



# Degradation and Performance Studies of ALD Stabilized SOFC Nano-Composite Cathodes (NCCs)

## 18 Month Update

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Dr. Kevin Huang,<sup>1</sup> Dr. Praveen Cheekatamarla  
and Dr. Jason D. Nicholas<sup>2</sup>

Funded by the Department of Energy Solid Oxide Fuel Cell Core Technology Program through Agreement Number DE-FE0031672

Program Manager: Joe Stoffa

May 1, 2019



# High Performance Circuit Pastes for Solid Oxide Fuel Cell Applications

## 6 Month Update

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Ms. Jiyun Park,<sup>1</sup> Mr. Robert Termuhlen,<sup>1</sup> Dr. Yue Qi,<sup>1</sup>  
Dr. Thomas Beiler,<sup>1</sup> Dr. Timothy Hogan,<sup>1</sup>  
Dr. Hui-Chia Yu,<sup>1</sup> Dr. Praveen Cheekatamarla<sup>2</sup> and  
Dr. Jason D. Nicholas<sup>1</sup>

Funded by the Department of Energy Solid Oxide Fuel Cell Core Technology  
Program through Agreement Number DE-FE0031250

Program Manager: Venkat Venkatraman



The left side of the slide features three vertical SEM images showing the morphology of nano-composite cathodes. The top image shows a dense, interconnected network of dark, rounded particles. The middle image shows a similar structure with slightly larger, more distinct particles. The bottom image shows a similar structure with even larger, more rounded particles. Each image is partially overlaid by a black vertical bar on its right side.

# TALK Outline

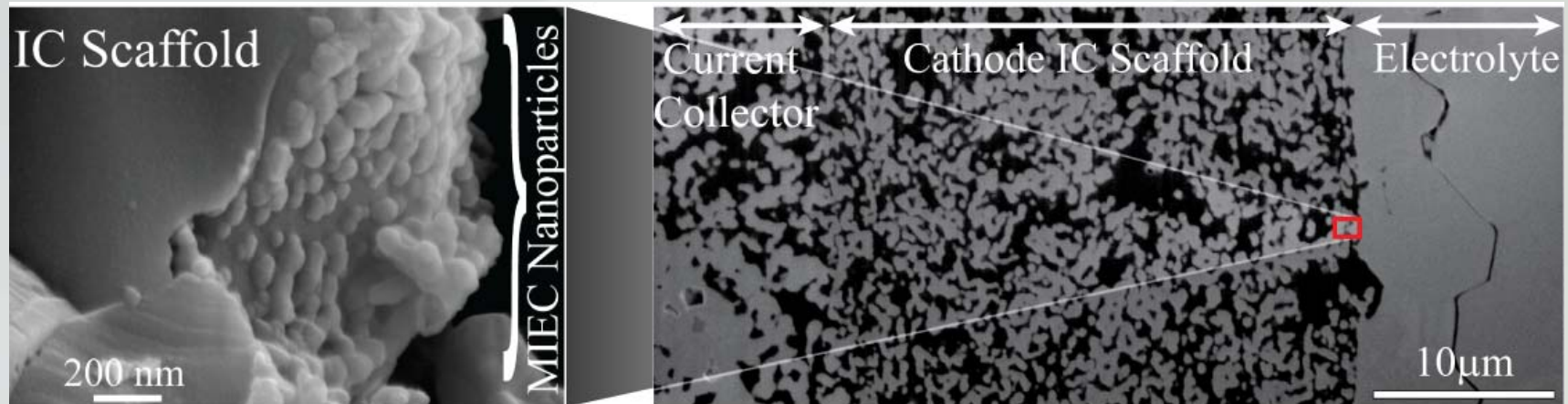
## ALD-Stabilized Nano-Composite Cathodes

1. Motivation
2. Experimental Methods
3. Results and Discussion

## Ni-Ag Circuits Pastes

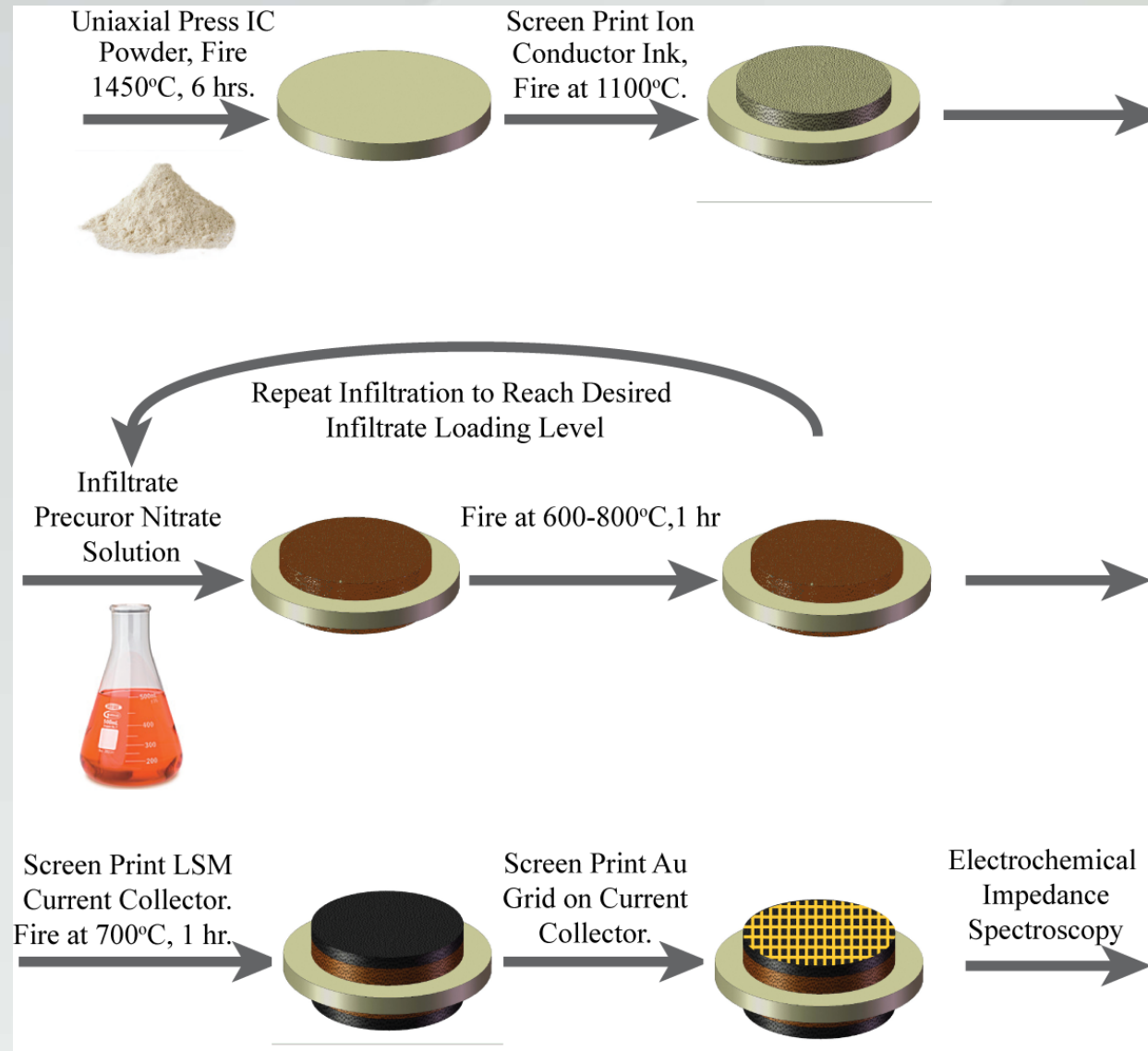
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# Motivation for MIEC on IC Nano-Composite SOFC Cathodes



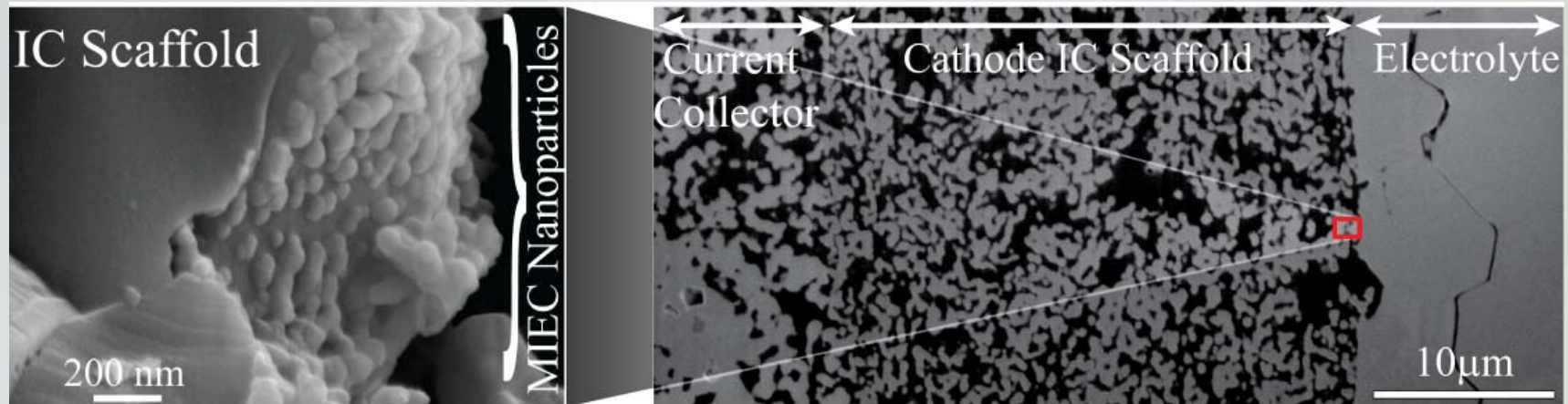
Material	Oxygen Surface Exchange Coefficient (k)	Bulk Oxygen Ion Diffusivity (D)	Electronic Conductivity ( $\sigma_e$ )
Mixed Ionic Electronic Conductor (MIEC)	High	Low	High
Ionic Conductor (IC)	Low	High	Low

# Standard Nano-Composite Cathode Fabrication



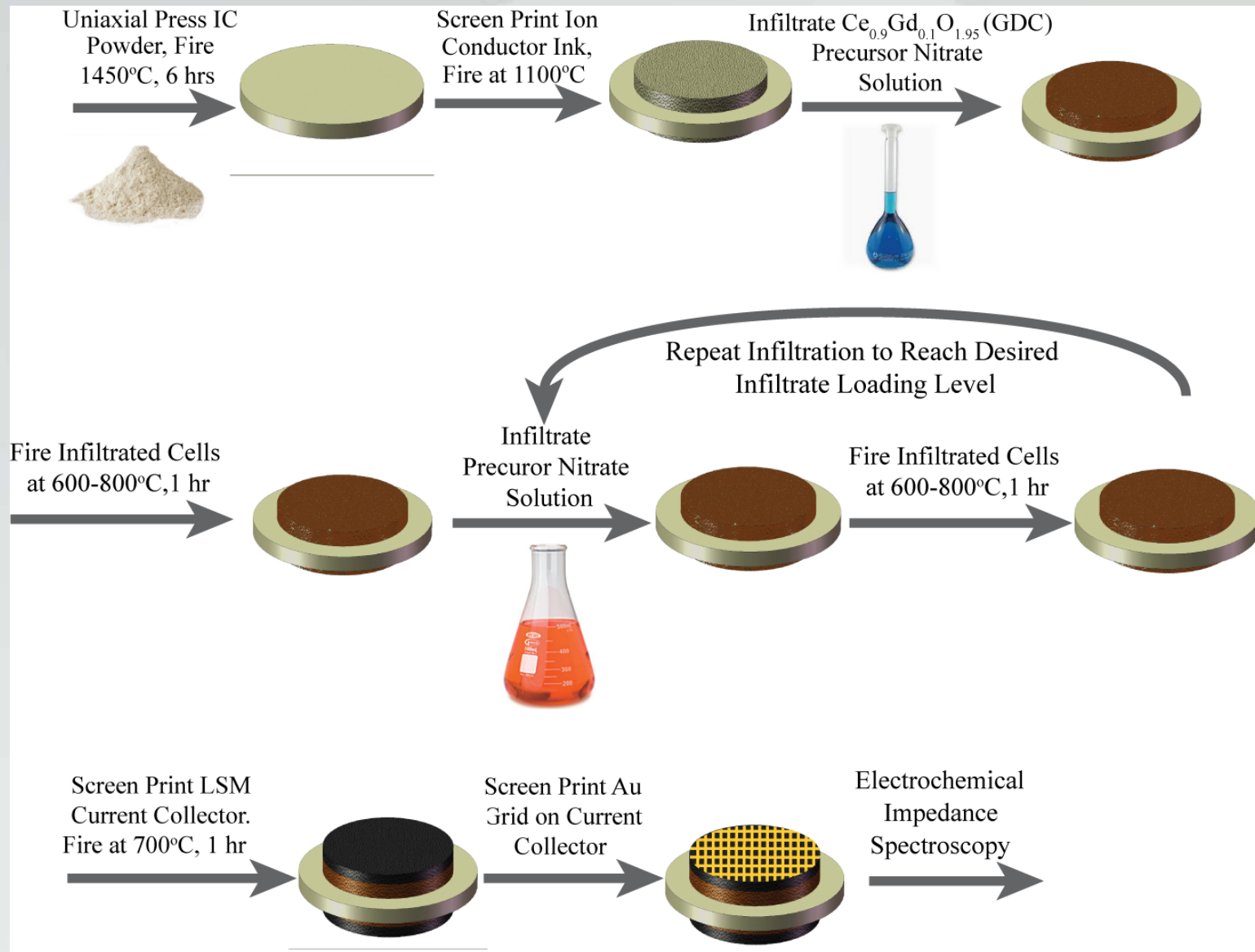


# Motivation for MIEC on IC Nano-Composite SOFC Cathodes



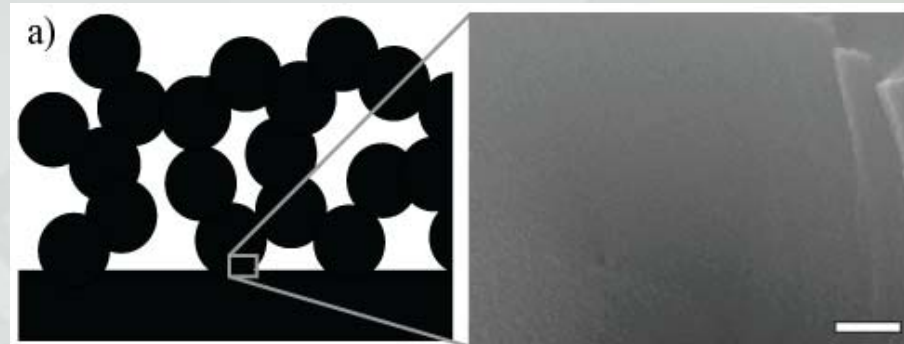
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Mixed Ionic Electronic Conductor (MIEC)	High	Low	High
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# GDC Pre-Infiltrated Nano-Composite Cathode Fabrication

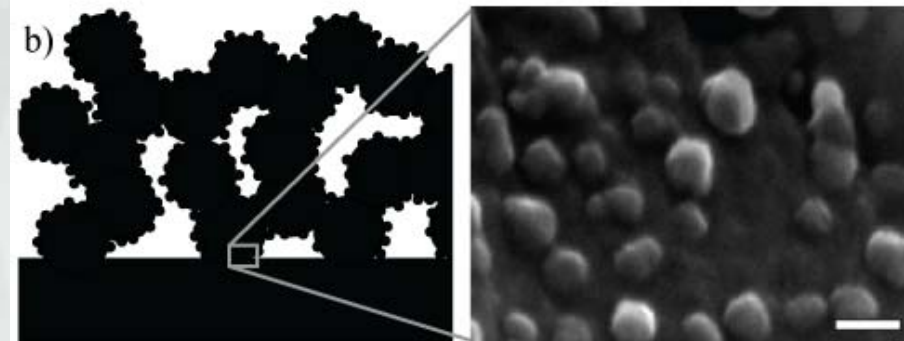


# Nano-Composite Cathode Fabrication With Ceria Pre-infiltration

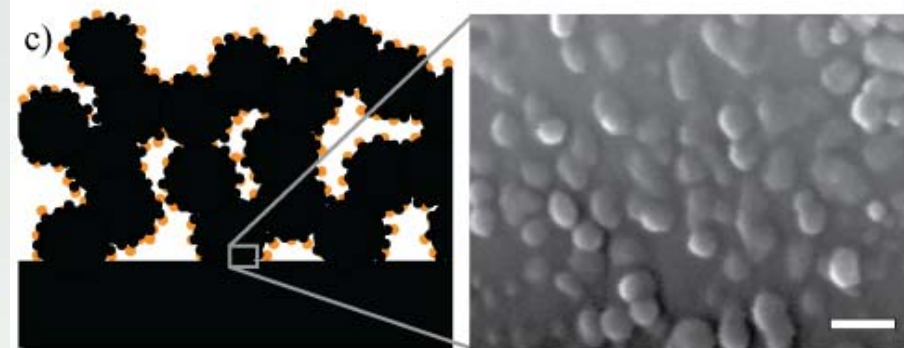
Uninfiltrated Surface



Ceria Pre-Infiltrated Surface



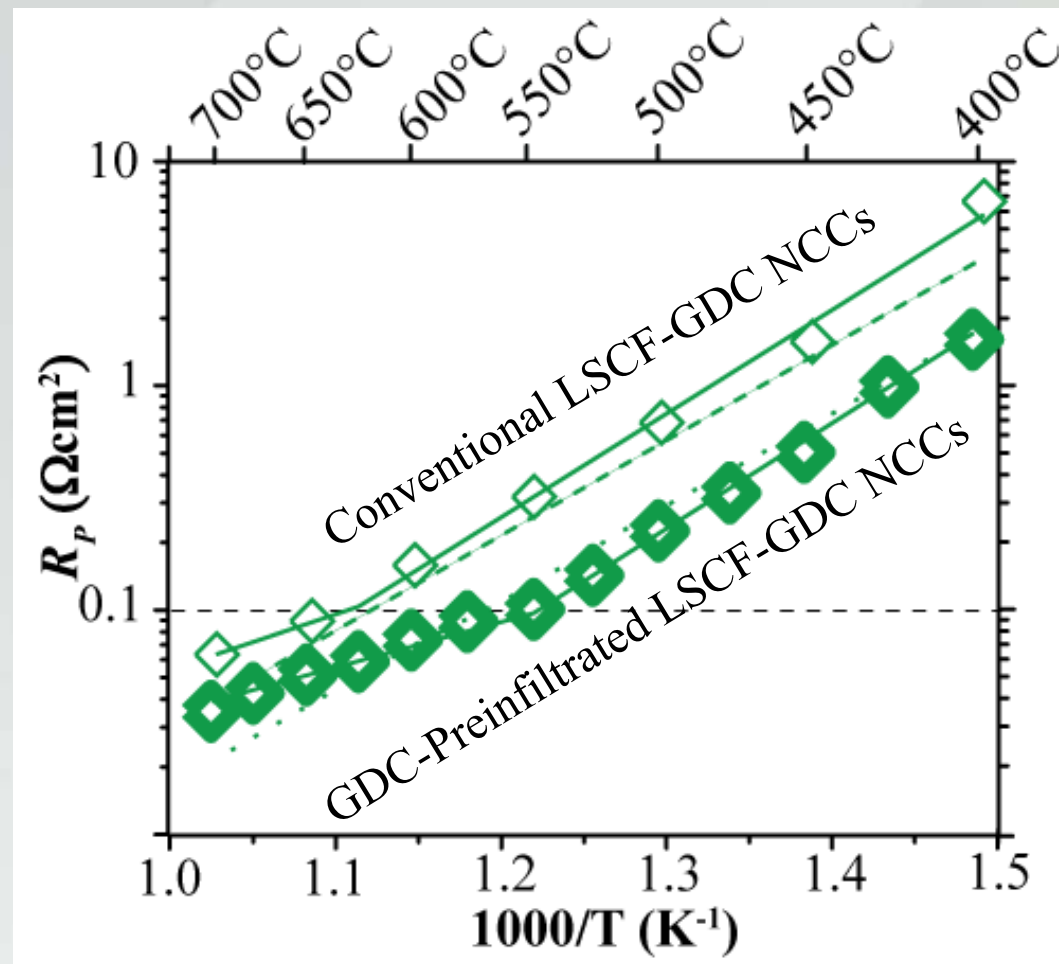
Ceria and LSCF Infiltrated Surface



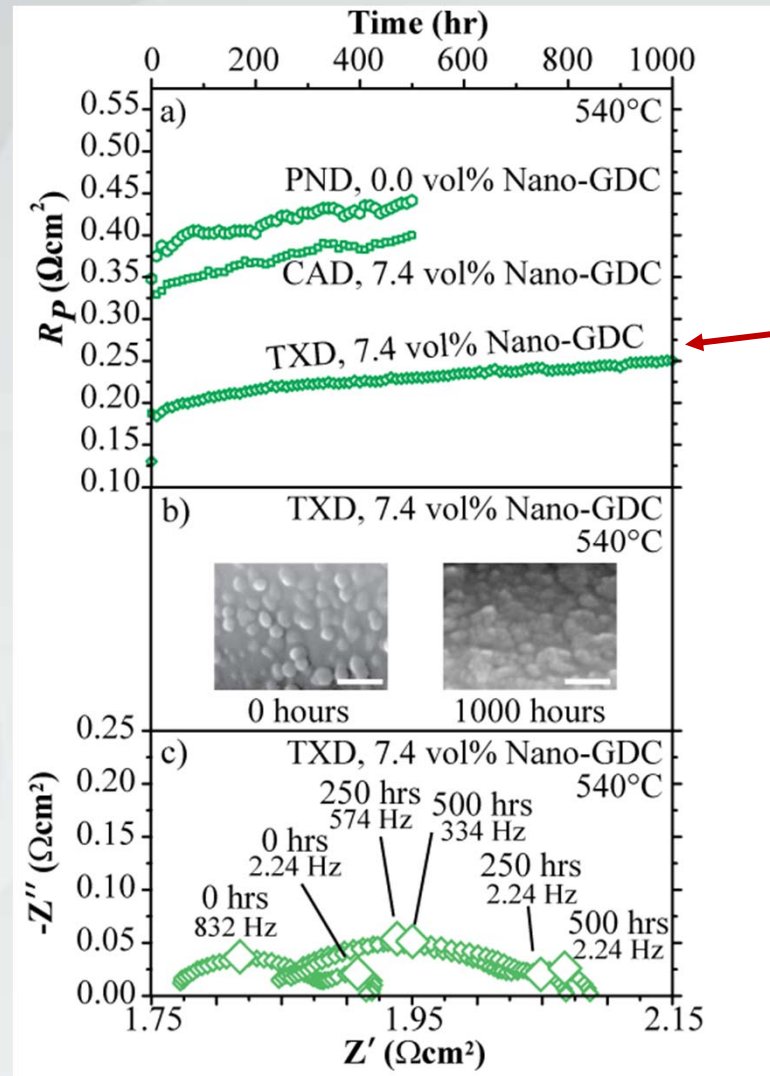
All scale bars are 50 nm wide



# Standard LSCF-GDC Shows Decent Performance and GDC Pre-Infiltration Decreases Cathode Resistance

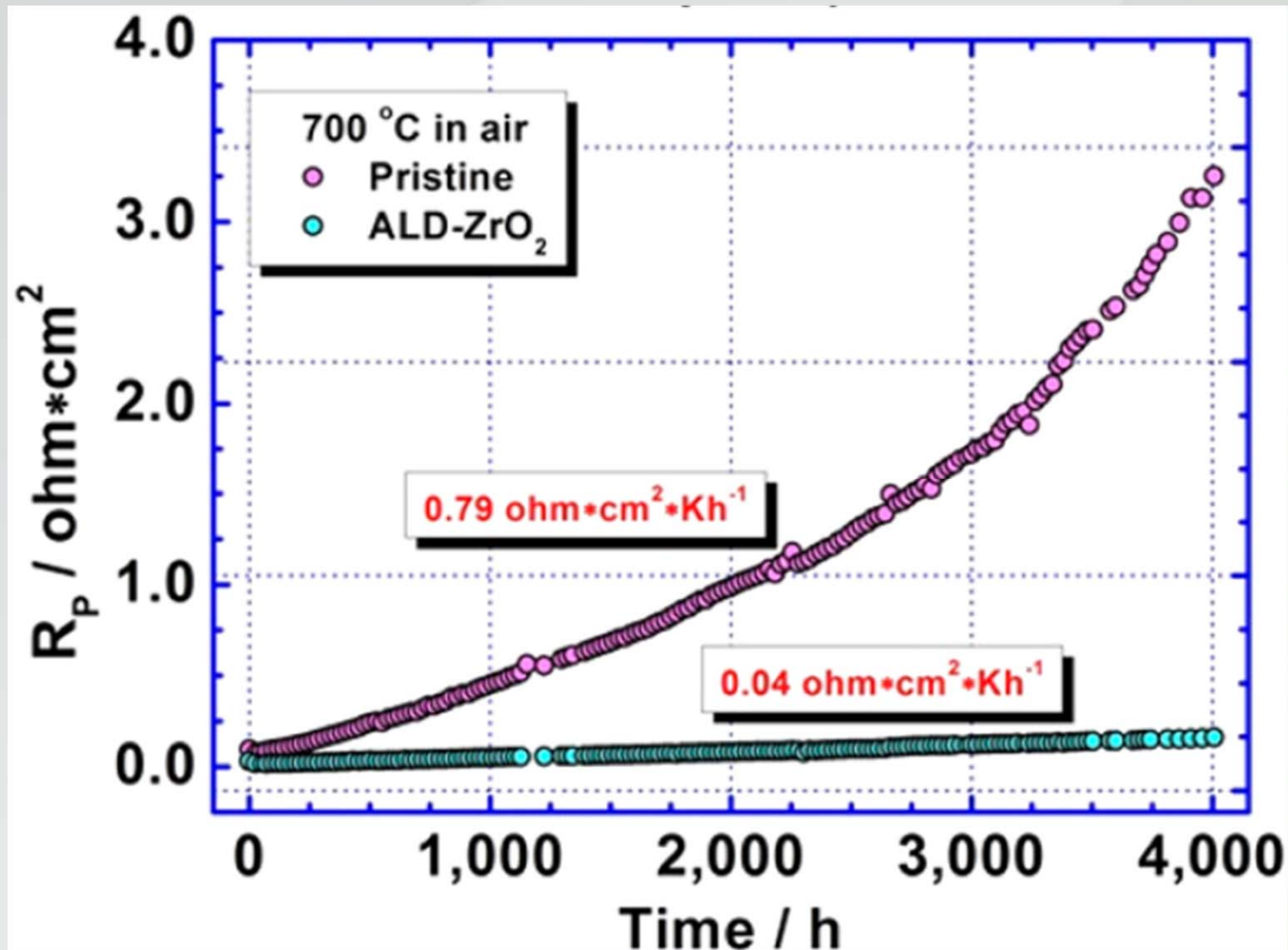


# Uncoated GDC Pre-Infiltrated LSCF-GDC Cells Have Stability Problems at 550°C Under Open Circuit Conditions



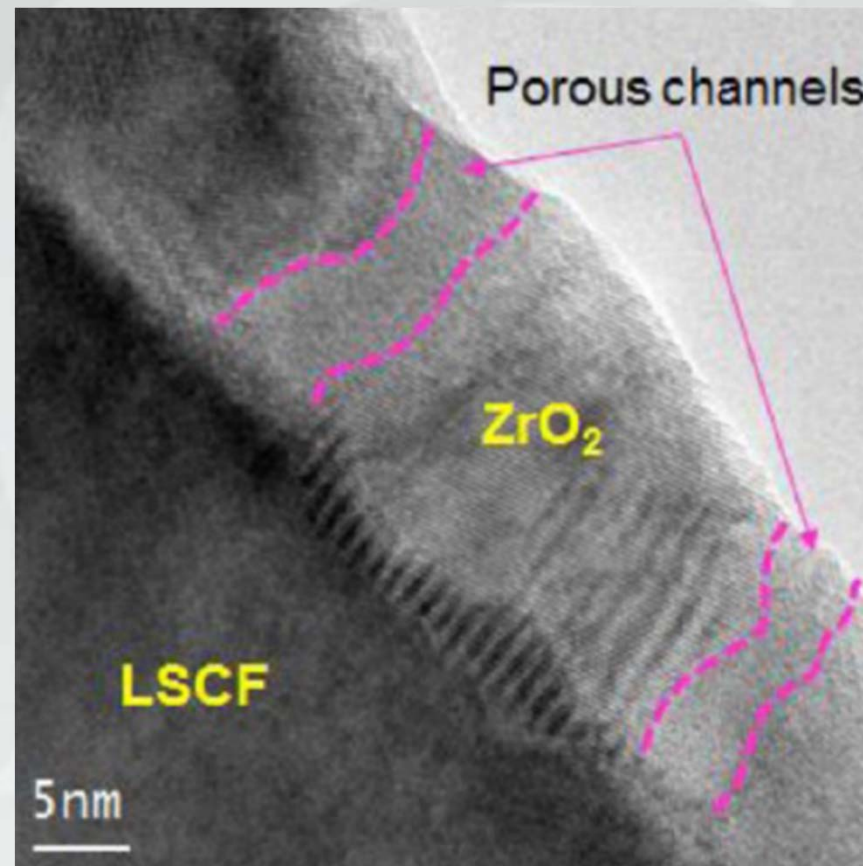
39 %/khrs after initial  
 “break-in”

## ZrO<sub>2</sub> Thin Film Deposited by ALD Could Greatly Improve the Stability of SOFC NCCs



LSC (La<sub>0.6</sub>Sr<sub>0.4</sub>CoO<sub>3-δ</sub>) nanoparticles supported on a porous LSGM (La<sub>0.8</sub>Sr<sub>0.2</sub>Ga<sub>0.83</sub>Mg<sub>0.17</sub>O<sub>3-δ</sub>) scaffold

## The Morphology of the ALD Coatings Grown by the Huang Group is More Porous than the Dense Conformal Coatings Reported by Other Groups



The left side of the slide features three vertical SEM images showing the morphology of nano-composite cathodes. The top image shows a dense, interconnected network of small, dark, rounded particles. The middle image shows a similar structure but with a more pronounced, porous appearance. The bottom image shows a similar structure with a more uniform, rounded morphology. Each image is partially overlaid by a black vertical bar on its right side.

# TALK Outline

## ALD-Stabilized Nano-Composite Cathodes

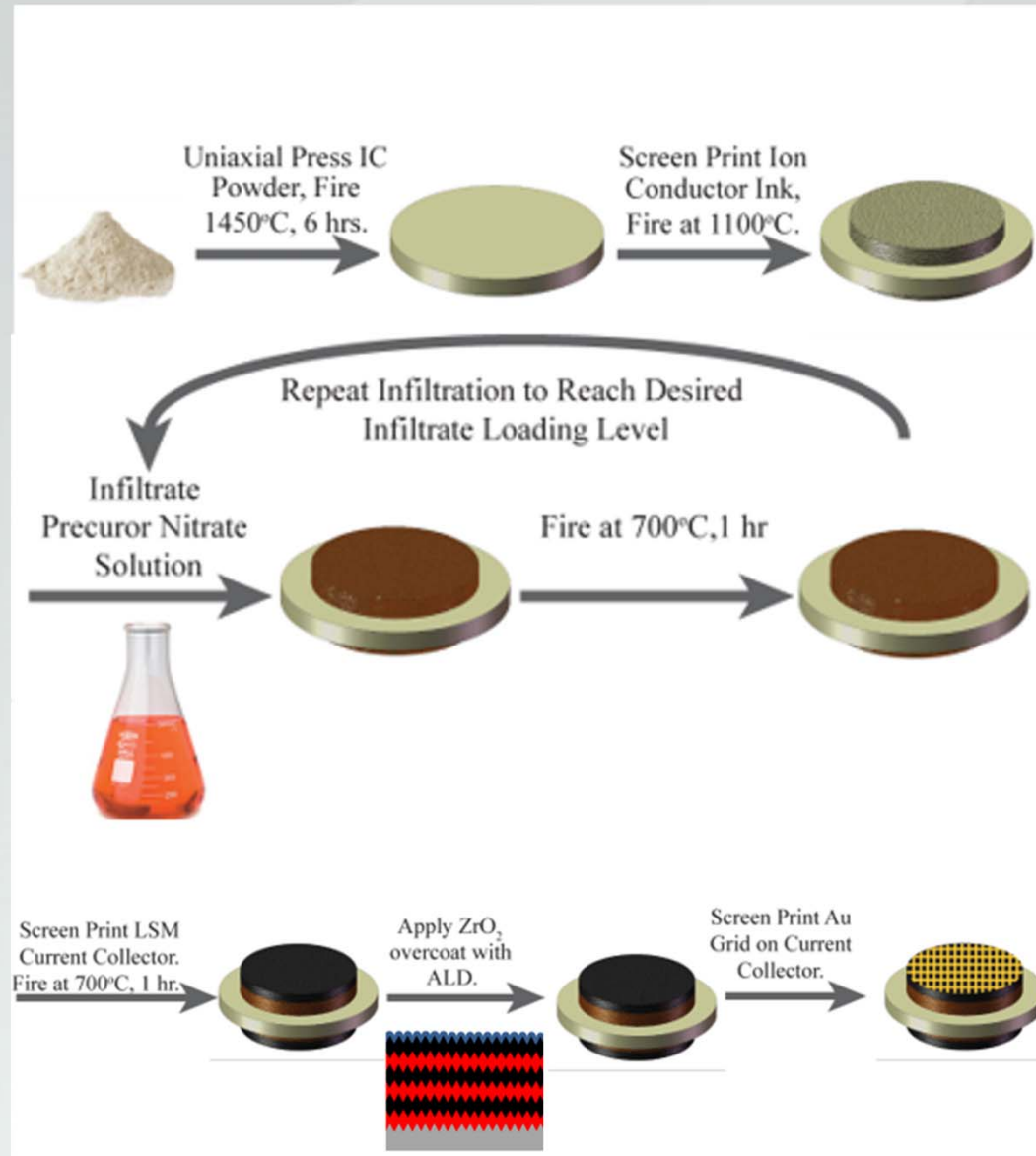
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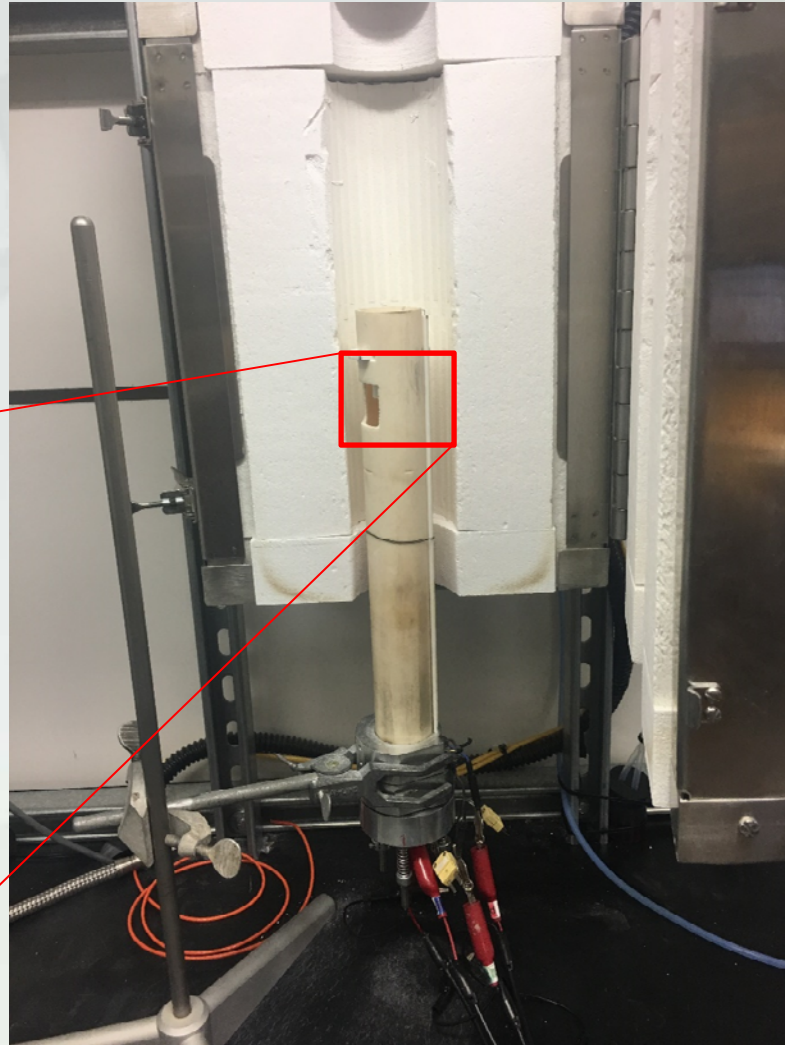
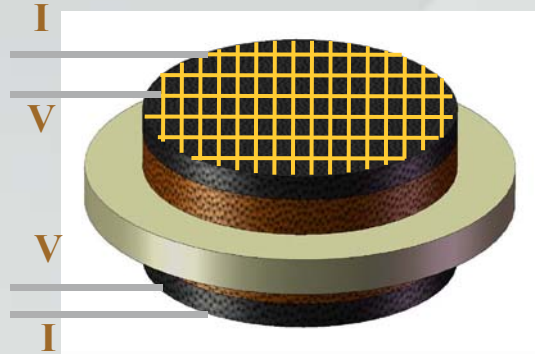
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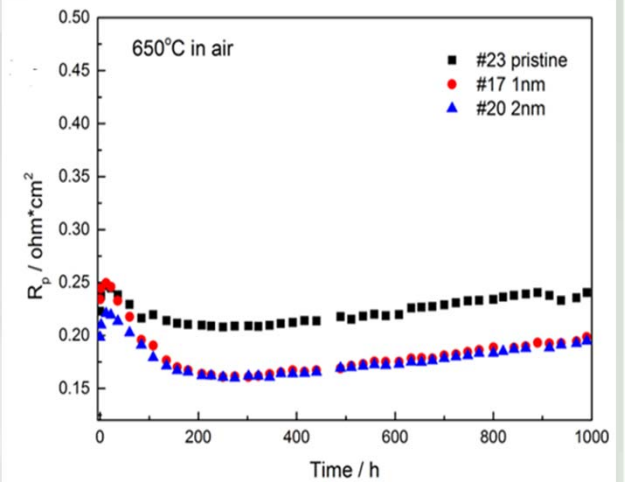
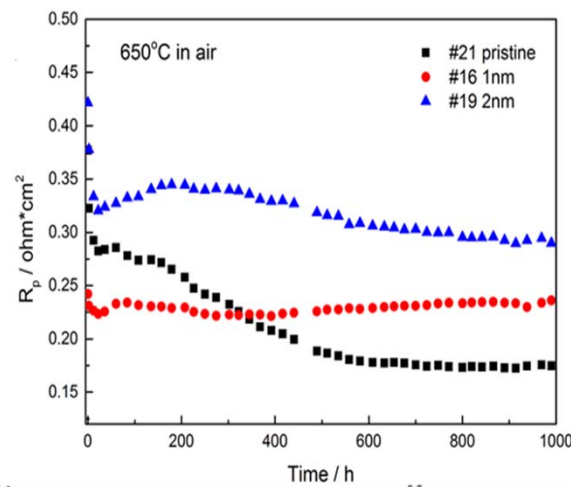
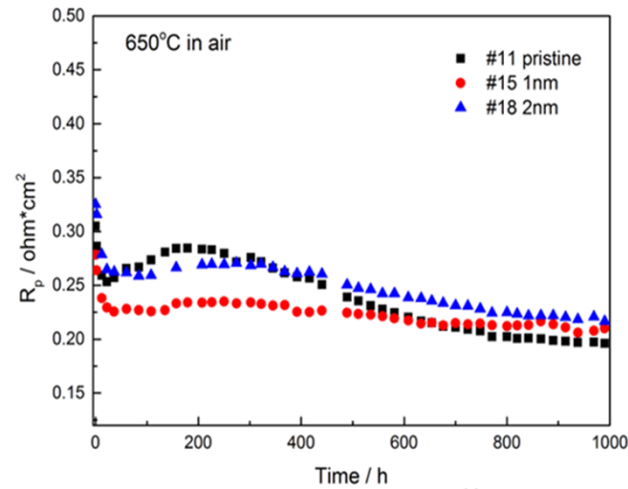
# ZrO<sub>2</sub> Coated Nano-Composite Cathode Fabrication



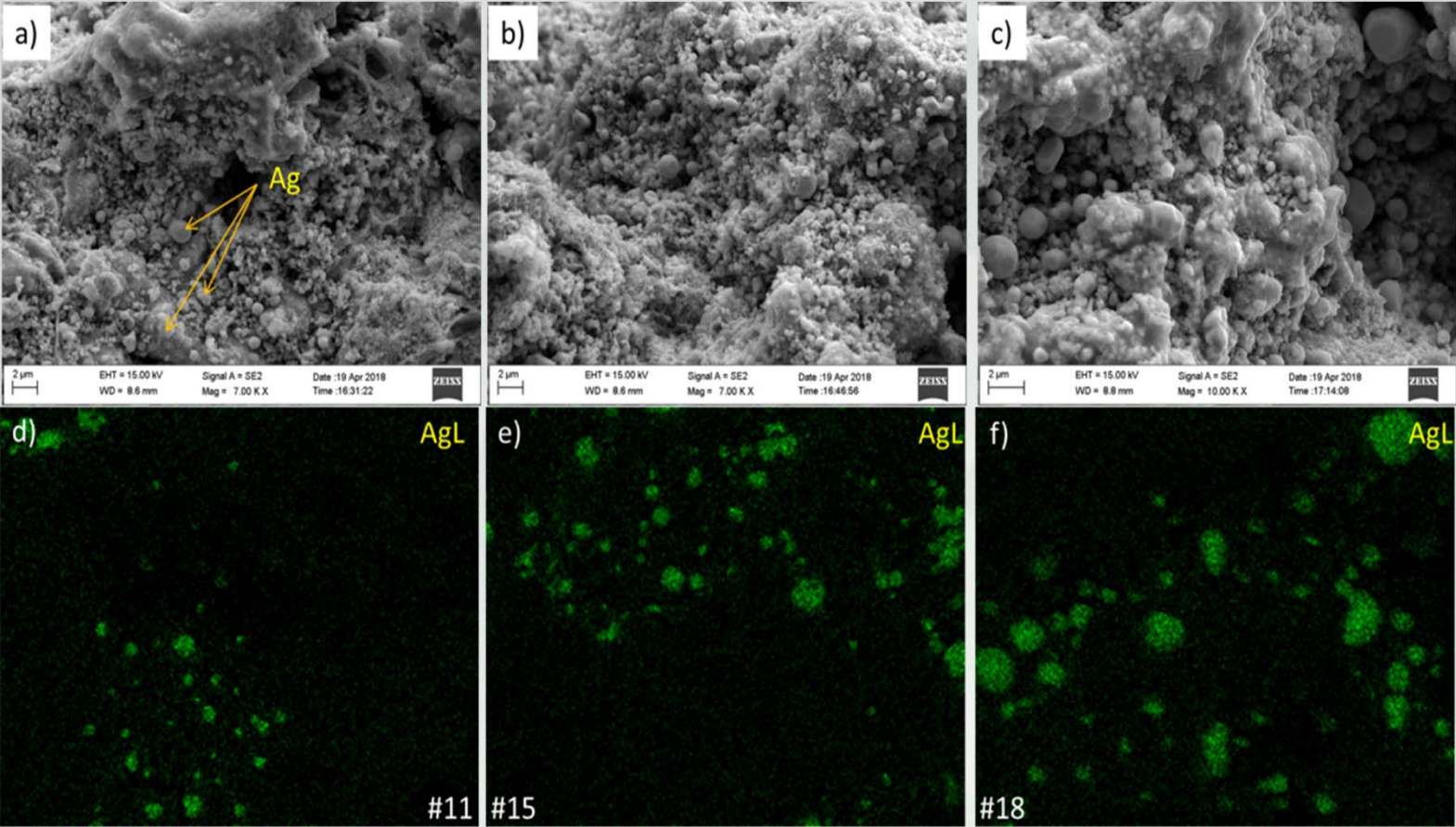
# Gold Current Collector Grids with Platinum Plate Push-Contacts were Used for the Measurements Here



# Silver Current Collectors Result in Strange, Unreproducible $R_p$ Behavior at Elevated Temperatures



# Silver Should NOT Be Used as Current Collector Because of Its Migration at Elevated Temperature



Electrode



Electrolyte

Electrode



Electrolyte



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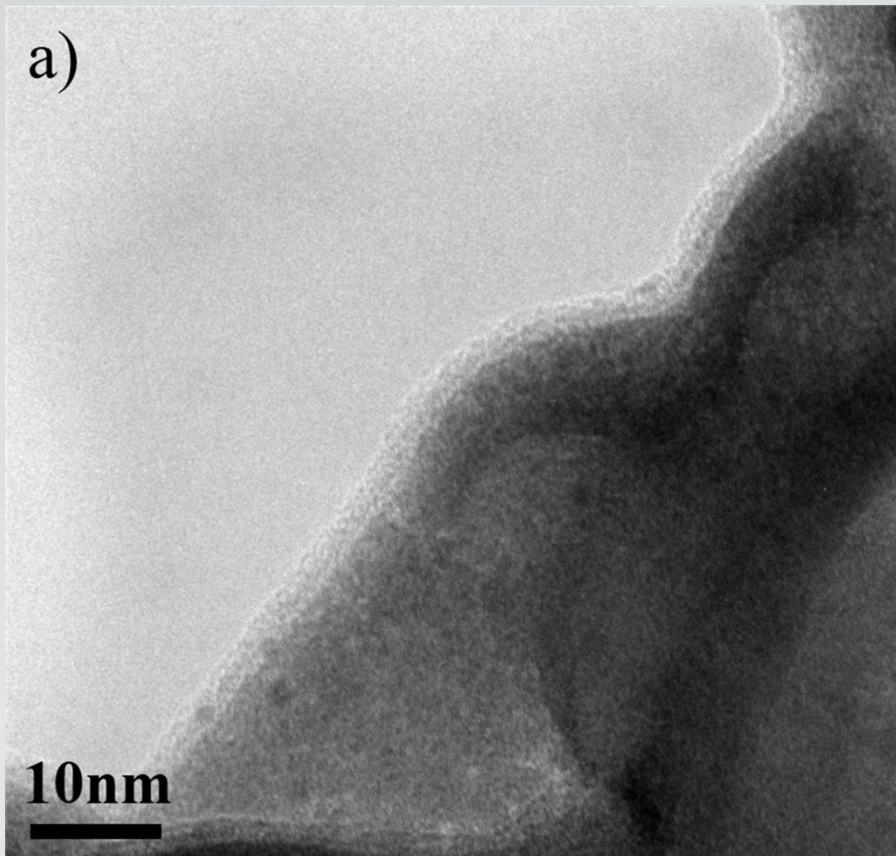
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## Ni-Ag Circuits Pastes

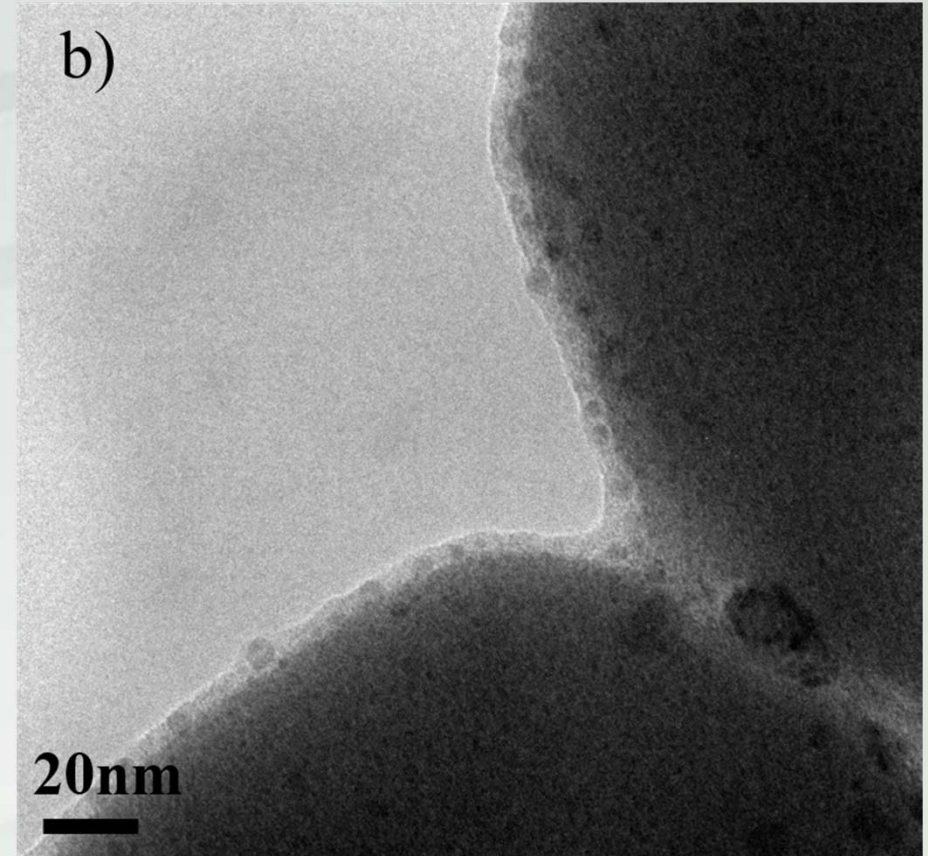
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## ZrO<sub>2</sub> Overcoat Maintained Conformal Target Thickness After 1000 Hours at 650°C

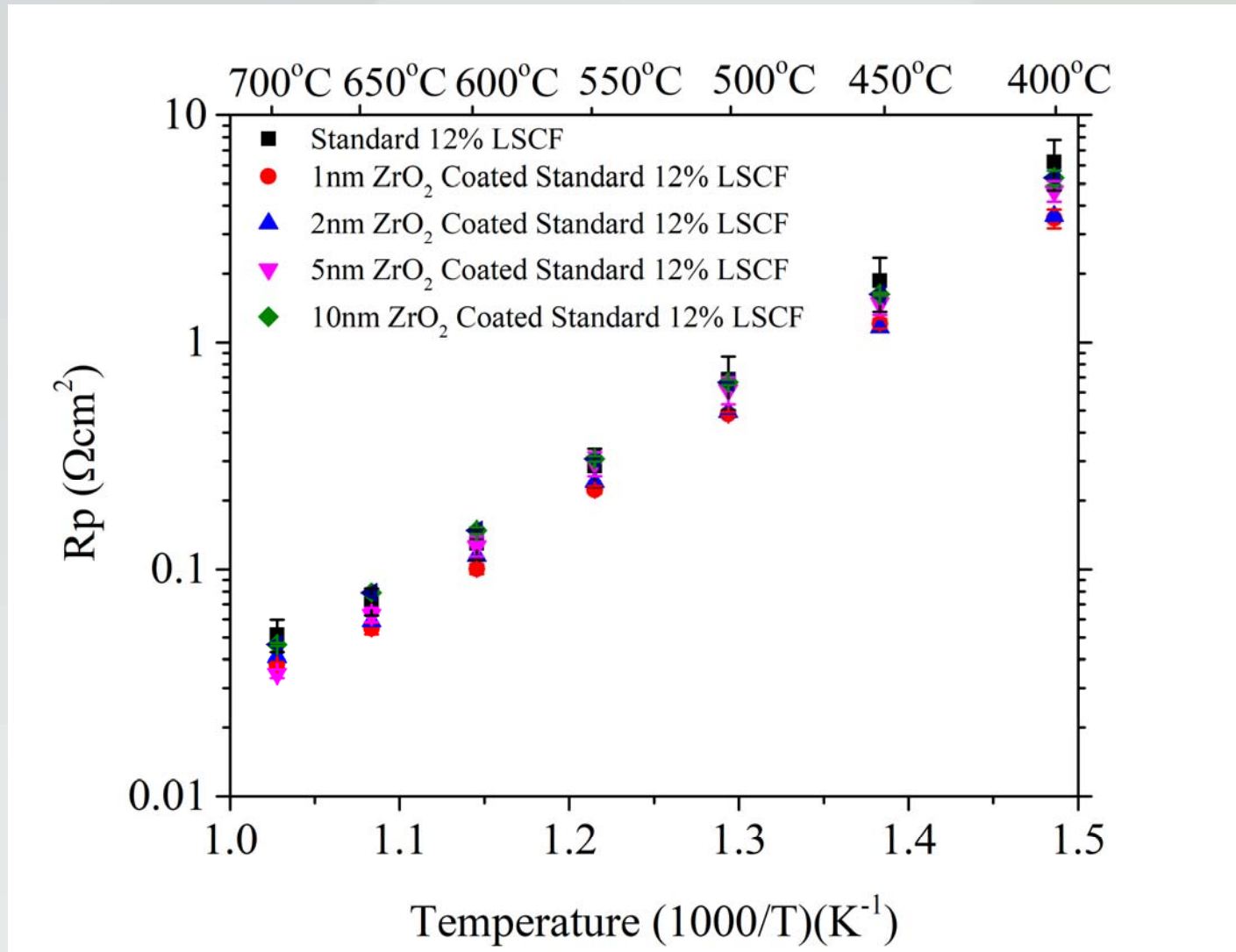


Tested cell with 3nm ZrO<sub>2</sub> overcoat (actual thickness ~3.2nm)



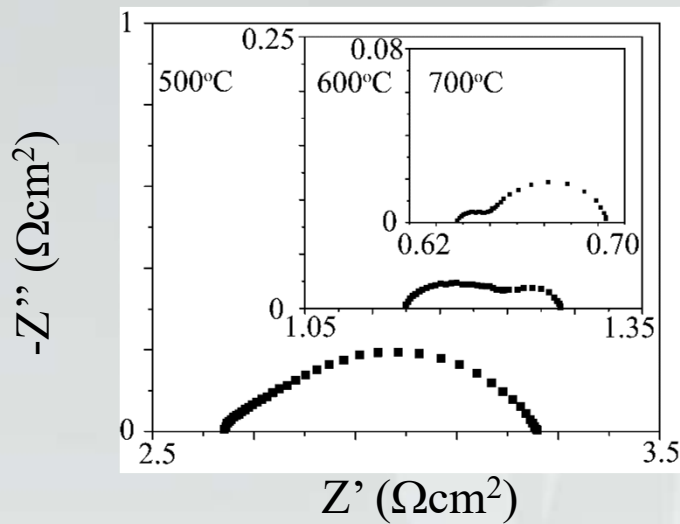
Tested cell with 5nm ZrO<sub>2</sub> overcoat (actual thickness ~5.3nm)

# 1-10 nm Thick Zirconia Overcoats Don't Dramatically Affect Conventional LSCF-GDC NCC Performance

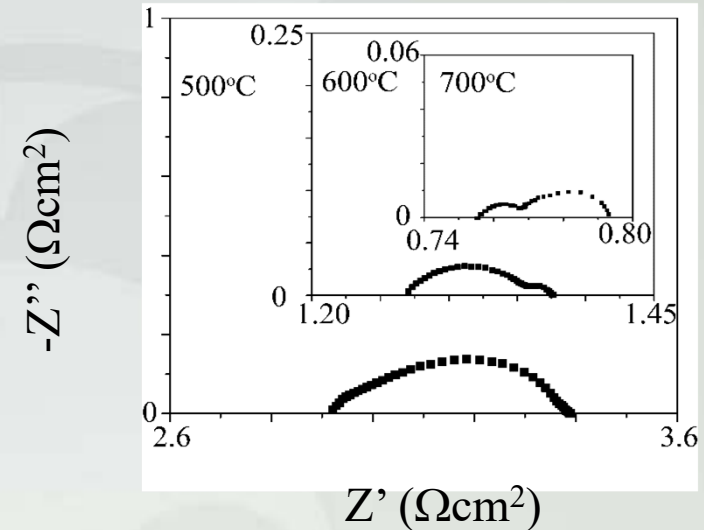


# 1-10 nm Thick Zirconia Overcoats Don't Dramatically Affect Conventional LSCF-GDC NCC Performance

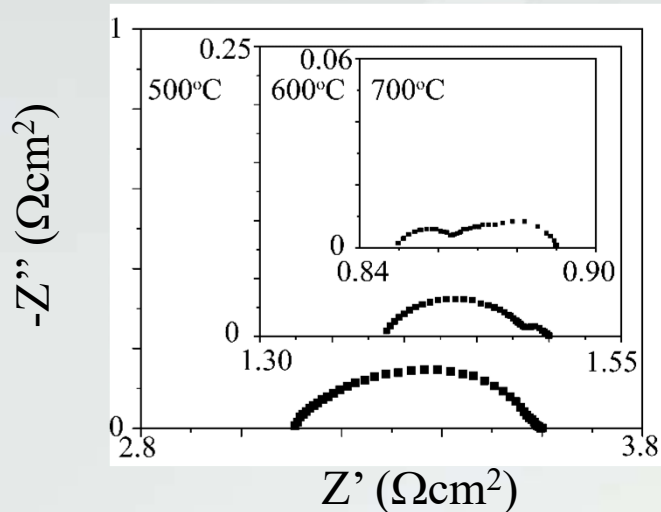
Standard LSCF-GDC



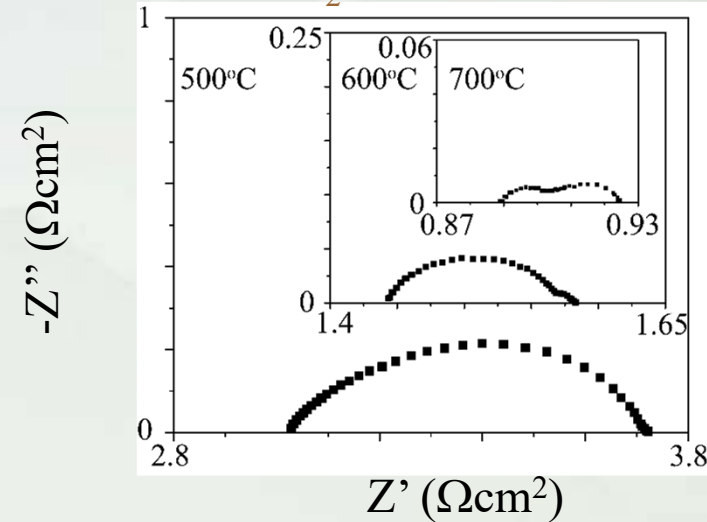
1nm  $\text{ZrO}_2$  coated LSCF-GDC



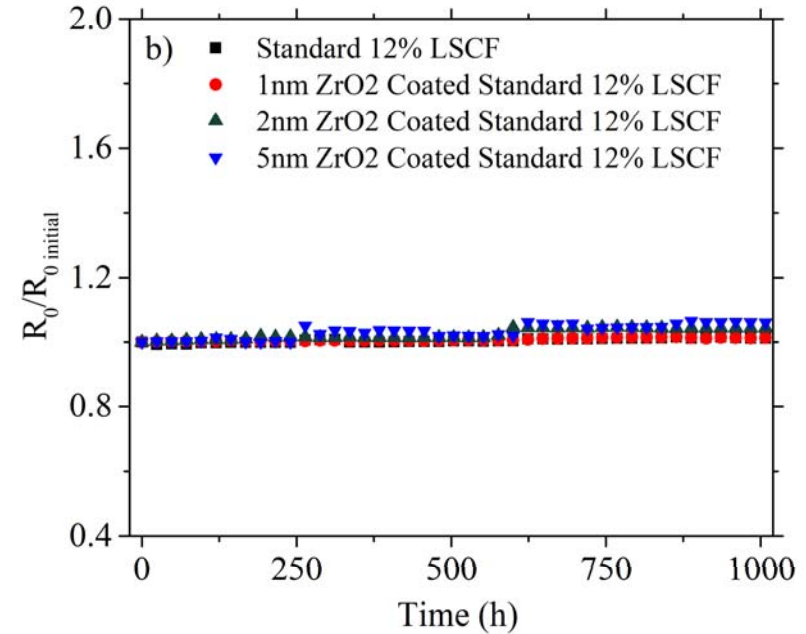
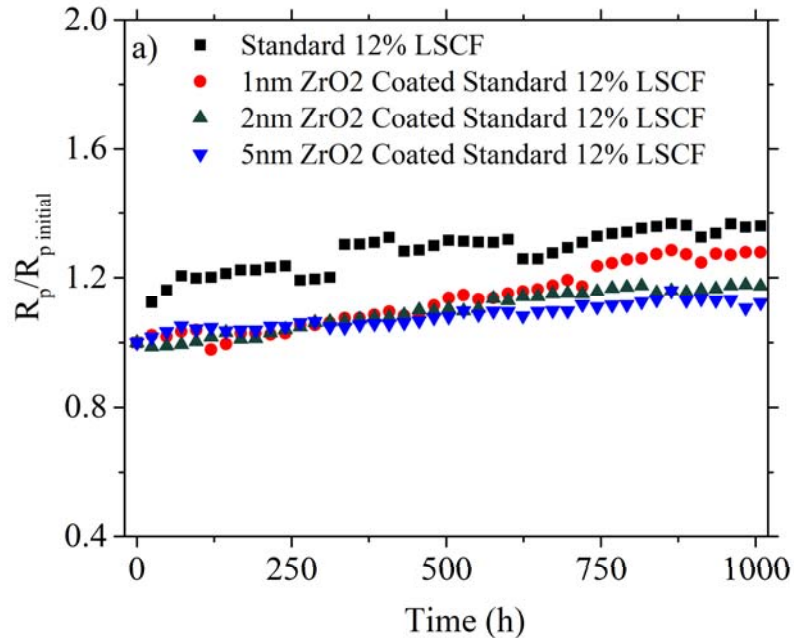
2nm  $\text{ZrO}_2$  coated LSCF-GDC



5nm  $\text{ZrO}_2$  coated LSCF-GDC



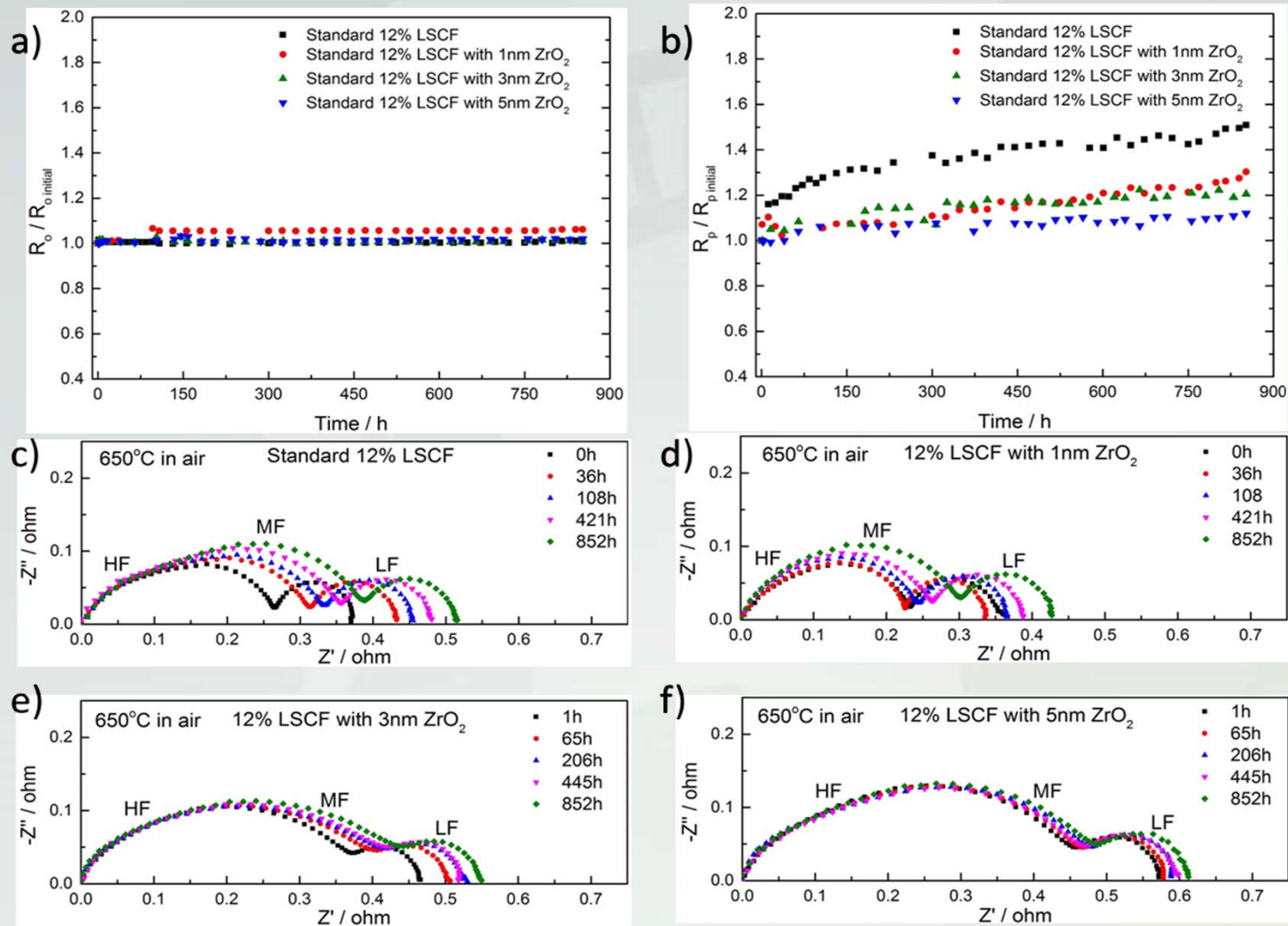
# Thicker Zirconia Overcoats Result in Lower 650°C $R_p$ Degradation Rates



	$R_p$ degradation rate (/khrs)	$R_0$ degradation rate (/khrs)
12% Standard LSCF	36.1% (0.043 $\Omega$ cm <sup>2</sup> /khrs)	1.9%
1nm ZrO <sub>2</sub> coated 12% LSCF	28% (0.015 $\Omega$ cm <sup>2</sup> /khrs)	1.4%
2nm ZrO <sub>2</sub> coated 12% LSCF	18% (0.015 $\Omega$ cm <sup>2</sup> /khrs)	4.2%
5nm ZrO <sub>2</sub> coated 12% LSCF	12% (0.011 $\Omega$ cm <sup>2</sup> /khrs)	6.3%

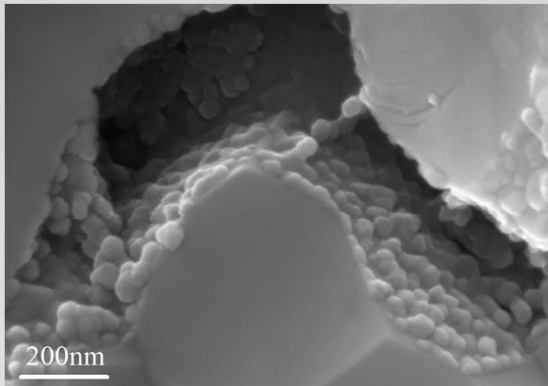


# Independent 650°C Degradation Tests at the University of South Carolina Showed Similar Results

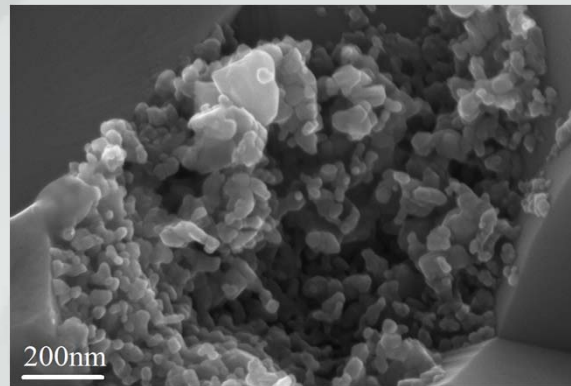




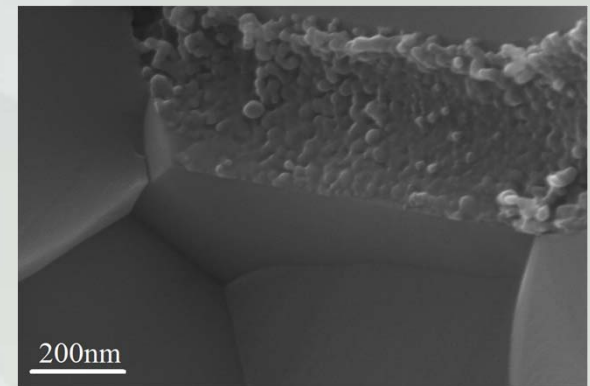
# No Obvious Particle Coarsening Was Observed for All Cells after 1000h Degradation at 650°C



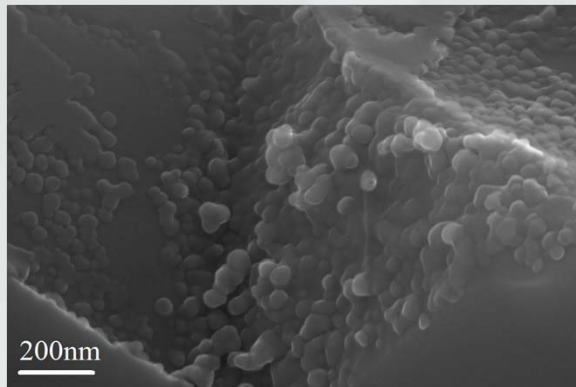
As-produced 12% standard LSCF-GDC NCC



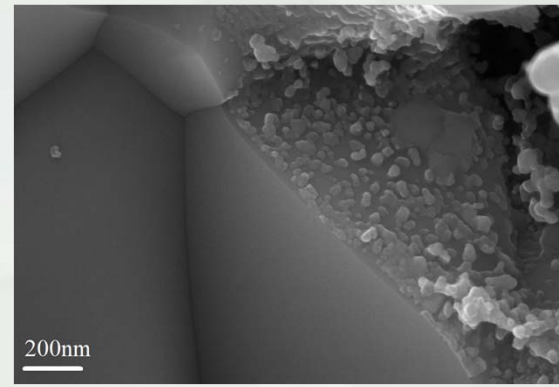
12% standard LSCF-GDC NCC after 1000h at 650°C



1nm ZrO<sub>2</sub> coated 12% standard LSCF-GDC NCC after 1000h at 650°C

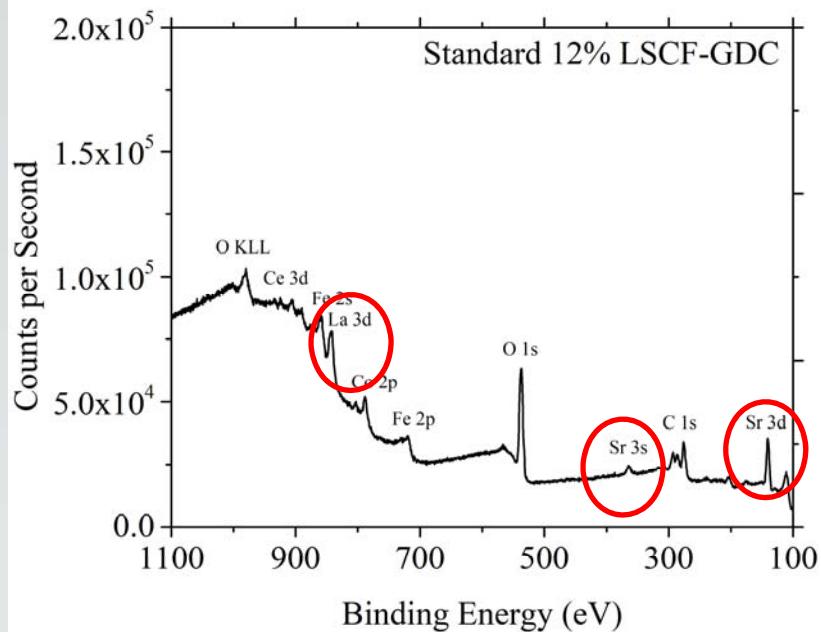


2nm ZrO<sub>2</sub> coated 12% standard LSCF-GDC NCC after 1000h at 650°C

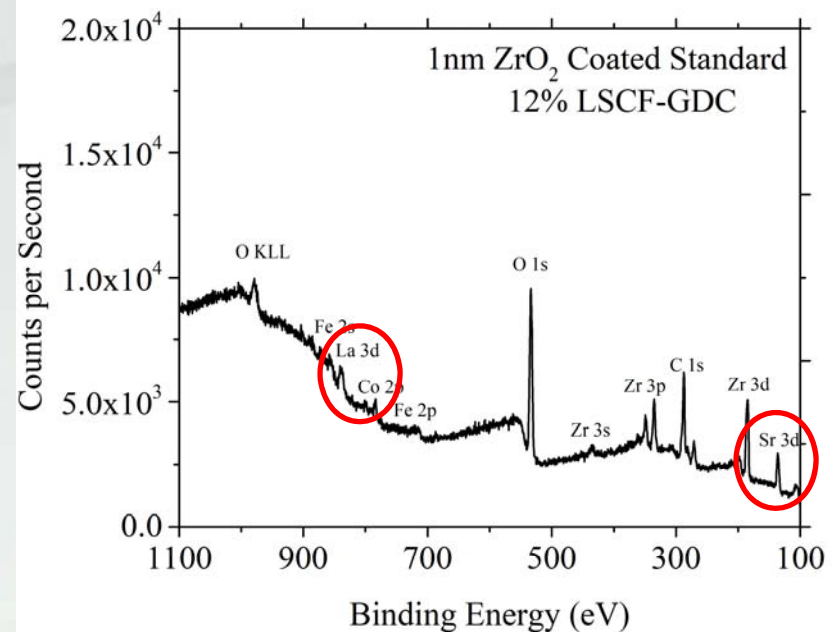


5nm ZrO<sub>2</sub> coated 12% standard LSCF-GDC NCC after 1000h at 650°C

# Suppression of Sr Segregation Is Suspected to Contribute to the LSCF-GDC NCC Performance Improvement with ZrO<sub>2</sub> Overcoat

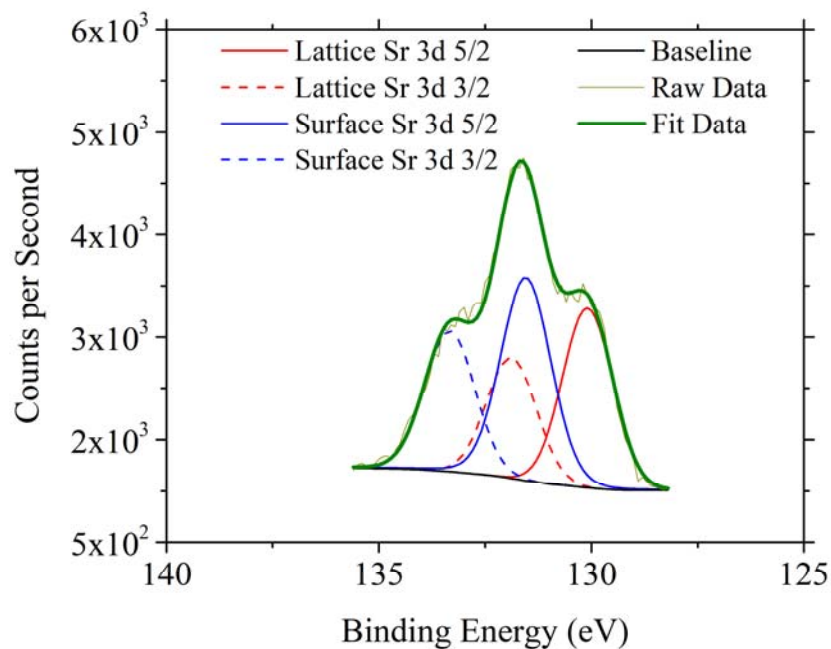


Standard 12% LSCF  
 $[\text{Sr}]/[\text{La}+\text{Sr}+\text{Co}+\text{Fe}]=51\%$



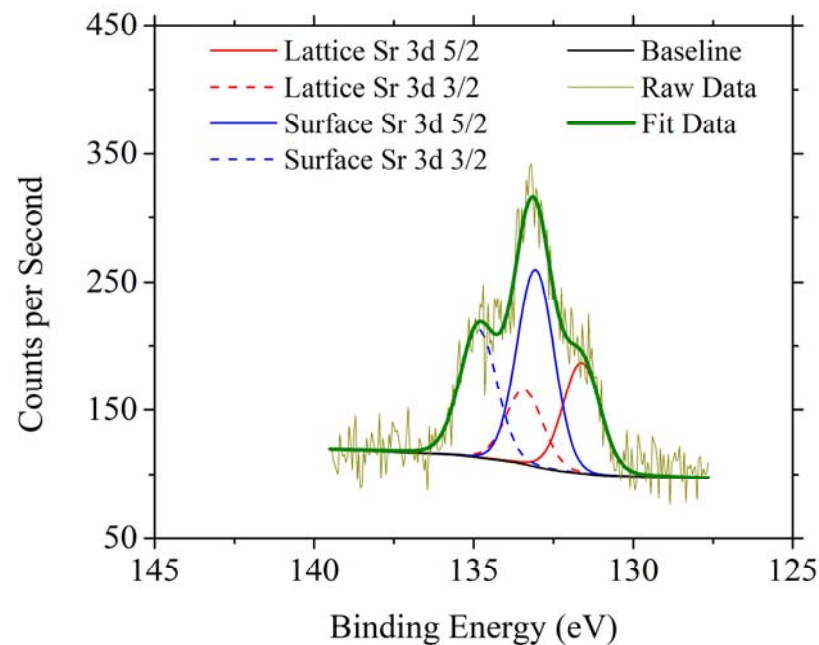
1nm ZrO<sub>2</sub> Coated Standard 12%  
 LSCF  
 $[\text{Sr}]/[\text{La}+\text{Sr}+\text{Co}+\text{Fe}]=42\%$

## XPS Deconvolution Suggests There is Not Much Change in the Partition Between Sr Surface Sites with ALD Overcoat



Standard 12% LSCF

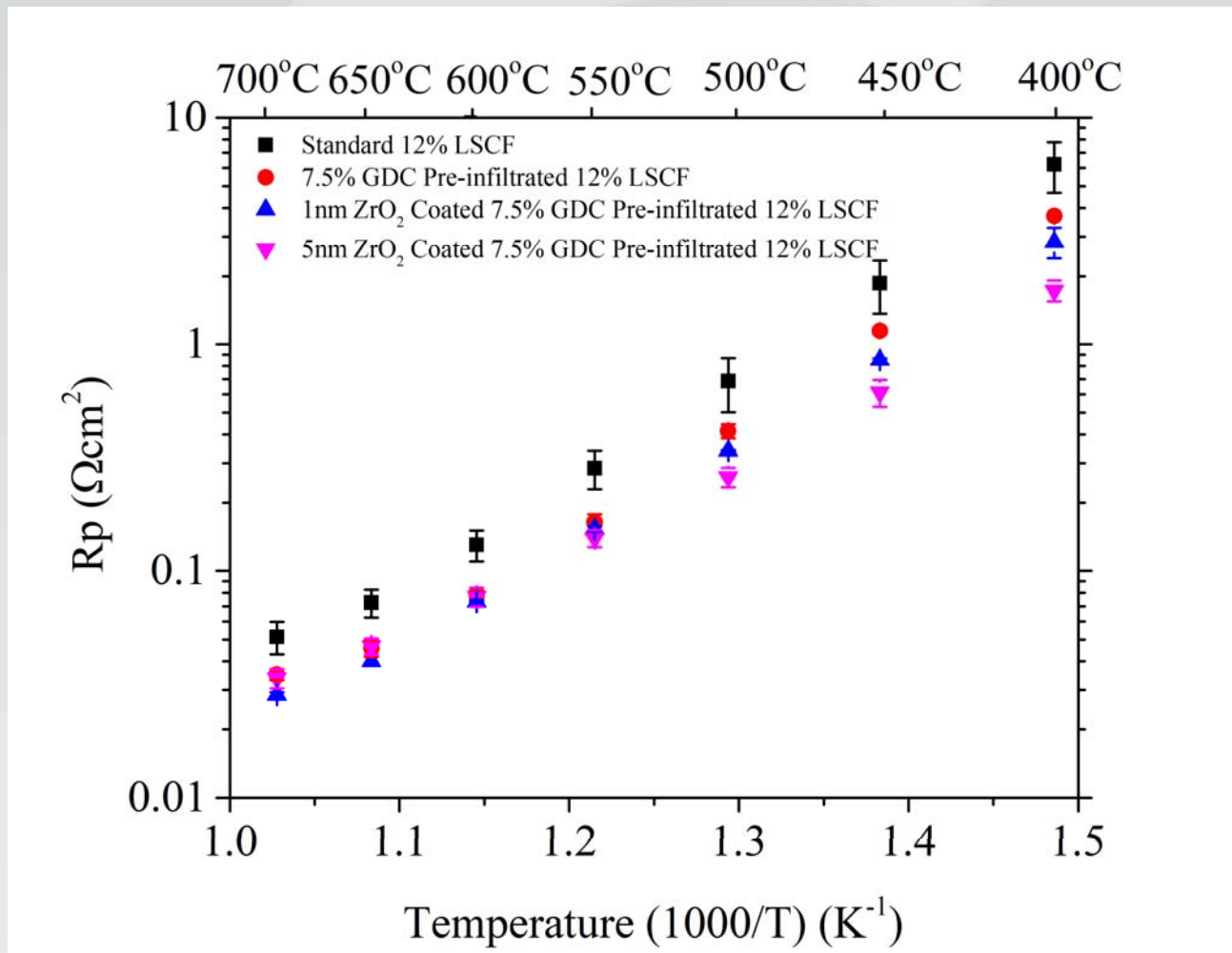
Surface Sr-- 57%



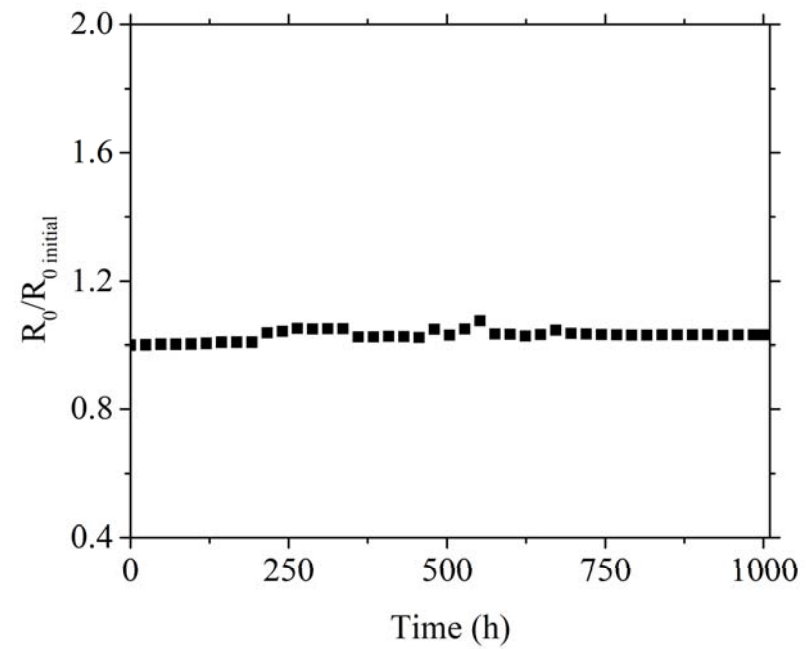
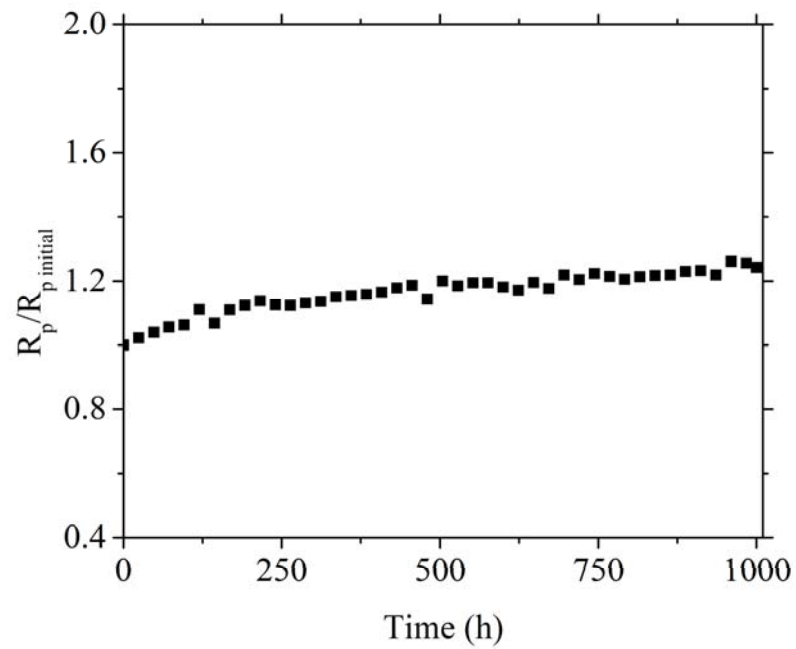
1nm ZrO<sub>2</sub> Coated Standard 12%  
LSCF

Surface Sr--63%

# $R_p$ Improvement in the Low Temperature Region Was Observed for Coated GDC Pre-infiltrated LSCF-GDC NCCs



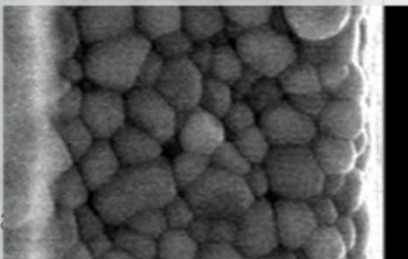
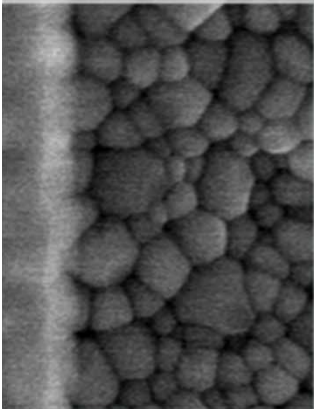
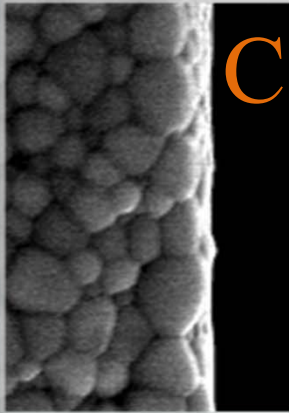
# 5nm ZrO<sub>2</sub> Coated 7.5% GDC Pre-infiltrated 12% LSCF Showed $R_p$ Degradation Rates of ~22%/khrs





## Conclusions

1.  $\text{ZrO}_2$  ALD overcoats of 1-5 nm in thickness improved the long-term stability of standard LSCF-GDC NCCs without significantly altering the 400 -700°C LSCF-GDC  $R_p$ .
2. A possible reason for the improved  $R_p$  is Sr surface segregation alterations induced by the zirconia overcoat.
3. ALD overcoats reduced the activation energy for oxygen incorporation into the GDC-preinfiltrated LSFC-GDC NCCs.
4. Although the results are promising, more work is needed to realize cathode degradation rates meeting DOE targets of <1%/khrs.





# High Performance Circuit Pastes for Solid Oxide Fuel Cell Applications

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Program Manager: Venkat Venkatraman

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# TALK Outline

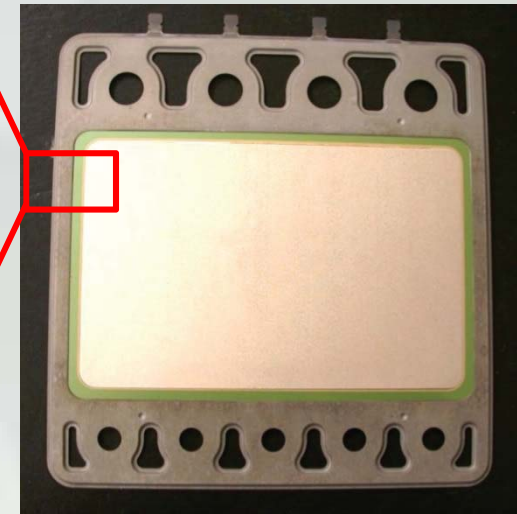
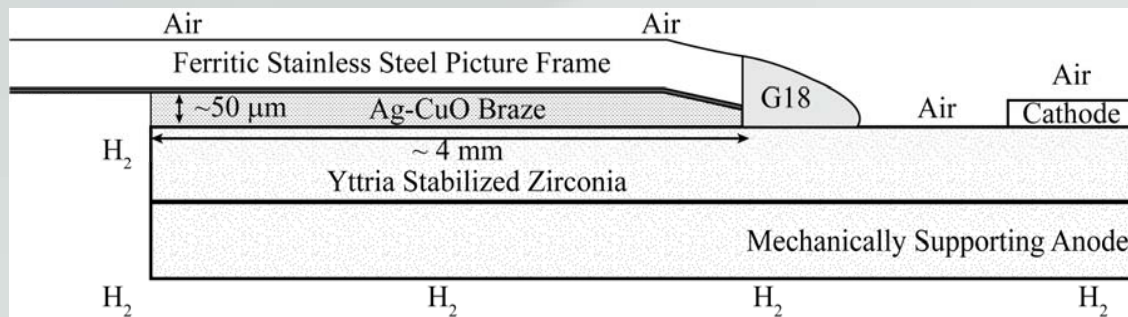
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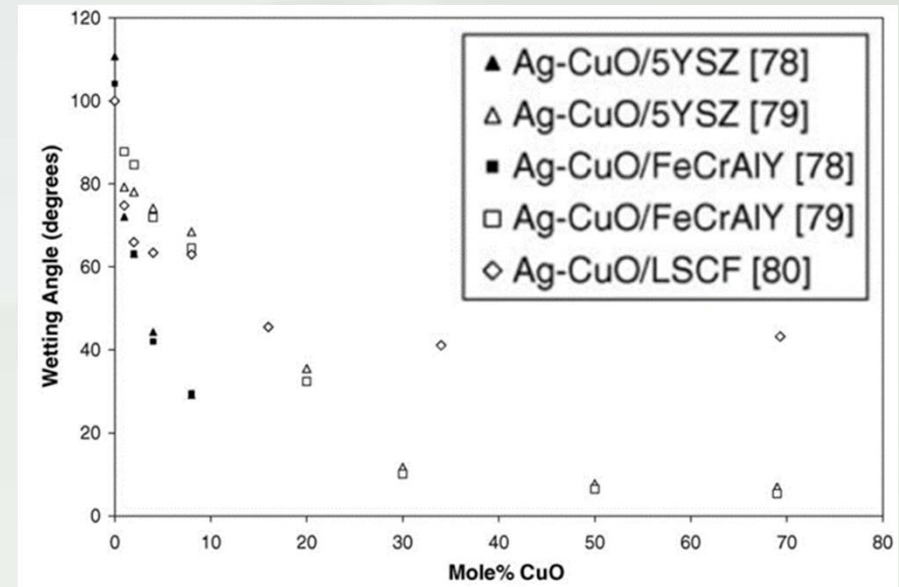
## Ni-Ag Circuits Pastes

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## Conventional Silver-Copper Oxide Reactive Air Brazes Have many Benefits



- Brazes are less Permeable to  $H_2$  and  $O_2$  than Glass Seals
- Brazes are more ductile than glass seals, so CTE Mismatch is not a concern
- Reactive air brazing can be performed in air
- The enhanced wetting provided by CuO allows Ag-CuO brazes to be used on a variety of ceramics



Delphi Technologies Inc, U.S. Patent No US7855030B2

Kim, J.Y., J.S. Hardy, and K.S. Weil, *Journal of the American Ceramic Society*, 2005. 88 (9): p. 2521-2527.

## Reactive Air Brazes Have Several Fatal Flaws

1. Braze joint will be exposed to dual atmospheres ( $H_2$ /Air) in SOFC operation.
2. Reactive air silver brazes are only partially wetting, resulting in occasional manufacturing defects (**Type I Pores**);
3. Reduction of reactive air additions (CuO) by hydrogen during SOFC operation can result in **Type II Pores**;
4. **Type III pore** formation due to  $H_2$  and  $O_2$  reaction. CuO additions do not prevent the formation of **Type III Pores** produced when hydrogen and oxygen dissolved in the braze meet and form water pockets.

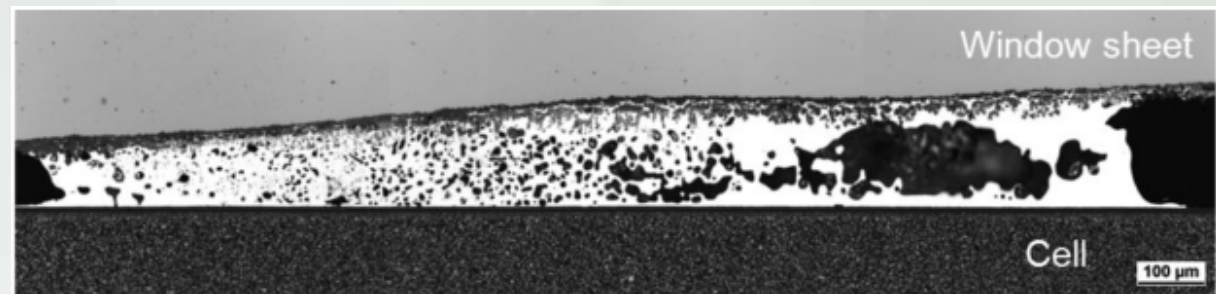
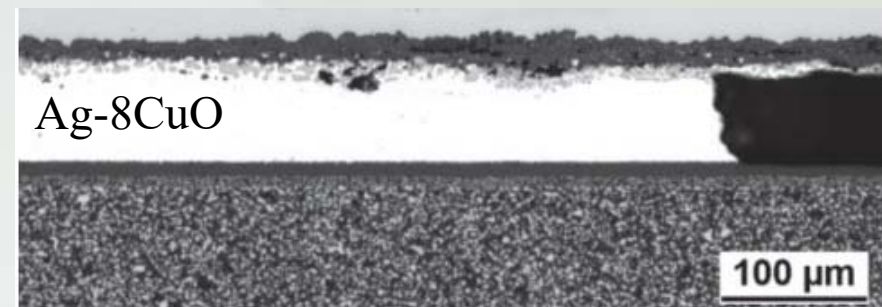
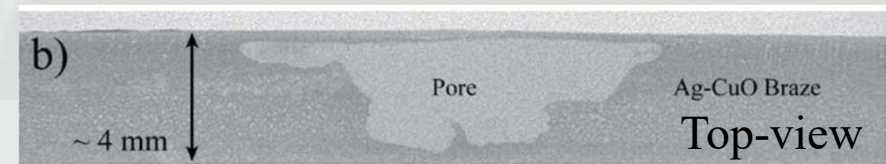
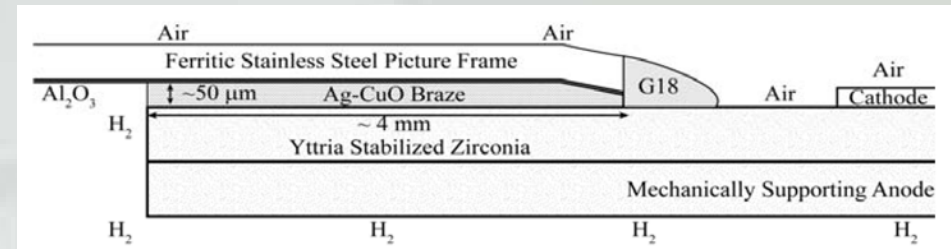
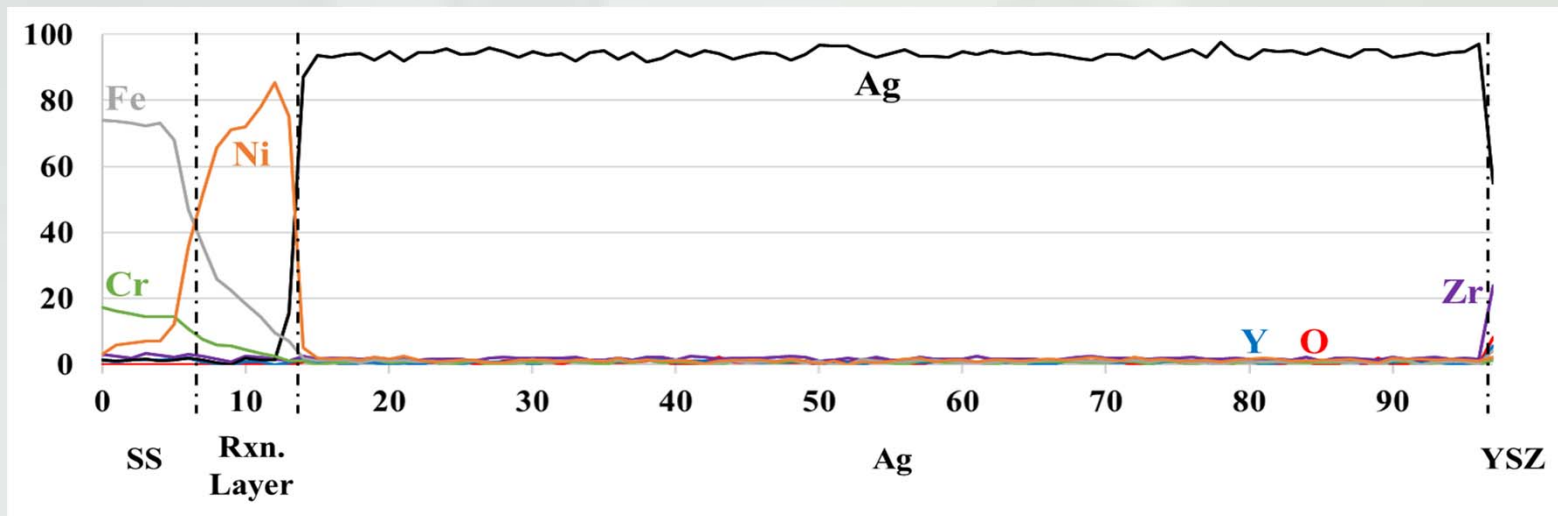
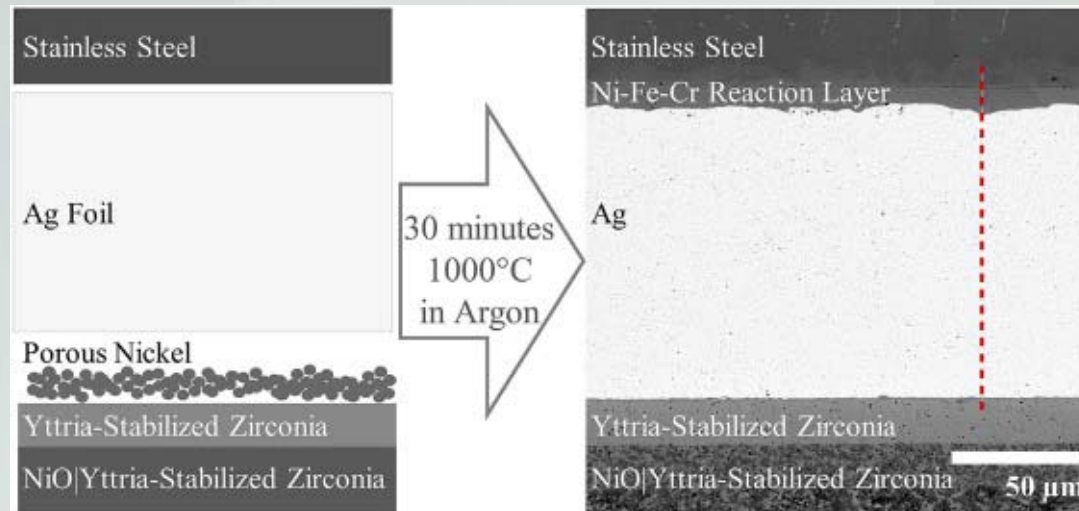


Image b) Courtesy of Delphi Automotive.

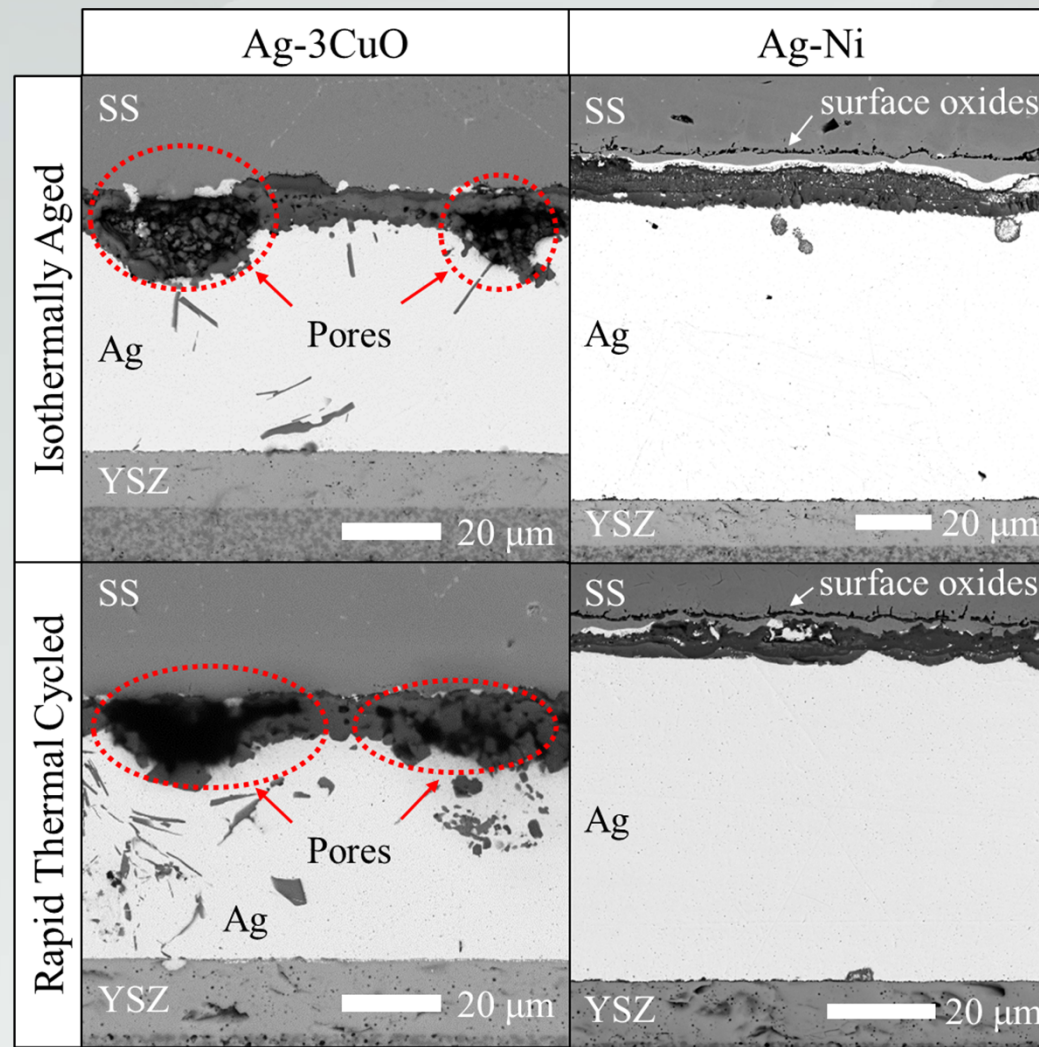
Bause, T., et al., Damage and Failure of Silver Based Ceramic/Metal Joints for SOFC Stacks. Fuel Cells, 2013. 13 (4): p. 578-583.



## Dense Braze Joints Can be Produced and the Porous Nickel Layer is Transient



## Ag-Ni Brazes Show Superior Dual Atmosphere tolerance After 300 hrs of Isothermal 750°C Aging, or 300 25°C/min 35-830°C Rapid Thermal Cycles



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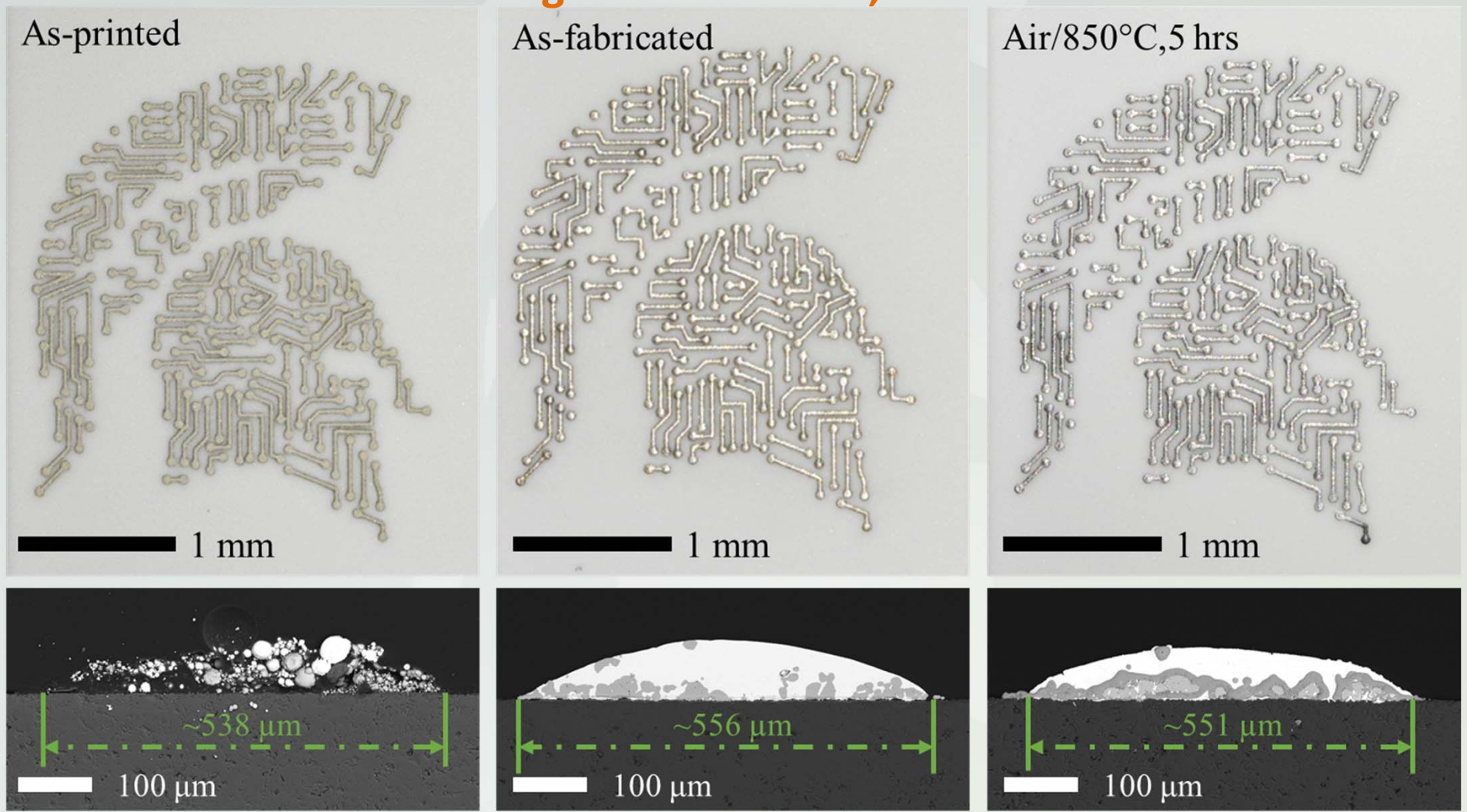
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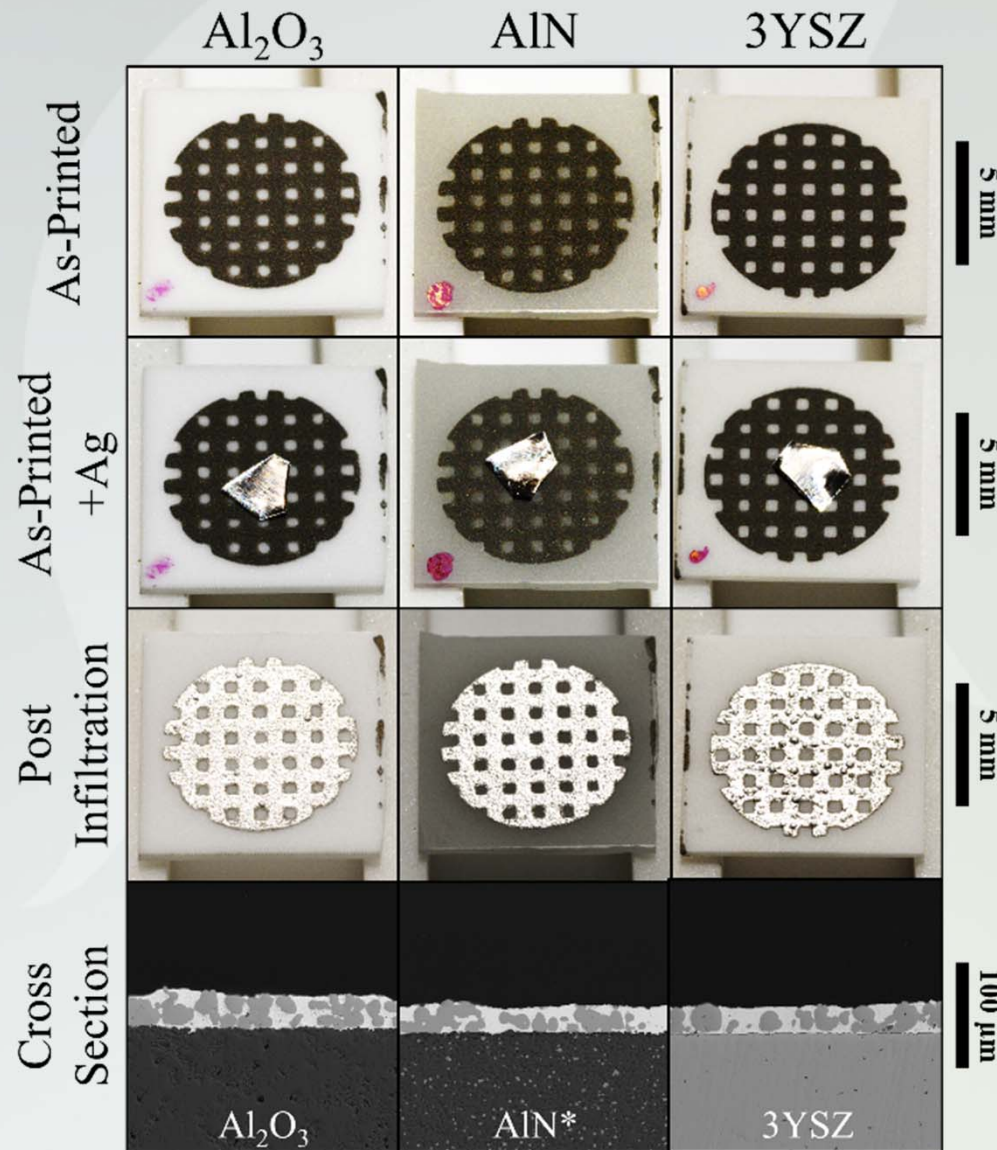


# The Controlled Wetting and Spreading of Ag can be used for High Temp Semiconductors, High Power SOFCs, Etc.



Zhou et al., *Controlled Wetting and Spreading of Metals on Substrates Using Porous Interlayers and Related Articles*, USPTO Provisional Patent (Submitted April 17, 2018)  
 Zhou et al., *Controlled Wetting and Spreading of Ag on Various Ceramic Substrates with Porous Ni Interlayers*, *Scr. Mater.*, 2018. (In Preparation)

# Molten Ag Will Also Infiltrate and Spread Through a Contiguous Ni Pattern

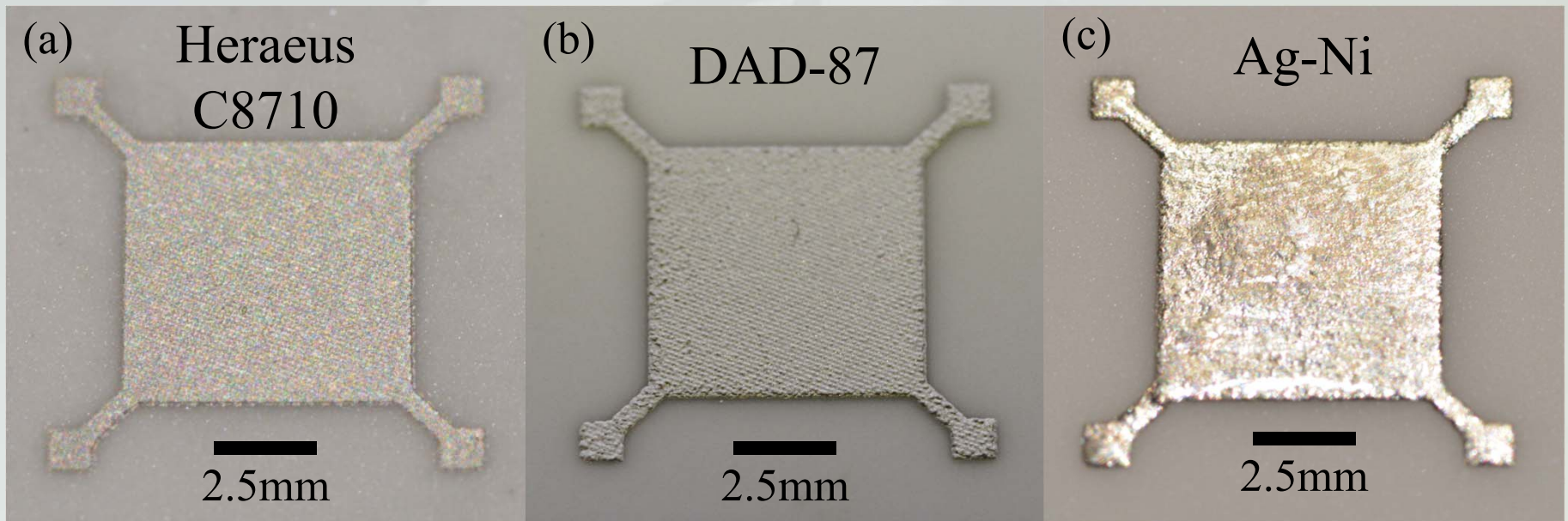


Zhou et al., *Controlled Wetting and Spreading of Metals on Substrates Using Porous Interlayers and Related Articles*, USPTO Provisional Patent (Submitted April 17, 2018)

Zhou et al., *Controlled Wetting and Spreading of Ag on Various Ceramic Substrates with Porous Ni Interlayers*, *Scr. Mater.*, 2018. (In Preparation)

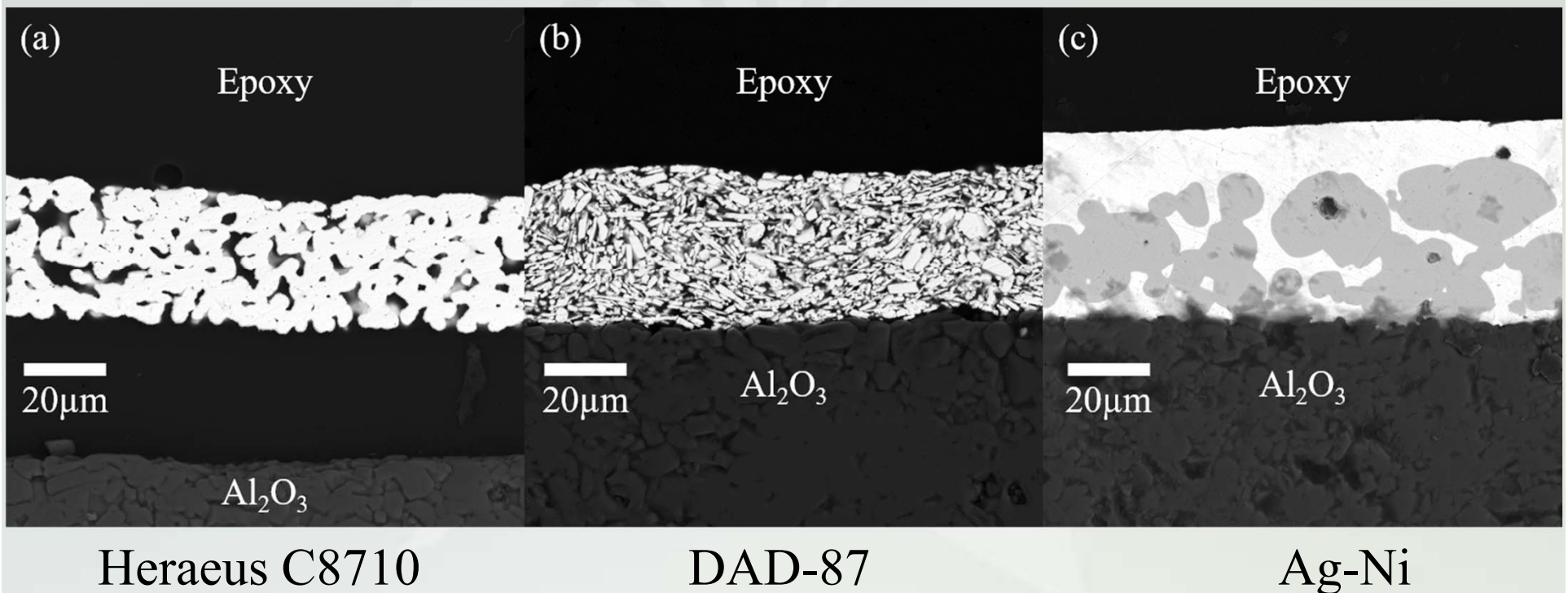


## The Electrical and Adhesion properties of Ag-Ni Circuit Pastes are Being Compared to Conventional Circuit Pastes

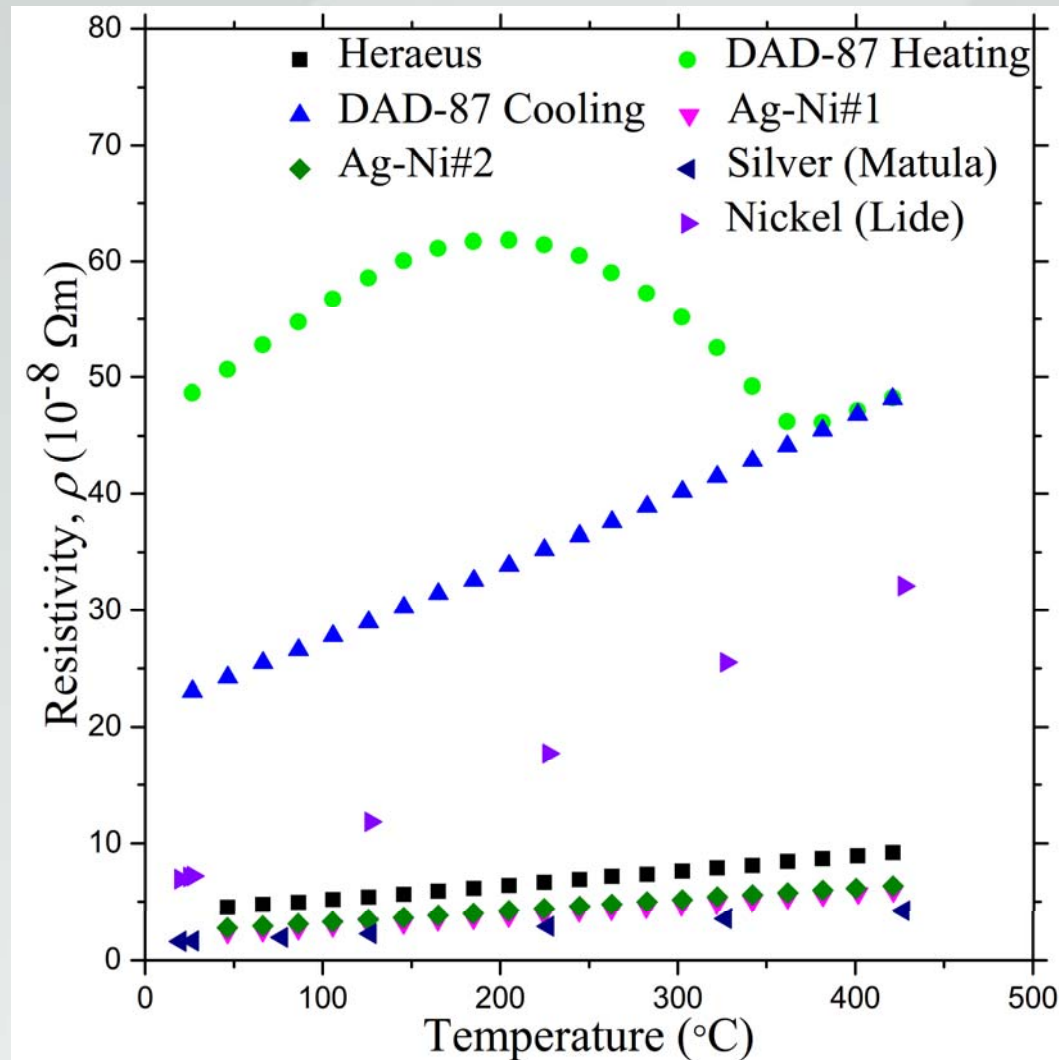


Screen printed samples for sheet resistance measurements

Following Manufacturer Suggested fabrication Procedures Results in Heraeus C870 Circuits that are delaminated, DAD-87 Circuits that are Nanoporous, and Ag-Ni Circuits that are Dense and Well Adhered



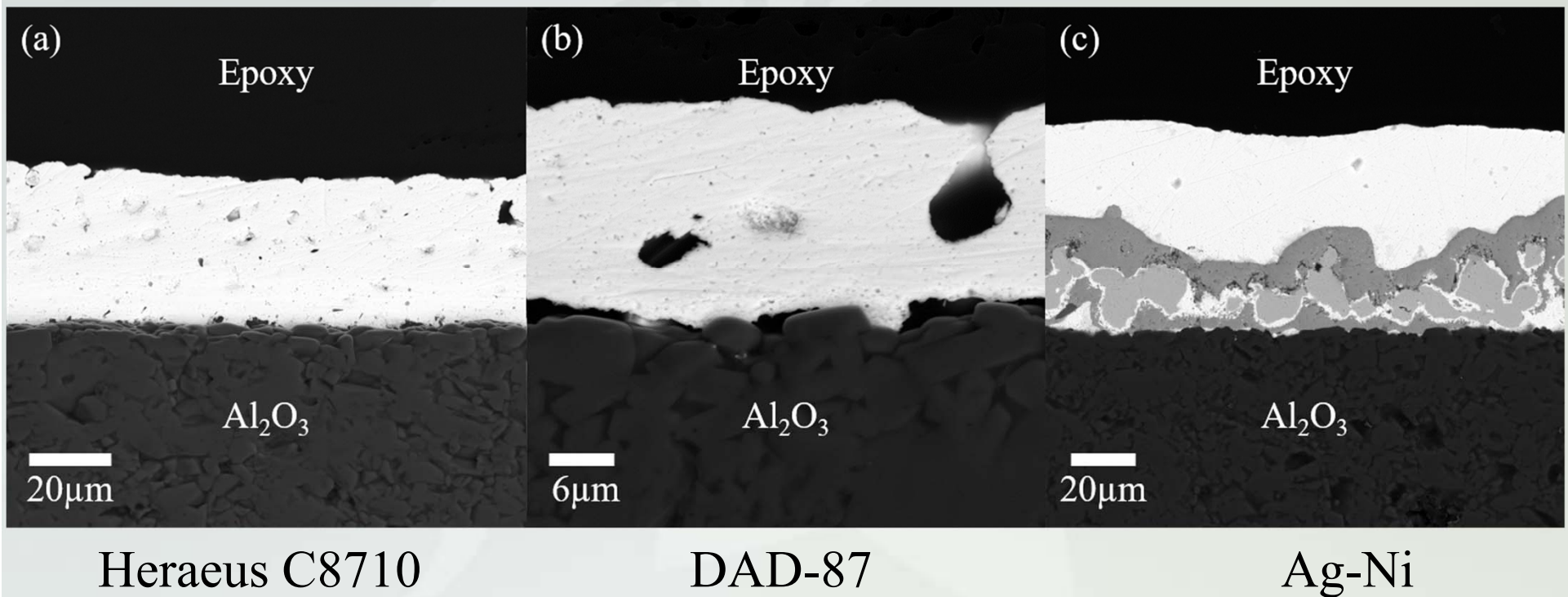
## The Ag-Ni Circuits Have a Resistivity Close to that of Pure Silver, as Expected



Matula, R.A., *Electrical resistivity of copper, gold, palladium, and silver*, *J. Phys. Chem. Ref. Data*, 1979. 8(4): p. 1147-1298.

Lide, D.R., *CRC Handbook of Chemistry and Physics, Internet Version*. 2005: CRC Press, Boca Raton, FL.

**10 Hours of 750°C Isothermal Aging Produces  
Dense, partially delaminated Heraeus C8710  
Porous, Well-Adhered DAD-87, and  
Dense, Well-adhered Ag-Ni with lots of Brittle NiO**

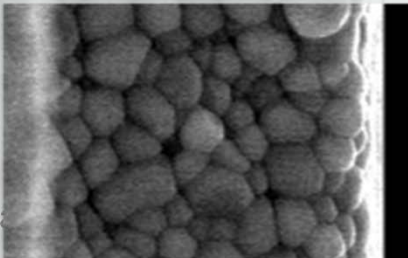
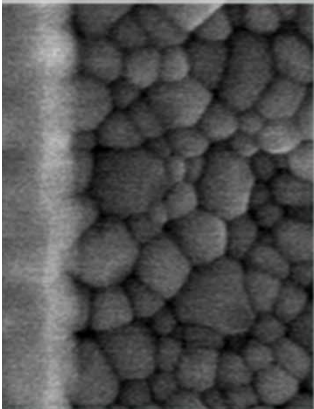
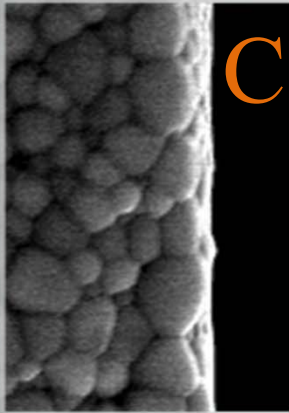


Note, the Heraeus C8710 is delaminated elsewhere



# Conclusions

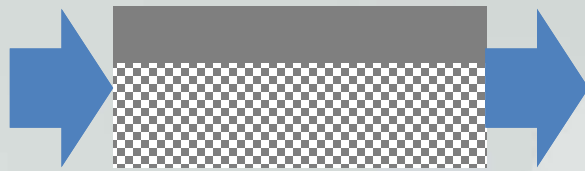
1. Conventional silver circuit pastes have significant delamination and/or conductivity problems
2. Ag-Ni circuits are one of the few ways to make dense, thick-film, well-bonded, low resistivity circuits for SOFCs, power electronics and other applications.
3. Techniques to eliminate NiO is needed in Ag-Ni circuits





# Backup Slides

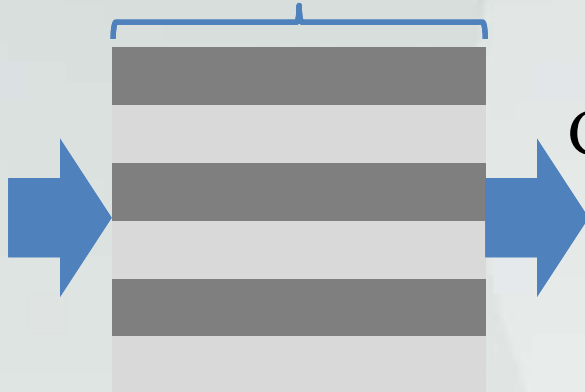
# Theoretical Resistivity of Ag-Ni Circuit



Two layers

Pure silver and silver-nickel mixture

Voltage,  $U$



Current,  $I$

$I$ , current through the circuit cross section  
 $U$ , voltage between two ends

Mixture layer minimum case

$R$ , resistance

$\rho$ , resistivity

$A$ , area

$l$ , length

$f$ , area fraction

$$R = \rho \frac{l}{A}$$

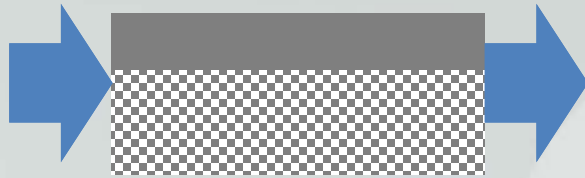
$$I = U \left( \frac{1}{R_{Ag}} + \frac{1}{R_{Ni}} \right)$$

$$\frac{1}{R_{mix}} = \frac{1}{R_{Ag}} + \frac{1}{R_{Ni}}$$

$$\frac{A}{\rho_{mix} l} = \frac{f_{Ag} A}{\rho_{Ag} l} + \frac{f_{Ni} A}{\rho_{Ni} l}$$

$$\frac{1}{\rho_{mix}} = \frac{f_{Ag}}{\rho_{Ag}} + \frac{f_{Ni}}{\rho_{Ni}}$$

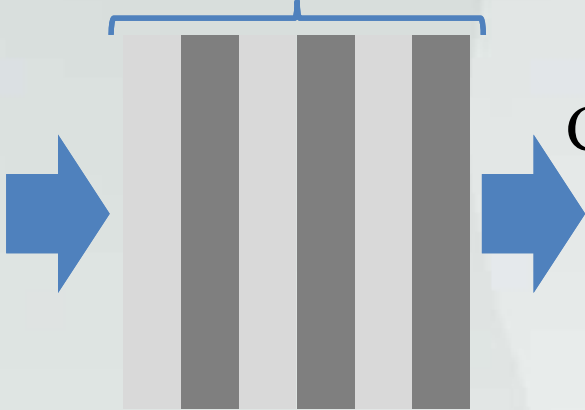
# Theoretical Resistivity of Ag-Ni Circuit



Two layers  
Pure silver and silver-nickel mixture

$$R = \rho \frac{l}{A}$$

Voltage,  $U$



Current,  $I$

$$U = IR_{Ag} + IR_{Ni}$$

$I$ , current through the circuit cross section  
 $U$ , voltage between two ends

$$R_{mix} = R_{Ag} + R_{Ni}$$

Mixture layer  
maximum case

$R$ , resistance

$$\rho_{mix} \frac{l}{A} = \rho_{Ag} \frac{lf_{Ag}}{A} + \rho_{Ni} \frac{lf_{Ni}}{A}$$

$\rho$ , resistivity

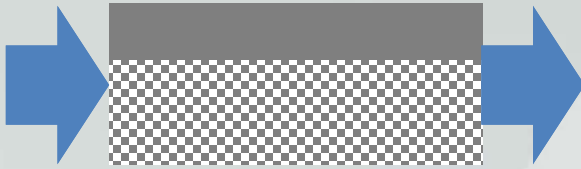
$A$ , area

$l$ , length

$f$ , area fraction

$$\rho_{mix} = \rho_{Ag}f_{Ag} + \rho_{Ni}f_{Ni}$$

# Theoretical Resistivity of Ag-Ni Circuit



Two layers

Pure silver and silver-nickel mixture

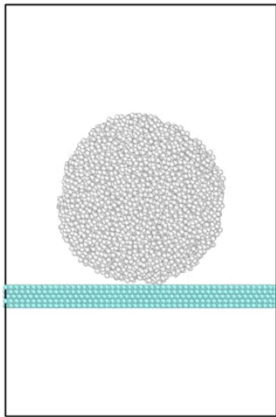
Overall circuit  
minimum case

$$\frac{1}{\rho_{overall}} = \frac{f_{Ag}f_{mix}}{\rho_{Ag}} + \frac{f_{Ni}f_{mix}}{\rho_{Ni}} + \frac{f_{pure}}{\rho_{Ag}}$$

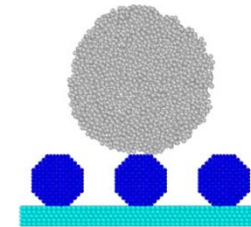
Overall circuit  
maximum case

$$\frac{1}{\rho_{overall}} = \frac{f_{mix}}{\rho_{Ag}f_{Ag} + \rho_{Ni}f_{Ni}} + \frac{f_{pure}}{\rho_{Ag}}$$

# Ni Particles Improve Ag Wetting on YSZ



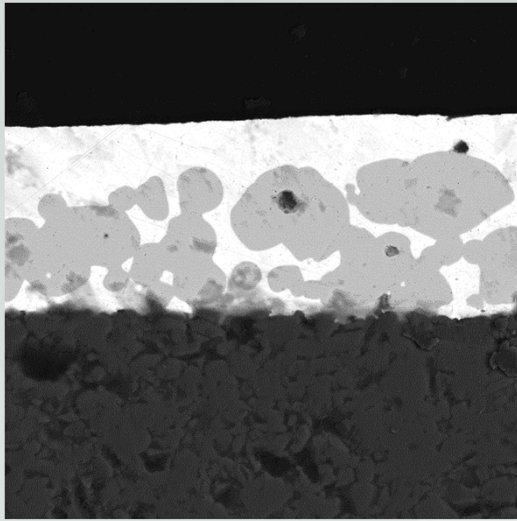
Ag on bare YSZ



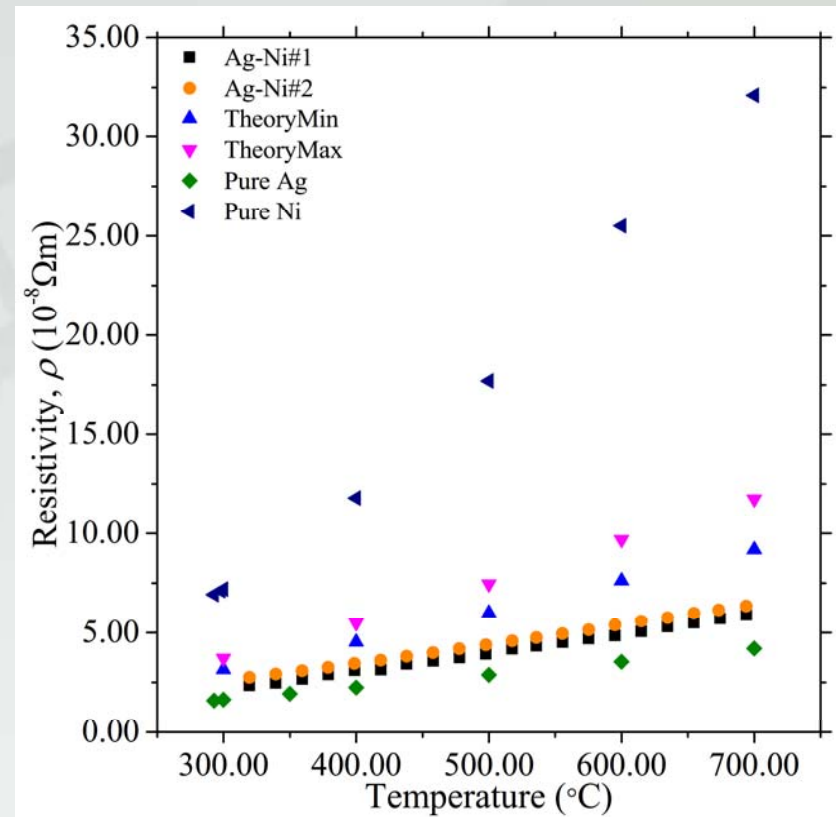
Ag on YSZ with  
Ni particles



# Theoretical Resistivity of Ag-Ni Circuit

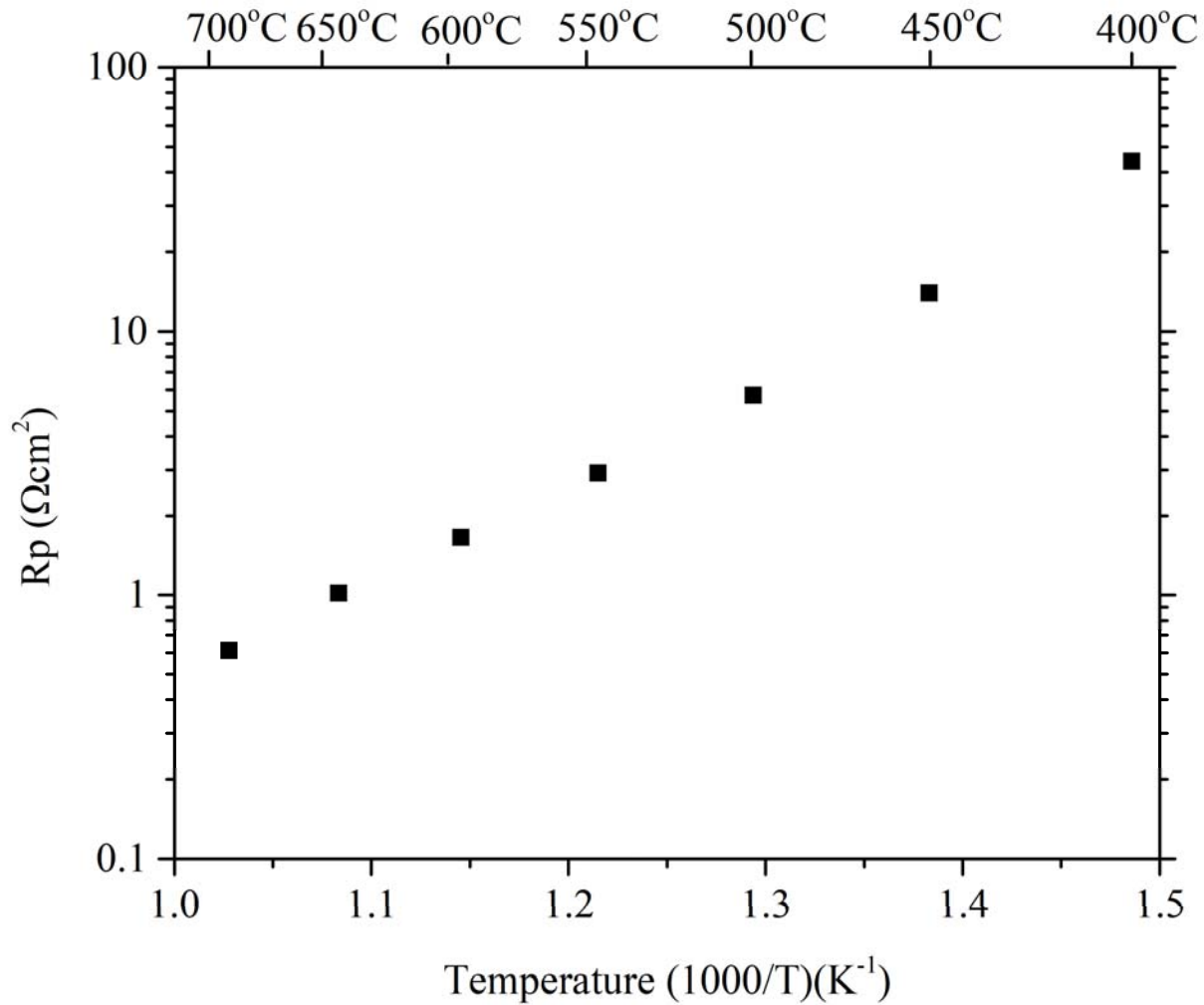


In calculation, area fraction is transformed to atomic fraction.

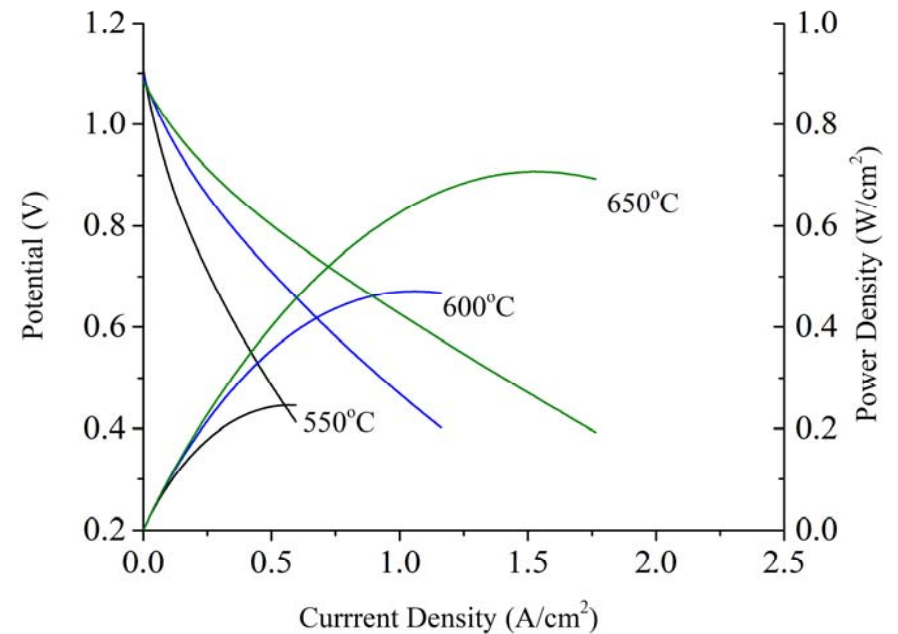
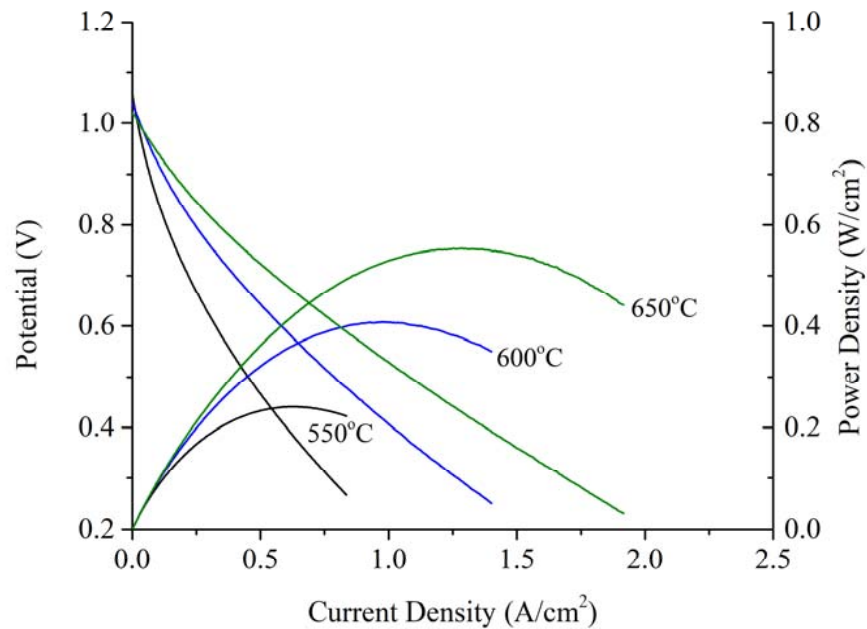


Explanation for deviation of theory: the fractions of Ag and Ni are measured in local areas, different from the overall relationship between Ag-Ni in the circuit.

## $R_p$ for Infiltrated $\text{Co}_3\text{O}_4$ -GDC Is Not Satisfactory



# Anode NiO Infiltration Saw an Improvement in I-V Performance for LSCF-GDC NCC/YSZ/Ni-YSZ



12% LSCF-GDC NCC/YSZ/Ni-YSZ

12% LSCF-GDC NCC/YSZ/Ni-YSZ  
with ~8% NiO anode infiltration