Operating Stresses and Their Effects on Degradation of LSM-Based SOFC Cathodes

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Project objectives

LSM* cathodes in solid oxide fuel cells (SOFCs)

- Seeking correlations between microstructure and performance changes
- Testing under aggressive conditions
 - Show effects of non-ideal operating conditions
 - Replicate effects of much longer conventional conditions?

Microstructure

3-D reconstruction: **3DR**

Phase fraction profiles

Three-phase boundaries: **TPB density**

Transmission electron microscopy & elemental mapping: *TEM/EDXS*

*) Lanthanum strontium manganite, $(La_{1-x}Sr_x)_yMnO_{3\pm\delta}$



Performance

Area specific resistance vs. time: **ASR(t)**

Durability testing under aggressive conditions

Electrochemical impedance spectroscopy:

Outline

- Effect of *Mn excess* (A-site deficiency) on performance
- EIS analysis
- **Conventional** vs. **aggressive** testing
 - Microstructural evolution
 - Cell performance

- New activities
 - Testing under *low poz*
 - **Aging** tests baseline for effects of T, t
 - Reproducibility
 - Effects of *current load cycling*
 - Effects of *ambient conditions*
 - Humidity
 Barometric pressure
 Inlet air temperature



Cell specifications; testing procedures

This study:

- Durability and aging tests
- Conventional or aggressive
 conditions
- LSV sweeps and EIS runs ⇒
 current cycling every 24 h

Button cells:

- 8YSZ electrolyte Ni/8YSZ anode
- Cathodes: LSM / 8YSZ
 - (La_{0.85} Sr_{0.15})_{0.90} MnO_{3±δ} (LSM 85-90) **11%** Mn excess
 - (La_{0.80} Sr_{0.20})_{0.95} MnO_{3±δ} (LSM 80-95) 5% Mn excess
 - (La_{0.80} Sr_{0.20})_{0.98} MnO_{3±δ} (LSM 80-98) 2% Mn excess



temperature [°C]	current density [mA cm ⁻²]	cathode p _{O2}
900	380	0.2
	OCV (aging)	
1000	760	
	OCV (aging)	
900	380	0.1
	OCV (aging)	
1000	760	
	OCV (aging)	



Prior work, conventional conditions: TEM/EDXS

cells tested at 800 °C; 11% Mn excess cathodes



*) Left side of each image

1) H.-J. Wang et al., 14th SECA Workshop, Pittsburgh, Pennsylvania, July 2013. 2) H.-J. Wang et al., Metall. Mater. Transactions E: Materials for Energy Systems 1 [3] 263-271 (2014).



11% Mn excess: TEM w/EDXS, 0-500 h testing in air

• **As received** (0 h)

 MnO_x seen sparingly across entire cathode





• 493 h aggressive test

- MnO_x near cathode / electrolyte interface
- MnO_x coarsened in CCC (cathode current collector)

For 11% excess Mn, 500 h aggressive testing reproduced some of the microstructural changes of 8–16 kh of conventional testing.



5% and 2% Mn excess: little or no MnO_x



Phase profiles across cathodes from 3DR



As-received and 500-h conv'l testing: uniform phase profiles *Porosity gradients,* lowest at e'lyte interface

Microstructural parameters from 3DR



11% Mn excess:

Pore coarsening; lowest TPB density •

5% Mn excess:

Some pore coarsening; stable TPB • density

2% Mn excess:

Pore coarsening; drop in TPB density ٠



2%



ASR and TPB density: role of Mn excess (aggr. testing, air)

- As Mn excess ↓,
 ASR↓
- As test time ↑:
 - Active TPB ↓
 - Total ASR ↑
- Effects on ASR diminish as Mn excess ↓



active **cathode** TPB density [μ m⁻²]



Summary: microstructural evolution and performance

- LSM 85–90 (11% Mn excess), 500 h of aggressive* testing in air:
 - Reproduced microstructure changes of 8 kh, conventional operation
 - *Loss of porosity* at cathode–electrolyte interface
 - *MnO_x accumulation* near cathode–electrolyte interface
 - Highest loss of active TPB density
 Highest rise in electrode ASR
- Do these findings extrapolate to conventional test conditions?

*) 1,000 vs. 900 °C; 760 vs. 380 mA cm⁻²



ASR and TPB density: role of Mn excess (conv'l testing)

- All compositions:
 ASR ↑ as t ↑, but microstructure change was slight
- 11% Mn xs: expected trend
- 5% Mn xs, ASR ↑
 as t ↑ at const.
 TPB density
- 2% Mn xs:
 inverse of expected trend





Electrochemical impedance spectroscopy

Fitting an arc with a single semicircle

Fitting multiple arcs with overlapping semicircles





Effect of Mn excess: air, aggressive testing



ASR EIS is the sum of the gray, orange, and green, minus red curves.

- ASR EIS gave good agreement with ASR DC from durability testing:
 - $(\pm 0.03 \ \Omega \ cm^2 \ for \ 5\% \ Mn \ xs)$ $(\pm 0.02 \ \Omega \ cm^2 \ for \ 11\% \ Mn \ xs)$
- Rise in ASR DC with time comes from series resistance R_s, not from R_p

Reproducibility: LSM 85–90 (11% Mn xs), aggressive, air



• ASR EIS gave good agreement with ASR DC $(\sim -0.04 \ \Omega \ cm^2)$.

• ASR EIS gave excellent agreement with ASR DC $(\pm 0.01 \ \Omega \ cm^2)$.

~20% difference in total ASR (0.1 Ω cm²) between identical cells



Aggressive vs. conventional: LSM 80–98 (2% Mn xs)



- All ASR components rose ~50% at 900 °C vs. 1,000 °C.
- All ASR components *rose with t*, but more strongly at 900 °C
- High frequency: inductive component ⇒ lower ASR EIS vs. ASR DC
- EIS and DC ASR still agree within 0.06 Ω cm².



Conclusions: ASR, microstructure, and EIS analysis

- Microstructure and ASR
 - *Electrode ASR* and *active TPB density:* mostly inverse relationship
 - ... but does not separate the *anode and cathode losses*, ...
 - ... nor the *contribution of YSZ* to the electrode ASR part of R_s ?
 - 11% Mn excess LSM cathodes:
 - largest microstructure changes
 - strongest ASR rise (500 h aggressive testing in air)
- Cautions about ASR comparisons between 900 and 1,000 °C
 - ASR degraded at 900 °C, despite stable cathode microstructures
 - Rise in $R_{\rm s}$ not $R_{\rm p}$ accounts for rise in ASR
 - Cathode (low-frequency) R_p was not the major source of ASR consistent with the higher testing temperature



Test fixture for controlled cathode atmosphere



• Testing in progress



Test fixture for controlled cathode atmosphere





Optical profilometry of SOFCs

- Nanovea ST400
 - Quantitative topographical information
 - Scan much larger areas than electron microscopy $mm^2 vs. \mu m^2$
 - Optical Pen 1: lateral accuracy = $1.1 \, \mu m$
 - Optical Pen 3: lateral accuracy = $2.6 \,\mu$ m





Optical profilometry: cross section (fracture surface)





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