



Crosscutting Review Meeting, April 21st, 2016, Pittsburgh, PA

New Mechanistic Models of Long Term Evolution of Microstructure and Mechanical Properties of Nickel Based Alloys

Jay Kruzic, Matt Evans (OSU)
Alex Greaney (UCR)



- Postdoctoral researchers:
 - Dr. Qin Yu
 - Dr. Agnieszka Truszkowska
- Collaborators
 - Dr. Jeff Hawk (NETL, Albany)
 - Dr. Kyle Rozman (NETL, Albany)



College of Engineering



This material is based upon work supported by the Department of Energy National Energy Technology Laboratory under Award Number(s) DE-FE0024065.

This report was prepared as an account of work sponsored by an agency of the United States Government. Neither the United States Government nor any agency thereof, nor any of their employees, makes any warranty, express or implied, or assumes any legal liability or responsibility for the accuracy, completeness, or usefulness of any information, apparatus, product, or process disclosed, or represents that its use would not infringe privately owned rights. Reference herein to any specific commercial product, process, or service by trade name, trademark, manufacturer, or otherwise does not necessarily constitute or imply its endorsement, recommendation, or favoring by the United States Government or any agency thereof. The views and opinions of authors expressed herein do not necessarily state or reflect those of the United States Government or any agency thereof.

- Current approaches to predicting creep and creep-fatigue behavior are highly empirical
 - Power law creep constitutive laws
 - e.g., Norton Law
 - Linear creep-fatigue damage accumulation

- Not easily extendible outside measured ranges
 - Longer lifetimes
 - Variable operating conditions

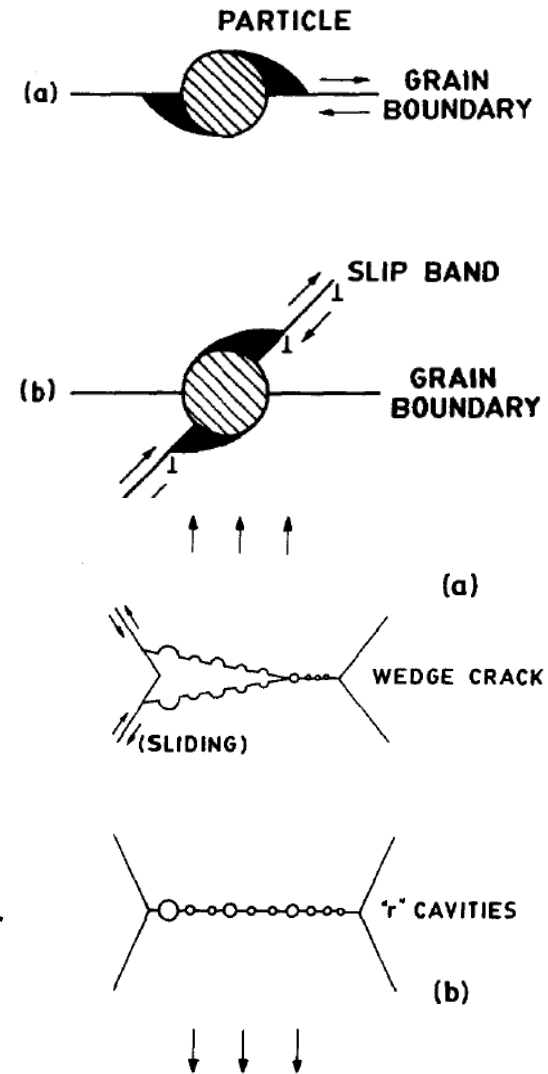


ruptured steel boiler pipe
<http://sirius.mtm.kuleuven.be/>

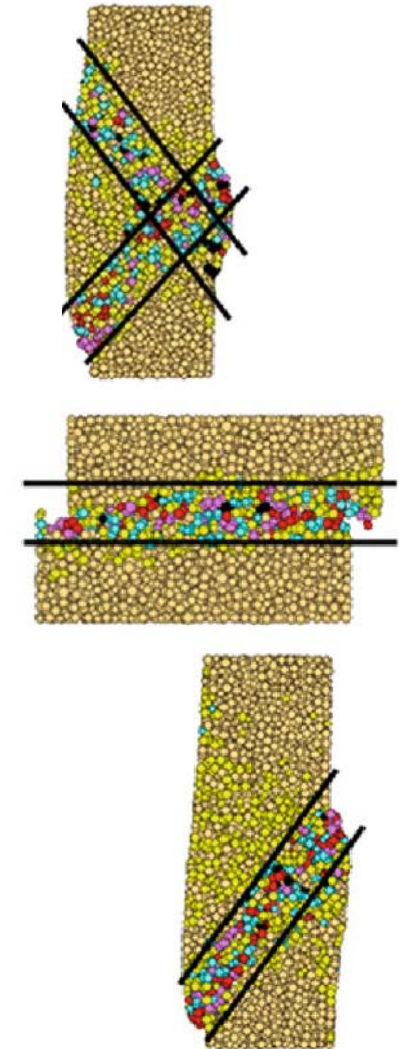
$$\epsilon = At\sigma^n \exp\left(-\frac{Q}{RT}\right)$$

$$\underbrace{\sum_j \left(\frac{n}{N_d}\right)_j}_{\text{Cyclic Damage}} + \underbrace{\sum_k \left(\frac{\Delta t}{T_d}\right)_k}_{\text{Creep Damage}} \leq D$$

- Review article by Pineau and Antolovich (2009):
 - Generalized approaches to lifetime prediction likely not successful
 - Suggest inclusion of critical damage mechanisms into material and service condition specific models
- Challenge
 - Difficult to include stochastic damage mechanisms in current modeling methods (e.g., FEM) in predictive manner

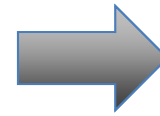


- Discrete element method widely used for granular media
 - Each particle is modeled as a discrete element
 - One-to-one correlation between element and particle
 - Sands, mined materials, and powders are commonly modeled
- Properties modeled include:
 - Granular body deformation
 - Granular body creep
 - Granular sintering and microstructure evolution
- Stochastic phenomena naturally emerge in DEM
 - Shear bands
 - Fracture nucleation and propagation
 - Void formation and growth



Zhao & Evans (2011)

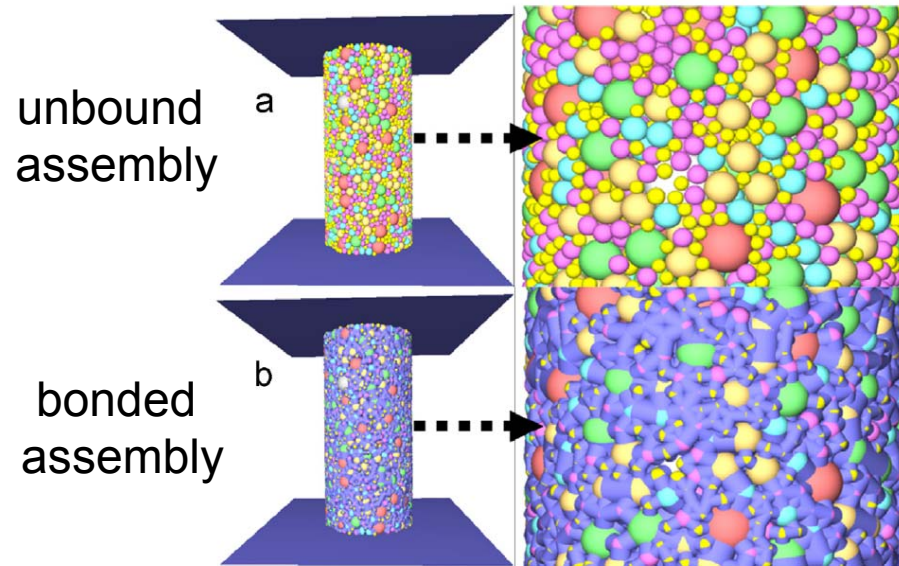
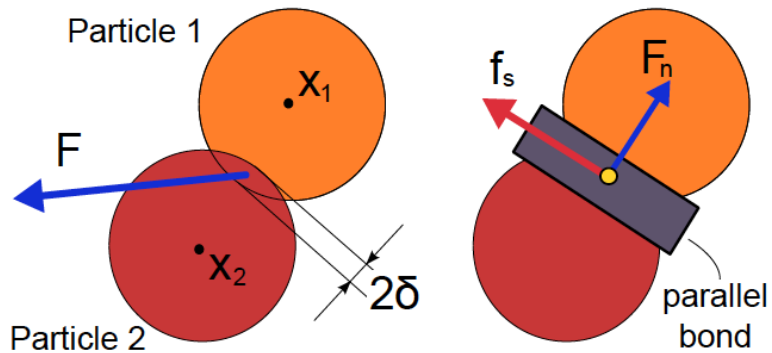
- Traditional DEM
 - Granular materials
 - Significant motion of discrete elements
 - Compression loading is straightforward



Oregon sand dunes

- Solid material DEM
 - Bond elements using parallel solid bonds
 - Full range of loading configurations can be simulated (tension, bending, etc.)

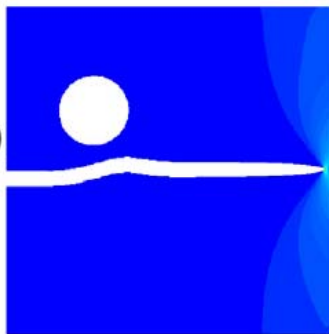
Now an element is meso-scale domain



- Solid materials DEM has been used for:
 - Amorphous materials (silica glass, polymers)
 - Particle reinforced composites
- No need to predefine crack location/path
 - Emerge naturally from DEM model

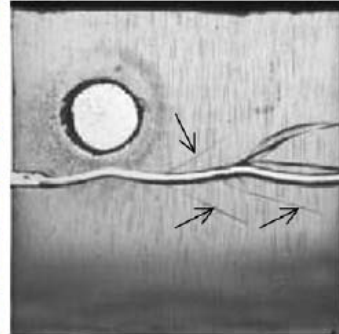
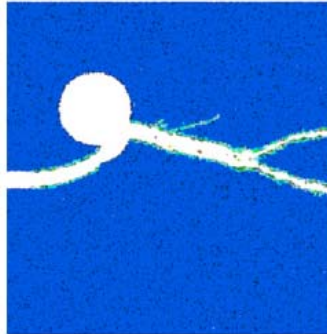
FEM Model:

No crack branching predicted



DEM Model:

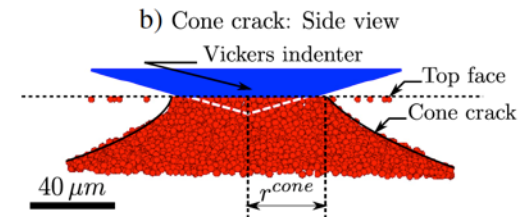
Crack branching matches experiment



Hedjazi et al. (2012)

DEM Model:

Cone crack emerges under indent in silica glass



c) Cone crack: Perspective view



Jebahi et al. (2013)

DEM started like this:



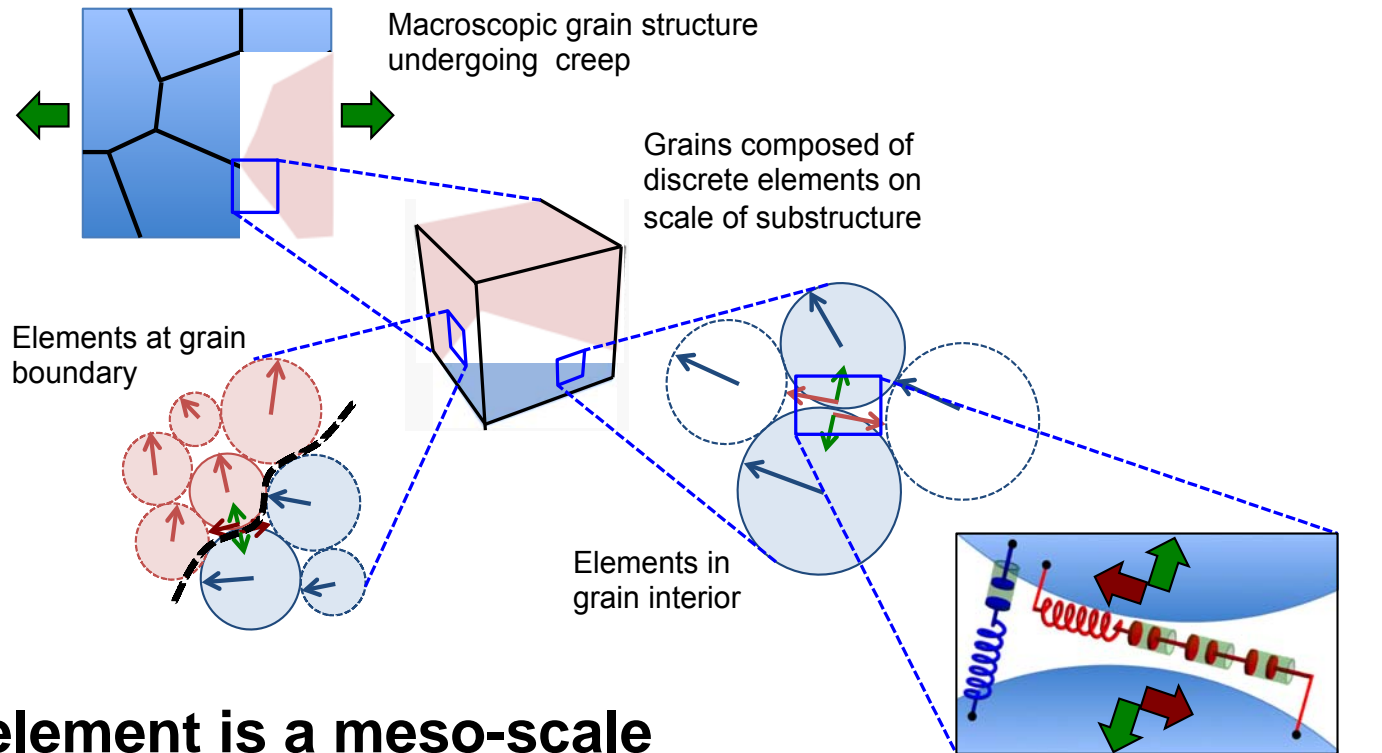
Oregon sand dunes

Next we want to model this:



Turbine blisk

- DEM crystal plasticity model for predicting creep and creep-fatigue of nickel based alloys

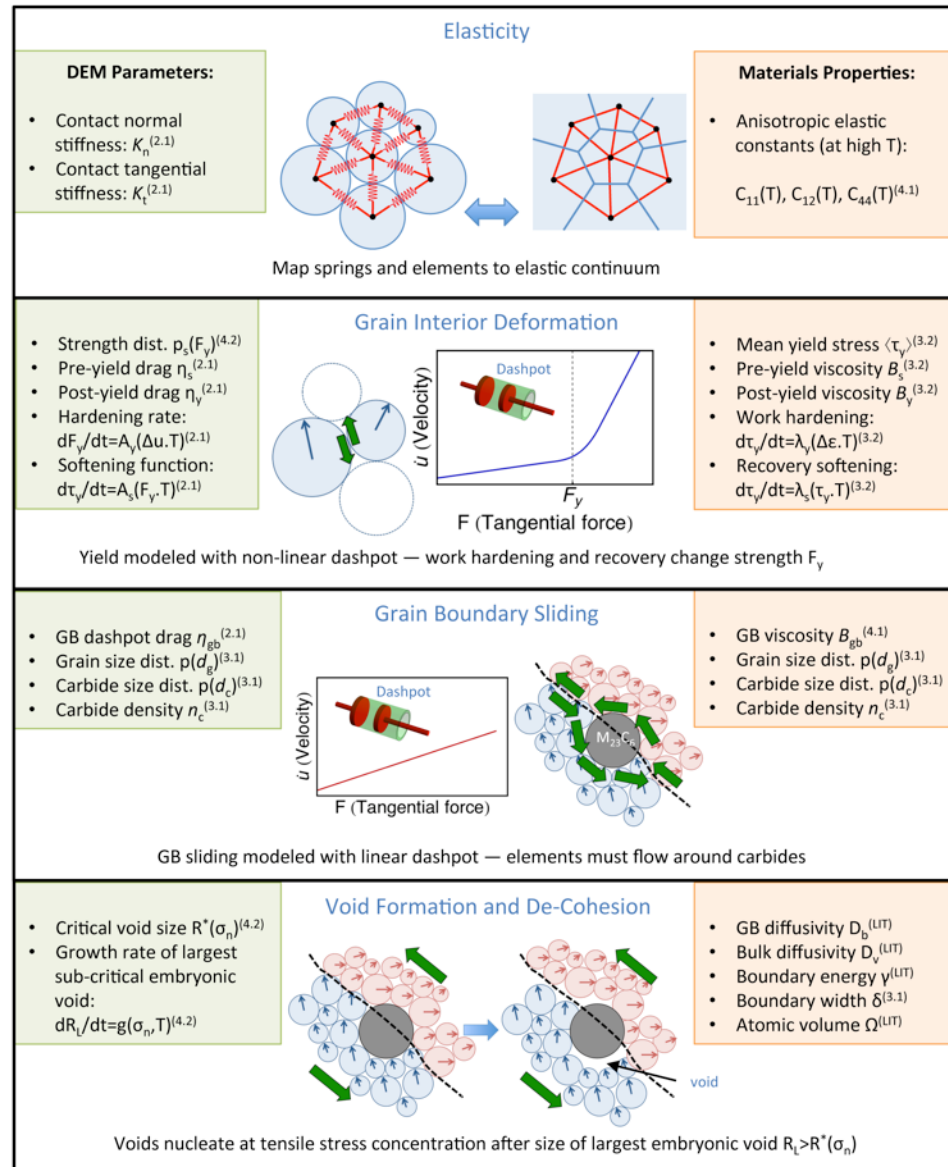


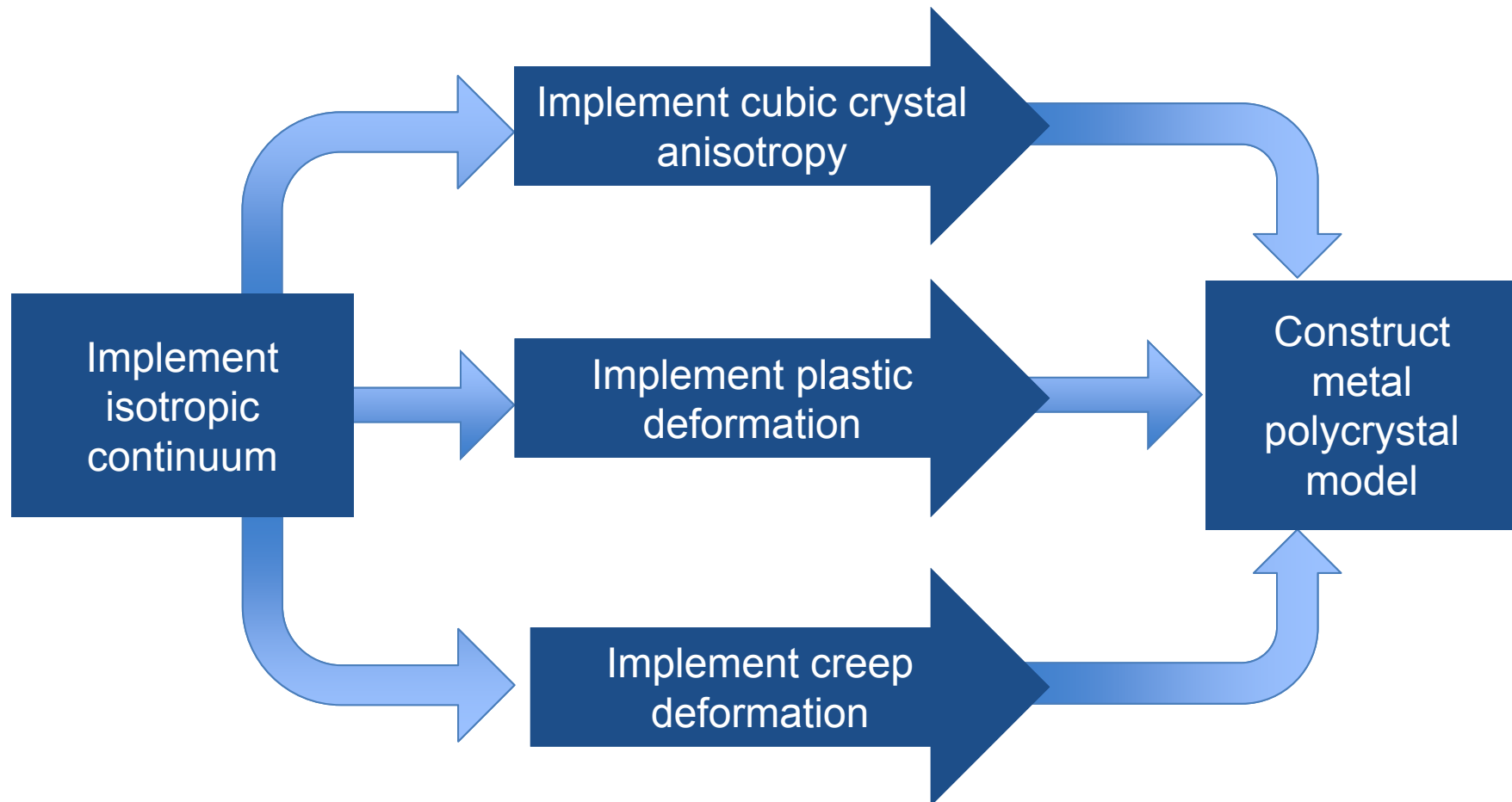
- **An element is a meso-scale domain**
 - e.g., a sub-grain or part of sub-grain

Contacts between grains modeled with springs and series dashpots

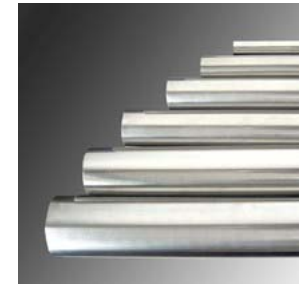
- We propose we can adapt DEM to correctly capture:

- Polycrystal deformation
- Microstructure evolution
- Stochastic damage evolution

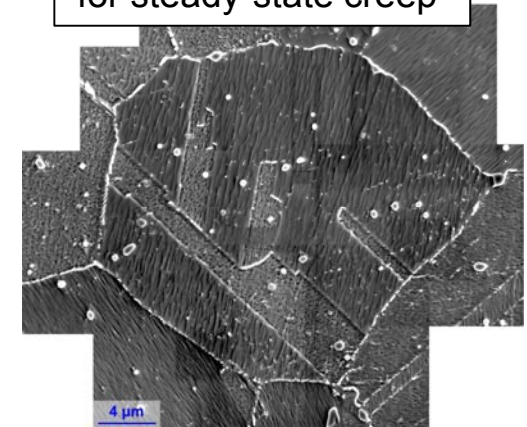




- Nimonic 75 chosen as model alloy
 - Simple Ni-20Cr solid solution microstructure represents many superalloys
 - Austenitic, solid solution grains
 - Chromium rich, globular grain-boundary carbides normally of the type $M_{23}C_6$
 - Certified tensile and creep reference material
 - We purchased a standardized microstructure certified to have specific tensile and creep properties
 - Model will be developed for 600°C deformation
 - Creep behavior certified at 600°C



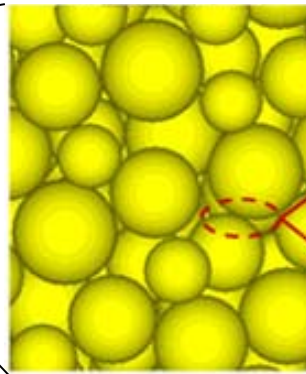
typical microstructure for steady-state creep





Oregon sand dunes

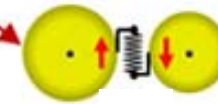
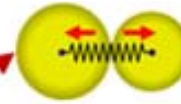
Particle Assembly



<http://www.ngi.no/>

Normal Stiffness

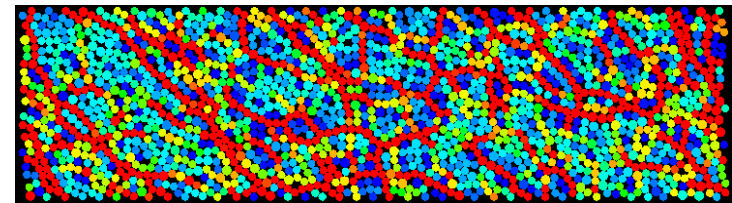
$$k_n$$



$$k_s$$

Shear Stiffness

But Sand is not
homogeneously elastic!

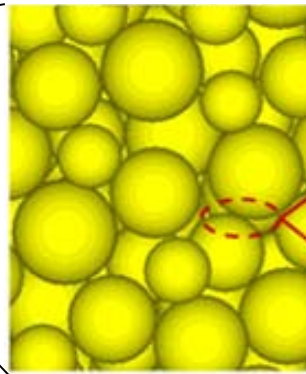


Intrinsic heterogeneous: force chains
and jamming in granular materials
(Image Banigan)



Oregon sand dunes

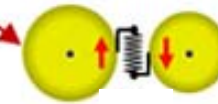
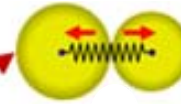
Particle Assembly



<http://www.ngi.no/>

Normal Stiffness

$$k_n$$



$$k_s$$

Shear Stiffness

We must developing contact behaviors to make sand elastic

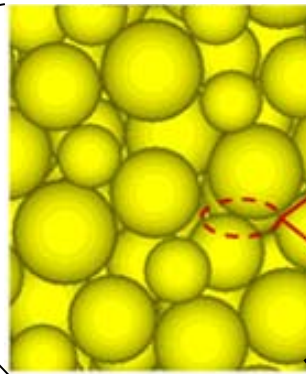


Homogeneous and isotropic in elastic response



Oregon sand dunes

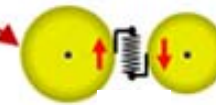
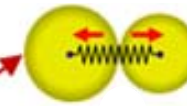
Particle Assembly



<http://www.ngi.no/>

Normal Stiffness

k_n



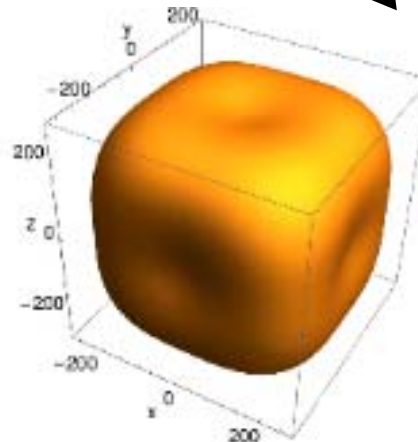
k_s

Shear Stiffness

We must define particle interactions k_n and k_s to produce full stiffness tensor

More than that we must make sand anisotropic elastic!

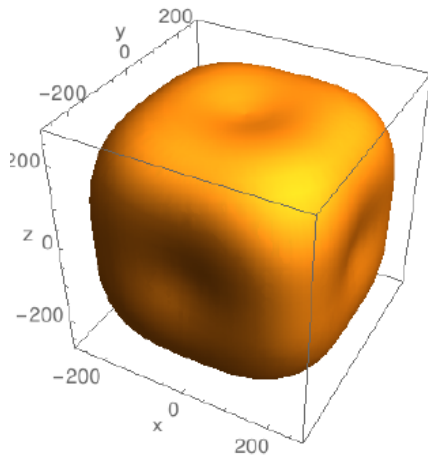
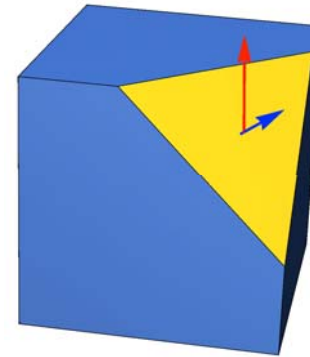
Ni-Cr: homogeneously elastic, but anisotropically elastic



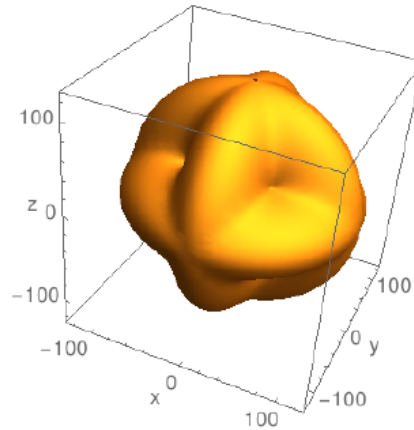
$$C_{ij} = \begin{pmatrix} C_{11} & C_{12} & C_{12} & 0 & 0 & 0 \\ C_{12} & C_{11} & C_{12} & 0 & 0 & 0 \\ C_{12} & C_{12} & C_{11} & 0 & 0 & 0 \\ 0 & 0 & 0 & C_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & C_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & C_{44} \end{pmatrix}$$

The subtleties of anisotropic elasticity...

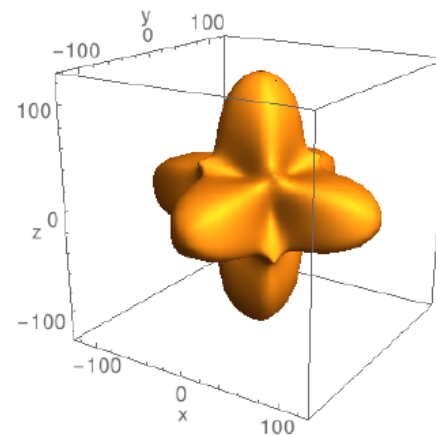
- Directionally dependent elastic response of single crystal Ni



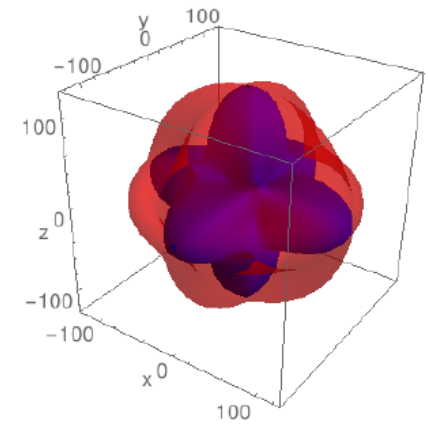
Directionally dependent normal stiffness



Shear stiffness in stiff direction

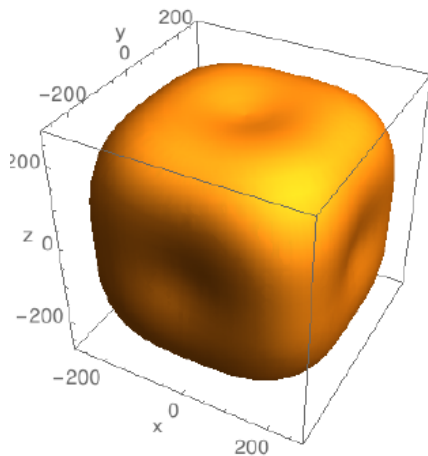
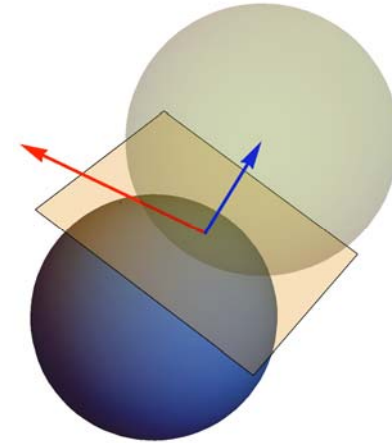


Shear stiffness in soft direction

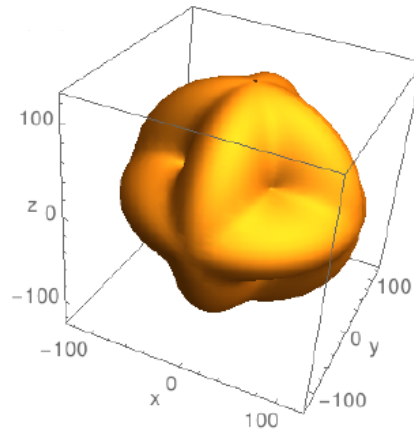


Stiff and soft shear stiffness overlaid

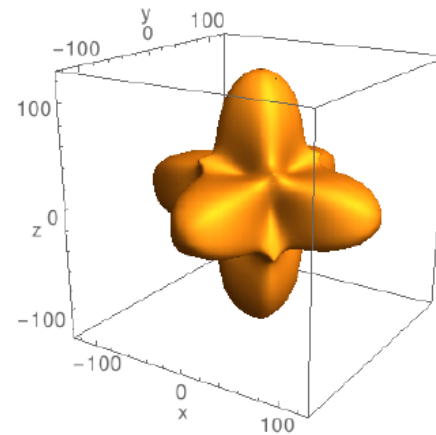
Derive a set of contact interactions that produce this angular response.



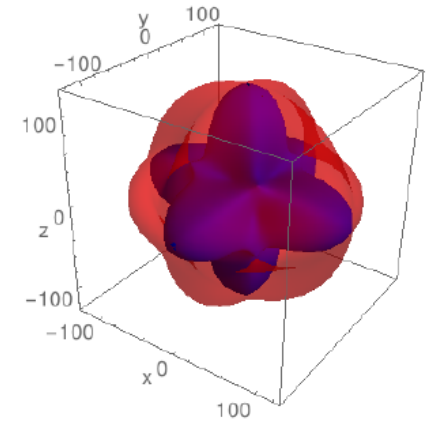
Directionally dependent normal stiffness



Shear stiffness in stiff direction



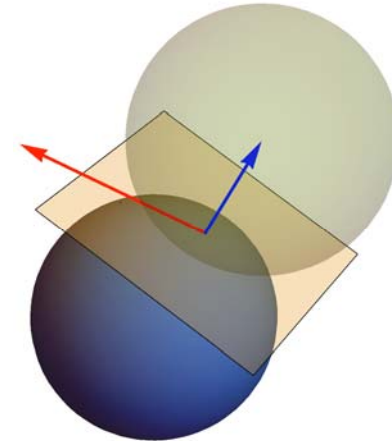
Shear stiffness in soft direction



Stiff and soft shear stiffness overlaid

Derive a set of contact interactions that produce this angular response.

Two different approaches to defining angularly dependent contact stiffness $k_n(\theta, \Phi)$ and $k_s(\theta, \Phi)$...



Emergent

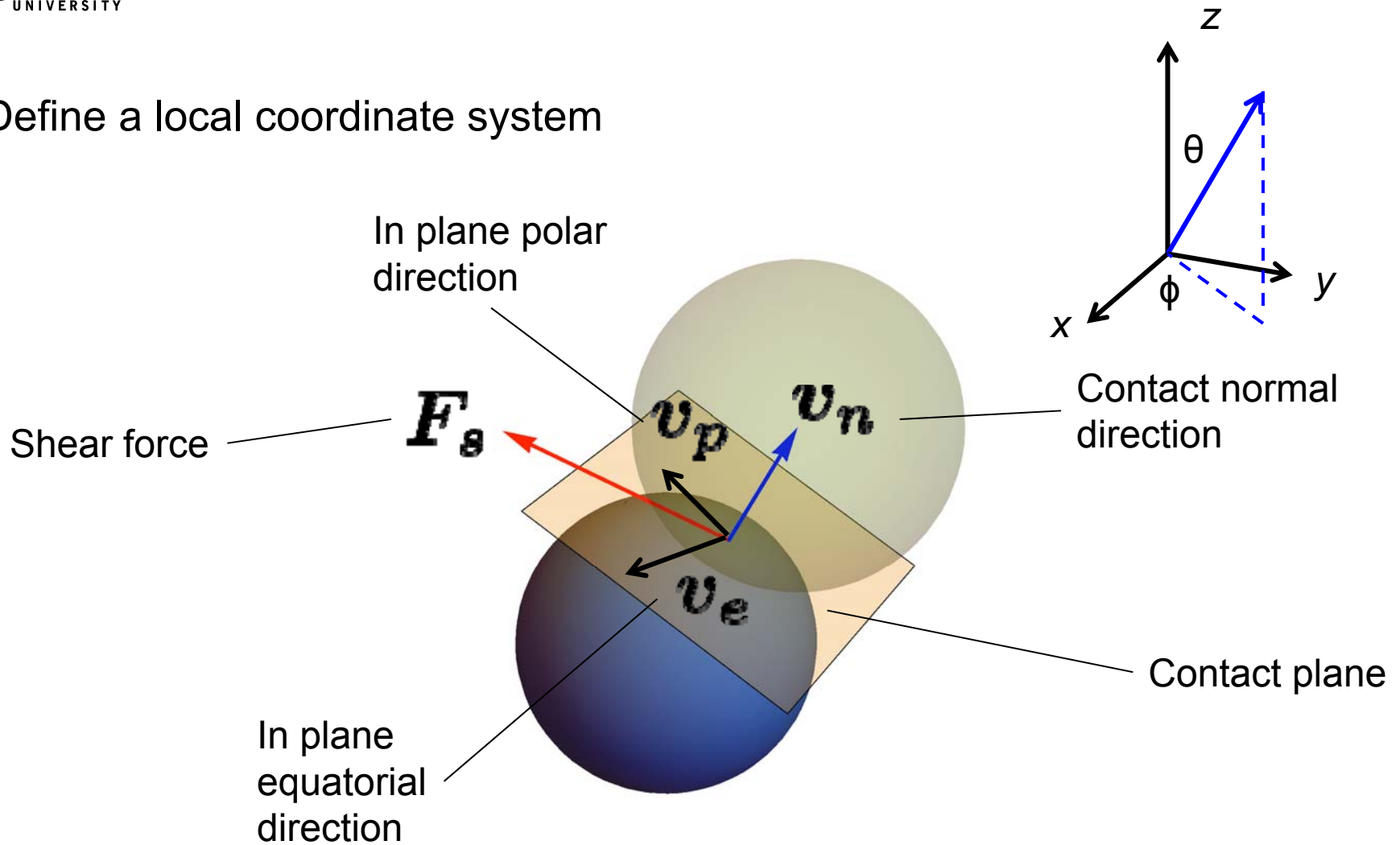
- Simple form of $k_n(\theta, \Phi)$ and $k_s(\theta, \Phi)$
- Correct cubic symmetry
- Cubic elasticity emerges from collections of particles

Imposed

- More complex form of $k_n(\theta, \Phi)$ and $k_s(\theta, \Phi)$
- Correct cubic symmetry
- Each contact mimics cubic elasticity

We have developed both approaches

Define a local coordinate system



Define bond contact laws
based on coordinate system

$$K(\theta, \phi) = \frac{1}{2} \pi R_c C_{11}^* \begin{bmatrix} A & F & E \\ F & B & D \\ E & D & C \end{bmatrix}$$

$$\alpha = \frac{2C_{44}^*}{C_{11}^* - C_{12}^*}$$

$$\beta = \frac{C_{44}^*}{C_{11}^*}$$

$$A = \frac{(-(\alpha - 1)\beta (8 \sin^4(\theta) \cos(4\phi) + 4 \cos(2\theta) + 7 \cos(4\theta)) + 11(\alpha - 1)\beta + 16\alpha)}{16\alpha}$$

$$B = \frac{\beta (-8(\alpha - 1) \sin^2(\theta) \cos^2(\theta) \cos(4\phi) + 7(\alpha - 1) \cos(4\theta) + 9\alpha + 7)}{16\alpha}$$

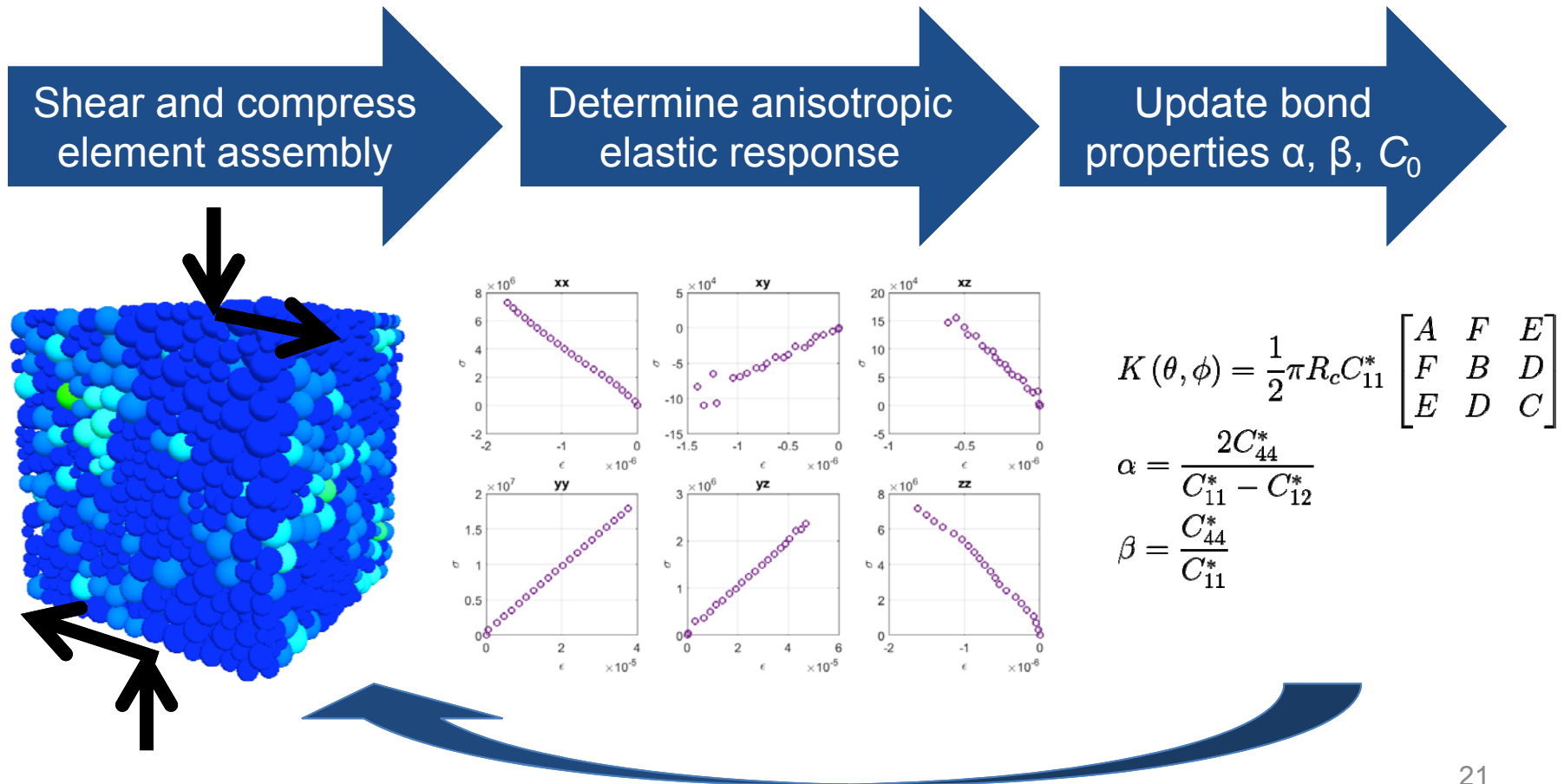
$$C = \frac{\beta (2(\alpha - 1) \sin^2(\theta) \cos(4\phi) + (\alpha - 1) \cos(2\theta) + 3\alpha + 1)}{4\alpha}$$

$$D = \frac{(\alpha - 1)\beta \sin^2(\theta) \cos(\theta) \sin(4\phi)}{2\alpha}$$

$$E = -\frac{(\alpha - 1)\beta \sin^3(\theta) \sin(4\phi)}{2\alpha}$$

$$F = \frac{(\alpha - 1)\beta (8 \sin^3(\theta) \cos(\theta) \cos(4\phi) - 2 \sin(2\theta) - 7 \sin(4\theta))}{16\alpha}$$

- Optimization routine is used to find bond property constants α , β , C_{11}^* that give desired assembly response:

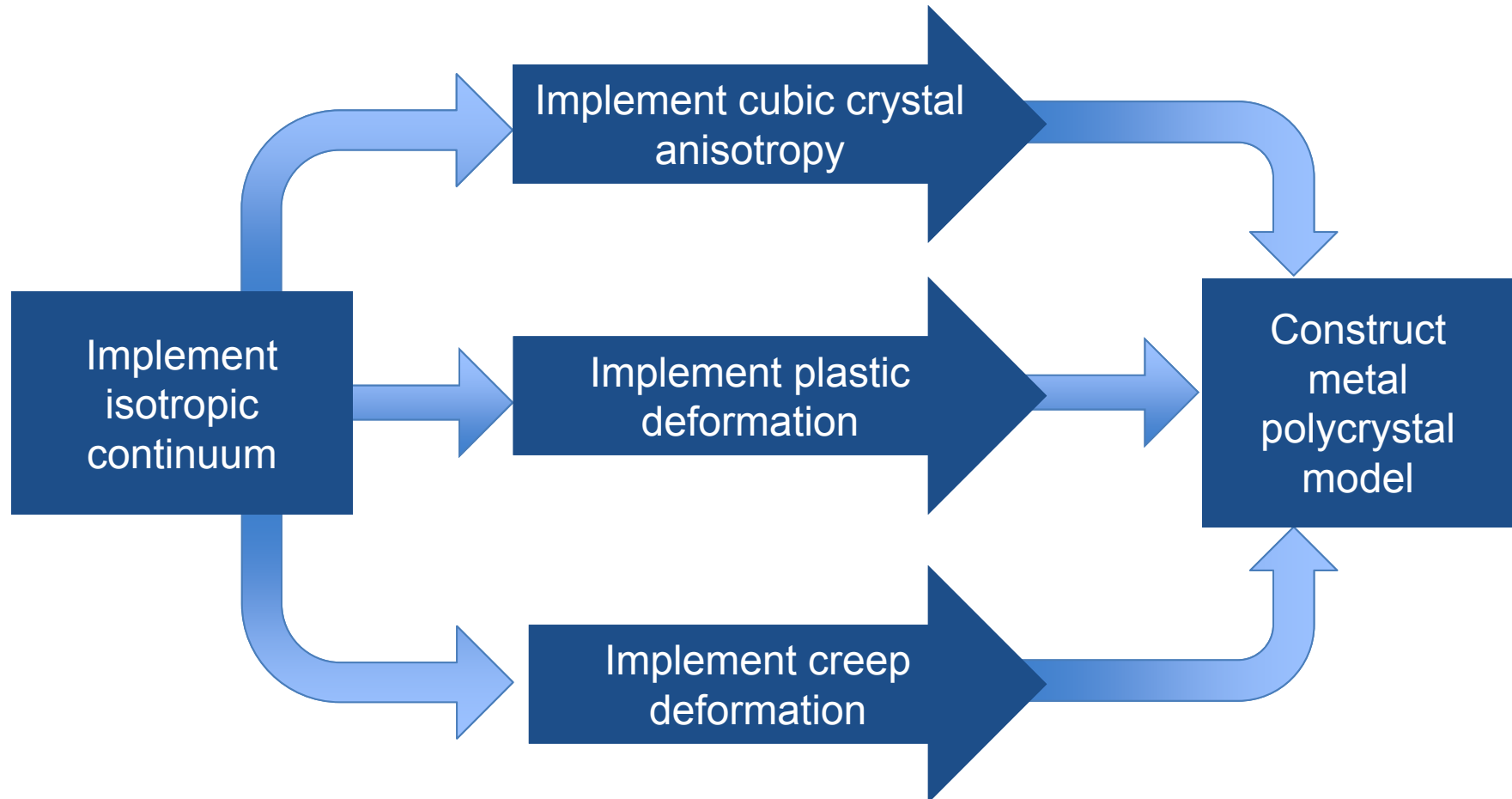


- Progress to date:

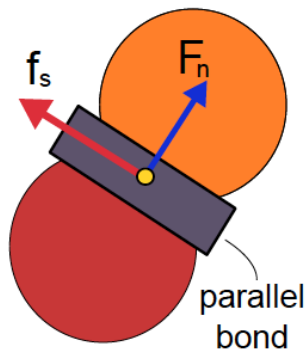
	C_{11} (GPa)	C_{12} (GPa)	C_{44} (GPa)
Current Model	237	77	73
Ni	251	150	124
Ni (600°C)	221	145	102
Fe	231	116	135
Cr	340	99	59
Si	166	80	64

↑
 C_{11} in line
with Ni-
alloys

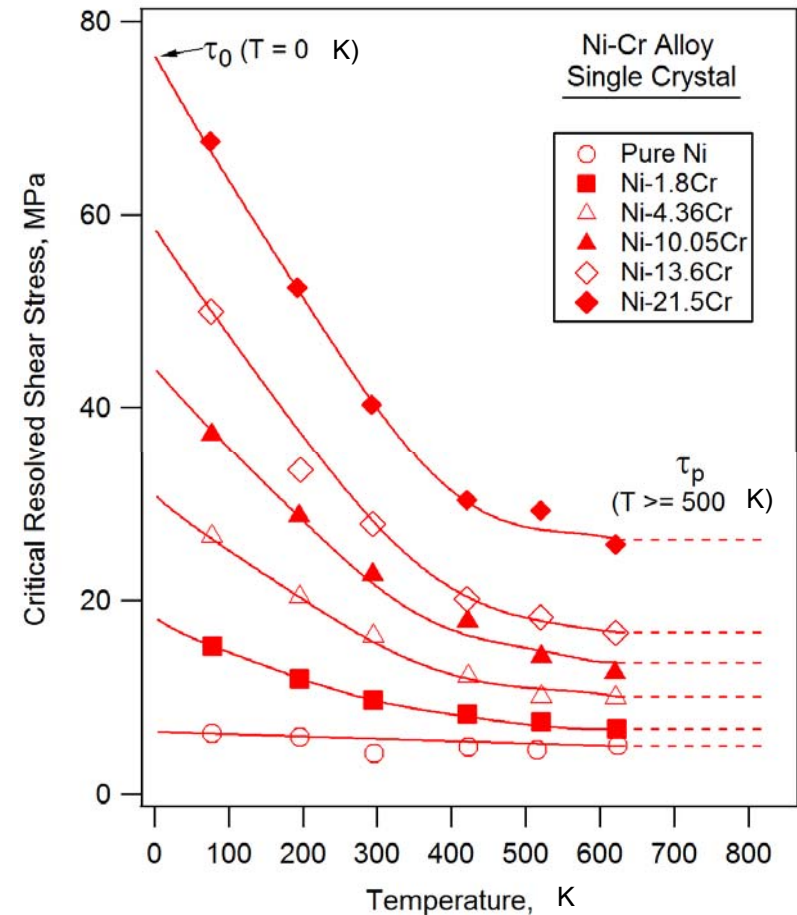
↑
 C_{12} and C_{44}
in line with
Si



- Critical resolved shear stress has a minimum plateau above ~500 K
- Literature data is being used to define parallel bond shear strengths in DEM model

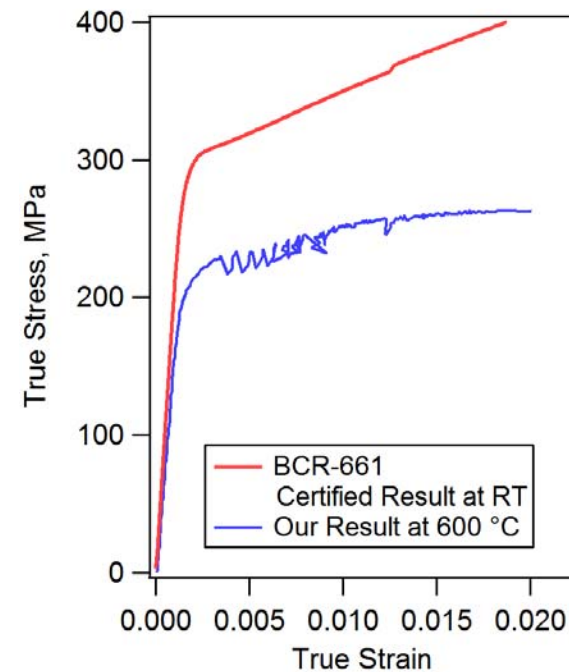
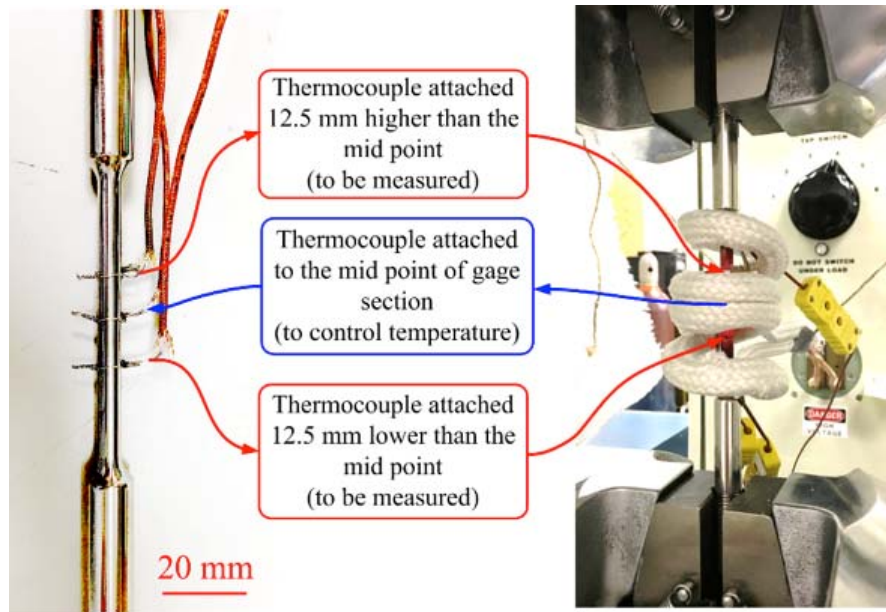


Various Ni-Cr alloys

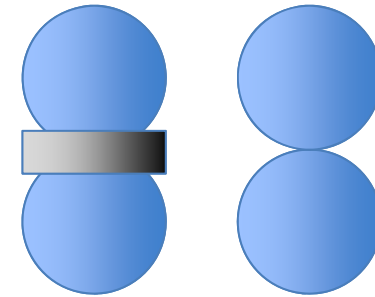


(Akhtar & Teghtsoo, 1971)

- Tensile testing underway on Nimonic 75
 - Determine hardening behavior at 600°C
 - Determine dynamic softening during stress relaxation at 600°C

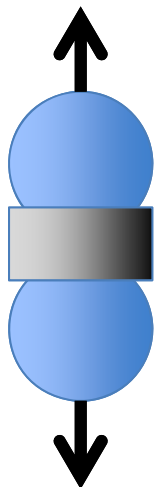


- Parallel bonded discrete elements:
 - Consider as meso-scale domains
 - Potential sub-grains

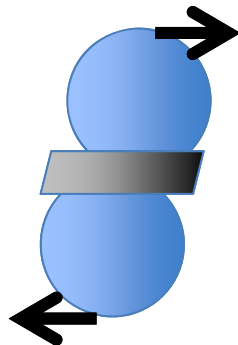


Potential Bond Breaking Phenomena

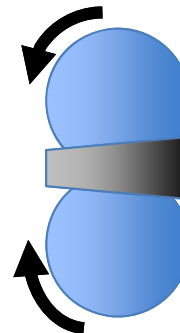
Normal stretch



Shear stretch

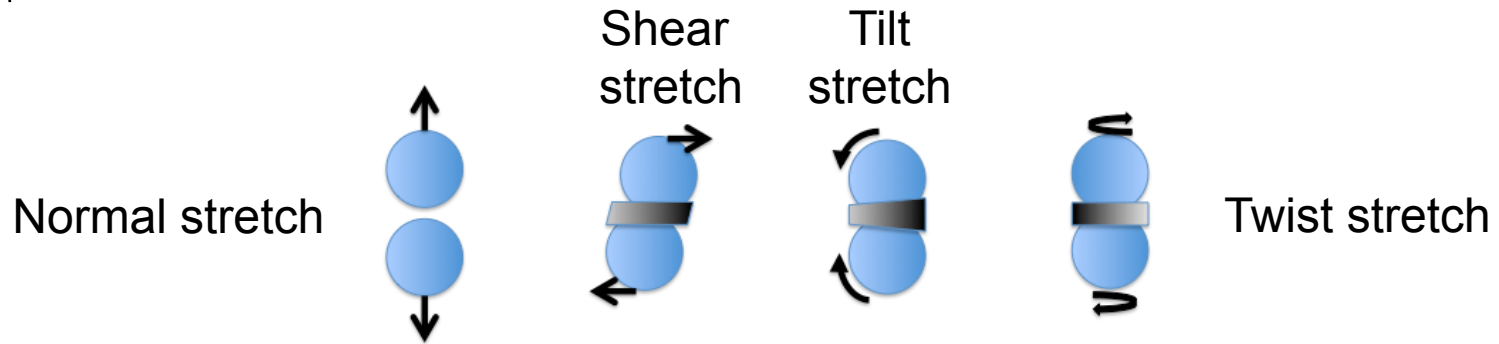


Tilt stretch



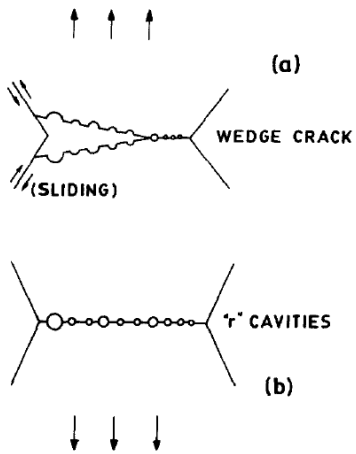
Twist stretch



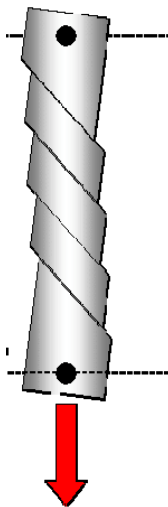


Corresponding Physical Phenomena

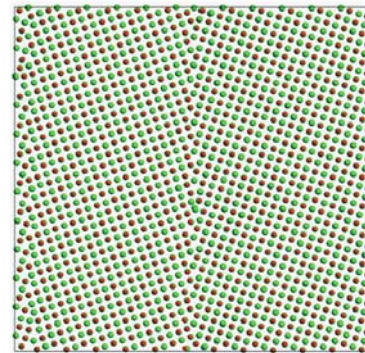
Crack or
Void Formation



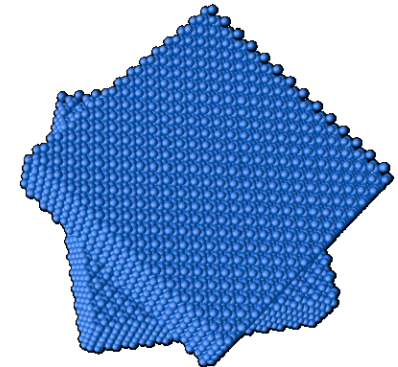
Plastic slip



Tilt Boundary
Formation

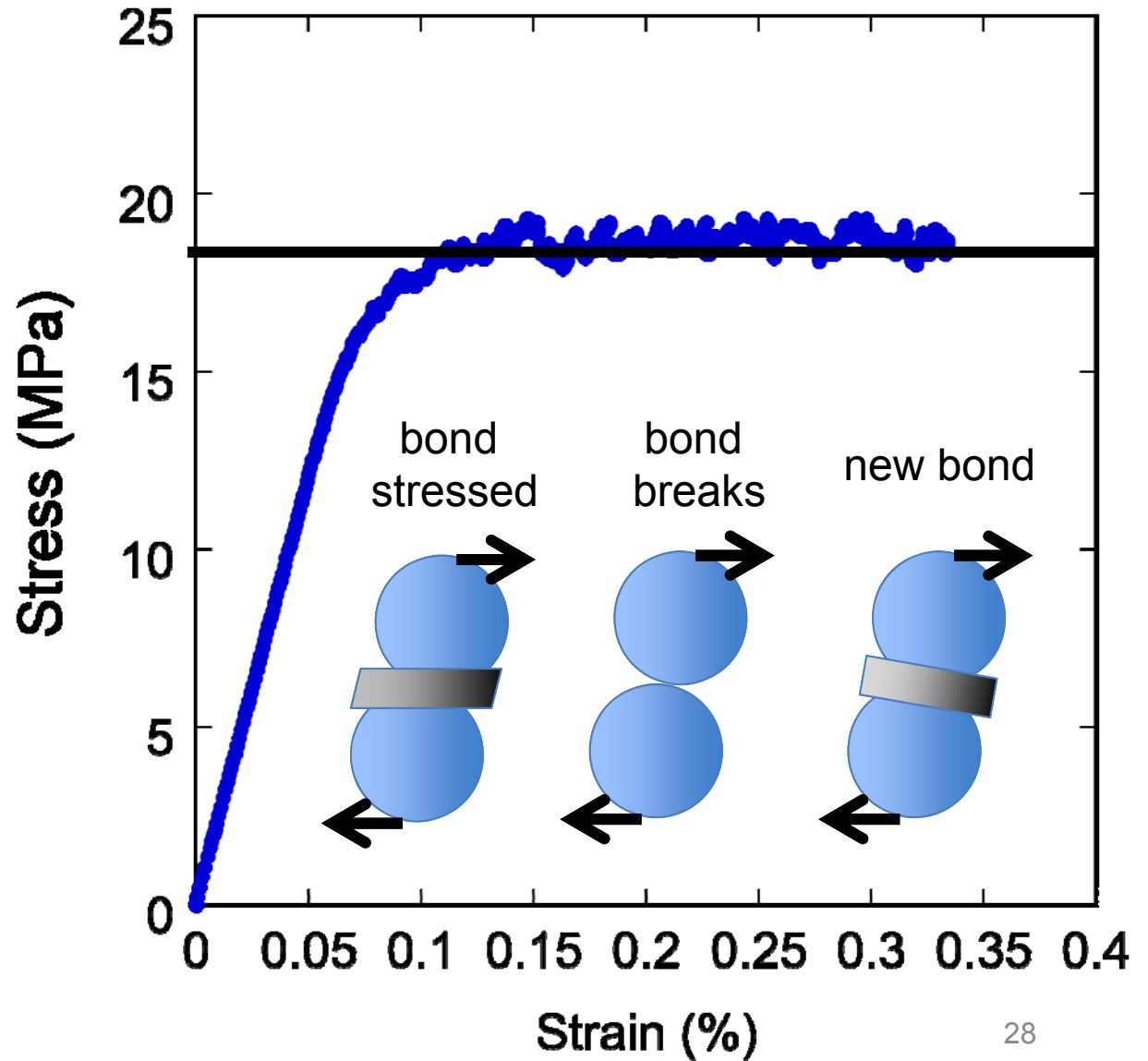
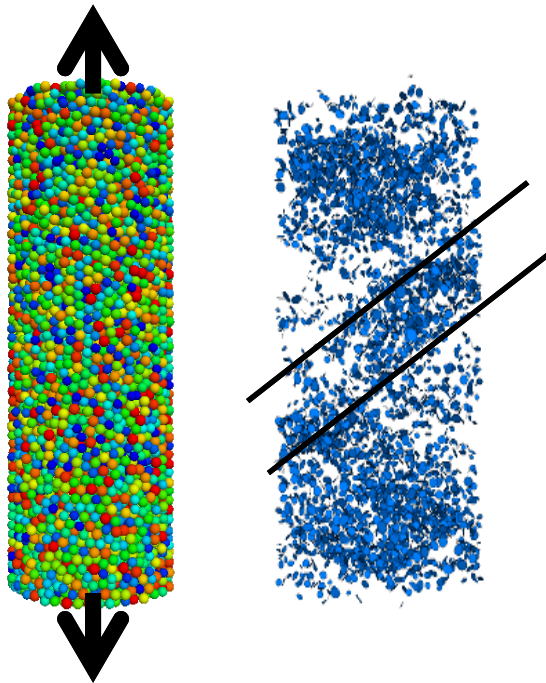


Twist Boundary
Formation

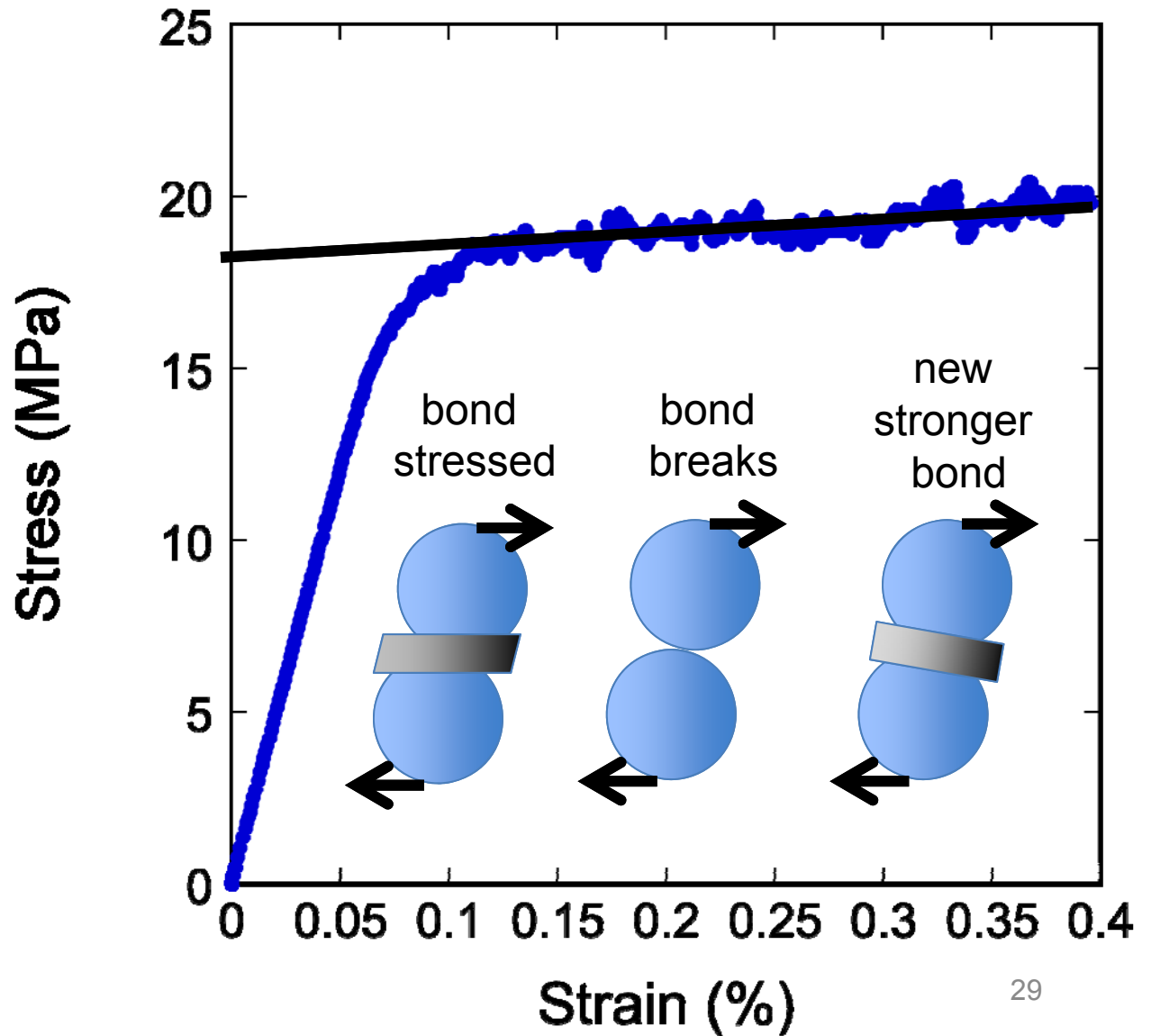
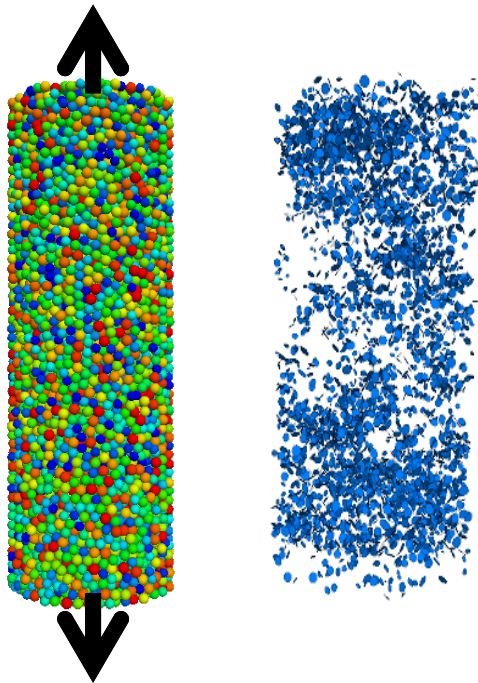


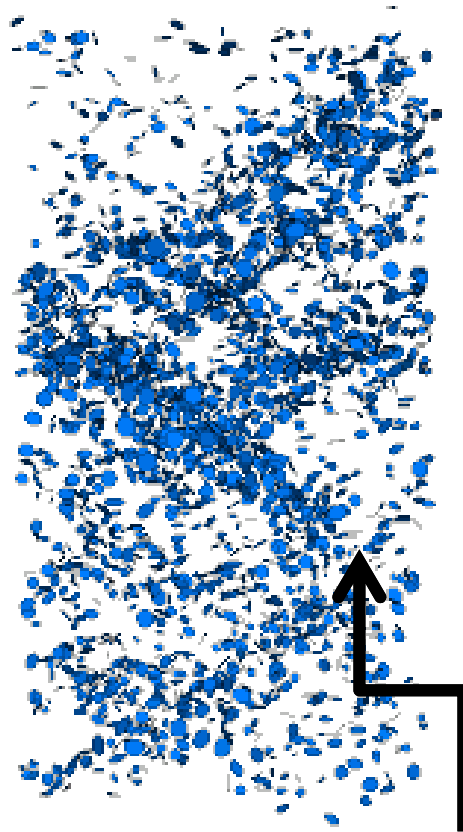
sub-grain evolution

- Non-hardening deformation
 - perfect plasticity
 - shear localization evolves

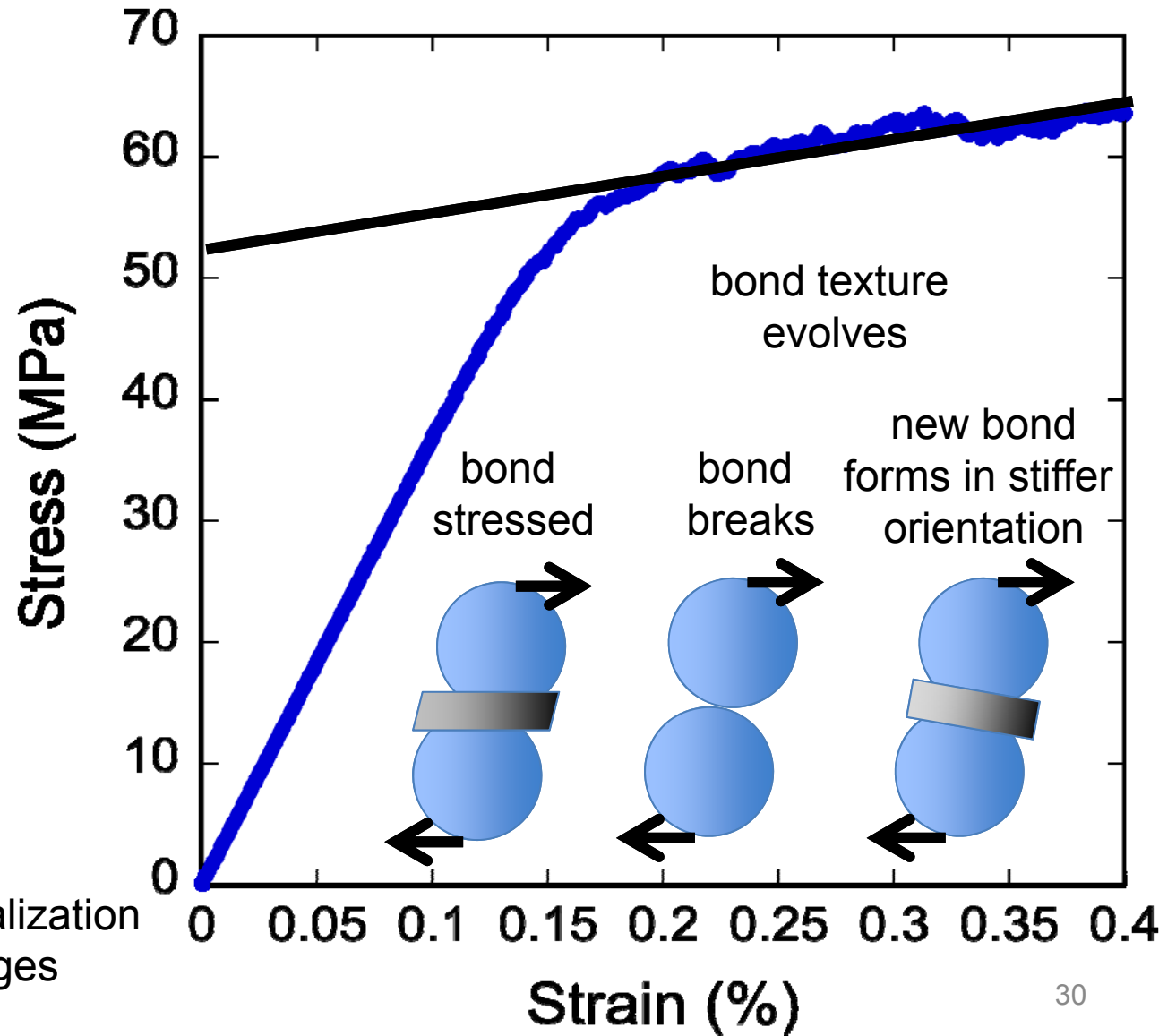


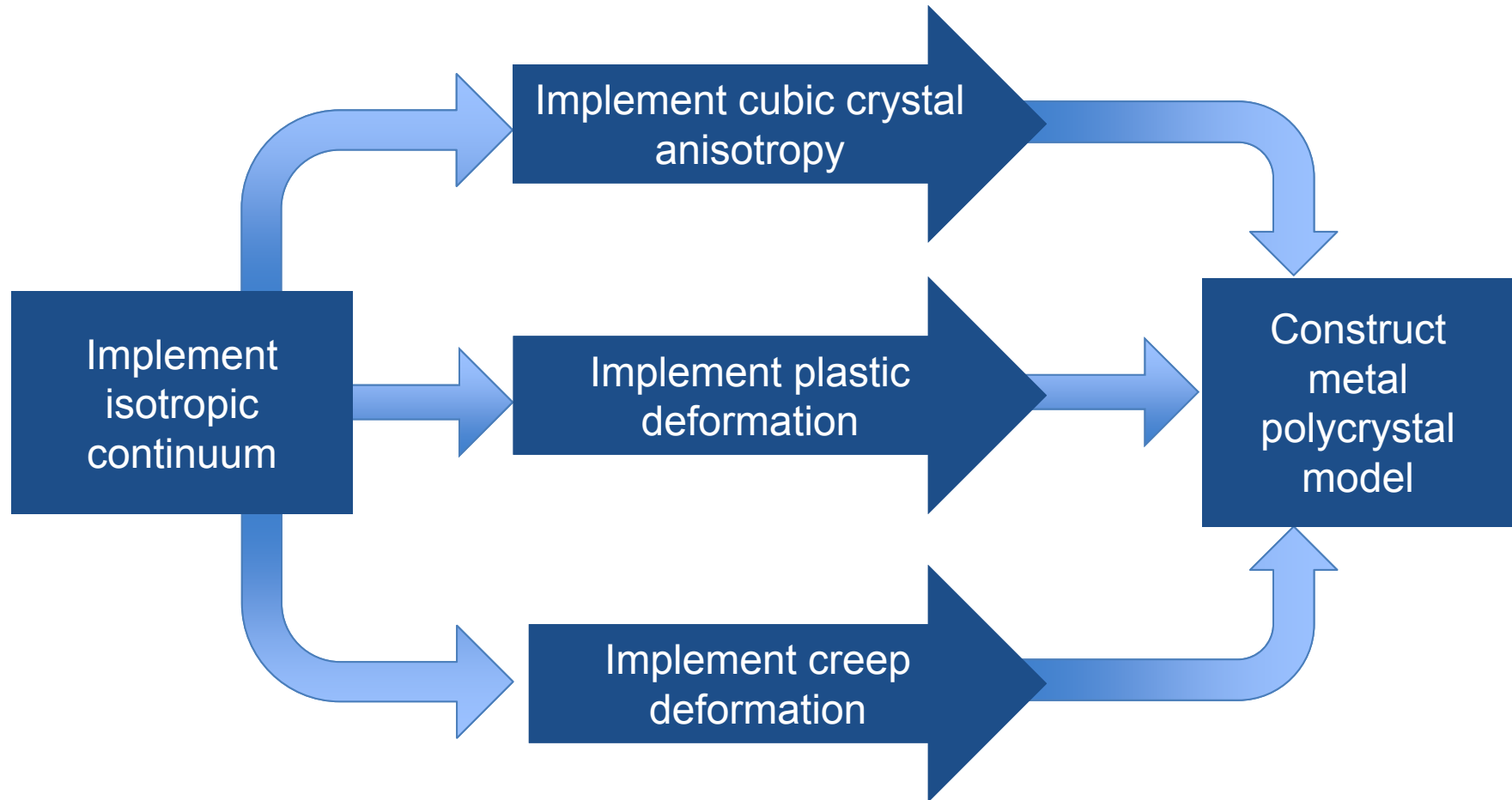
- Hardening deformation
 - Strain hardening
 - localization suppressed



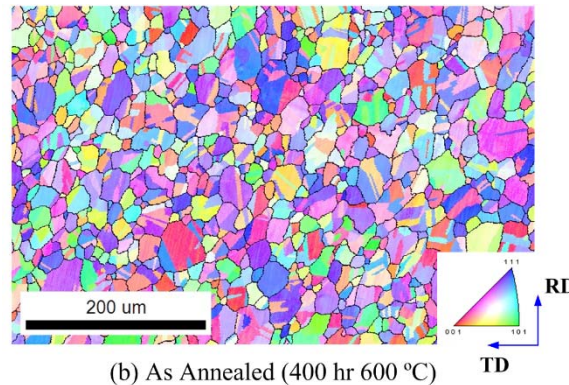


shear localization emerges



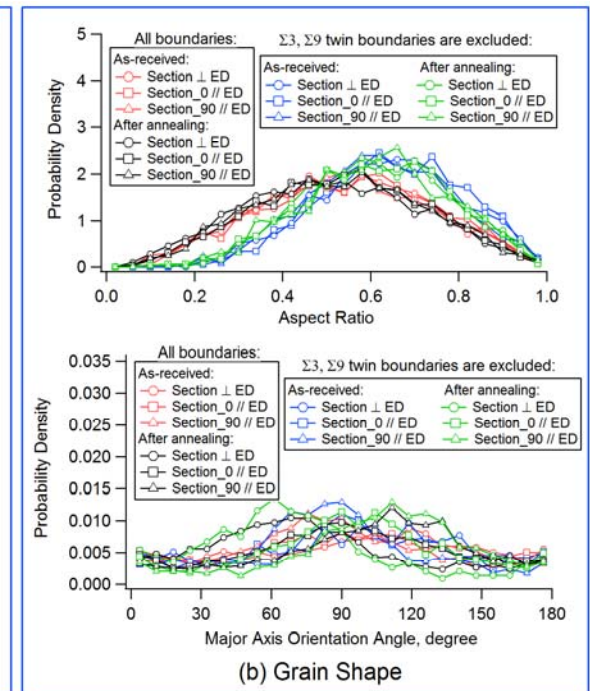
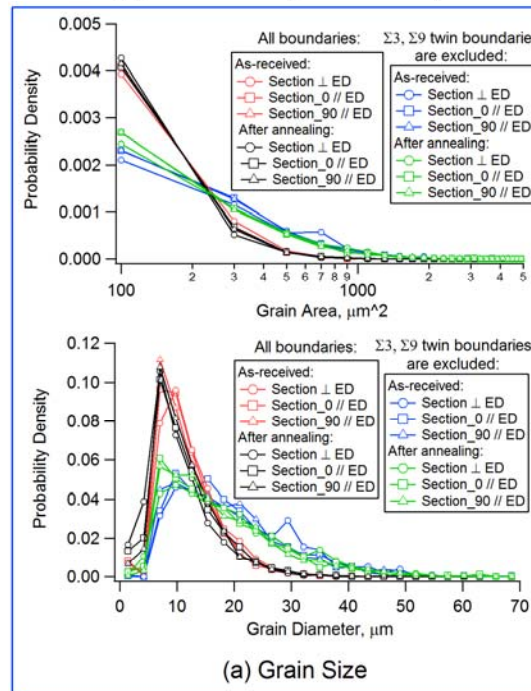
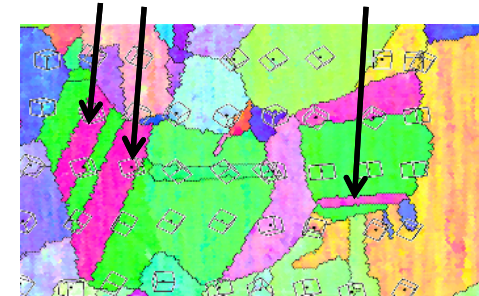


- EBSD used to quantify grain structure
 - Presence of twins skews apparent distributions
 - $\Sigma 3$ and $\Sigma 9$ annealing twin boundaries are unlikely damage sites (Zhang & Field, 2013)
 - Initially a twin-free microstructure is created for our DEM model



(b) As Annealed (400 hr 600 °C)

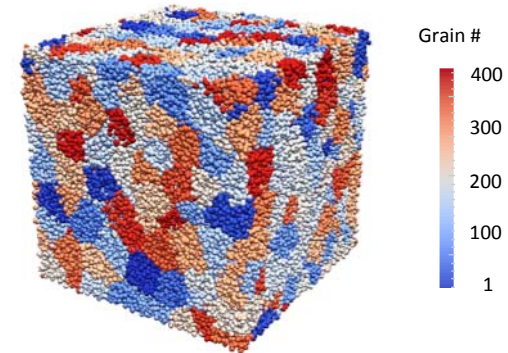
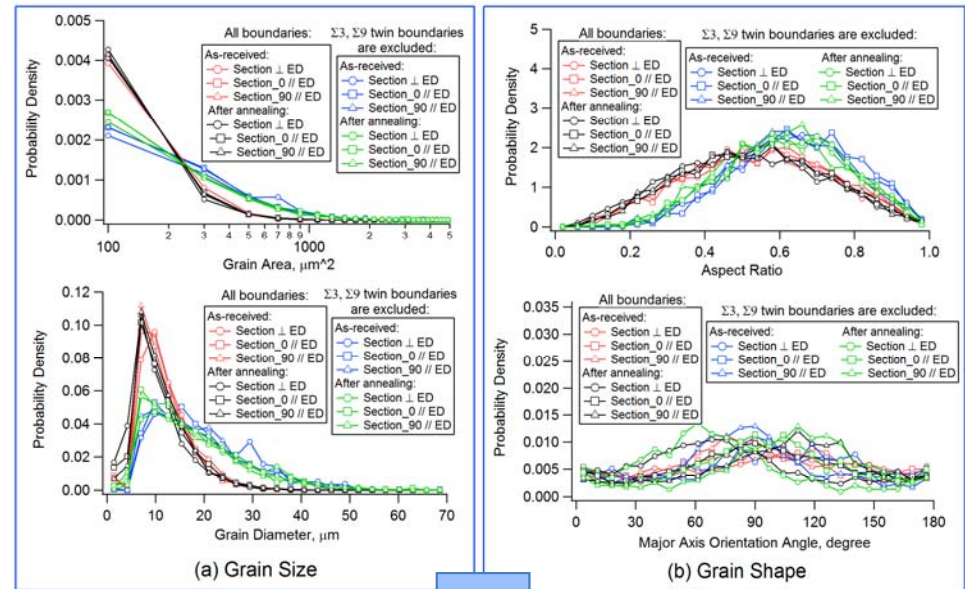
Twins are prevalent



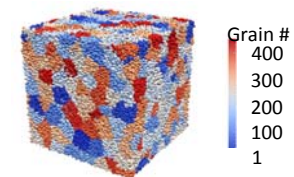
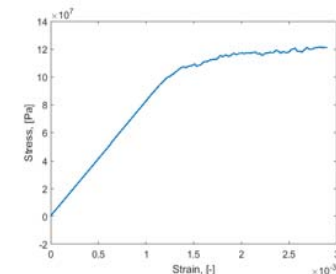
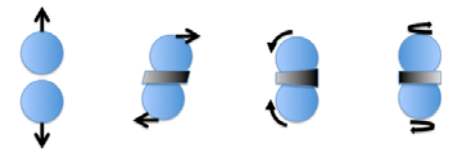
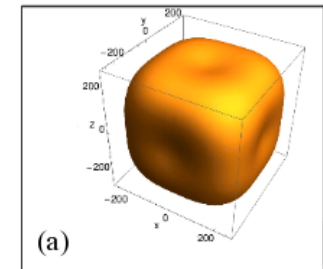
- A 3-D Voronoi algorithm for crystal plasticity has been adapted for making a polycrystalline DEM assembly
- Assembly captures essential grain size/shape statistics
- Microstructure also being measured in steady state creep regime
 - Steady state microstructure will be used for model



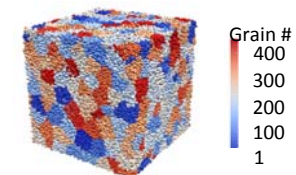
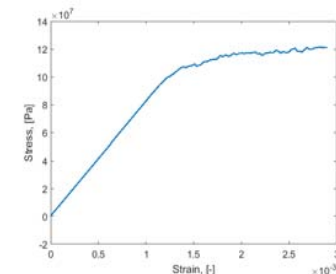
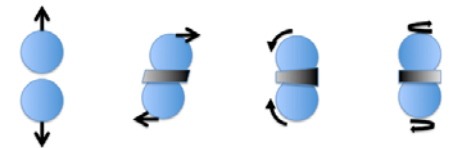
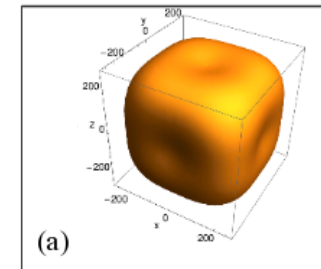
10 mm
creep samples tested at NETL



- An anisotropic elasticity formulation can be developed to mimic cubic anisotropy
 - We will tweak formulation to access the exact desired stiffness tensor
- Bond breaking and reforming can be used to simulate metal plasticity
 - Elastic and plastic behavior is coupled based on our initial results
 - Will refine model to capture desired strength, hardening, and slip plane orientation behavior



- Elastic Anisotropy:
 - Run large DFT-MD simulation to verify constants at 600°C for Ni-20Cr
 - Refine element stiffness formulation to get correct tensor for Ni-20Cr alloy
- Plasticity:
 - Determine strain hardening law at 600°C for Nimonic 75 using tensile samples as function of strain rate
 - Refine bond breaking and reforming scheme to capture desired strength, hardening, and slip plane orientation behavior
- Creep:
 - 600°C creep response well known from literature
 - Determine softening laws at 600°C for Nimonic 75 using tensile stress relaxation experiments
 - Develop time dependent bond breaking and reforming scheme to capture creep behavior
- Final Model Assembly:
 - Combine above elements into a crystal plasticity DEM model for Nimonic 75
 - Validate and refine model based on experimental creep results

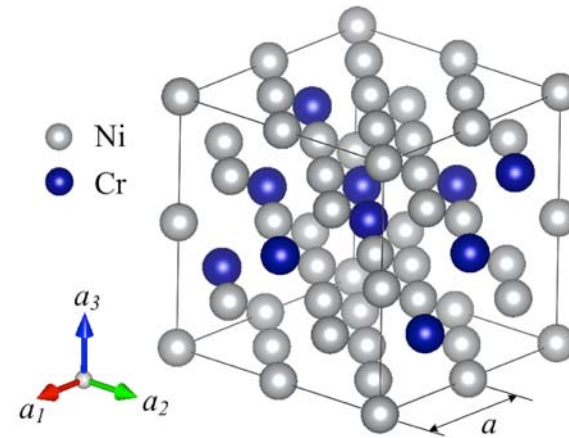




Questions?



- Density Functional Theory (DFT) used to calculate C_{11} , C_{12} , C_{44} for Ni-20Cr at 0 K
 - Special quasi-random structure (SQS) supercell
 - Only small deviation from pure Ni seen in C_{11}



Material	Equilibrium lattice parameter, a , Å	Bulk modulus B , GPa	Zener's modulus C' , GPa	C_{11} , GPa	C_{12} , GPa	C_{44} , GPa
Pure Ni (2x2x2 supercell)	3.518	200.1	55.7	274.3	163.0	128.8
Ni-20Cr (2x2x2 SQS supercell)	3.523	204.1	60.0	284.1	164.1	131.7

- MD simulations at $T > 0K$ using embedded atom method and a 3 x 3 SQS supercell
- For pure Ni, we can compare elastic softening of C_{11} , C_{12} , C_{44} to published data
- In both cases, fitting elastic softening to Watchman functional is appropriate
- Future DFT-MD simulation at 600°C will verify extrapolation

