
Modified Simulation 3	8.75	.10	.215	.05-.25	.95	.05	-	-
Modified Simulation 4	8.75	.10	.215	.08	.95	.025-.24	-	-
Modified Simulation 5	8.75	.10	.215	.08	.95	.05	0.0-.80	-
Modified Simulation 6	8.75	.10	.215	.08	.95	.05	-	0.0-.80

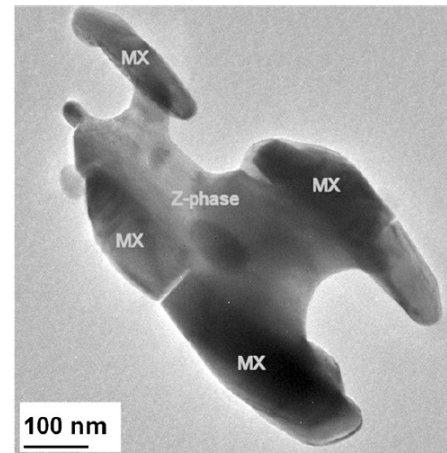
# Cr's Role (Exp)

- Prevent oxidation and corrosion
- Formation of  $M_{23}C_6$  & Z-Phase
  - $Cr_6C$  forms at lower wt.%,  $Cr_{23}C_6$  forms at higher wt.%.
  - More Z-Phase is observed at higher wt.% Cr (*Sawada, K., 2008*)

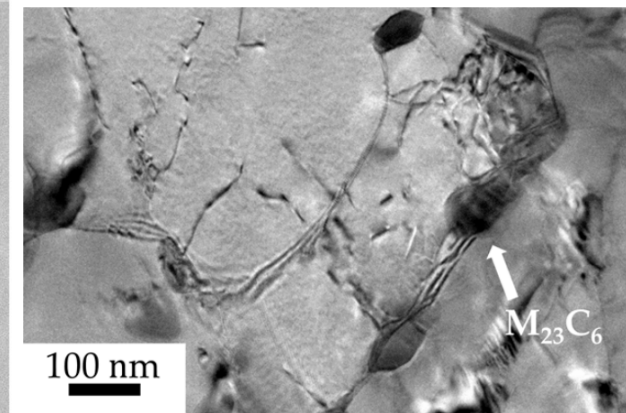


2.25wt.%Cr  
Gr.22

Yong Yang, 2010



Danielsen 2009

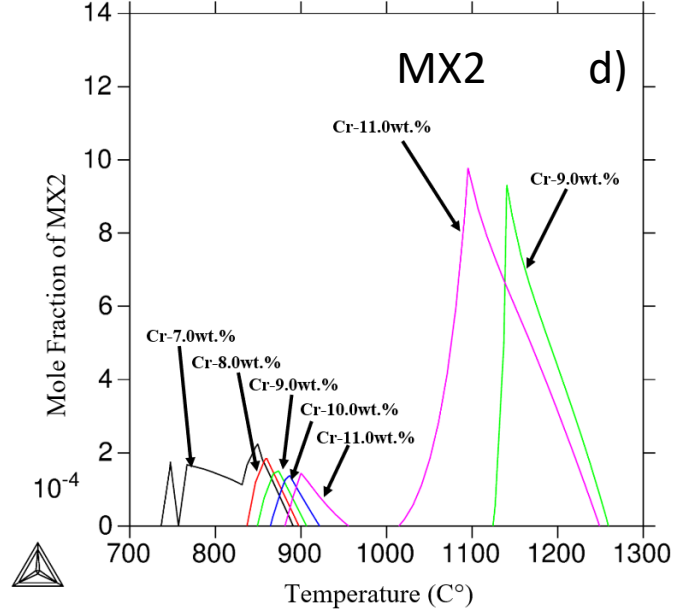
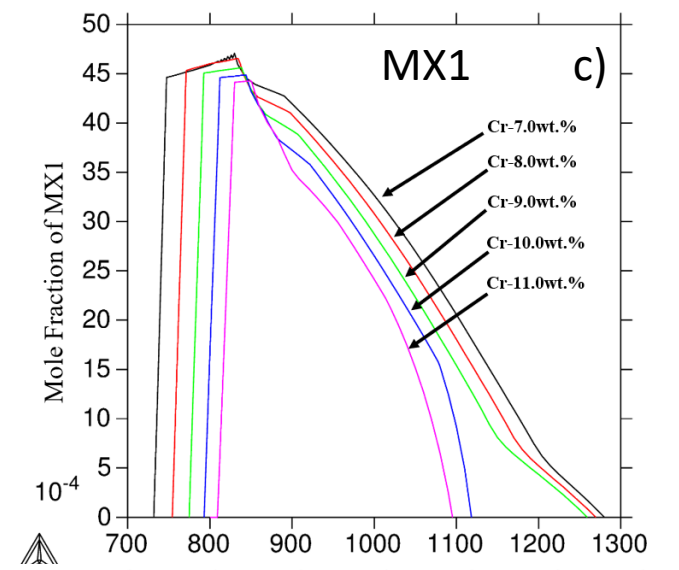
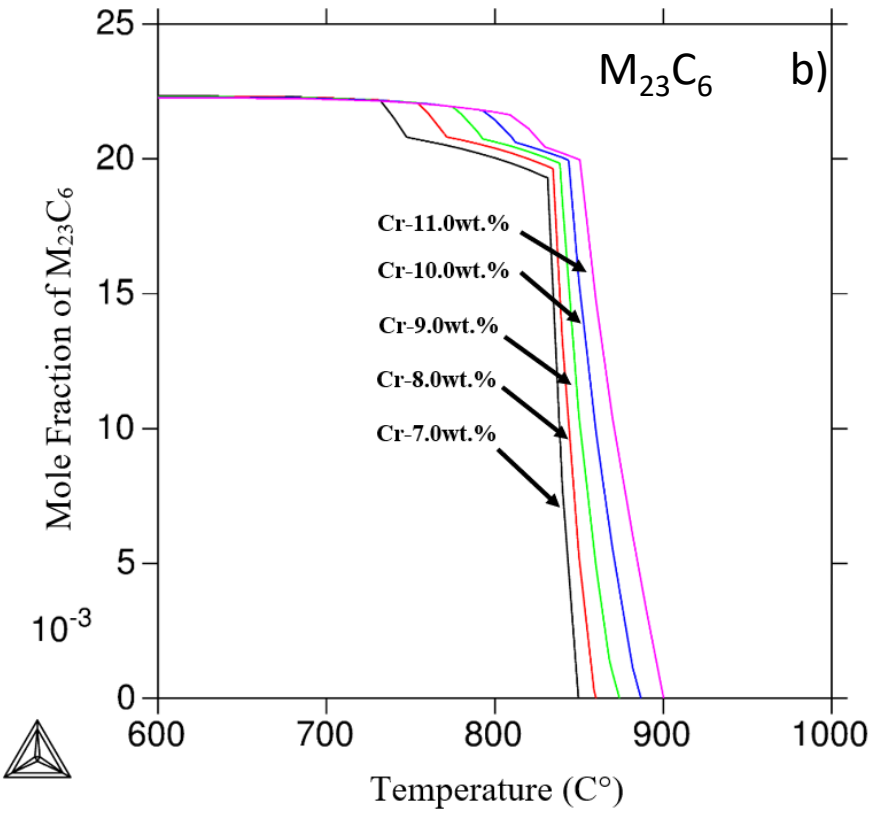
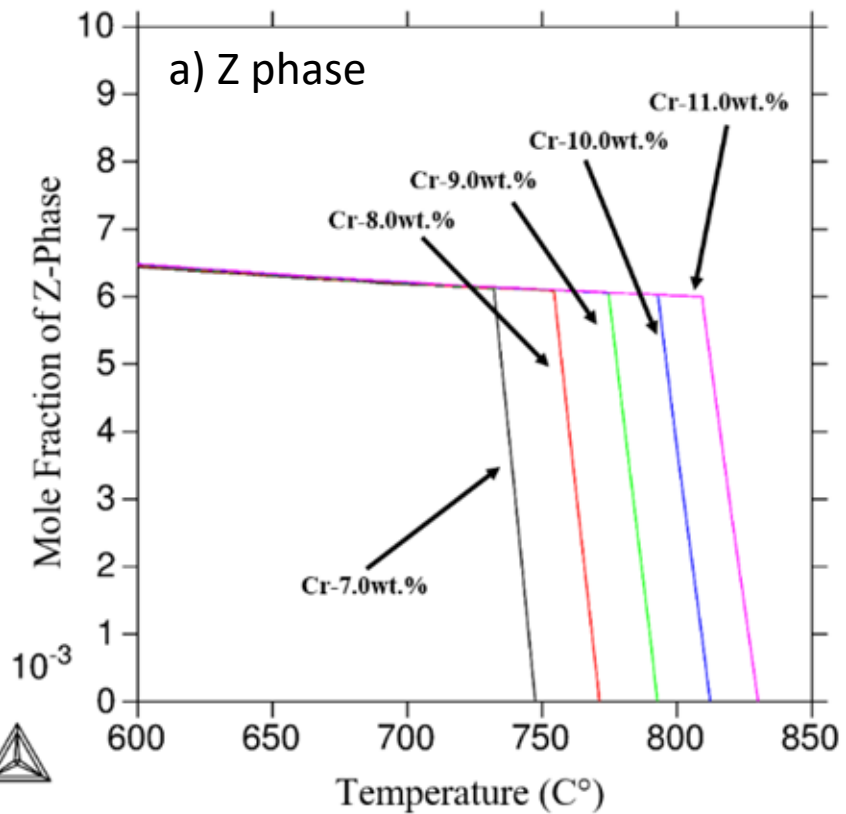


9wt.%Cr  
Gr.91

Javier Vivas, 2017

# Cr's Role (Simulation)

Mainly affect the Z phase formation



# V & Nb's Role (Exp)

## Vanadium (V)

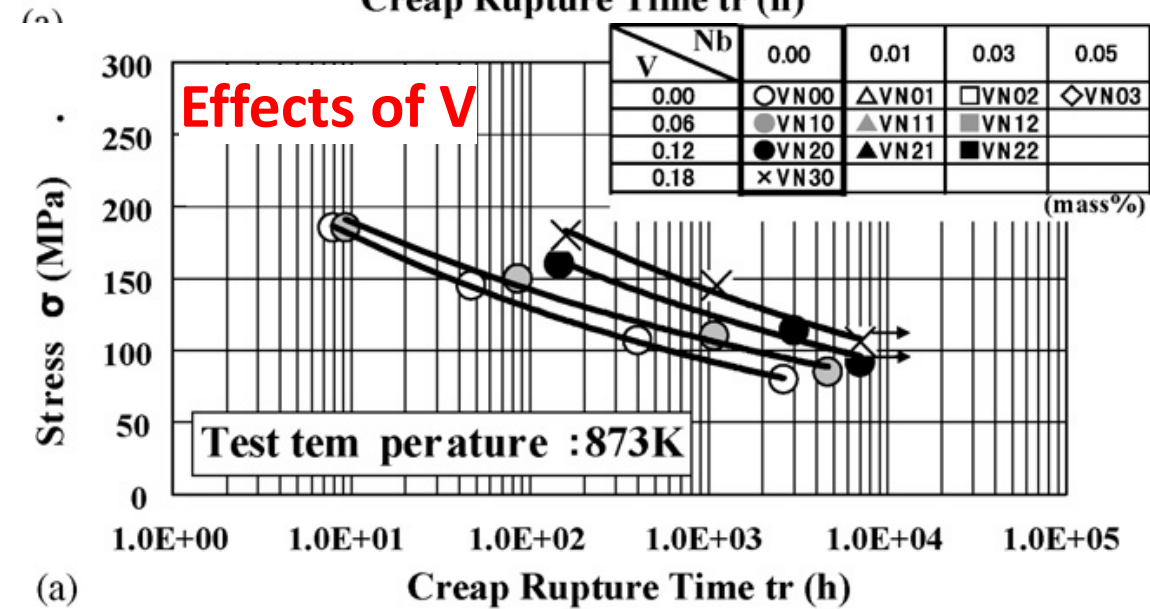
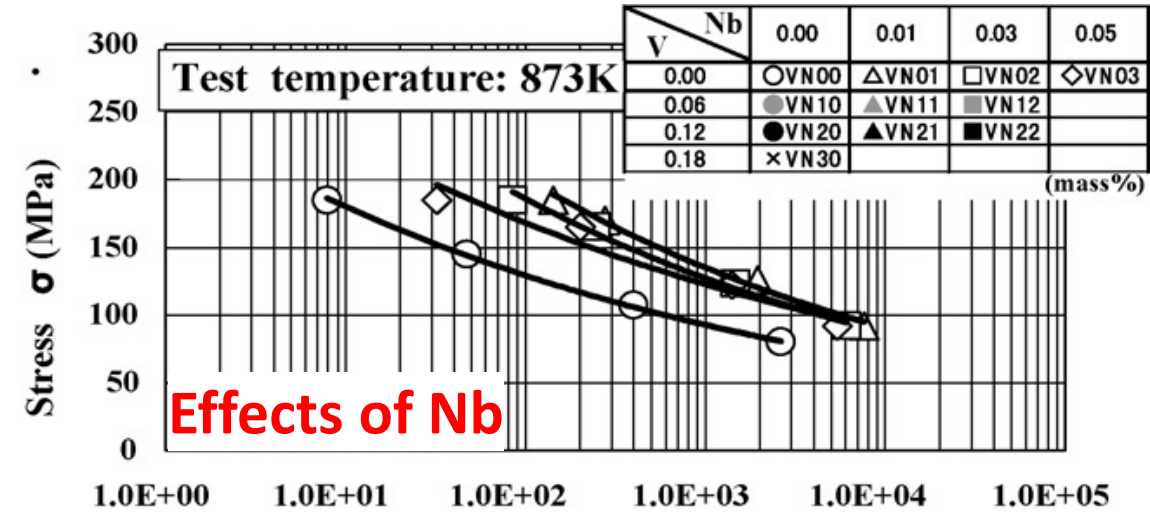
- MX Phase V(C,N) (MX1)

## Niobium (Nb)

- MX Phase Nb(C,N) (MX2)

## Both

- increase in creep life and MX density in matrix.
- reduce  $M_{23}C_6$  coarsening with increased wt.% V
- by replacing Mo and W (Yuanta Xu, 2016)

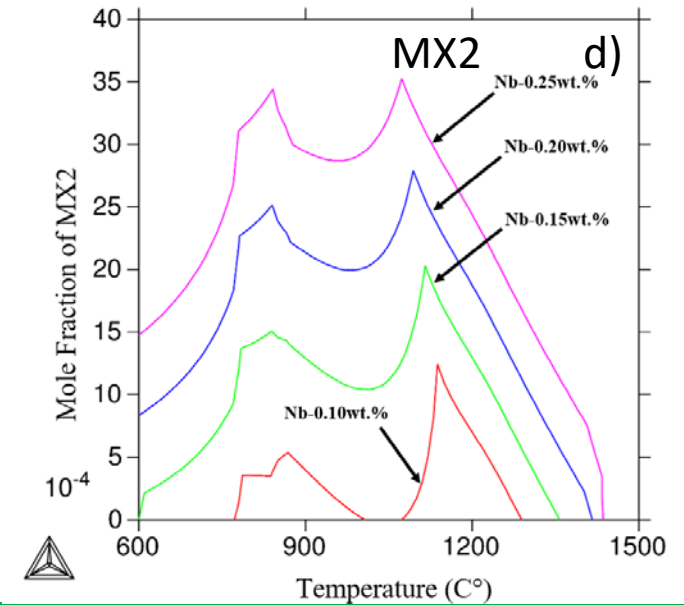
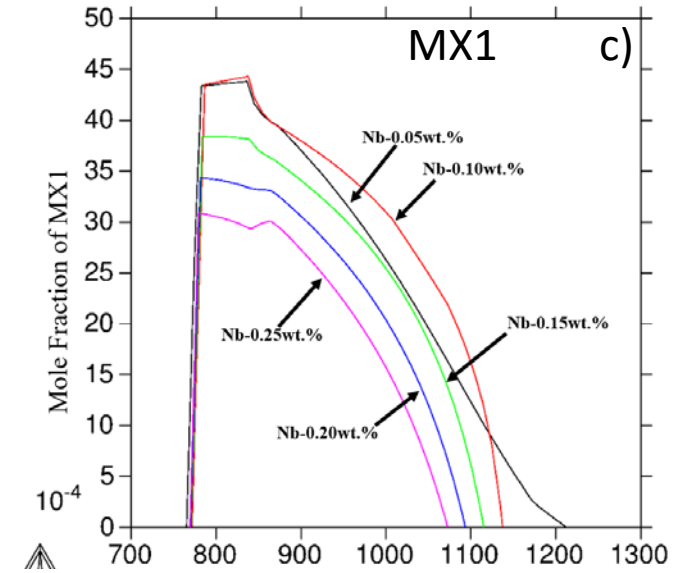
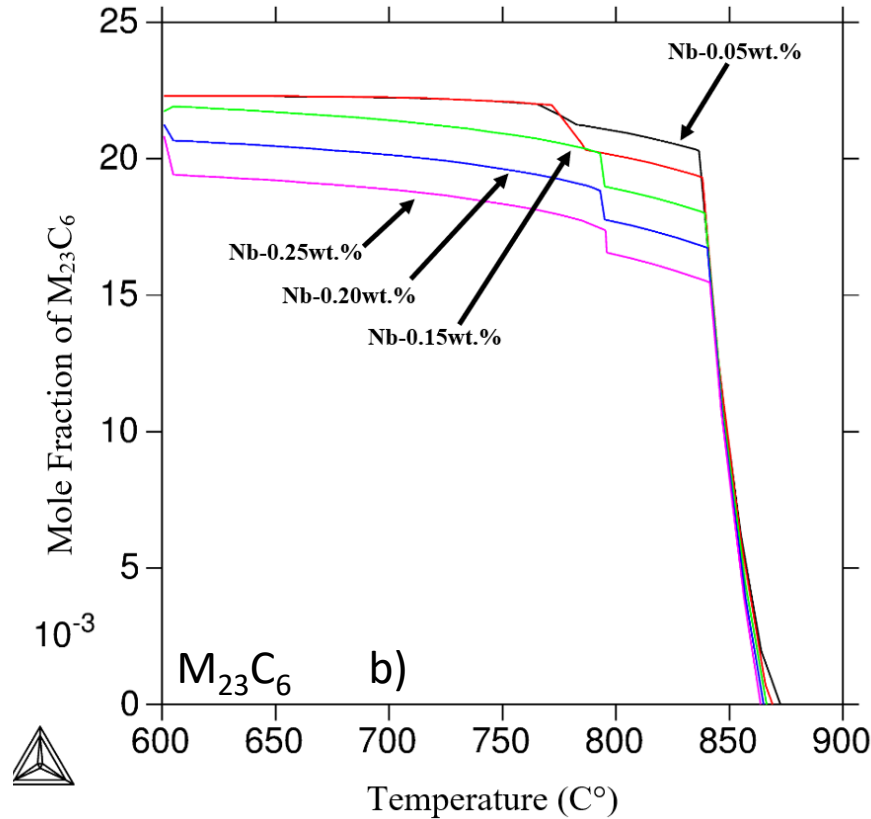
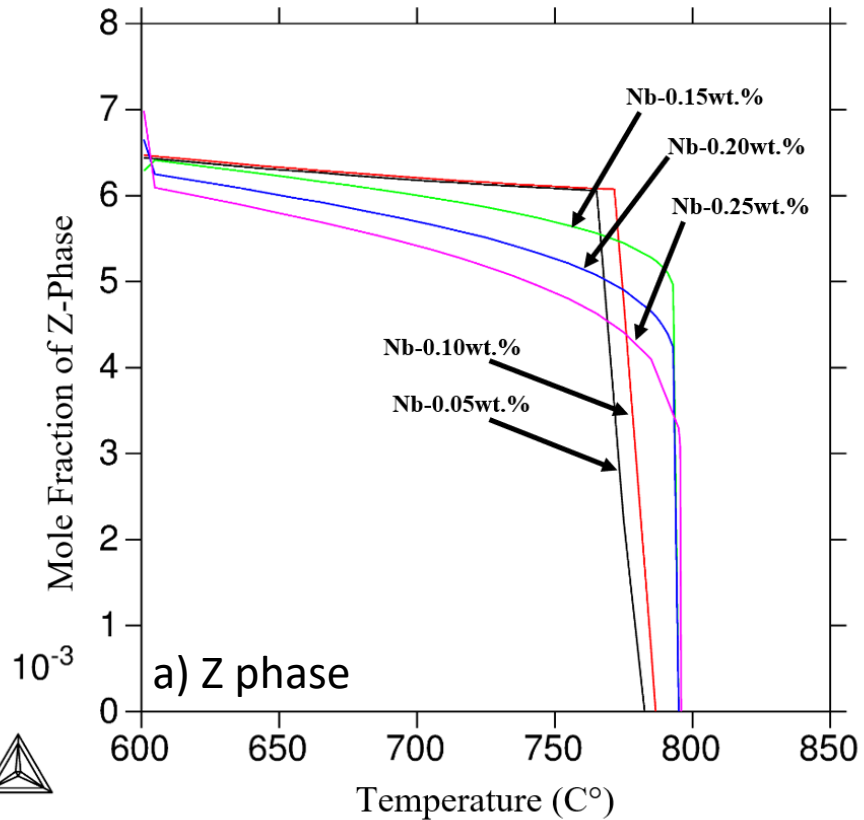




# Nb's Role (Simulation)

With the increase of Nb:

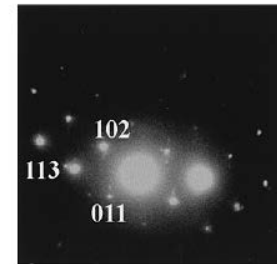
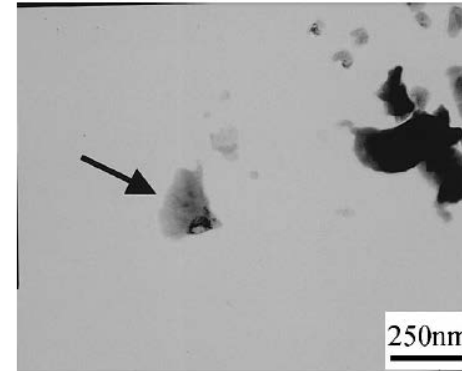
- M<sub>23</sub>C<sub>6</sub> is less stable
- MX<sub>2</sub> phases become more stable at lower T



# N's Role (Exp)

## Nitrogen (N)

- Nitrogen helps the formation of MX and Z-phase formation
- Increasing N (up to 0.07wt.%N) helps **short-term creep tests** with more MX nitrides [Vaclav]
- increasing N is detrimental to For **long-term creep tests** as it helps the precipitation of Z-phase and  $Cr_2N$  [Sawada]



B=[211]

Fig. 10. TEM image of carbon extraction replica in 7N steel and electron diffraction pattern from Z-phase. Arrow indicates Z-phase. The replica was obtained from specimen crept for 1187 h at 923 K.

Kota Sawada, 2004

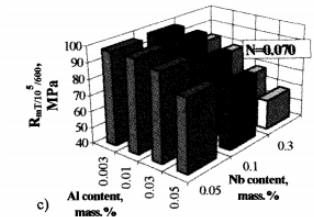
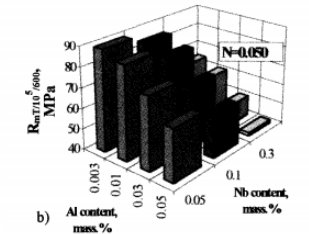
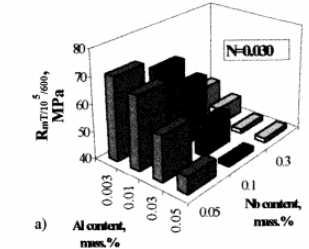
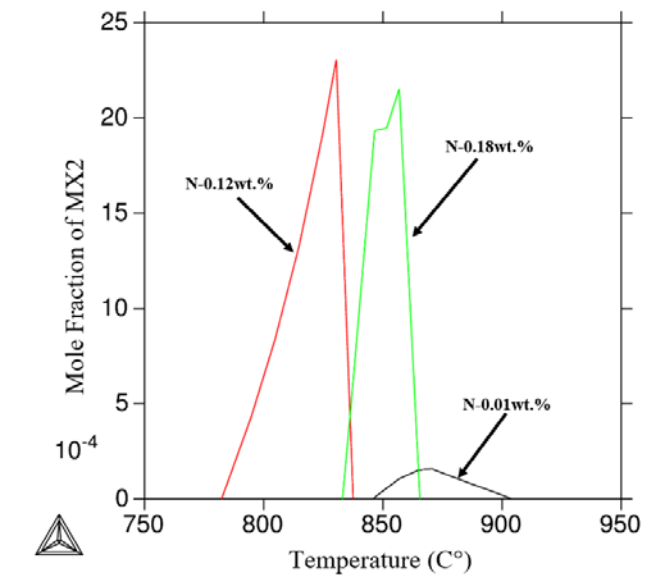
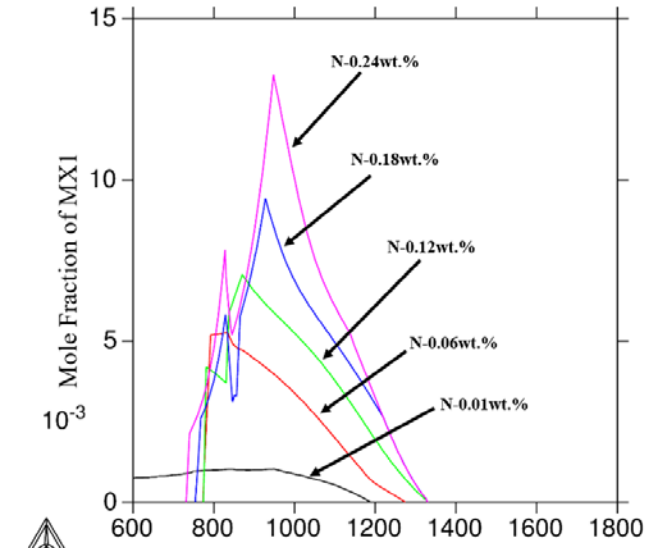
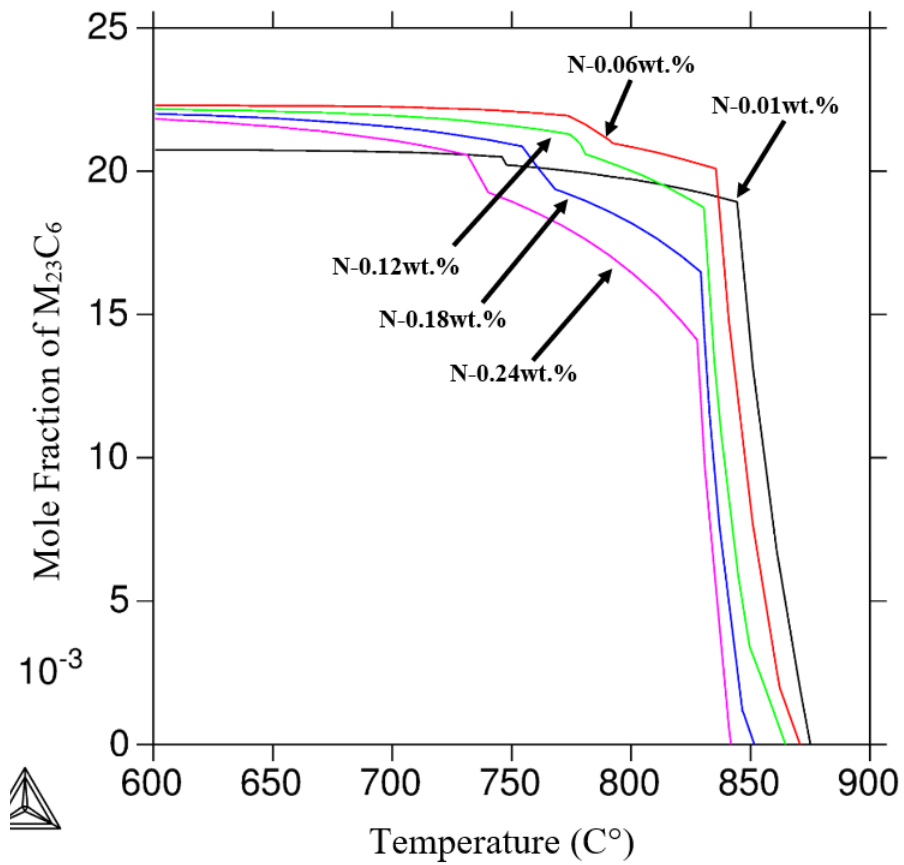
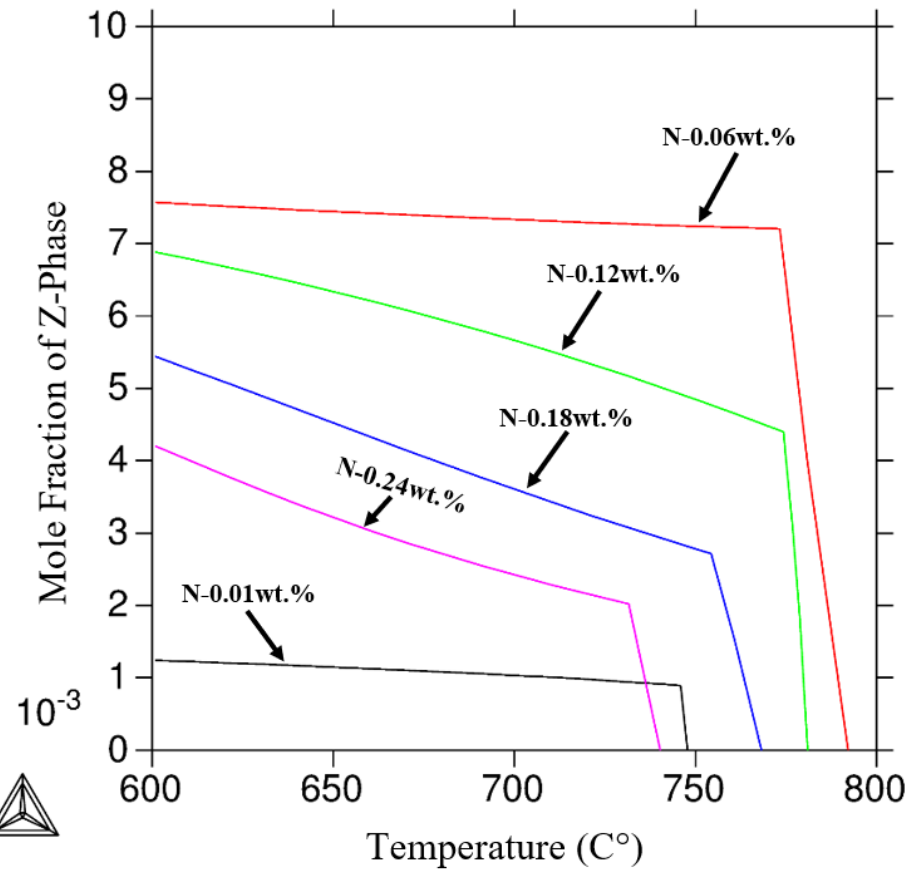


Fig. 1: Creep rupture strength of the model 0.10C-9Cr-1Mo steel in dependence on nitrogen, aluminium and niobium contents  
 a) for N=0.030%  
 b) for N=0.050%  
 c) for N=0.070%

Vaclav Foldyna, 2001

# N's Role (Simulation)

Extremely low N is beneficial to reduce the formation of Z phase  
 Extra addition of N will cause the formation of detrimental Hcp (Cr<sub>2</sub>N)



# Compositional Changes – Simulation Setup Part 2

## Optimized Simulation

- To give an optimized version of Grade 91 based on results and *within the parameters*.
  - Min Cr
  - Max V, Nb, Mn, and Ni

## Modified Simulation

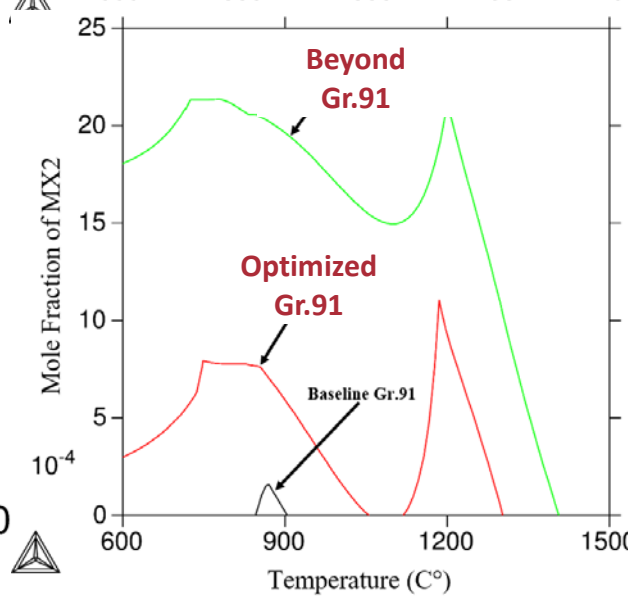
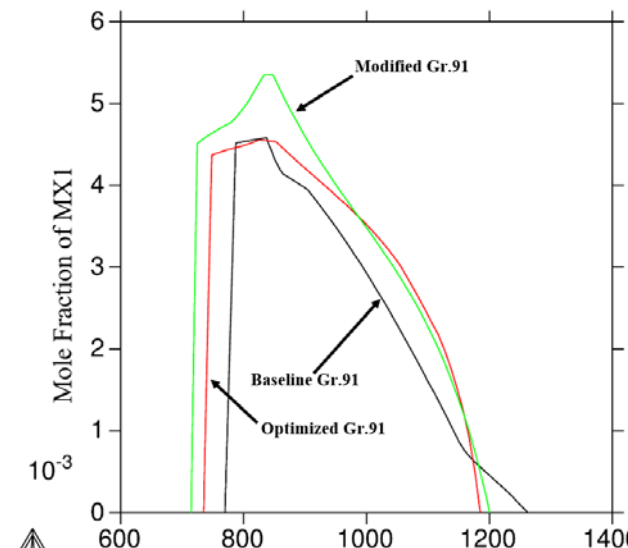
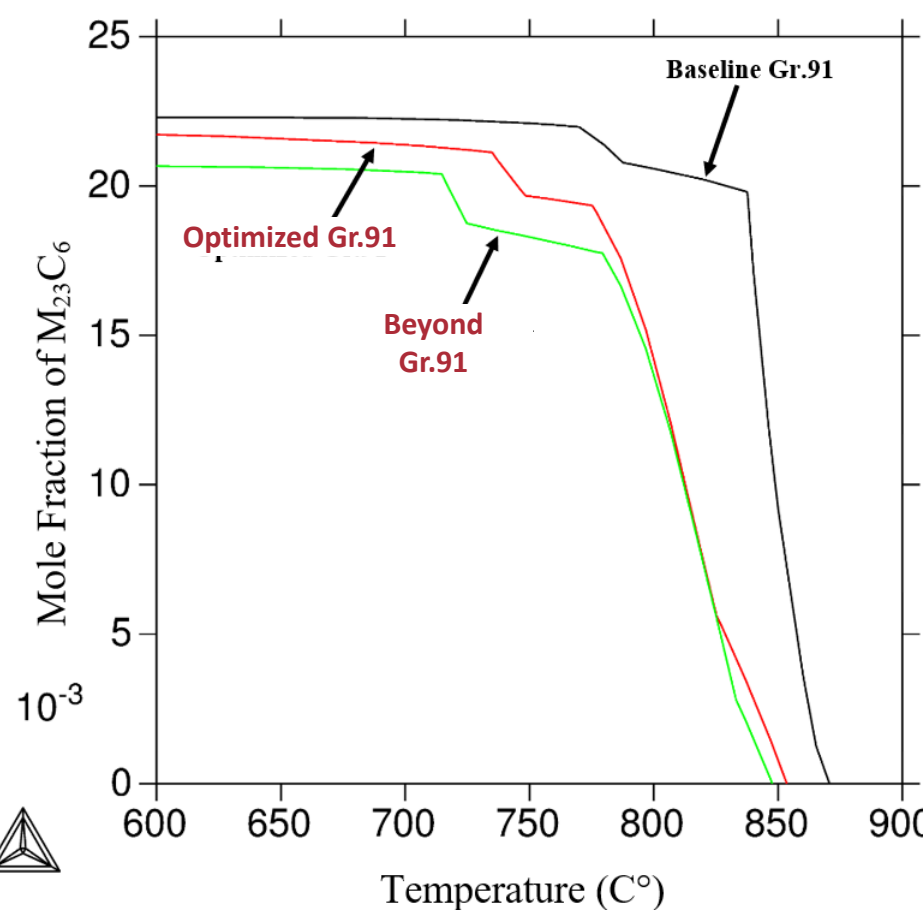
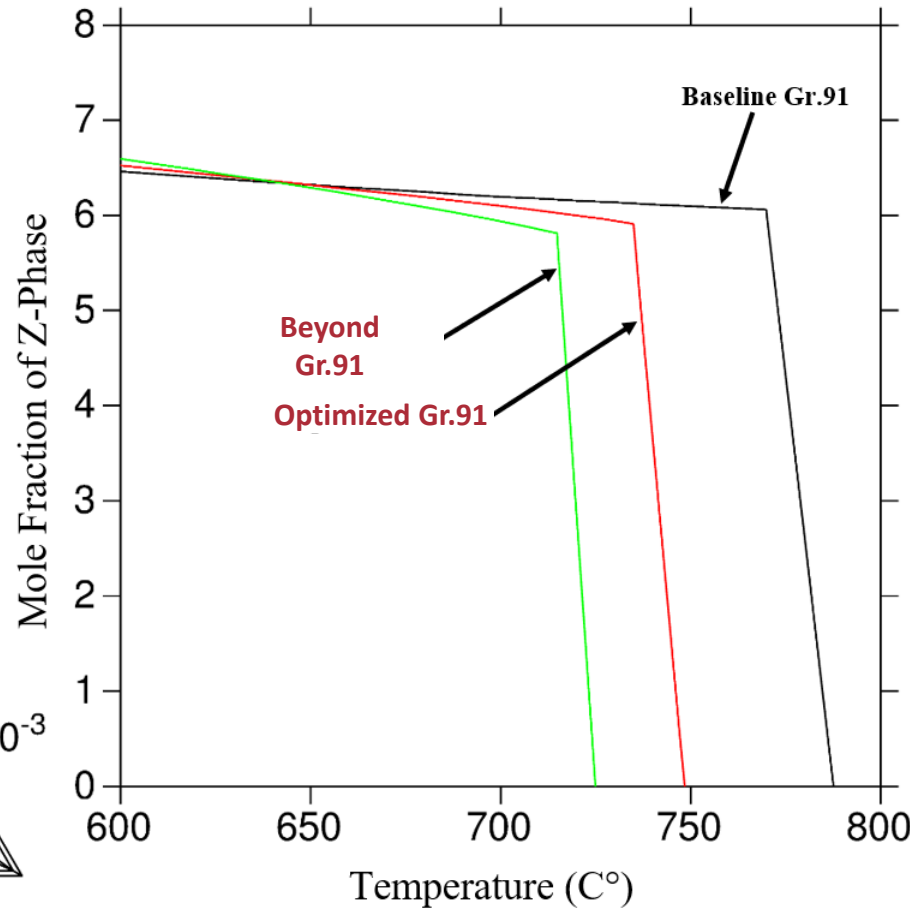
- To give an optimized version of Grade 91 based on results and *outside the parameters*.
- This is to established that certain elements in the system, when increased, suppress certain secondary phases establishing a pattern.

Simulation Setup Part 2

Elements (wt.%)	Cr	C	V	Nb	Mo	N	Mn	Ni
ASME standard	7.90-9.60	0.06-0.15	0.16-0.27	0.05-0.11	0.80-1.10	0.025-0.080	0.25-0.60	0.43(max)
Baseline Simulation	8.75	.10	.215	.08	.95	.05	-	-
Optimized Gr.91	7.9 ↓	.10	.27 ↑	.11 ↑	.95	.05	.66 ↑	.43 ↑
Beyond Gr.91	7.9 ↓	.11	.40 ↑	.20 ↑	.95	.05	.66 ↑	.43 ↑



# Compositional Changes Optimized & Modified





## Part III

# • Experimental Investigation

- *Processing Parameter Design*
- *Real creep test*
- *Microstructure analysis*



# Conclusions

From the optimized version of Gr.91:

- V & Nb = Increase MX
  - Mn & Ni = decrease Ac temperatures
- 
- It is clear that V & Nb play an important role in the stability of MX phases, however it decreases the overall ductility and toughness of the steel due to previous works.
  - Decreasing Cr seems to be beneficial for the creep resistance of Gr. 91 alloy, however, this can affect other properties like oxidation resistance.



# Acknowledgements

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