

Microchannel Heat Exchangers Based on Alloy 230 – Exposure Characteristics and Mechanical Behavior

FWP 1022406 –Advanced Alloy Development
Period of performance: 10/1/2016 – 9/30/2017

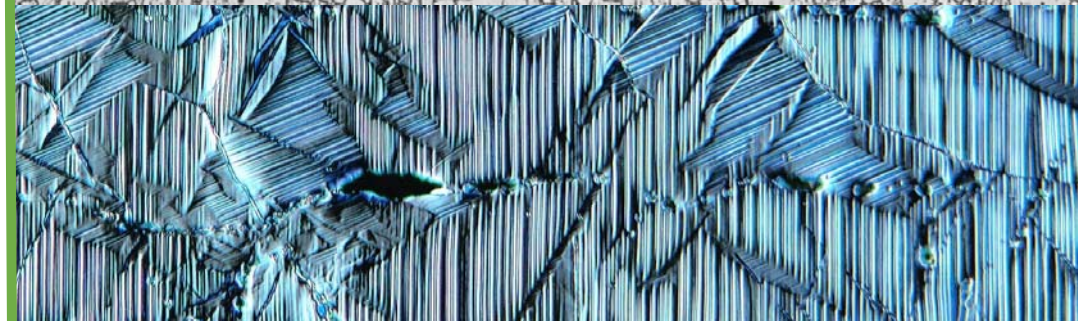
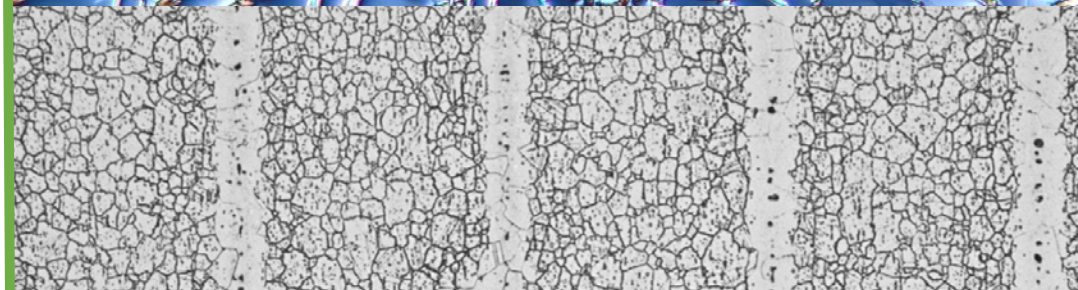
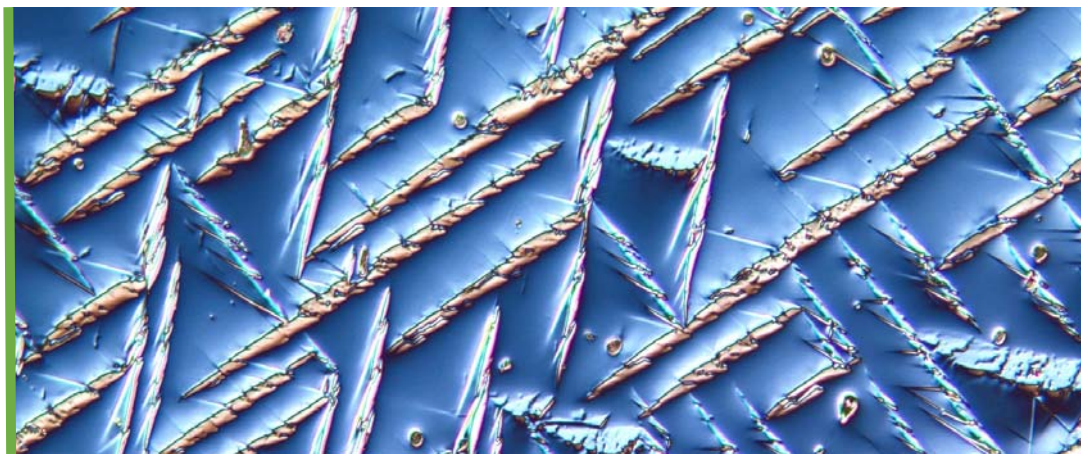
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Crosscutting Technology Research Project Review
March 22, 2017 – Pittsburgh, PA



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Project Goals and Objectives



- **Accelerating commercialization of supercritical carbon dioxide power cycle technology by addressing materials and manufacturing challenges for components of sCO₂ power systems.**
 - Joining of similar and dissimilar power plant alloys
 - Performance of joints in sCO₂ power cycle environments
- **Milestones**
 - Demonstrate mechanical and environmental performance of joints in supercritical CO₂ - 09/30/2017
- **Deliverables**
 - Technical report (either presentation or publication) on the high temperature corrosion of the joined power plant materials. - 09/30/2017
 - Technical report (either presentation or publication) on the cross-joint strength of typical power plant materials. - 09/30/2017

Outline

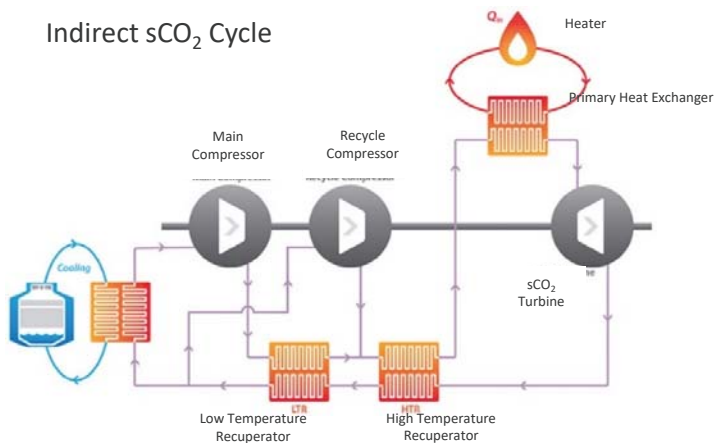


- **Supercritical CO₂ power cycles and compact heat exchangers**
- **Materials issues in manufacturing compact heat exchangers**
 - Diffusion bonding (DB)
 - Transient liquid phase bonding (TLPB)
- **Mechanical strength of bonded structures**
- **High-temperature corrosion of bonds in sCO₂**

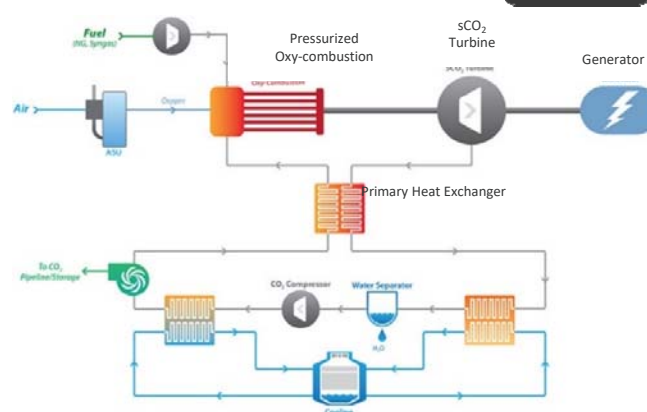
Supercritical CO₂ Power Cycles



Indirect sCO₂ Cycle



Direct sCO₂ Cycle

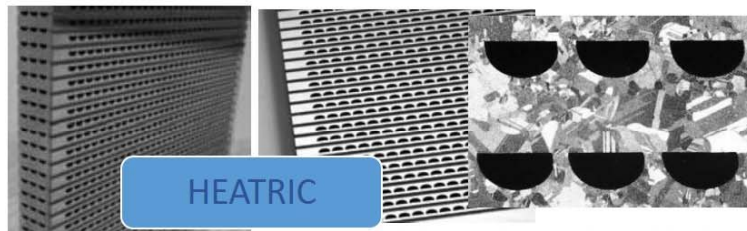


Cycle/Component		Inlet		Outlet	
		T (C)	P (MPa)	T (C)	P (MPa)
Indirect	Heater	450-535	1-10	650-750	1-10
	Turbine	650-750	20-30	550-650	8-10
	HX	550-650	8-10	100-200	8-10
Direct	Combustor	750	20-30	1150	20-30
	Turbine	1150	20-30	800	3-8
	HX	800	3-8	100	3-8

Essentially pure CO₂

CO₂ with combustion products including O₂, H₂O and SO₂

Compact Heat Exchangers for sCO₂ Cycles

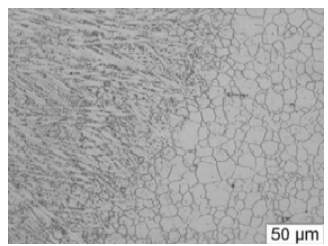
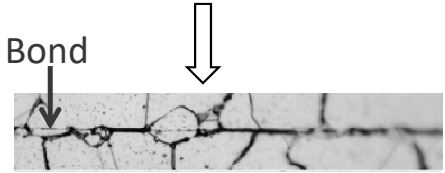
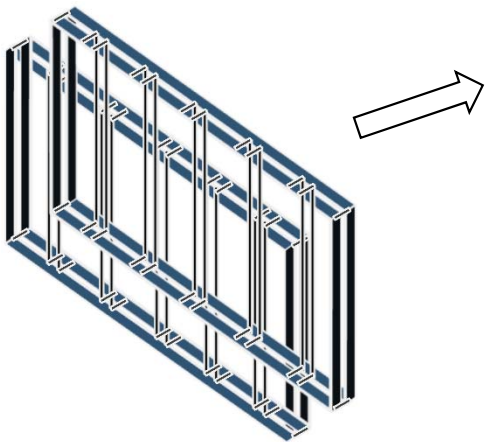


- Higher efficiency
 - Due to much shorter heat diffusion lengths in fluid
- Smaller size
 - Use of less materials (expensive superalloys)
 - Takes less space
- Modular design
 - Expandable to large power plants

Heat Exchangers

Micro channel heat exchangers

- Higher heat transfer efficiency due to shorter heat diffusion lengths
- Smaller size
- Modular Design

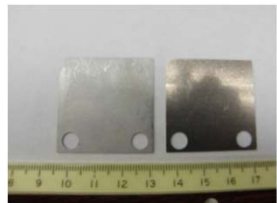


- Dimensional Tolerances
 - Pressure & Temperature Drop
- Uniform microstructure
 - Environmental Resistance

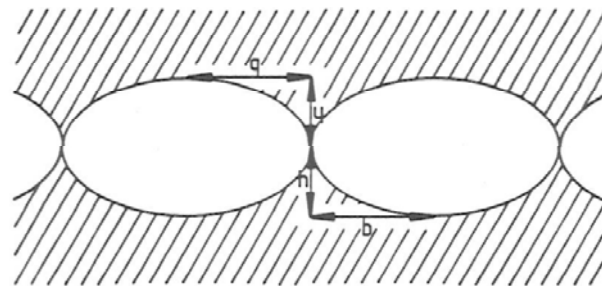
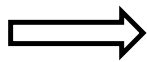
Diffusion bonding, brazing, and transient-liquid-phase (TLP) bonding are the most robust approaches for sCO₂ cycles

- Pattern microscale flow paths
- Join these using laser welding, diffusion bonding or brazing

Schematic of diffusion bonding process



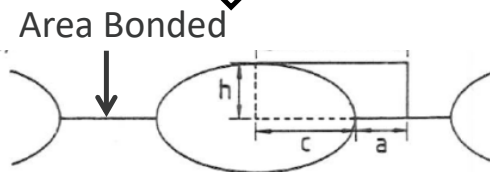
550 μm shims



Parallel, elliptical voids, contact between ridge tops

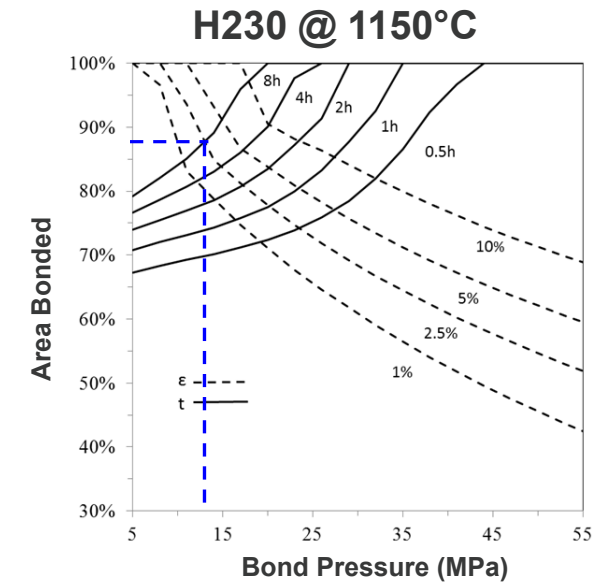


Pressure & Temperature



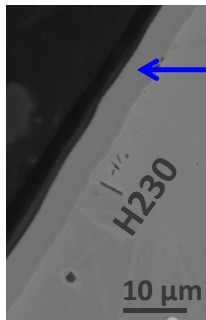
Area Bonded

- Solid state process
- Applied high pressure at high temperature
- Void closure due to diffusion of constituent atoms and creep processes by application of high pressure

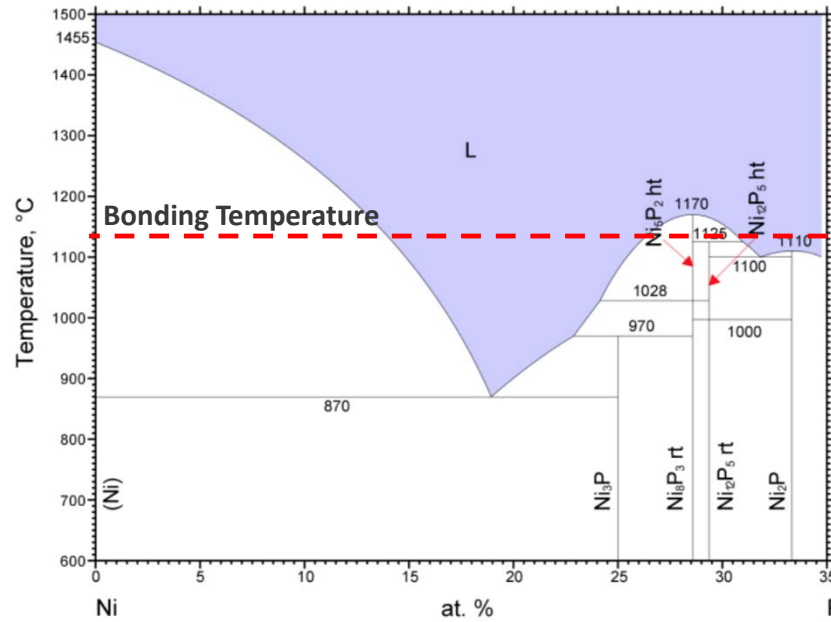


Balance creep, time, pressure and bonded area

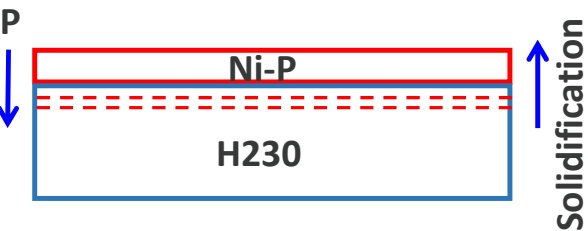
Transient-Liquid-Phase Bonding



Ni-P plating acts as a low-melting interlayer

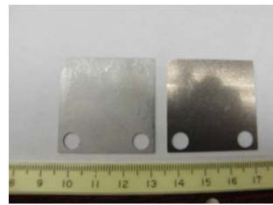


- Both solid state and liquid state reactions
- Less pressure than diffusion bonding
- Lower melting point interlayer
- Isothermal melting and solidification of interlayer

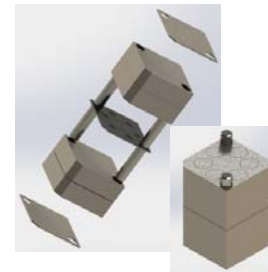


Output of Bonding - stacks

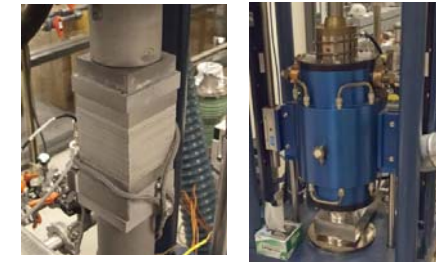
- Downselect between H230 & H282
- H230 – NiCrW solid solution strengthend alloy
- H282 – Gamma prime strengthened alloy
- H230 was selected and challenges with H282 will be discussed later



Cold rolled and 1232 °C solution annealed - 550 μ m H230 shims



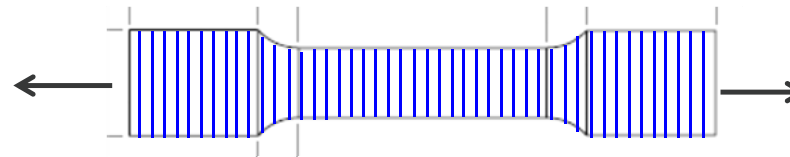
Stacked onto a fixture



Pressed at 1150°C, 12.7MPa in vac



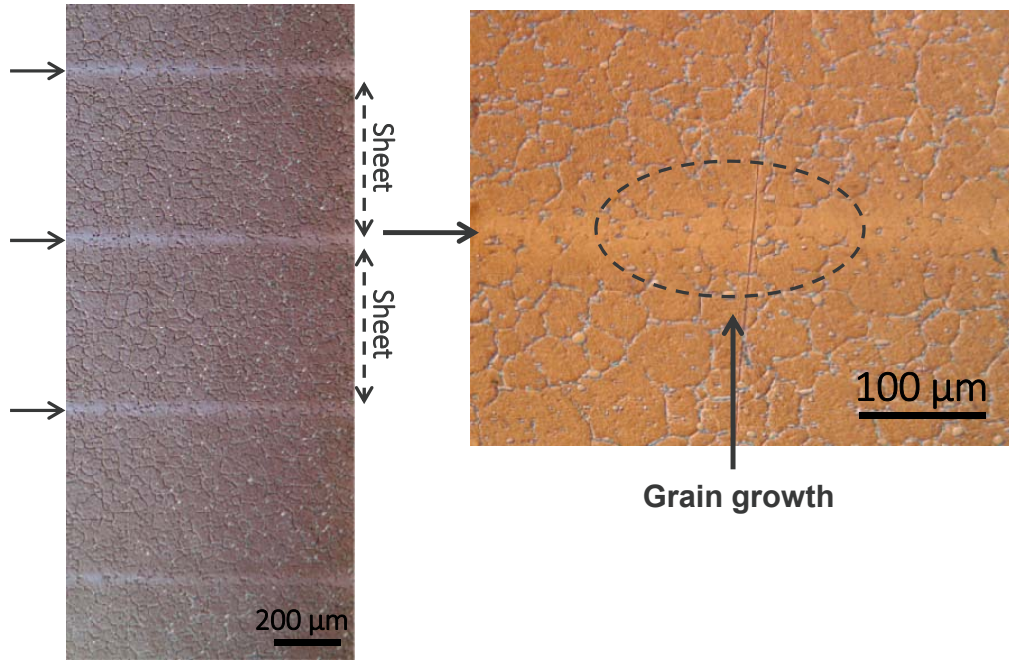
Stacks



Tensile Samples

Diffusion Bonding of Alloy H230

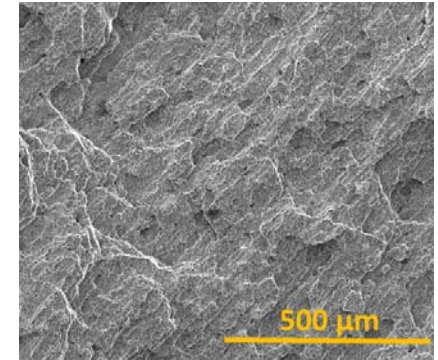
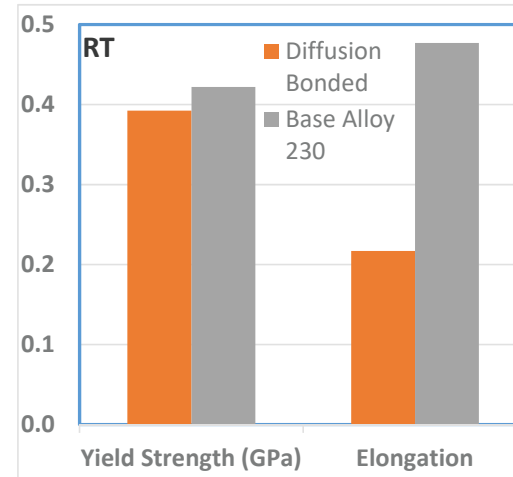
Microstructure



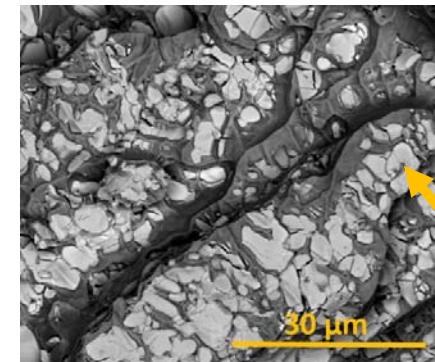
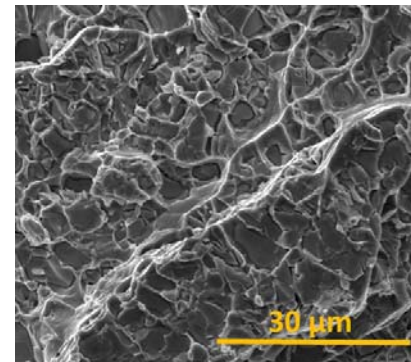
Grain growth

Etched microstructure to observe grain growth through the bond line

Mechanical Behavior

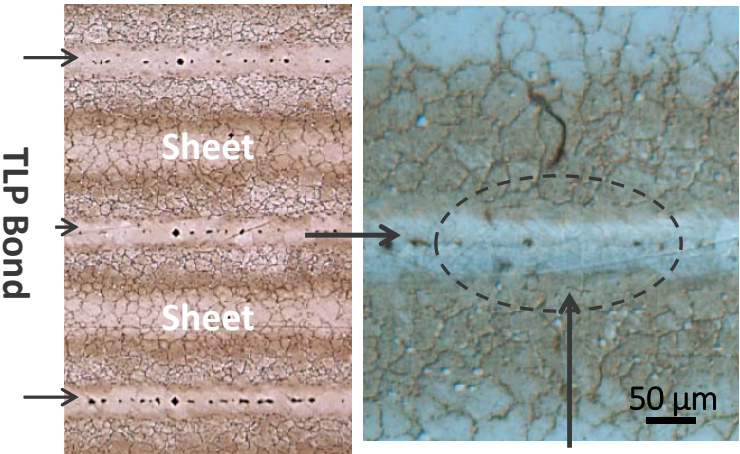


Ductile fracture along the precipitate bands.

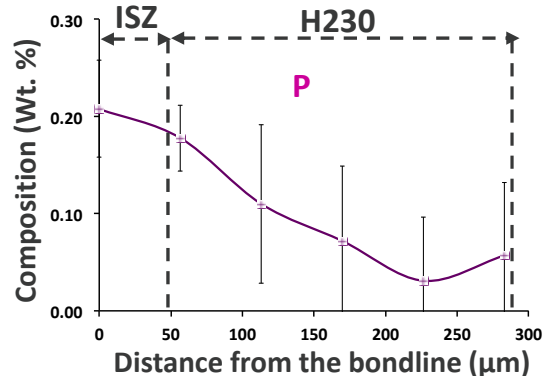
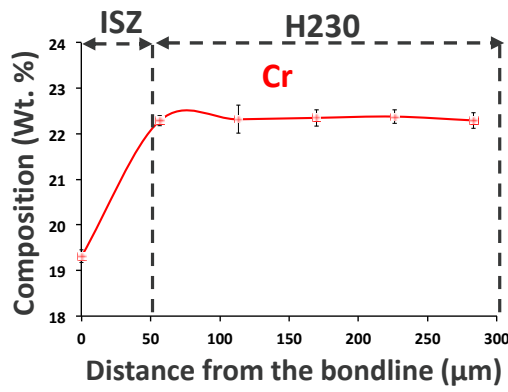
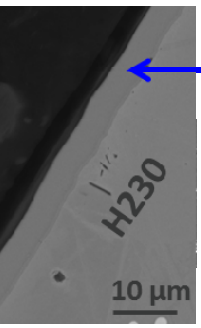
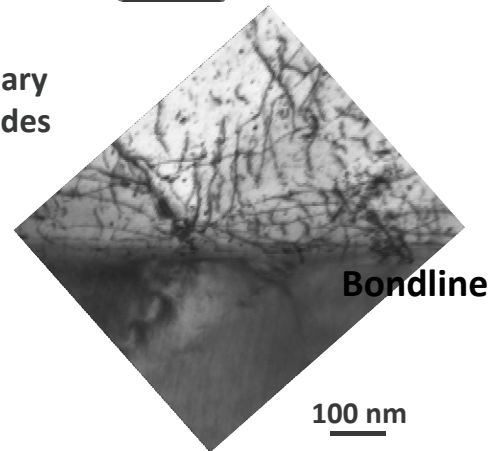
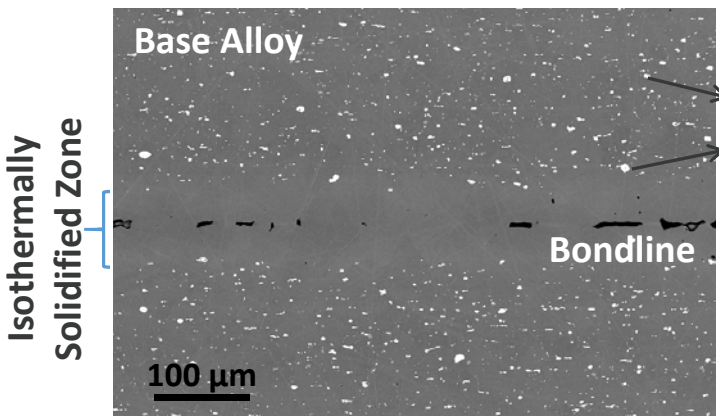


W, C

TLP Bonding of Alloy H230



Grain growth across the bond

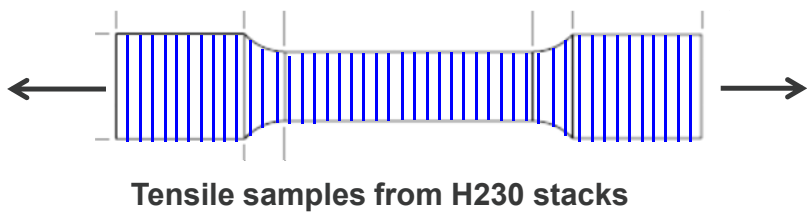
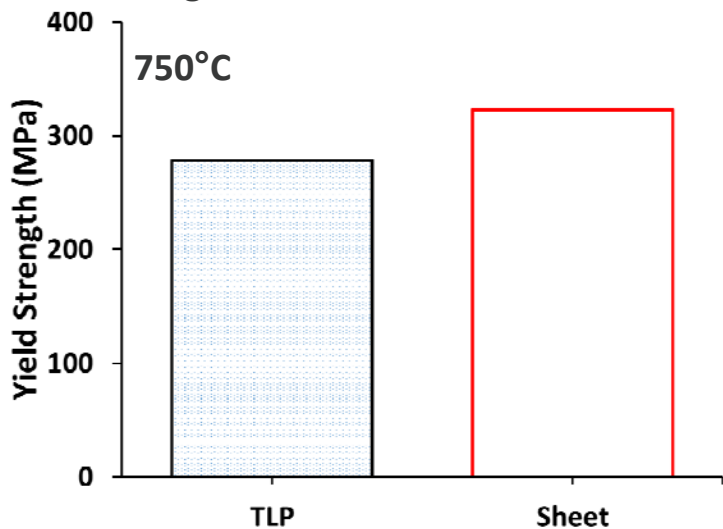


No nanoscale phase near the bondline

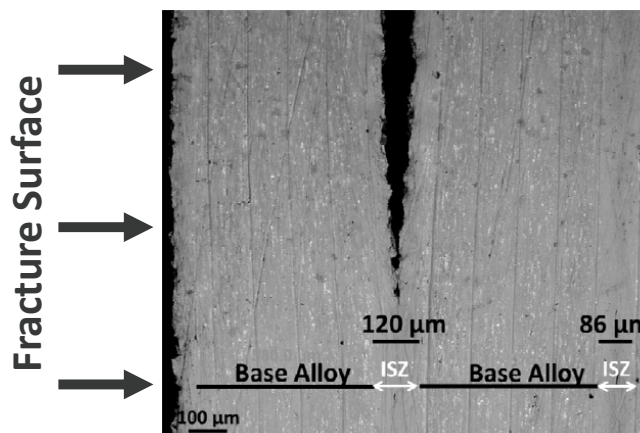
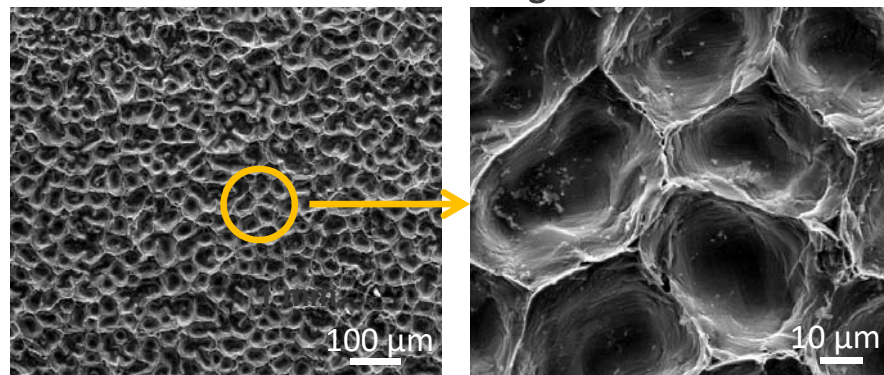
- Cr important for oxidation resistance - Slight reduction in Cr near the bondline
- P should be as low as possible to avoid formation of brittle phases - P is ~0.2 wt. % near the bondline

Tensile Testing

Yield strength of TLP stacks is ~86% of bulk H230



Ductile fracture through the ISZ

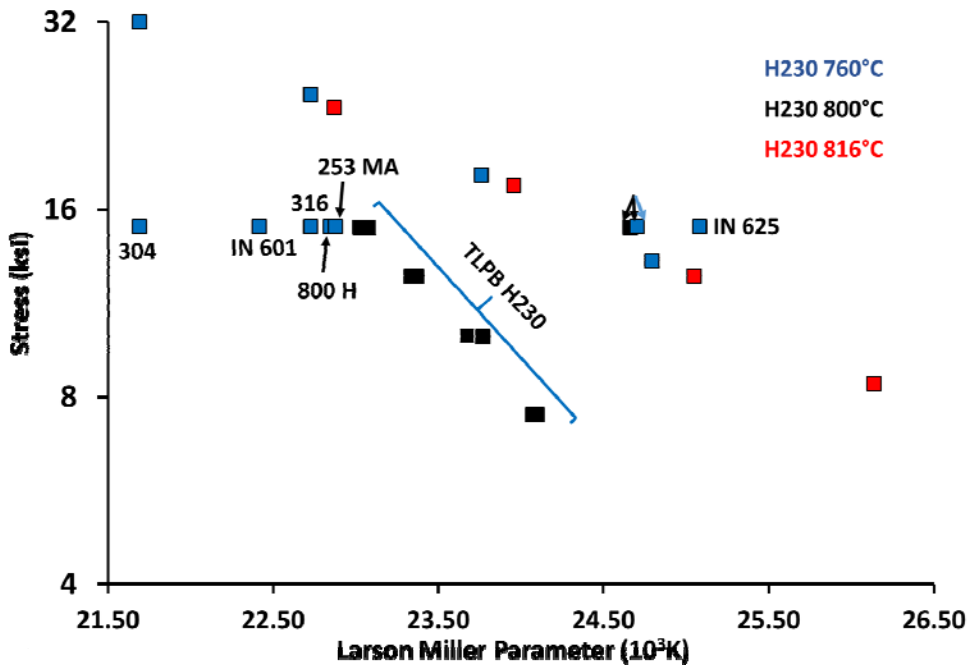


Plastic strain is constrained in the ISZ

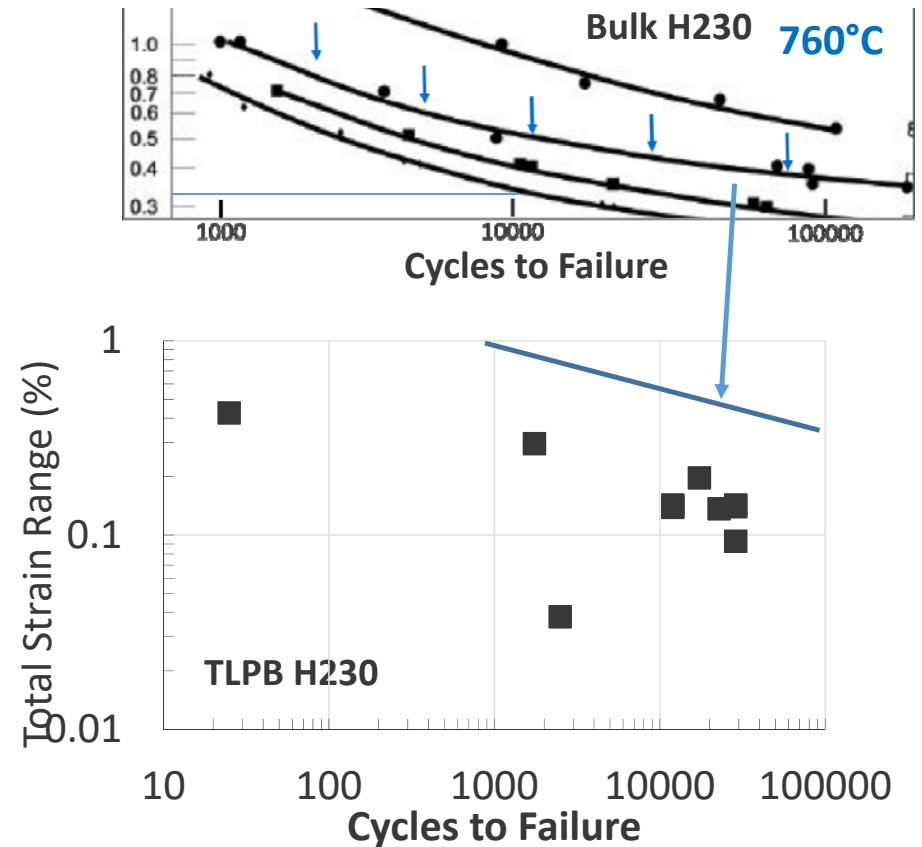
High temperature mechanical properties



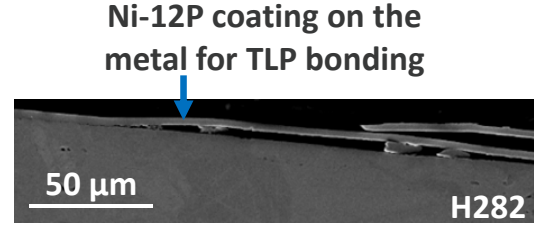
Creep properties



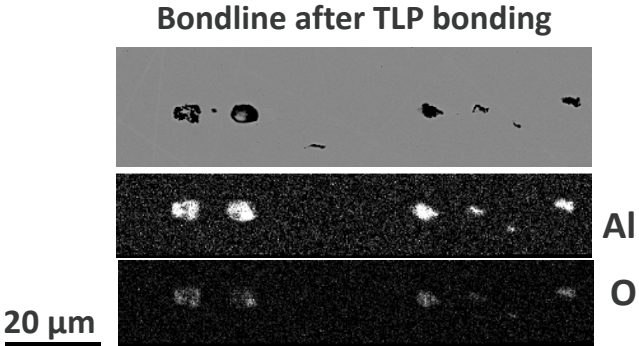
Low cycle fatigue properties @ 760°C



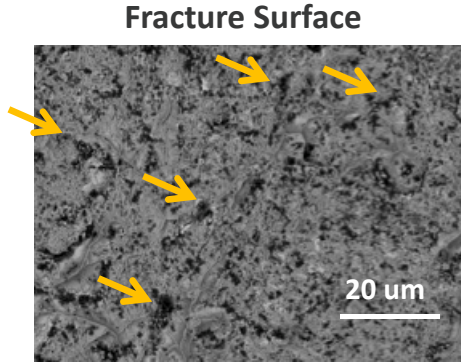
Challenges in bonding of H282



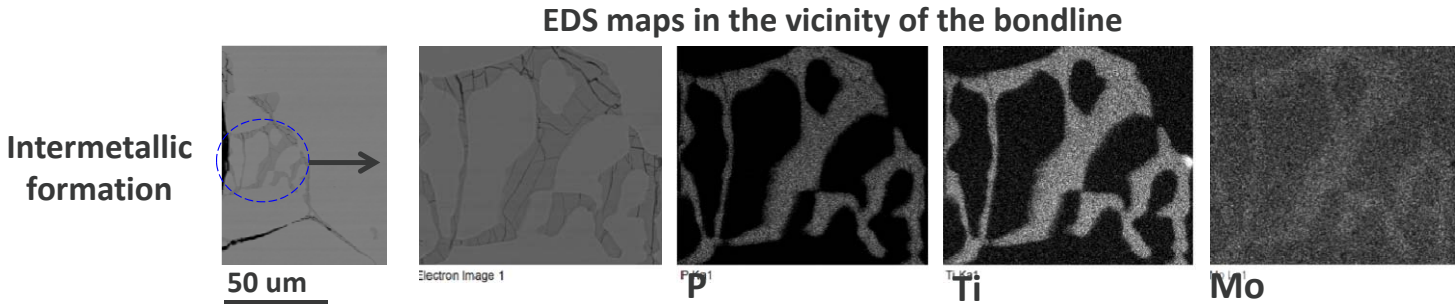
Ni-12P coating on the metal for TLP bonding
Coating does not adhere to the metal



Al₂O₃ particles (dark) along the bondline

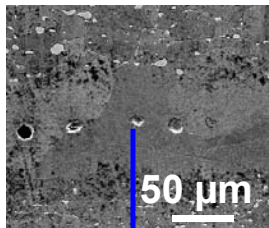


Al₂O₃ particles (dark) on the fracture surface

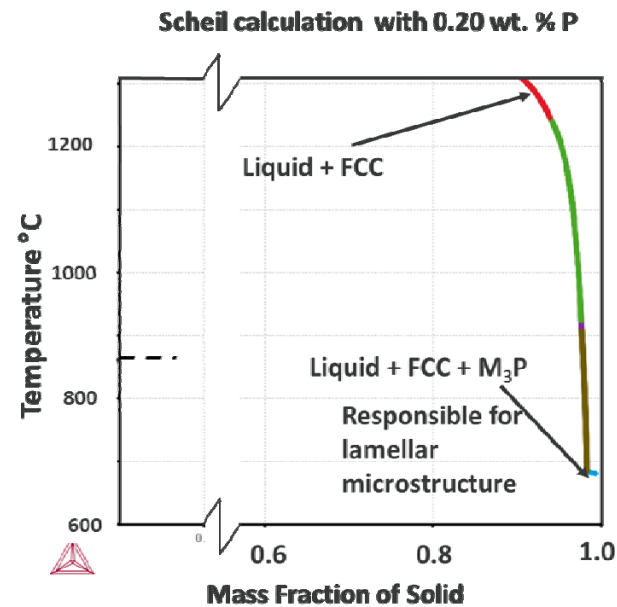
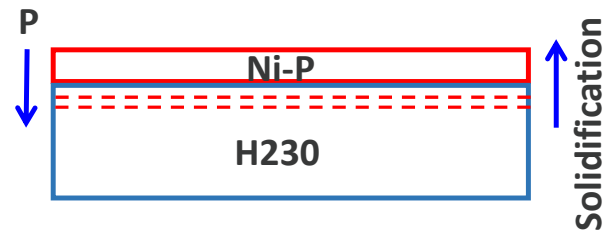
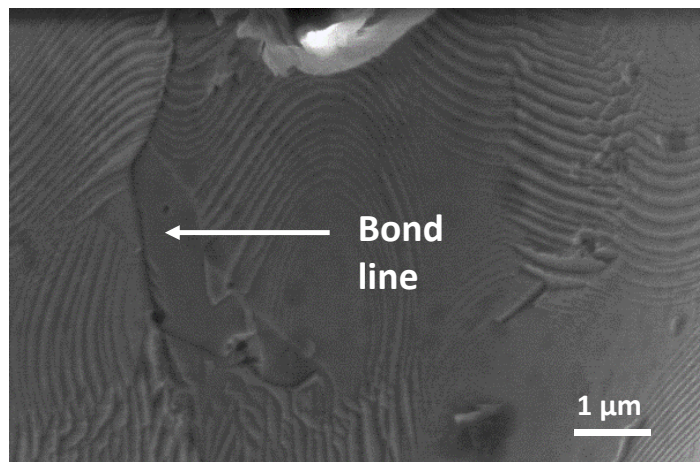


Challenges with TLP bonding of H282 – Surface oxides & Intermetallic Formation

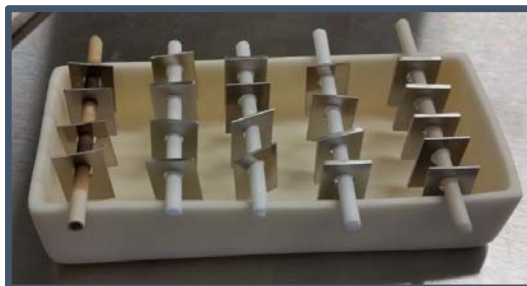
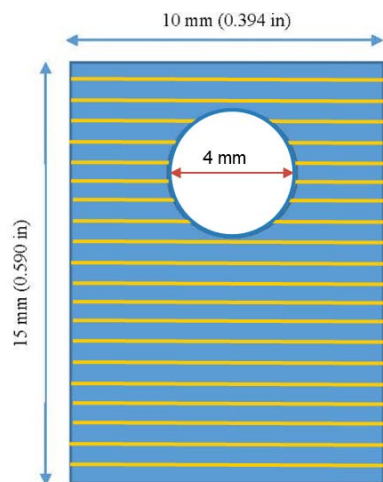
Shrinkage Pores in TLP Bonding



Pores in the vicinity of the bondline with a lamellar structure.



High-Temperature Oxidation of Bonded Regions in CO₂



1 bar CO₂ Exposures

- Gas: 1 bar CO₂ (99.999% purity)
- O₂ level in furnace tube: <12 ppm
- Gas flow rate: 0.032 kg/h
- Temperature: 700°C
- Duration: 4000 h in 500 h increments
- 24 h purging with CO₂ before heating

Characterization

Mass Change

XRD

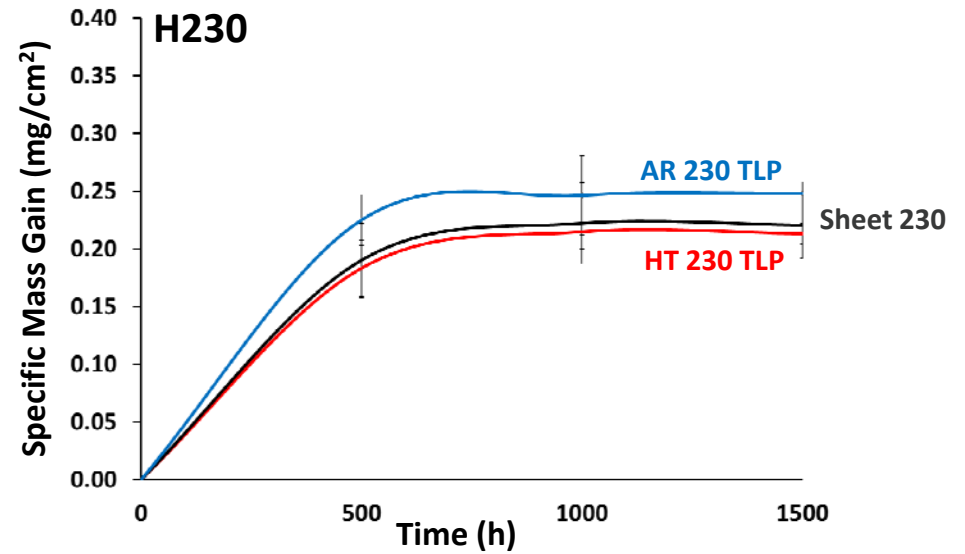
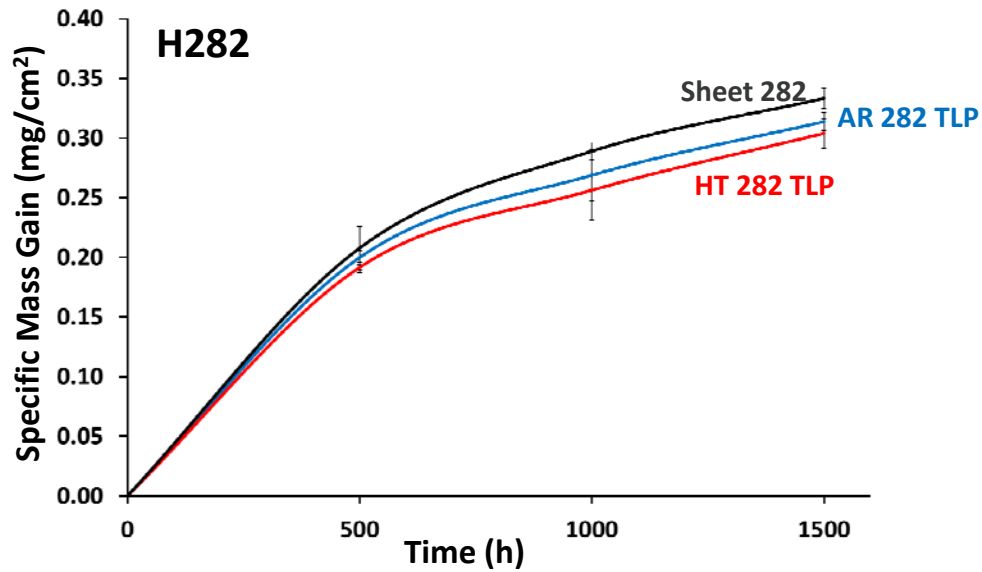
SEM



250 bar CO₂ Exposure

- Gas: CO₂ (99.999% purity)
- Flow rate: 2 ml/min
- Temperature: 720°C
- Duration: 1500 h in 500 h incr.
- Argon purging before heating

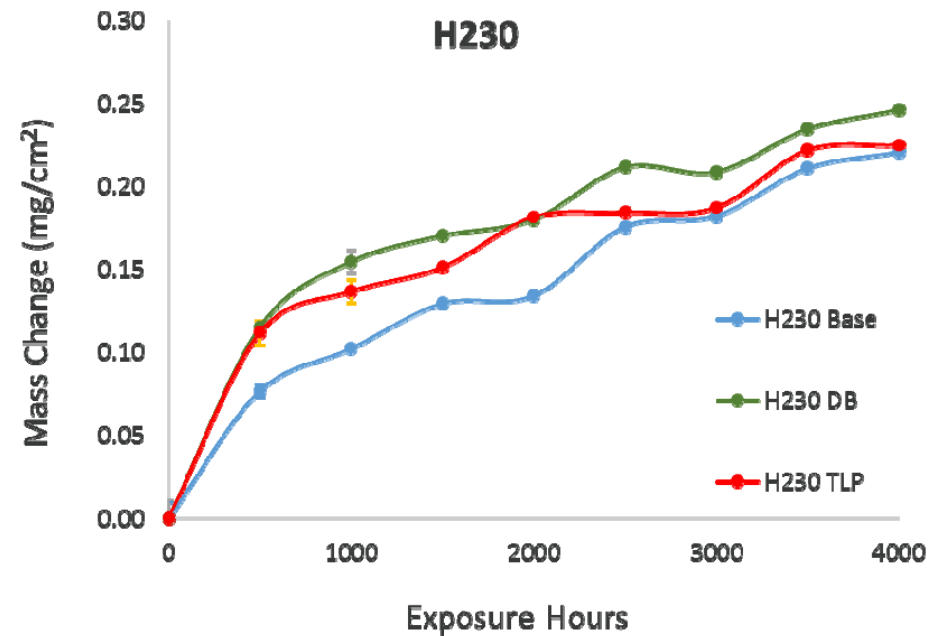
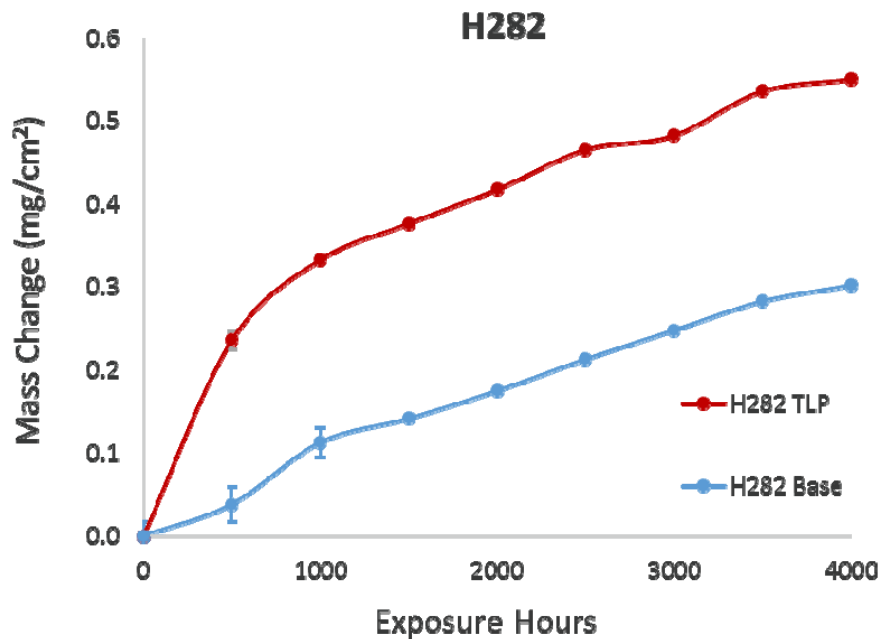
Oxidation in sCO₂ (720°C – 250 bar)



H282 continues to have mass gain up to 1500 hrs BUT H230 mass gain tapers off after 500 hrs

Atmospheric CO₂ Exposure

Fastest oxidation rate within first 500 hrs, then slower, but steady continued oxidation



Glancing angle XRD

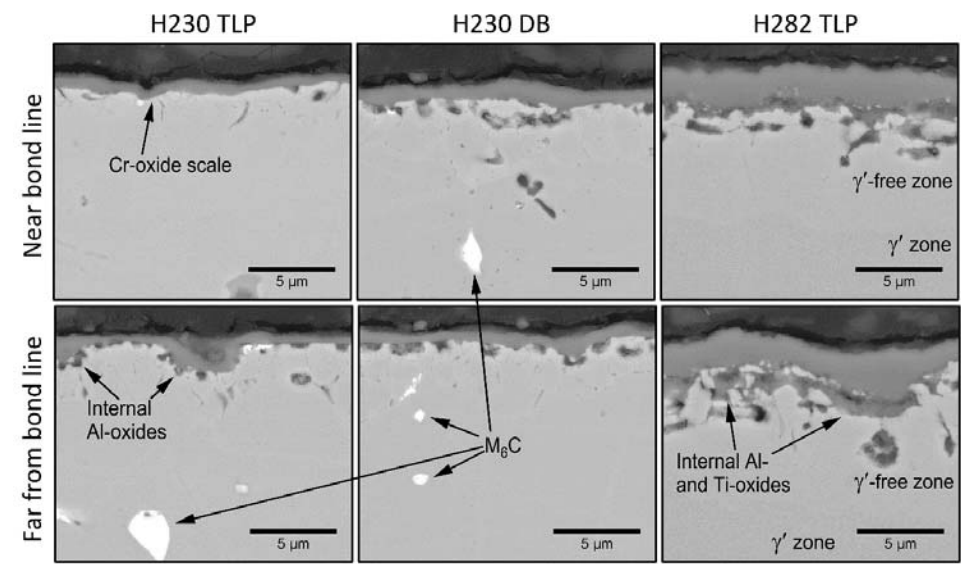
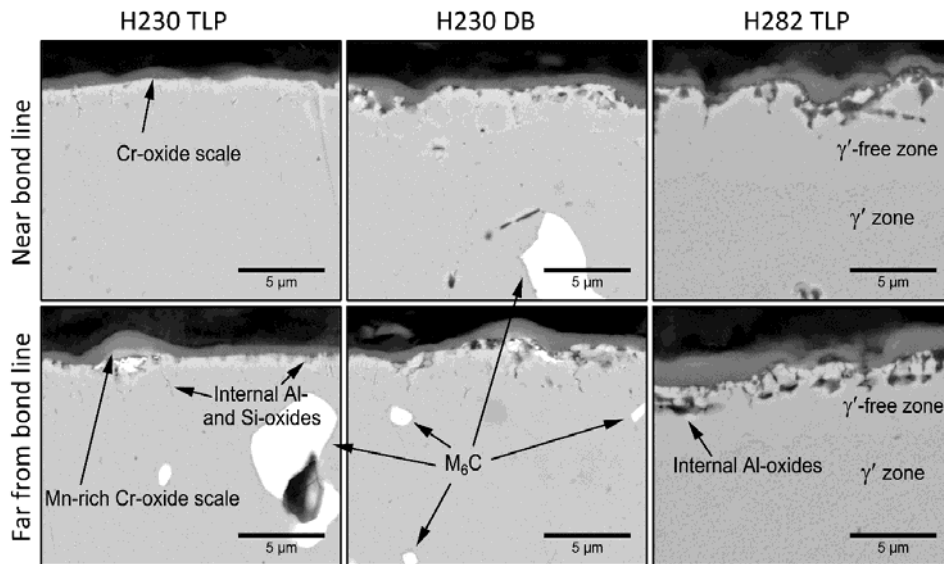
Surface is predominantly Cr₂O₃, with some M₃O₄ phase appearing at longer exposure times

Surface is primarily Cr₂O₃, with a smaller portion of M₃O₄ phase

Bonded alloys exposed to 700 °C aCO₂

500 h exposure time

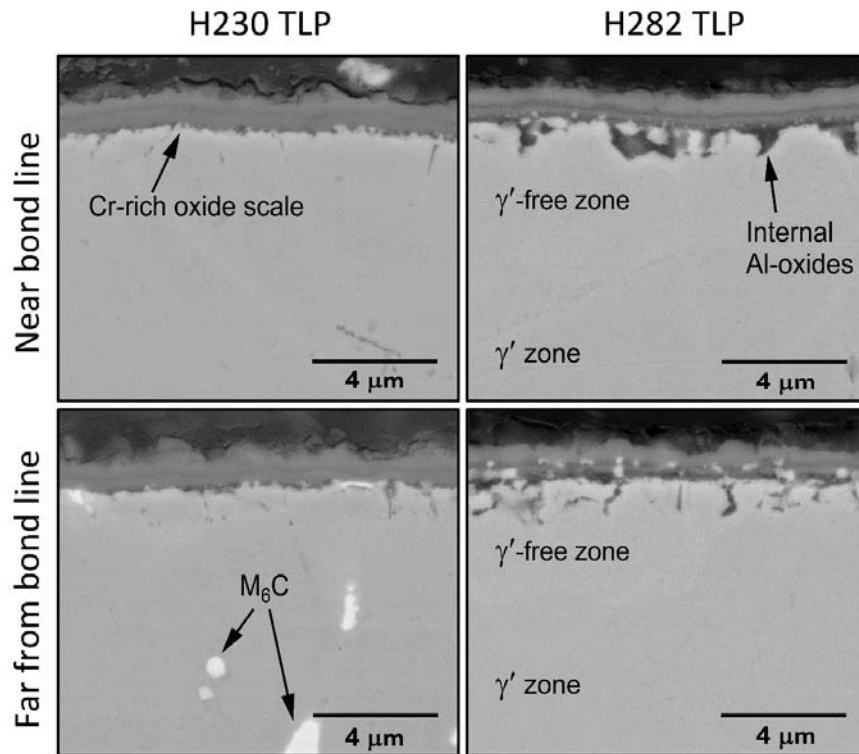
1500 h exposure time



- Thin, protective Cr-rich oxide layers are formed for H230 and H282 during exposure to 700 °C aCO₂.
- Internal oxidation of Al leads to a sub-surface layer depleted of γ' phase in H282.
- No difference in oxidation resistance is observed near/far from the bond layer for either alloy.

Bonded alloys exposed to 720 °C sCO₂

500 h exposure time



Similar results as H230 and H282 bonded samples exposed to 700 °C aCO₂:

- Thin Cr-rich oxide layers.
- Internal oxidation of Al leads to a sub-surface layer depleted of γ' phase in H282.
- No difference in oxidation resistance observed near/far from the bond layer.

One difference is that sCO₂ exposures show a 2-layer oxide structure containing some Mn (H230) or Ti (H282) in addition to Cr. This is the subject of ongoing investigation.

Performance of welded alloys in sCO₂

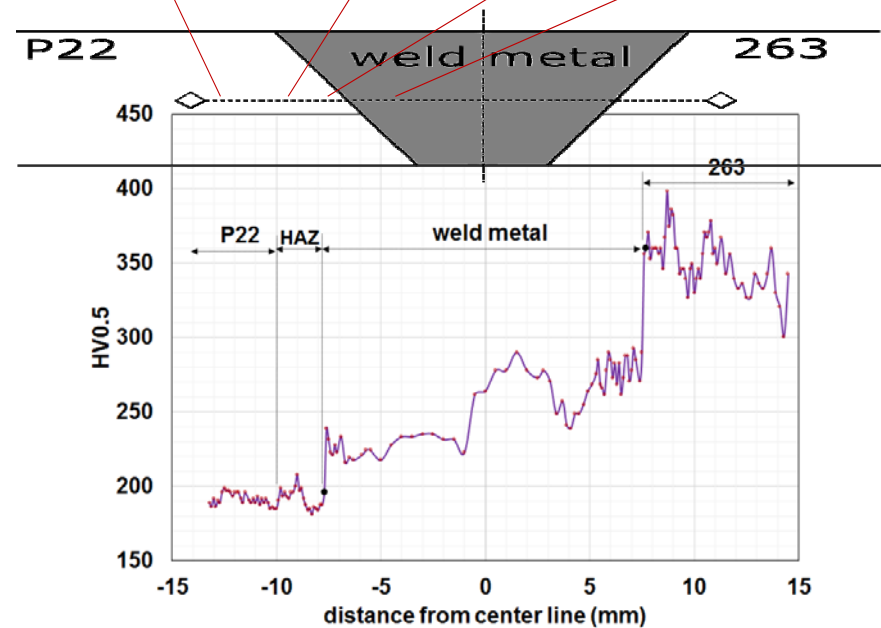
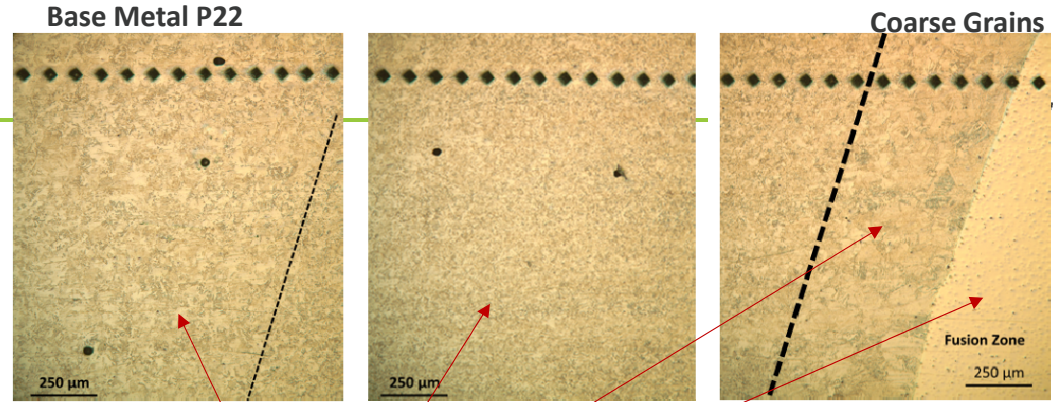
• Similar and dissimilar metal welds (1 inch thick plates) were done with gas tungsten arc welding (GTAW) and post weld heat treated at Edison Welding Institute.

- P22 – P22,
- P91 – P91,
- 347H – 347H,
- Alloy 625 – Alloy 625,
- Alloy 263 – Alloy 263,
- P22 – P91,
- P91 – 347H,
- P22 – Alloy 263,
- Alloy 625 – Alloy 263 ,
- 347H – Alloy 263

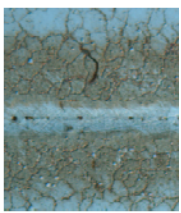
• Changes in microstructure (Heat affected zone – HAZ) due to welding were characterized using optical microscopy and hardness testing

• sCO₂ exposures of weld samples will be performed

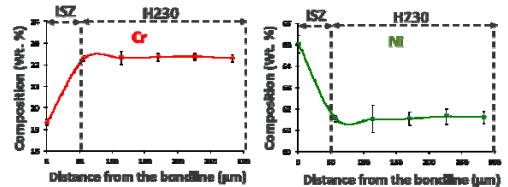
- Corrosion performance
- Mechanical performance



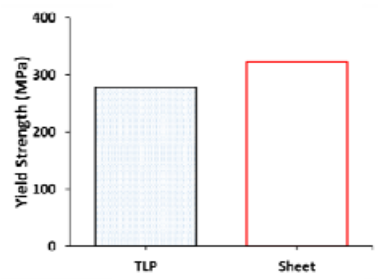
Summary



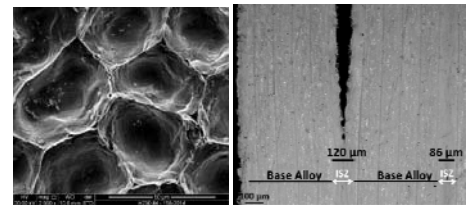
1) Uniform bond with grain growth across the bondline



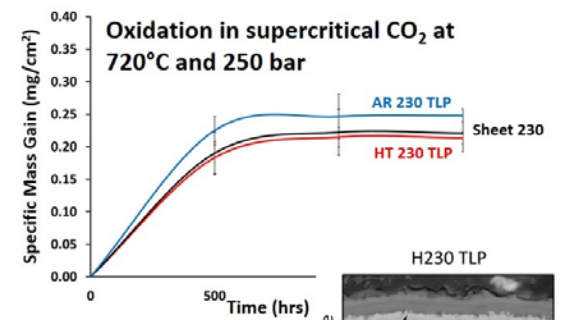
2) Ni increase, Cr dip through the bond



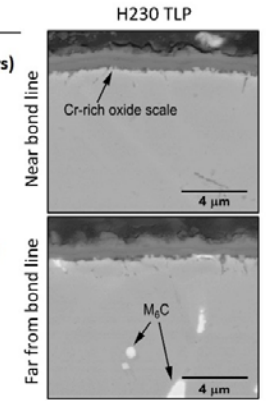
3) TLP-bonded H230 with ~ 86% strength of bulk @ 760°



4) Ductile fracture through the bond, plastic strain constrained in the bond region



- Thin Cr-rich oxide layers.
- No difference in oxidation resistance observed near/far from the bond layer.



5) Similar oxidation behavior between the bond region and base metal